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Volume 13 Bioelectrochemistry

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Volume 13 Bioelectrochemistry

Edited by Richard C. Alkire, Dieter M. Kolb, and Jacek Lipkowski



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Preface

The intent of this edition is to provide an up-to-date account of the recent development in the fast-growing field of bioelectrochemistry. Significant methodological advances in studies of model biomembranes supported at electrode surfaces provided new tools for drug screening, detection of proteins, and electrochemical DNA screening. Significant progress in the understanding of the enzymatic reactions at electrode surfaces lead to development of biofuel cells. The knowledge of membrane and cell electroporation by high electric fields finds application in tumor therapy. This volume reviews the progress in bioelectrochemical science with a particular emphasis on the recent methodological developments and new applications of biochemistry for analytical detection, in medicine, and for energy conversion in biofuel cells. The volume should be of interest to students and researchers working in several fields such as electrochemistry, biochemistry, and analytical and medicinal chemistry. Each chapter provides sufficient background material so that it can be read by a non-specialist and specialist alike.

August 2011 University of Guelph, Guelph, Ontario, Canada Jacek Lipkowski

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1

Amperometric Biosensors

Sabine Borgmann, Albert Schulte, Sebastian Neugebauer, and Wolfgang Schuhmann

1.1 Introduction

The scope of this chapter is to review the advancements made in the area of amperometric biosensors. It is intended to provide general background about biosensor technology and to discuss important aspects for developing and optimizing biosensors. A major concern of this chapter is also to critically review the benefits, limitations, and potential of the different approaches to biosensor research and its applications. An introduction to biosensor research is given (Section 1.1) before criteria of "good to excellent" biosensor research are outlined (Section 1.2), and a standard for characterizing biosensor performance is defined (Section 1.3). Endeavor has been made to define what "good to excellent" biosensor research represents.

Because of the volume of the literature regarding amperometric biosensors as well as space limitations it is not possible to cite any substantial contribution to the field. We selected – to the best of our knowledge – representative work that can be of use not only for beginners but also for advanced researchers in the field as a basis for discussion. Examples of success stories accomplished in biosensor research are given as case studies in Section 1.4. General milestones and achievements relevant to biosensor research and development are listed in Table 1.2. The final conclusions are given in Section 1.5.

A way to address the current impact of biosensor research on analytical chemistry, biochemistry, biology, and medicine is to have a look at the number of publications. Table 1.1 contains the number of articles and reviews with the keyword "biosensor" and related keywords published between 2005 and 2010. About 11 345 papers and 549 reviews have been published containing the keyword "biosensor."

Almost 2000 papers dealing with glucose or employing glucose oxidase as biological recognition element have been published during the last five years. Glucose sensing is one of the success stories of biosensing. The health and the quality of life of diabetes patients depend on the accurate monitoring of their blood glucose levels by means of glucose biosensors. [59–65] The widespread use of glucose

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Table 1.1 Numbers of papers published in important fields of biosensor research between 2005 and 2010.

Keywords	Number of papers published	Number of reviews published
"Biosensor"	11 345	549
"Biosensor" and "glucose"	1974	96
"Biosensor" and "glucose oxidase"	1 3 3 1	37
"Biosensor" and "laccase"	109	7
"Biosensor" and "cellobiose dehydrogenase"	11	0
"Biosensor" and "DNA"	2166	156
"Biosensor" and "disposable"	287	6
"Biosensor" and "amperometric"	1931	86
"Biosensor" and "electrochemistry"	1715	63
"Biosensor" and "reagentless"	177	3
"Biosensor" and "direct electron transfer"	570	25
"Biosensor" and "mediated electron transfer"	53	2
"Biosensor" and self-assembled monolayer"	389	12
"Biosensor" and "conducting polymer"	220	20
"Biosensor" and "osmium"	40	0
"Biosensor" and "PQQ"	26	6
"Biosensor" and "NADH"	190	5
"Biosensor" and "biofuel cell"	73	8
"Biosensor" and "microsensor"	45	2
"Biosensor" and "microelectrode"	185	16
"Biosensor" and "microarray"	257	26
"Biosensor" and "biochip"	155	12
"Biosensor" and "protein chip"	304	14
"Biosensor" and "microfabrication"	57	3
"Biosensor" and "microfluidics"	184	15
"Biosensor" and "scanning electrochemical microscope"	51	
"Biosensor" and "nano"	476	29
"Biosensor" and "nanobiosensor"	33	4
"Biosensor" and "nanomaterial"	56	16

Database search for publications of the latest five years was performed on 10 May 2010 with Web of Science (Thomson Reuters).

oxidase (GOx, EC 1.1.3.4) as analytical reagent has been reviewed in detail. [63, 66, 67] The success of GOx as biological recognition element for biosensors is not only due to the importance of its substrate glucose and its enzymatic performance but also to its outstanding high stability and relatively low price.

Thus, it is not surprising that GOx has also evolved into an initial testing tool for the primary evaluation of new biosensor architectures. It seems to be almost the indestructible "working horse" as a model system. However, one needs to be careful with the general applicability for transferring the findings from initial studies to other more challenging biological recognition elements without providing substantial experimental evidence. This highlights the importance of design-

ing smart electron transfer (ET) pathways allowing the use of a general biosensor design for more than one biological recognition element (and analyte).

The number of papers for the different keywords from Table 1.1 gives hints on the current trends in the field of biosensor research. As mentioned above, glucose sensing is an ongoing trend. Whereas "biosensing and DNA" (2166 papers, 156 reviews) is still a hot topic, especially for the area of low-cost diagnostic devices. The use of biosensor approaches to biofuel cells (about 80 papers in the last five years) is increasingly of interest, and the use of nanomaterials is just evolving and becoming a hot topic. Although, to the best of our knowledge, nanobiosensors not only fabricated out of nanomaterials but also with a transducer surface confined to nanometric dimensions have not yet been realized.

The number of publications, however, does not address the level of quality of the presented research. What, in fact, represents the outcome of all these publications? What represents the resulting scientific advancement? This may not be so easy to answer as it as first seems. Thus, we will first give a general introduction to amperometric biosensors in the rest of this section, before we address the quality issue of biosensor research in Sections 1.2 and 1.3 and before we present some of the success stories of biosensor research (Section 1.4) in order to address the questions mentioned at the beginning of this paragraph.

1.1.1 Definition of the Term "Biosensor"

The use of enzyme electrodes was reported for the first time in 1962 [68]. The term "biosensor" was introduced by Cammann in 1977 [6]. The IUPAC definition of a biosensor, however, was introduced as recently as 1999 to 2001 [3–5]. Figure 1.1 schematically summarizes the set-up of a biosensor. A biosensor is a device that enables the identification and quantification of an analyte of interest from a sample matrix, for example, water, food, blood, or urine. As a key feature of the biosensor architecture, biological recognition elements that selectively react with the analyte of interest (e.g., antibody–antigen or enzymatic reactions) are employed. It is important to note that the biological recognition element is either integrated within or in close proximity to the transducer. The transducer enables the transformation of the analyte recognition and/or catalytic conversion event into a quantifiable physical signal, for example, a current in an amperometric biosensor.

As outlined in Figure 1.1, a biosensor consists of different components. Examples of these components are given in Figure 1.2. It is obvious that there are many ways to design a biosensor architecture. A variety of biological recognition elements ranging from enzymes to antibodies can be employed. The compilation given in Figure 1.2 helps one to understand which parameters change during a biological recognition event in a biosensor. This knowledge is fundamental for developing and optimizing biosensors. The choice of the transduction process and transduction material is dependent on this knowledge as well as the chemical approach to construct the sensing layer on the transducer surface.

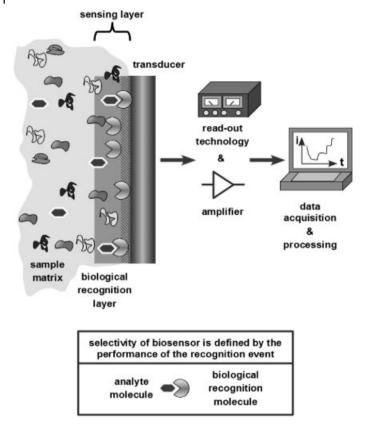


Figure 1.1 Typical biosensor set-up.

The choice of the biological recognition element is the crucial decision that is taken when developing a novel biosensor design. It is important to define criteria for, for example, a suitable redox enzyme for a specific biosensor. Most importantly, the enzyme needs to selectively react with the analyte of interest. The redox potential of the primary redox center needs to be within a suitable potential window (usually between -0.6 and $0.9\,\mathrm{V}$ vs. Ag/AgCl). The enzyme needs to be stable under the operation and storage conditions of the biosensor and should provide a reasonable long-term stability. It is advantageous if the chemical structure of the enzyme allows the introduction of additional functionalities for chemical modification with redox mediators, binding, or crosslinking with the immobilization matrix. In addition, the potential for tuning the properties of the redox enzyme by means of genetic or chemical techniques can be helpful for biosensor optimization. An important factor, especially with respect to potential commercialization, is that the redox enzyme is available at reasonable costs and effort.

biosensor components

biological recognition analyte element substrate of enzyme, enzymes, pH, antibodies, antigens, ions. receptors. small molecules, radicals, tissues, cells, peptides, proteins, bacteria, yeast, DNA. RNA. nucleic acids. toxins. biomimetics such as e.g. aptamers, ribozymes antigens, antibodies, haptens, viruses, bacteria change upon physical transduction recognition element principle - examples product or reactant, electrochemical pH, amperometric redox activity, potentiometric impedance ET. heat, capacitance, optical absorbance mass. electric potential, current, fluorescence conductance, or impedance chemiluminescence

Figure 1.2 Examples for biosensor components.

viscosity

The advantages of employing enzymes in biosensor architectures are the following:

- They exhibit a very high catalytic activity with a turnover on a per mole basis which makes them not only exceptional bioelectrocatalysts for effective signal amplification in biosensors but also for biofuel cells. Good turnover frequencies k_{cat} are in the range of up to at least $100 \,\text{s}^{-1}$.
- Typically, enzymes have a high selectivity for their substrates.
- In addition, the driving force, the redox potential that is needed to achieve enzymatic biocatalysis, is often very close to that of the substrate of the enzyme. Therefore, biosensors can operate at moderate potentials.

iv) In several cases, an improvement of the enzyme stability was found when enzymes were immobilized on transducer surfaces [25, 69].

The disadvantages of using enzymes in bioelectrochemical devices are the following:

- i) Enzymes are rather large molecules. Thus, despite the high catalytic turnover at the active site of the enzyme, the overall catalytic (volume) density is low. As an example, at most about a few picomoles of enzyme molecules per square centimeter are contained in a monolayer of enzymes. Barton and coworkers calculated that the theoretical current density in such a monolayer is about $80\,\mu\text{A}\,\text{cm}^{-2}$ under the assumption that the "footprint" of the enzyme is about $100\,\text{nm}^2$ and the turnover frequency is about $500\,\text{s}^{-1}$ [70].
- Often the active site of the enzyme is deeply buried within the surrounding protein shell. Thus, direct ET is often not possible and artificial redox mediators are required.
- iii) Enzymes have a limited lifetime and, therefore, biosensors exhibit only a limited long-term stability. So far, operational lifetimes of biosensors have been realized to up to 30 to 60 days [71, 72].

Mainly oxidoreductases have been employed for biosensors [73]. However, especially in the context of biofuel cell development, the spectrum of enzymes employed as bioelectrocatalysts is increasing [25]. For biosensor applications, it is important that the catalytic activity strongly depends on the substrate concentration which corresponds to an operating range of about the $K_{\rm M}$ value or below. This is important for obtaining a suitable dynamic range of the envisaged biosensor. In the case of blood glucose, for example, normal glucose levels are between 4 and 8 mM [74]. Typically, sugar-oxidizing enzymes have rather high $K_{\rm M}$ values (about 10 mM). Thus, if such enzymes are employed, the resulting biosensor can operate below substrate saturation. In contrast, in the case of biofuel cells the substrates are often present at concentrations well above the $K_{\rm M}$ value.

Electroanalytical techniques (also in combination with other techniques, e.g., optical techniques such as photometry and Raman spectrometry) can be employed to investigate many functional aspects of proteins and enzymes in particular. It is possible to study the biocatalytic process with respect to the chemistry of the active site, the interfacial and intramolecular ET, slow enzyme activators or inhibitors, the pH dependence, the transport of the substrate, and even more parameters. For example, slow scan voltammetry can be used to determine the relation of ET rates or of protonation and ligand binding. In contrast, fast scan voltammetry allows the determination of rates of interfacial ET. In addition, it is also possible to investigate chemical reactions that are coupled to the ET process, such as protonation. The use of direct ET for mechanistic studies of redox enzymes was recently reviewed by Léger and Bertrand [27]. Mathematical models help to elucidate the impact of different variables on the entire current signal [27, 75, 76].

1.1.2

Milestones and Achievements Relevant to Biosensor Research and Development

Biosensors have been studied extensively during the last fifty years. Hence, a number of milestones mark the progress made in biosensor research. Table 1.2 summarizes the main scientific milestones that are relevant to biosensor discovery and further development of this technology.

1.1.3

"First-Generation" Biosensors

Though many highly complex detection schemes can be found in biosensor designs, the simplest approach to a biosensor is the direct detection of either the increase of an enzymatically generated product or the decrease of a substrate of the redox enzyme. Additionally, a natural redox mediator that is participating in the enzymatic reaction can be monitored. In all three cases it is necessary that the compound monitored is electrochemically active. The use of GOx as biological recognition element for a "first-generation" biosensor design is the typical case and has been employed numerous times (Figure 1.3). Here, the increasing concentration of the product H2O2 or the decrease in O2 concentration as natural co-substrate can be electrochemically detected in order to monitor glucose concentration [68, 103, 110, 150, 151].

The major drawbacks of the first-generation biosensor approach are the following: (i) if the O₂ concentration is monitored, it is challenging to maintain a reasonable reproducibility due to varying O2 concentrations within the sample and (ii) working electrode potentials for either the oxidation of H₂O₂ or the reduction of O₂ are not optimal because these potentials are prone to the impact of interferences present in biological samples, such as ascorbic acid or dopamine.

1.1.4

"Second-Generation" Biosensors

In order to achieve biosensors which operate at moderate redox potentials the use of artificial redox mediators was introduced for the "second-generation" biosensors [135, 152–157]. Following the pioneering work by Kulys and Svirmickas [124, 125], Cass et al. were the first to show that an artificial redox mediator, ferrocene, could be employed for an amperometric glucose biosensor [135]. Figure 1.4 schematically explains how such a redox mediator can be used to read out the analyte concentration within a sample. The employed redox enzyme for the analyte of interest is able to donate or accept electrons to or from an electrochemically active redox mediator. It is important that the redox potential of this mediator is in tune with the cofactor(s) of the enzyme. Preferably, the redox mediator is highly specific for the selected ET pathway between the biological recognition element and the electrode surface. Note that the difference in potential between the different cofactors and the introduced artificial redox mediator should not be less than $\Delta E \sim 50 \,\mathrm{mV}$

 Table 1.2
 Milestones and achievements relevant to biosensor research and development.

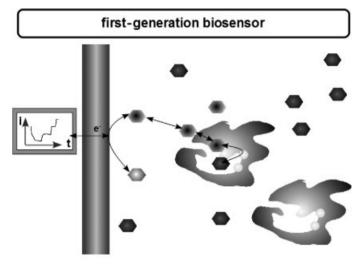
Year	Contribution
1800s	Alessandro Giuseppe Anastasio Volta (1745–1827) introduced modern electrochemistry, and found out at that the frog legs employed in the 1791 experiments of Luigi Galvani (1737–1798) to generate currents were not the true source for the stimulation. Actually, it was the contact between two dissimilar metals. He termed this type of electricity "metallic electricity" and demonstrated the first electrochemical battery using his voltaic piles [77].
1839	The principle of the fuel cell was discovered by Christian Friedrich Schönbein (1799–1868) presenting a hydrogen–oxygen fuel cell [78, 79]. Sir William Robert Grove (1811–1896) created one of the first fuel cells which he called a "gas battery" [80]. He also wrote one of the first books that stated the principle of conservation of energy in 1846. Grove is known as the "father of the fuel cell." Friedrich Wilhelm Ostwald (1853–1932), a founder of the field of physical chemistry, contributed significantly to the operation principles of fuel cells [81]. The term "fuel cell" became fashionable around 1889.
1889	Walther Nernst (1864–1941) introduced the Nernst equation [82].
1894	Emil Fischer (1852–1919) introduced the key-lock-principle (specific binding between enzyme and substrate) [83].
1913	Leonor Michaelis (1875–1949) and Maud Leonora Menten (1879–1960) developed the basis for enzyme kinetics and defined a mathematical model, the Michaelis–Menten kinetics [84].
1916	Immobilization of proteins (adsorption of invertase on activated charcoal) reported for the first time by Nelson and Griffin [85].
1922	Jaroslav Heyrovský (1890–1967) invented polarography and the use of the dropping mercury electrode for electroanalysis [86, 87]. Heyrovský and Masuro Shikata (1895–1965) developed a polarograph that was able to automatically record cyclic voltammograms and that was the first automated analytical instrument [88]. In 1959, Heyrovský received a Nobel prize for the development of polarography [89].
1925	George E. Briggs and John B.S. Haldane re-evaluated the Michaelis–Menten equation and contributed to the modern view on the steady-state treatment of enzyme-catalyzed reactions [90].
1926	Otto Warburg (1859–1938) discovered cytochrome c oxidase ("Warburg ferment"). This represents the basis for the description of the mechanism of cellular respiration (Nobel prize in 1931) [91]. Later, Warburg discovered the cofactors (NADH) and the mechanism of dehydrogenases. This leads to optical tests for NADH and NADPH which allows for testing the activity of dehydrogenases. This indicator reaction can be coupled with other enzyme reactions. These advancements were the basis of the work of Hans-Ulrich Bergmeyer (Boehringer Mannheim) promoting enzymatic analysis in the 1960s [92].
1950	Erwin Chargaff (1905–2002) discovered that the ratio of adenine to thymine and guanosine to cytosine is in all living creatures about 1 (Chargaff's rules) [93].
1953	James Dewey Watson (born 1928) and Francis Harry Compton Crick (1916–2002) developed a model for the structure of the double helix of DNA [94].
1955	Frederick Sanger (born 1918) determined the complete amino acid sequence of the two polypeptide chains of insulin. He received a Nobel prize in 1958 for his work on the structure of proteins, especially insulin [95].

Table 1.2 (Continued)

Year	Contribution
1956	Rudolph A. Marcus (born 1923) introduced a theory of electron transfer, named Marcus theory. He received the Nobel Prize in Chemistry for this achievement in 1992 [19, 29, 30].
1956	Leland C. Clark Jr. (1918–2005) presented his first paper about the oxygen electrode, later named the Clark electrode, on 15 April 1956, at a meeting of the American Society for Artificial Organs during the annual meetings of the Federated Societies for Experimental Biology [96].
	In 1962, Clark and Ann Lyons from the Cincinnati Children's Hospital developed the first glucose enzyme electrode. This biosensor was based on a thin layer of glucose oxidase (GOx) on an oxygen electrode. Thus, the readout was the amount of oxygen consumed by GOx during the enzymatic reaction with the substrate glucose [68]. This publication became one of the most often cited papers in life sciences. Due to this work he is considered the "father of biosensors," especially with respect to the glucose sensing for diabetes patients.
1957	The first crystal structures of proteins were resolved [97].
1959	Rosalyn Sussman Yalow (born 1921) and Solomon Aaron Berson (1918–1972) developed the radioimmunoassay (RIA) which allows the very sensitive determination of hormones such as insulin based on an antigen–antibody reaction [98, 99]. In 1997, Yalow received the Nobel Prize in Medicine for developing RIA. Today the RIA technology is surpassed by enzyme-linked immunosorbent assay (ELISA) because the colorimetric or fluorescent detection principles are favored over radioactive-based technologies.
1960s	General Electric (GE) developed a fuel cell-based electrical power system employing the so-called "Bacon cell" in order to maintain the Gemini and Apollo space capsules of NASA.
1963	Garry A. Rechnitz together with S. Katz introduced one of the first papers in the field of biosensors with the direct potentiometric determination of urea after urease hydrolysis. At that time the term "biosensor" had not yet been coined. Thus, these types of devices were called enzyme electrodes or biocatalytic membrane electrodes [100].
1964	For the first time, enzymes were used as fuel cell catalysts by Yahiro $\it et~al.$ in a glucose/O $_2$ biofuel cell [101].
1967	G.P. Hicks und S.J. Updike introduced the first practical enzyme electrode immobilizing the enzyme within a gel [102, 103].
1969	George Guilbault introduced the potentiometric urea electrode [104].
1970	Bergveld introduced the ion selective field effect transistor (ISFET) [105].
1970s	ELISA was introduced by Stratis Avrameas (Institut Pasteur, France) und G. Barry Peiers (University of Michigan, USA) and others [106, 107].
1972	Betso <i>et al.</i> showed for the first time that direct electron transfer (ET) of cytochrome c could be realized at mercury electrodes. This breakthrough suffers from nonreversible electrochemistry due to protein denaturation on this electrode material [108].
1973	Ph. Racinee and W. Mindt (Hoffmann La Roche) developed a lactate electrode [109].
1973	G.G. Guilbault and G.J. Lubrano introduced an amperometric glucose enzyme electrode that was based on the detection of the product of the enzymatic reaction, hydrogen peroxide [110].
1975	The first commercial biosensor (YSI analyzer) was introduced [60, 61]. A review by Newman and Turner summarized the commercial development of blood glucose biosensors used at home by disbeter periods [50].
	home by diabetes patients [59]. (Continued)

Table 1.2 (Continued)

Year	Contribution
1976	First microbe-based biosensors [111–113].
1976	The first bedside artificial pancreas was introduced. The glucose analyzer allows one to control an insulin infusion system (the Biostator) [114–116].
1977	Karl Cammann introduced the term "biosensor" [6].
1977	First realization of reversible ET of cytochrome c employing tin-doped indium oxide electrodes [117] and 4,4'-bipyridiyl as a promoting monolayer on gold electrodes [118, 119].
1979	First steps towards biofuel cells were realized [120-123].
1979	Pioneering work by J. Kulys using artificial redox mediators [124, 125].
1980s	Self-assembled monolayers (SAMs) start to receive considerable attention in the scientific community and are employed in biosensor research [49–52].
1981	Oxidation of NADH at graphite electrodes is described for the first time [126, 127].
1982	First needle-type enzyme electrode for subcutaneous implantation by Shichiri [128].
1982	First biologically engineered proteins using site-directed mutagenesis, enabling work on specific mutants of enzymes [129–131].
1983	First surface plasmon resonance (SPR) immunosensor [132–134].
1984	First ferrocene-mediated amperometric glucose biosensor by Cass <i>et al.</i> [135]. The work led to the development of the first electronic blood glucose measuring system which was commercialized by MediSense Inc. (later bought by Abbott Diagnostics) in 1987.
1988	Adam Heller and Yinon Degani introduced the electrical connection ("wiring") of redox centers of enzymes to electrodes through electron-conducting redox hydrogels [47, 136]. This work was the basis for continuous glucose monitoring employing subcutaneously implanted miniaturized glucose biosensors [137–139].
1988	Direct ET by means of immobilized enzymes was introduced [22, 120, 122, 123, 140-142].
1990	Bartlett et al. introduce mediator-modified enzymes [143].
1980s to 1990s	Nanostructured carbon materials such as C_{60} and nanotubes were discovered [144, 145].
1997	IUPAC introduced for the first time a definition for biosensors in analogy to the definition of chemosensors [3–5].
2002	Schuhmann <i>et al.</i> introduced the use of electrodeposition paints (EDPs) as immobilization matrices for biosensors [17]. Following work enabled the incorporation of redox mediators into the polymer structure of EDPs [18, 146].
2003	An enzymatic glucose/ O_2 fuel cell which was implanted in a living plant was presented by Heller and coworkers [147].
2006	The first H_2/O_2 biofuel cell based on the oxidation of low levels of H_2 in air was introduced by Armstrong and coworkers [148].
2007	An implanted glucose biosensor (Freestyle Navigator System) operated for five days [149].



direct oxidation or reduction of the electroactive product or reactant of the biological recognition process

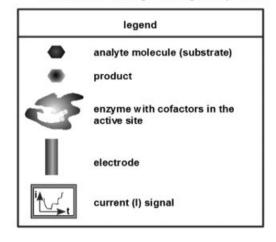


Figure 1.3 Schematic representation of the architecture of a "first-generation" biosensor.

in order to provide a reasonable driving force of the reaction. As free-diffusing mediators, a large variety of compounds such as ferrocene derivatives, organic dyes, ferricyanide, ruthenium complexes and osmium complexes have been used [158].

What are the most important properties of redox mediators suitable for biosensors? First of all, the electrochemistry has to be reversible and they need to be stable in the oxidized and reduced forms. No side reactions should occur. The redox potential needs to be compatible with the enzymatic reaction. It is helpful if the basic structure of the redox mediator also allows for chemical modifications

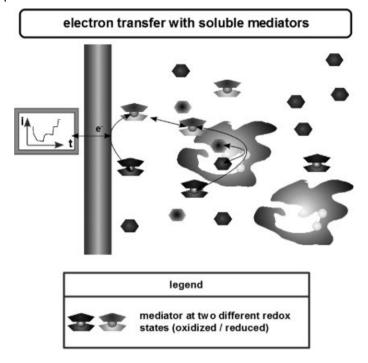


Figure 1.4 Schematic representation of a biosensor operating with soluble mediators.

enabling tuning of the desired redox potential. The immobilization of the redox mediator on the electrode surface and/or the redox enzyme needs to be possible. For instance, functional side chains for, for example, covalent binding to the polymer backbone, redox enzyme, or electrode surface need to be available. The redox mediator should not be toxic and available at reasonable cost and experimental effort. Note that the $K_{\rm M}$ value of the enzyme for a specific redox mediator also impacts the sensor response.

The major drawback of using either a natural or an artificial free-diffusing redox mediator in a biosensor design as illustrated in Figures 1.3 and 1.4 is that sufficient natural (e.g., O_2) or artificial mediator needs to be available to the active site of the enzyme and, subsequently, at the electrode surface for generating a detectable current signal. In addition, and of more importance to the accuracy and long-term stability as well as product safety, artificial mediator molecules that are not securely fixed within the sensing film can leak from the electrode surface [159]. This will change the sensor performance over time. In addition, not all redox mediators are biocompatible. The described problems with the use of free-diffusing redox mediators are not critical for single-use devices. For example, self-monitoring devices for monitoring blood glucose levels are very successfully used by diabetes patients at home [59, 61, 65, 160–163].

1.1.5

"Third-Generation" Biosensors

A different approach to realize biosensor architectures is the immobilization of a redox enzyme on the electrode surface in such a manner that direct ET is possible between the active side of the enzyme and the transducer [22, 140]. Thus, freediffusing redox mediators are not necessary for these types of biosensors [164-169]. Biosensor designs based on direct ET have been investigated thoroughly and comprehensively reviewed [22-26, 170-182]. Figure 1.5 schematically illustrates how such direct ET can be realized within "third-generation" biosensors.

Proteins can spontaneously adsorb on many electrode materials [176] as schematically shown in Figure 1.5a. The interaction is mainly governed by hydrogen bonds as well as electrostatic, dipole-dipole, or hydrophobic interactions. It is important to take into account spontaneous adsorption on the electrode surface because it might also contribute to the overall current signal of a biosensor based on a more complex architecture. The impact of this effect can be evaluated by performing suitable control experiments.

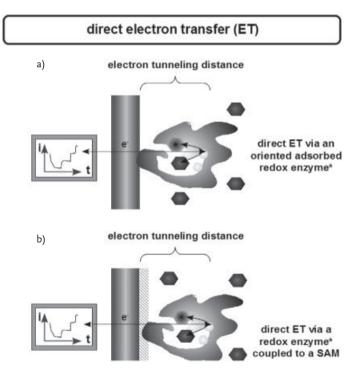


Figure 1.5 Schematic representation of biosensor architectures based on direct ET: (a) via an oriented adsorbed redox enzyme; (b) via a redox enzyme coupled to a self-assembled monolayer (SAM).

It is important to note that proteins tend to denature during such an adsorption process on noble metals or carbon electrodes. In addition, the stability of the adsorbed sensing layer is highly dependent on the pH value and ionic strength of the solution as well as the temperature, the electrode material, and other additional factors. For instance, as early as 1972 direct ET was observed on mercury electrodes employing cytochrome c as redox protein [108]. Reversible electrochemical behavior of cytochrome c was not observed because the protein denatured on the surface.

Therefore, it was a milestone when reversible ET of cytochrome c was achieved for the first time by employing tin-doped indium oxide electrodes [117] and 4,4'-bipyridiyl as a promoting monolayer on gold electrodes [118, 119]. Starting from the 1980s self-assembled monolayers (SAMs) have received considerable attention in the scientific community and have been successfully employed in biosensors [49-52]. The advantages of SAM-based biosensor architectures are the following: (i) the technology is easy and straightforward to use because the formation of SAMs is relatively fast [52]; (ii) enzymes can be adsorbed on SAMs providing the enzyme has an overall surface charge opposite to that of the SAM [176, 183]; (iii) the covalent attachment of redox enzymes and mediators is possible [184–186]; (iv) the design of alkanethiols can be tailored to particular needs (e.g., length of spacer, type of functional groups, mixture of alkanethiols with different properties for integrating different chemical functionalities, generation of multilayers); and (v) SAMs are sufficiently stable with respect to temperature and pH and can be operated in a rather broad potential range (about -1.4 to 0.8 V vs. SCE). This stability window with respect to applied redox potential can also be utilized to generate structured biosensor designs by, for example, stripping of confined areas within the SAM using a scanning electrochemical microscope (SECM) [187– 189]. Thus, complex biosensor architectures even employing multiple redox enzymes or mediators can be realized.

It needs to be taken into account that direct ET between a redox enzyme in very close vicinity to the electrode surface (e.g., first monolayer) is normally very slow. It might even be impossible due to the shielding of the active side and/or redox-active cofactors of the enzyme by the surrounding insulating protein shell. Therefore, observation of direct ET has so far been restricted to either small redox proteins or redox enzymes that are characterized by the location of their cofactor(s) close to the protein shell. If the distance is significantly longer than 10 to 15 Å the chance for efficient direct ET is also significantly reduced according to Marcus theory.

Biology has solved this problem by introducing multi-cofactor enzymes in which the overall distance between two redox sites is divided into a number of shorter distances (multiple cofactors with different redox potentials instead of just one cofactor) or by introducing small redox shuttle proteins such as cytrochromes in the respiratory chain. Thus, ET cascades have been proven to be very efficient and useful. This principle was borrowed from nature not only for direct ET-based biosensors but also for mediated ET-based biosensors. Note that the overall efficiency of ET cascades depends on the entire architecture of a biosensor. A striking

option is to reduce the distance between the active site of the redox enzyme and the transducer surface. Figuratively speaking, one could bring the enzyme closer to the electrode surface or vice versa. For example, the enzyme could be genetically or chemically modified to reduce the impact of the protein shell on the ET distance between the electrode and the active site, or the orientation of the active site towards the electrode surface is controlled by chemically (or genetically) modifying the enzyme or the electrode surface. Strategies for such modifications have been extensively used and evaluated [186]. An example of bringing the electrode surface closer to the enzyme could be the introduction of conducting nanoparticles or nanostructures into the sensing layer in order to increase the probability of ET taking place [190].

However, ET efficiency is not only dependent on the distance of the involved redox relays but also on the properties of the electrode material, the nature of the enzyme, the properties of the immobilization matrix, and the redox mediator (if any) in a complex manner.

For the evaluation of a biosensor design based on direct ET, one needs to take into account that not all the enzymes immobilized on the transducer surface are at productive ET distance. A portion of the immobilized enzymes may be oriented in such a way that direct ET is not possible due to a longer distance between the catalytic side of the enzyme and the transducer in comparison to the ideal oriented distance. A certain percentage of the enzymes immobilized may lose or change their catalytic activity during the immobilization procedure. Therefore, one approach to estimate the ratio of enzyme molecules that communicate via direct ET and enzyme molecules that are fully functional but do not contribute to the overall current response of the biosensor is to measure the current in the absence and presence of a suitable free-diffusing redox mediator. Another approach would be to estimate the catalytically active enzyme concentration on the electrode surface by means of a standard optical enzyme activity test. This is also helpful in case no direct ET can be detected and it is not clear if the enzyme undergoes denaturation during the sensor fabrication and operating process.

1.1.6

Reagentless Biosensor Architectures

For a variety of applications it is useful to employ artificial redox mediators for biosensor architectures. In the case of mediated ET, redox mediators shuttle electrons between the active side of the enzyme and the electrode. The main advantages of employing mediated ET within a biosensor device are that the ET process is independent of the presence of natural electron acceptors or donors. For instance, oxygen is ubiquitously present in biological systems. Given that the redox mediator is appropriately selected, the influence of possible interfering compounds can be reduced because the working potential of the biosensor is defined by the formal potential of the redox mediator. In addition, the pH dependence of the sensor response can be better controlled. Furthermore, the sensitivity and overall current response can be increased. Multicomponent ET cascades can be designed.

Due to the drawbacks of free-diffusing redox mediators, especially with respect to continuous monitoring of the analyte of interest, the development of reagentless biosensors has become of importance over the last 15 years [191]. The outstanding feature of a reagentless biosensor is that all components required for the electroanalytical reaction are securely immobilized on a transducer surface. This is a nontrivial task because the immobilization approach needs to ensure a microenvironment within the biosensor film that facilitates the biological recognition reaction and an efficient ET cascade. The only free-diffusing component of the overall assay reaction is the analyte which is provided by the sample solution. The most appropriate approaches towards reagentless biosensor architectures are the use of an appropriate redox enzyme and an ET pathway that either uses direct ET or mediated ET via securely immobilized redox mediators within the biosensor film. There are several technologies available for immobilizing biological recognition elements on transducer surfaces: adsorption, microencapsulation, entrapment, covalent attachment, and crosslinking. These techniques have been comprehensively reviewed [173, 192–197].

What does a suitable immobilization matrix for a reagentless biosensor need to provide? First of all, the selected biological recognition element needs to be securely fixed at the electrode surface. Furthermore, it needs to provide a microenvironment that either maintains or tunes the enzyme activity at a desired level. This is important not only for the efficiency of the reaction with the analyte and the ET but also for the operation, storage, and long-term stability of the biosensor. Matching charges and hydrophobicity/hydrophilicity as well as hydrogen bonds, complexation, or covalent binding sites are important. It is advantageous if the nature of the immobilization matrix is intrinsically open for optimization, for example, by means of adapting the chemical structure, tuning the chemical and/ or physical properties as well as the immobilization technique. With respect to miniaturization and automation of biosensor production as well as the reproducibility within the production process, it becomes of importance that the immobilization approach can be done in a way that enables exclusive addressing of the electrode surface. A certain flexibility of the backbone of the immobilization matrix has proven to be advantageous, especially with respect to the aspect of diffusion limitations. Note that one limiting case for biosensor performance can be the diffusion limit. In addition, suitable binding sites for redox mediators, spacers, or (multiple) redox enzymes are useful. Redox mediators also need to be securely immobilized within the biosensor film. If in vivo or in vitro use of the device is intended the film needs to be biocompatible and compatible with special needs such as sterilization.

Figure 1.6 describes what reagentless biosensor structures based on mediated ET can look like.

Designing appropriate and efficient ET pathways within a biosensor intrinsically requires that the generated environment is suitable for the chosen biological recognition element. Its catalytic activity and stability might be tuned to the desired performance by the immobilization matrix. The immobilization procedure itself needs to be compatible with the ET pathway strategy. For instance, the stress on

mediated electron transfer (ET)

a) ET via a conductive polymer chain



ET via a redox-relay modified polymer

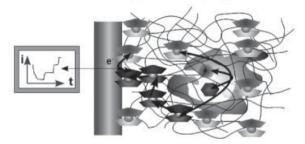


Figure 1.6 Schematic representation of reagentless biosensors: (a) ET via conductive polymer chains; (b) ET via redox-relay modified polymer chains.

an enzyme by either the chemical nature of the immobilization reaction or the applied potential for electrodeposition on the transducer surface may lead to a significant loss of enzyme activity (enzymatic sensor) or binding reaction (affinity sensor). Efficient ET between a redox enzyme and the redox relays in the immobilization matrix to the electrode surface is mainly controlled by the distance between the individual redox couples [198] participating in the overall ET reaction. Thus, for further optimization, it is crucial to elucidate the rate-determining steps of all involved processes.

For instance, if one would like to design a suitable redox polymer for a certain enzymatic reaction, it helps to think about the following factors. First, the redox polymer needs to create a three-dimensional network that allows secure immobilization of the enzyme with a reasonable pore size. In addition, fast diffusion of the analyte, products, or counter-ions and fast ET kinetics need to be ensured. The polymer film deposited as sensing layer creates a diffusion barrier which often prolongs the response time, shifts the linear measuring range, and decreases the sensitivity of the sensor. Second, the redox polymer should create a local

microenvironment that is beneficial for enzyme immobilization, functionality, selectivity, kinetics, and stability. Good interaction between the active site of the enzyme and the redox relays, especially for the first ET step, is vital. For example, ET transfer distances have to be reasonably short (<10 Å).

The general advantages of reagentless biosensor structures can be summarized as follows. Since all components of the assay are securely immobilized on the electrode surface, there is no or just a negligible loss of redox mediators, cofactors, and/or enzymes over the time of operation. This is of importance for the performance and safety of a device because the impact of free-diffusing possibly toxic substances is minimized. Therefore, reagentless biosensor architectures are often used for *in vitro* and *in vivo* measurements as outlined in Section 1.4.5.

Which advantage of a certain generation of biosensors outweighs the advantages of the other generations will, however, depend on the analytical task. The specifications of requirements (e.g., type and concentration range of analyte, composition of sample matrix and occurrence of possible interferences, official regulations for the final application, overall cost limitation for the device) need to be defined. A comprehensive review of the state of the art for the specific analytical task helps to develop a suitable strategy. Sampling and sample preparation are additional important aspects. After a preliminary testing of the envisaged biosensor design, it has to be thoroughly tested to determine if it is capable of identifying and quantifying the analyte of interest. For applications with high throughput and/or commercial interest, approaches have to be evaluated to properly deal with data collection, processing, interpretation, documentation, and reporting. It might be necessary to provide suitable instrumentation to operate the sensing device, and features such as a self-referencing system for calibration, temperature control, etc., might become assets to be considered.

1.1.7 Parameters with a Major Impact on Overall Biosensor Response

The choice of biosensor architecture depends to a major extent on the (bio)chemical processes involved in the biorecognition process. The processes in close proximity to the electrode surface that are involved in a typical biosensor reaction are rather complex. It is important to have an overview about the variables that affect the performance of a biosensor (Figure 1.7) and which of these parameters may have a major impact on the signal response. As a matter of fact, the design of appropriate sensor architecture depends on the specific demands arising from the particular analytical task. Thus, a sound understanding of the challenges of the analytical task with respect of the main reactions involved in the overall sensing process is mandatory. It is important to note that due to the complexity of a biosensor architecture one deals with a multiparameter space of which, most likely, only a limited number of parameters can be controlled. This implies that it may be helpful to visualize the analytical task and the envisaged biosensor design at molecular dimensions in order to better understand the processes taking place, thus being able to identify at an early stage potential pitfalls with

variables affecting the electroanalytical performance of a biosensor

electrode variables

- material
- surface area
- geometry
- surface condition

electrical variables

- potential
- current
- charge
- impedance

external variables

- temperature
- pressure
- time
- composition of sample matrix
- occurrence of interferences and/or electrode fouling processes

potential noise variables

- electrochemical noise
- thermal noise
- environmental noise (e.g. electromagnetic)

mass transfer variables

- convection
- diffusion
- surface concentration
- adsorption

electrolyte variables

- bulk concentration of electroactive species
- ⊌ pH
- solvent
- trace impurities

reaction variables

- kinetic and thermodynamic parameters of ET & reactions(s) (bulk, surface, biosensor film)
- rate-determining step (rds) of the reaction
- stability of reaction partners/ products

Figure 1.7 Parameters influencing the overall response of a specific biosensor architecture.

respect to general device layout as well as the interpretation of the obtained data. It is indispensable to always have in mind the impact of diffusion, enzyme, and ET kinetics. In addition, the impact of temperature is not negligible. Diffusion, kinetics, selectivity, and overall (bio)sensor performance are highly temperature dependent [199, 200].

A detailed list of variables affecting the electroanalytical performance of a biosensor (Figure 1.7) enables an awareness of the main "adjusting screws" of a particular system for further optimization.

Knowing the most influential parameters of a specific biosensor architecture is the basis to understand and fine tune the performance of these devices in a rational manner. Figure 1.8 summarizes the key features of typical biosensors and lists several that are of additional importance for commercial devices. Among these, selectivity, sensitivity, accuracy, response, and recovery time as well as operating lifetime are some of the most important key factors. Keeping in mind the needs of the specific analytical task of interest, it seems to be necessary to characterize at least the key parameters mentioned in Figure 1.8 in order to specify the analytical performance of a biosensor design.

It is indispensable to elucidate the rate-limiting steps of the overall reaction sequence in order to develop an appropriate optimization strategy. Thus, mathematical and chemometric approaches are expected to promote a deeper understanding of the processes involved. As a useful source of information for modeling biosensor responses, a related book chapter by Bartlett *et al.* [201] is recommended.

An important aspect of biosensor optimization is the elimination of interferences, or at least a reduction of the impact of interferences. In many samples there are components that either directly react at the electrode surface or the involved redox centers or interfere with the biological recognition reaction (e.g., inhibitors or other substrates for the enzyme). In addition, leakage from the sensing layer, loss in enzyme activity or electrode fouling may occur. Thus, changes in sensitivity and baseline drifts may occur during biosensor operation. Therefore, suitable strategies for calibration are needed to ensure reproducible and quantitative results. For real-world applications it is imperative to characterize and optimize the biosensor architecture under actual measuring conditions. A useful review by Phillips and Wightman [202] discusses critical guidelines for the validation of *in vivo* microsensors.

Electroanalytical methods and biosensor architectures offer various strategies to tune selectivity for the analyte of interest [203]. Among the diverse approaches to tackle problems arising from interferences and electrode fouling are the following. (i) Size exclusion and/or charge repulsion are utilized when additional films or membranes are placed as upper layers on top of the actual sensing layer. Typical membranes are polymethylcellulose, Nafion, hydrogels, polypyrrole, ophenylenediamine, polyeugenol, and other electrodepositable films (conducting or nonconducting). However, this approach is used at the expense of biosensor response time. (ii) Use of suitable redox mediators allows operation at moderate potentials below the potential of abundant interferences such as ascorbic acid. (iii) The applied potential is a useful tool to discriminate between different electroactive species under the assumption that the redox waves are distinguishable. Electroanalytical techniques such as cyclic voltammetry (CV), fast-scan CV, differential pulse voltammetry (DPV), square wave voltammetry (SWV) or differential pulse amperometry (DPA) as well as multiple pulse amperometry [204, 205] are useful for determining several species in parallel and discriminating between them.

key features of a biosensor

key parameters of biosensor performance

- analyte
- employed biocomponent for recognition of analyte
- type & composition of sample matrices
- selectivity
- sensitivity
- dvnamic range
- detection limit
- response time
- reproducibility
- precision
- stability (in use, long-term, storage)
- calibration approach (requirements, intervals, drift)
- duration to reach baseline

additional parameters of biosensor performance, e.g. relevant for commercialization

- cost per measurement
- duration of measurement
- turnover of measurements (time resolution)
- warm-up time
- sample volume
- measuring temperature
- size, weight & price of the device
- costs, size & weight of required instrumentation
- delivery time
- development status of the device/procedure (commercial product, standard procedure (in house or commercial, established research procedure, basic research, proof-of-principle)
- time between maintenance checks
- potential to be coupled with other analytical techniques
- data storage and processing

Figure 1.8 List of key characteristics of a biosensor.

(iv) The choice of the immobilization matrix can be important for the susceptibility to interferences. (v) Mathematical models and chemometrics can be employed. One approach utilized the impact of temperature on biosensor performance to improve selectivity [199, 200]. A recent review summarizes for example some of the strategies towards the elimination of interference of glucose biosensors [206].

1.1.8

Application Areas of Biosensors

Today biosensors are mainly used for healthcare applications, controlling industrial processes, and environmental monitoring, as outlined in Figure 1.9. In all cases the biosensor design, packaging, and instrumentation required are dependent on the purpose of the measuring approach. In several cases, the type of sample dictates the biosensor design. For example, if potentially harmful samples such as blood or contaminated waste water are of interest, disposable sensor formats are preferred. Samples can be analyzed off-line in a laboratory, such as glucose testing of patients' blood samples in a hospital laboratory or water samples from rivers. In addition, off-line analysis can also be performed close to the operation side of an industrial plant or process or glucose monitoring can be performed at home by patients themselves. For several applications, however, there is a need for online analysis in real time, such as quality control in the food or drug industries or metabolite monitoring at the bedside or during surgery. The requirements for

applications of biosensors - examples

clinical

- single-use
 - glucose monitoring (by the patient at home)
 - → lactate (sport event/training)
- multi-analysis
 - glucose monitoring (hospital)
 - pathogen detection (pathology, Internal medicine)
- short-term invasive
 - glucose monitoring (hospital, bedside)
- long-term implantable
 - glucose monitoring (artificial organs)

Figure 1.9 Areas of application for biosensors.

non-clinical

- single analysis
 - glucose, alcohol, aldehyde monitoring (food industry)
- continuous monitoring
 - glucose, other small molecules, pathogens, pollutants (food & water industry, fermentation, quality control)
- environmental monitoring
 - pathogens, e.g. plaque, anthrax (ecological agencies, military, quality control)

a single-use device differ from those for multi-analysis and continuous monitoring and have to be taken into account when considering overall biosensor architecture

1.2 Criteria for "Good" Biosensor Research

It is obvious that science does not always lead to ground-breaking advancements that are worth publishing. This is also very much true for publications from biosensor-related research. The area of biosensor research is even more susceptible to publications that do not significantly contribute to the present state of the art, since the basic equipment for doing high-level biosensor research is comparatively cheap. Moreover, nearly any modified electrode with an immobilized biorecognition element will show a certain response upon the addition of a specific analyte, leading to a calibration graph, the possibility of determining the pH optimum, etc. This is not intrinsically a sign of low quality if the contribution is otherwise scientifically sound. An example would be a publication that only slightly changes an existing biosensor design, but otherwise is unambiguously supported by technically sound data and interpretation. However, there are also a large number of publications that suffer from technically wrong or biased data acquisition, processing, and interpretation or an insufficient amount of data for the hypothesis proposed.

To do research is basically to generate knowledge which is made available to the scientific community via publication. The main aim is that other scientists will be convinced by the scientific approach and they can adopt the strategy or scientific principle for answering their own research questions. Thus, criteria for "good to excellent" biosensor research have to be measured in terms of the following questions:

- i) Does the research work introduce a novel sensing principle, a novel signal amplification strategy, a novel specifically adapted redox mediator with improved properties, a novel immobilization scheme, a novel sensor architecture with tunable parameters? Does the proposed research work contribute to an increase in fundamental knowledge, an in-depth evaluation of the signal transduction mechanism, or an in-depth physicochemical evaluation of the rate-determining steps and the interplay of the parameters in the complex parameter space?
- ii) Does the research work introduce novel aspects to an already known sensing architecture, to an already known application, or does it extend a sensing principle to be more general? Does the work include the discovery of surprising results by combining a specific biological recognition element with an already known sensing principle and is there a rational way to understand this surprising result? Does the adaptation of an already known sensing principle to a specific application require innovative features?

- iii) Is the proposed research work just a variation of an existing principle by varying the biological recognition element, the electrode material, the size and integration of the electrode? Is there any predictable contribution of the elements of which the sensing layer is composed to the expected signal generation, interference elimination, improvement of long-term or operational stability, etc.?
- iv) Does the contribution solve a previously unsolved scientific question? Can the principle be the basis for improvements in sensitivity, selectivity, applicability?
- v) Is the selection of the compounds used for creating the sensor architecture based on buzzwords such as nanomaterial, nanosensor, etc.? Is the effect of the material used correlated with the meaning of the buzzword or is it just used because of the buzzword?
- vi) Does the complexity of the sensor architecture allow a rational investigation of the complex influence of all compounds used on the final sensor output? Is it an effect or a scientific result? If a novel effect is discovered, can it be explained by a scientifically sound argumentation chain? Is it possible to design control experiments to provide evidence for the hypothesis concerning the sensing mechanism?
- vii) Is there any possibility of reproducing the measurements in the same laboratory at another time or even in a different laboratory? Are all results derived from one sensor? Is there any statistical evaluation of the repeatability of the sensor fabrication protocol, and of the obtained signals?
- viii) If the results are sound and justified, do the authors try to benchmark the results with existing sensing strategies for the same analyte and in the same application?

Taking the above into account it seems to be straightforward to distinguish between fundamental biosensor research and biosensor development. For fundamental biosensor research, the discovery of novel sensing strategies or bioelectrochemical signal transduction schemes, the elucidation of the fundamental processes, and the understanding of the complex parameter interplay that finally leads to the observed sensor signal are the focus of the research. For biosensor development, an existing sensor principle has to be adapted to a specific application taking into account costs, storage time, reproducibility, calibration, validation, legal consequences, etc. Presently, many biosensor papers that predominantly deal with fundamental biosensor design try to include some application aspects by showing that some standard samples can be measured at a required quality. However, these results are most often obtained in the research laboratory using the standard addition method and well-trained personnel. On the other hand, papers on application-oriented research often try to include basic mechanistic studies at a limited depth. This is also reflected by the editorial policy of international journals accepting work on biosensors. Recently, fundamental studies are more often published in physical chemistry journals while the biosensor journals have shifted to be more application oriented.

Ideally, a biosensor design needs to be adaptable to a certain application or analytical task. If the design is a general principle, the biosensor performance needs to be tunable to the needs of a specific analytical task and be open for modifications leading to a broader range of analytes or applications. If the mechanism behind the biosensor function is at least partially understood, a fine tuning of the sensing layer or a rational adaptation of the sensor design may become possible. As pointed out above, the parameters influencing and limiting the overall sensing process of a biosensor are often not fully understood. In a classical approach, it was assumed that the parameters are linearly independent. Thus, it was assumed that one may independently vary one parameter while the others are kept constant. For example, parameters such as transducer type and pretreatment, enzyme and mediator concentration and their ratios within the film, type of immobilization matrix, immobilization parameters, and film thickness are varied. A rational optimization approach includes that the main parameters affecting the overall ET pathway and hence the final sensor response have to be investigated in order to find reasonable tools for tuning the performance of the selected biosensor design. As a matter of fact, it is well known that it is impossible to keep parameters constant while changing others. For example, if the enzyme loading is increased, the film thickness, the diffusional properties for the substrate and the products, the counter-ion movement, possible ET reactions, etc., may be altered simultaneously. Thus, a "pseudo" rational approach has to be complemented by combinatorial approaches in which the overall parameter space is addressed by means of a large number of measurements after permutation of all possible influencing parameters. The knowledge gained should be comprehensively summarized in a related publication. The final consideration should be whether the sensor fabrication process described in a paper can be directly repeated successfully in another laboratory leading to similar sensor responses.

1.3 Defining a Standard for Characterizing Biosensor Performances

Considering the workflow as proposed in Figure 1.10 it becomes obvious that it is essential to characterize a specific set of key parameters mainly determining biosensor performance. A selection of suitable key parameters is given in Figure 1.8. With respect to the particular needs of a specific analytical task one needs to decide which performance parameters make sense to be evaluated at a certain stage of the process. For example, Phillips and Wightman evaluated guidelines for the validation of *in vivo* microsensors [202]. In the following, the most common and most important characteristics of biosensors are discussed.

Though a biological recognition reaction is typically very selective, interferences may occur due to substances other than the analyte of interest. Such interferences can be converted by the biorecognition element or at the transducer surface and

workflow scheme for developing and optimizing biosensors

rational approach to sensor architectures

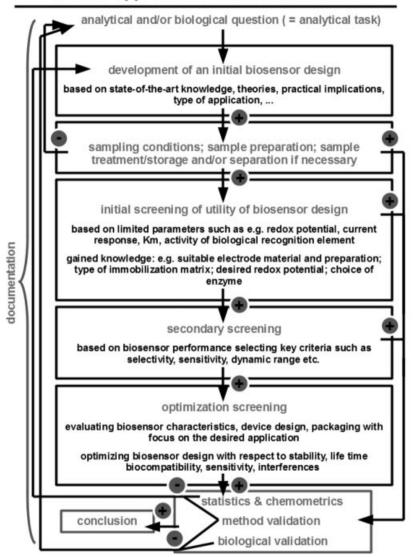


Figure 1.10 Workflow for successful biosensor research.

thus create false-positive results. The selectivity of a biosensor is characterized by the selectivity coefficient. The selectivity coefficient is defined as the quotient of the respective binding constants of the analyte of interest A and a potential interference I with the biorecognition element B:

$$\begin{split} \mathbf{B} + \mathbf{A} &\rightarrow \mathbf{B} \mathbf{A}; \, K_{\mathrm{BA}} \\ \mathbf{B} + \mathbf{I} &\rightarrow \mathbf{B} \mathbf{I}; \, K_{\mathrm{BI}} \\ \\ K_{\mathrm{AI}} &= \frac{K_{\mathrm{BA}}}{K_{\mathrm{BI}}} K_{\mathrm{AI}} \end{split}$$

In addition to selectivity, sensitivity (S) is a vital parameter of the performance of a biosensor. Sensitivity is defined as the slope of change in signal with change in concentration:

$$S = \frac{d(\text{signal})}{d(\text{concentration})}$$

This is, however, only straightforward if the sensor response is linear.

The linear range of a sensor is defined as the range in which the sensor signal is proportional to a change in concentration. Linear range should not be confused with dynamic range. Dynamic range describes the range in which a change in concentration will lead to any sort of noticeable change in signal. Typically, the whole dynamic range will not yield a signal suitable to determine the analyte of interest in a reliable way. In most cases the working range of a sensor corresponds to the linear range.

The limit of detection (LOD) of a biosensor is one of the most important parameters to be determined. That holds especially true when disease markers have to be determined. The LOD is typically defined as

$$LOD = k \times std_{background}$$

where k is the signal-to-noise ratio and $std_{background}$ is the standard deviation of the background signal. The value of k can be chosen deliberately depending on the desired accuracy of the LOD but is typically 3. Another definition describes the smallest detectable concentration of an analyte c_{LOD} as

$$c_{\text{LOD}} = \frac{\text{std}_{\text{background}}}{S}$$

where *S* is the sensitivity.

It has to be pointed out that the LOD cannot be properly discussed without any knowledge of the binding constant of the primary biorecognition process. If the binding constant of the biorecognition process is, for example, in the nanomolar range, a detection limit far below seems to be thermodynamically impossible. Thus, it is very important to gain a solid understanding about the difference in which signal can be measured and amplified and which limit of detection can be achieved based on a distinct biological recognition process. Hence, it would be very helpful if together with a LOD an estimation of the binding constant was given.

Terms rarely mentioned in the biosensor literature are *accuracy* and *precision*. Accuracy describes the agreement between the average of the measured value and a reference value. To determine accuracy is relatively straightforward. Normally sensors are tested using solutions with well-known concentrations of the analyte ("true value"). Obtained values can easily be compared to the true value.

Precision describes the scatter of measured values around the average of the measured values. Precision is much more important for biosensor performance than accuracy. Accuracy can be influenced by systematic errors that can be corrected. However, a sensor producing values that are scattered will not be regarded as very reliable. Some confusion can be found in the use of terms that are measures for the precision of a sensor, repeatability and reproducibility. Many papers report work as highly reproducible even though reproducibility is defined as the between-laboratory precision. That simply means that a sensor's measurements are reproducible if the same results are obtained in different laboratories with the same sensor architecture [207]. This is, however, rarely tested. Most authors really test the repeatability of a sensor. Repeatability is the in-laboratory precision. Inlaboratory precision means that the sensor yields the same value of, for example, concentration in repeated measurements. Repeatability also means that the same sensor architecture will yield the same result if manufactured in the same laboratory. It would hence make more sense to speak about the standard deviation of a single sensor and to analyze the repeatability of the respective sensor architecture than to speak about reproducibility if the latter has not been tested.

Another measure for sensor performance that is a potential source of confusion is the *stability* of a sensor. Sensor stability can mean different things including but not limited to *working stability*, *storage stability*, and *long-term stability*. Working stability (sometimes also called usage stability) describes the stability of the sensor during continuous operation. Storage stability obviously describes the stability of the sensor upon storage, while long-term stability describes the sensor stability during operation in a sample solution but not necessarily continuous operation. It already becomes clear that the "stability" of a sensor will rarely mean the same thing for different sensor architectures.

Sensor performance is also characterized by commercial means, most importantly *time per measurement* and *cost per measurement*. Again, there is no one-size-fits-all definition for these terms and one should emphasize on being transparent in the way in which these numbers are determined.

In conclusion, a number of parameters are helpful when characterizing biosensor performance. It is, however, extremely important to be precise in the use of these terms. A clear definition of the measured variable is mandatory in any case and needs to be reported in a transparent fashion.

1.4 Success Stories in Biosensor Research

This section aims at discussing how "good" biosensor research inspires the scientific community to achieve advancements that reach from novel basic concepts to

real-world applications. To do so, selected examples with a significant scientific impact are presented. This selection is not exhaustive, and there are many other possible success stories, but is just some examples selected by a very personal view.

1.4.1

Direct ET Employed for Biosensors and Biofuel Cells

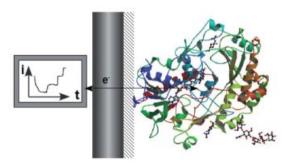
The fundamentals of biosensors that exhibit direct ET between biological recognition element and electrode have been discussed thoroughly in the section on third-generation biosensors (Section 1.1.5). This section concentrates on highlighting the major contributions made in the area of direct ET in biosensors and biofuel cells. Though the latter is not a sensing application it draws from the same concepts as third-generation biosensors, and biofuel cells are a striking example of the continued development of redox-enzyme electrodes.

The realization of direct ET poses some challenges that already have been outlined above. Figure 1.11 summarizes the key features and challenges of this approach. The protein shell may prevent ET processes due to a large distance between active site and electrode if the active site is deeply buried within the protein. Proteins with suitable characteristics for direct ET have to be securely fixed to the electrode surface in an orientation facilitating direct ET. The orientation of the redox enzyme towards the electrode surface is the main challenge in designing direct ET pathways. Basically, two cases can be distinguished: direct ET between the active site of the enzyme or direct ET via an internal electron pathway within the protein. The first case is relevant for rather small redox proteins or for redox enzymes exhibiting an active site closely located at the outer protein shell. The second case mainly applies to multi-cofactor enzymes. However, the orientation is critical in both cases since either the redox-active center or the cofactor closest to the protein shell has to be located within a productive ET distance [27, 208-211]. Recently, a study investigated the different orientations of recombinant horseradish peroxidases to gold surfaces [212].

Early work on the direct electrochemistry of redox proteins suffered from the poor stability of those proteins at electrode surfaces. Though a significant body of work demonstrates the direct electrochemistry of redox proteins at graphite electrodes [213], the major breakthrough came with the use of surface-modified electrodes that provided a substrate for the stable orientation and immobilization of redox proteins at the electrode surface [208, 214, 215]. Surface-modified electrodes allowed for the study of the direct electrochemistry of cytochrome c on 4,4'-bipyridyl-modified gold electrodes [216]. As previously mentioned, SAMs on electrode surfaces are useful tools to realize biosensors suitable for direct ET [52]. The creation of monolayers on electrode surfaces with immobilized recognition elements is not limited to SAMs; a review by Willner and Katz summarizes the different approaches to realize covalent binding of enzymes to functionalized electrode surfaces as well as strategies to employ modified enzymes (e.g., protein conjugates) [186].

Enzymes that have been much studied in direct ET configuration include peroxidases [217, 218], especially horseradish peroxidase [219, 220], laccase [120, 121],

analytical task direct electron transfer (ET)



key features & challenges for direct ET

- immobilization of the enzyme on the electrode surface (impact on the structure of the enzyme, on the enzyme activity, on the diffusion of the substrate, on the ET kinetics, ...)
- orientation of the active site of the enzyme to the electrode surface
- orientation of the cofactor(s) of the enzyme to the electrode surface
- distance(s) between the redox relays taking part in the direct ET pathway
- selection of a suitable redox potential for the measurement (driving force of the reaction, catalytic current should be observed at the redox potential of the redox relay (e.g. a cofactor) communicating with the electrode surface)
- efficiency of the overall ET process

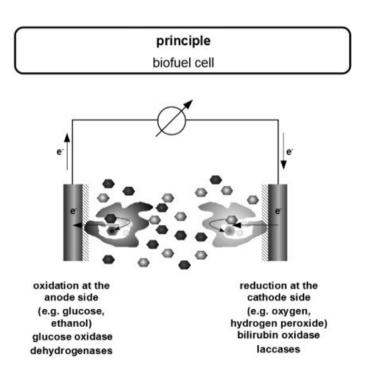
Figure 1.11 Biosensors based on direct ET.

and dehydrogenases [221] including fructose dehydrogenase [222], cellobiose dehydrogenase [223–225], and quinohemoprotein alcohol dehydrogenase [226]. It is important to keep in mind that for characterizing biosensor responses it is important to check if the enzyme employed is still able to efficiently catalyze the physiological reaction at a rational range of redox potential.

It is very important to define criteria to unequivocally proof a direct ET pathway between an immobilized redox protein and an electrode surface. The first important prerequisite is the occurrence of the direct electrochemistry of the redox cofactor inside the protein in the absence of the substrate. Hence, a reversible redox wave in a cyclic voltammogram of the protein-integrated cofactor has to be visible with a formal potential which clearly shows that the protein structure is not

disturbed during the immobilization process. The second and most important prerequisite is that upon addition of the substrate the catalytic current increases at the redox potential of the protein-integrated cofactor without any significant overpotential at least at slow scan rates. If these two features in the cyclic voltammogram are not seen, a direct ET pathway can be excluded.

The development of enzyme electrodes with immobilized redox enzymes in direct ET communication was the prerequisite for the design of enzyme-based biofuel cells. For representative recent reviews see [70, 227, 228]. A fuel cell generally converts chemical energy into electrical energy in a continuous process as long as fuel is supplied. Biofuel cells convert chemical energy by means of a biocatalytic process as can be seen in Figure 1.12. Typically, enzyme-based biofuel cells consist of at least one enzyme electrode on either the cathode or anode side of the fuel cell or enzyme electrodes on both the cathode and the anode sides. The high specificity of fuel conversion by enzyme electrodes allows for membrane-free



key features & challenges of biofuel cells

- high selectivity of the biocatalytic reaction
- membraneless design
- stability of the enzymes and choice thereof poses key challenge
- high current densities

Figure 1.12 Principle of biofuel cells.

designs in which the cathode and anode reactions proceed in a single compartment.

The typical reaction on the cathode side is the reduction of oxygen at either a platinum catalyst or an electrode. The ET pathway at the enzyme electrode can be mediated (Section 1.4.3) or direct. Enzymes that have been employed in biofuel cells relying on direct ET include laccases [229, 230] which, however, suffer from a pH optimum in the acidic range and inhibition by halide ions. Thus, despite their favorable high potential for oxygen reduction they show poor stability in human tissue and fluids. Alternatively, bilirubin oxidase has been used as oxygen reduction biocatalyst in biofuel cell cathodes due to the better pH optimum and the insensitivity towards chloride ions. However, bilirubin oxidase has an about 200 mV lower reduction potential for molecular oxygen [231]. Alternatively, reduction of peroxide can be the cathodic reaction in biofuel cells. Relying on the experience with peroxidase-modified electrodes in biosensor research, electrodes modified with peroxidases have been shown to be highly efficient biocatalysts in biofuel cells [232, 233]. Microperoxidases that are truncated forms of cytochrome c have also been employed in biofuel cells [234] in which they convert hydrogen peroxide that is supplied by the enzymatic reaction of GOx with glucose and oxygen.

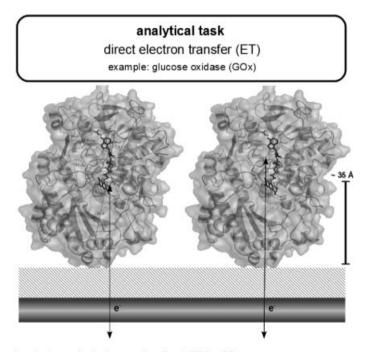
The most common reaction at the anodic side of biofuel cells is the oxidation of sugars which relies on the catalytic properties of oxidases. This class of enzymes has, however, usually poor potential for direct ET. Direct ET on the anodic site was, however, described for a number of hydrogenases [235, 236] and cellobiose dehydrogenase [225, 237, 238]. Enzymatic catalysis by means of direct ET was also realized on conducting graphite or TiO₂ particles [239, 240].

In conclusion, biofuel cells have a tremendous potential to be applied in, for example, implantable sensors or similar functional devices. They are a striking example of the continued development and application of the principles of biosensors employing direct ET.

1.4.2 Direct ET with Glucose Oxidase

Glucose sensors are *the* success story with respect to biosensor research and application. Today, diabetes patients are able to monitor their blood glucose levels on their own at home with commercial devices [59–65]. All these successful devices use a mediated ET pathway with natural or artificial redox mediators irrespective of whether GOx or other glucose-converting enzymes such as pyrroloquinoline quinone (PQQ)-dependent glucose dehydrogenase or nicotinamide adenine dinucleotide (NAD+)-dependent glucose dehydrogenases are used. Also, there have been continuing attempts to demonstrate direct ET between the active-site integrated flavin adenine dinucleotide (FAD) cofactor of GOx and an electrode surface. However, after solving the crystal structure of GOx [241] it becomes clear that the ET distance from the protein-integrated FAD is large and hence fast ET kinetics are unlikely.

Despite the knowledge about the large ET distance, there is an ongoing attempt to propose direct ET between GOx and specifically prepared electrode surfaces (Figure 1.13). However, the specific nature of GOx, namely its reaction with its natural electron acceptor O_2 and the probably unwanted generation of H_2O_2 , may lead to glucose-proportional current changes which are incorrectly attributed to direct ET reaction. Even if all traces of molecular oxygen are removed the source for the obtained current changes is often not clear. For example, if carbon nanotube-modified sensor surfaces are used, it is difficult to unequivocally confirm that no traces of the metal catalyst used for the growth of the nanotubes are left providing free-diffusing metal complexes which may serve as redox mediator for shuttling electrons between the active site of GOx and the electrode surface. As a



key features & challenges for direct ET by GOx

- glucose oxidase (GOx; EC=1.1.3.4) from Aspergillus niger;
 Source: PDB Database (DOI:10.2210/pdb1cf3/pdb); crystal structure at 1.8 Å, GOx expressed in Saccharomyces cerevisiae
- active as homodimer
- in black: cofactor FAD, in dark grey: Valine537 (active site)
- distance from the active site of the active site to the outer protein shell is about 20 to 40 Å
 - => rather long distance for direct ET!!!
- effective ET occurs usually ≤ 10 Å
- orientation of the active site with respect to electrode surface and substrate diffusion

Figure 1.13 Direct ET between GOx and an electrode surface.

matter of fact, these potential sources for free-diffusing redox species become increasingly unpredictable the more different components are used to fabricate the sensor.

Thus, as already pointed out, there are clear presuppositions which have to be met before a potential direct ET pathway may be discussed. First, the FAD/FADH₂ redox wave in a cyclic voltammogram has to be visible at the potential which is characteristic for the cofactor bound within the active enzyme. Moreover, upon addition of glucose a clear oxidation current has to commence at this redox potential without any significant overpotential. In our opinion, this is one of the main sources for falsely assuming direct ET. It is known that FAD is not covalently bound to the enzyme and hence can become dissolved in the electrolyte during denaturation of the protein. The redox potential of free-diffusing or surfaceadsorbed FAD differs from that of FAD located at the active site of the enzyme. Thus, due to the high surface area of the often-used sensor architectures and the comparatively large amount of GOx adsorbed on the electrode surface, the FAD/ FADH₂ redox couples may often be due to released FAD. Upon addition of glucose, the catalytic current is then not closely related to the observed redox wave and hence is no criterion for a potential direct ET between GOx and the electrode surface. In the following paragraphs a number of recent publications are briefly mentioned in which the source of the observed glucose-proportional current is not completely clear and there may or may not be alternative possibilities to a direct ET process to explain the observed effects.

In the following, a number of recent papers proposing direct ET of GOx will be discussed. A uniformly porous TiO2 material was synthesized using a carbon nanotube template-assisted hydrothermal method and GOx was adsorbed leading to glucose-proportional currents [242]. Similarly, three-dimensional macroporous inverse TiO2 opals were synthesized from a sol-gel procedure using polystyrene colloidal crystals as templates. Glucose oxidase was successfully immobilized on the surface of an indium tin oxide electrode modified using inverse TiO2 opals. Cyclic voltammetry showed stable and well-defined redox peaks for the direct ET of GOx in the absence of glucose. This redox peak increased upon addition of glucose [243]. Along the same lines, direct electrochemistry of GOx adsorbed on boron-doped carbon nanotubes/glassy carbon surfaces [244] or an oxidized borondoped diamond electrode [245], nitrogen-doped carbon nanotubes [246], exfoliated graphite nanosheets [247], single-wall carbon nanotubes in combination with an amine-terminated ionic liquid [248], and GOx incorporated into polyaniline nanowires on carbon cloth [249] was proposed. Entrapping GOx at the inner wall of highly ordered polyaniline nanotubes [250] or chemically synthesized multiwalled carbon nanotube-SnO2-Au composites [251], co-deposited GOx-NiO nanoparticles [252], and immobilization of GOx in a natural nanostructural attapulgite clay film-modified glassy carbon electrode [253] have been investigated. Biologically synthesized silica-carbon nanotube-enzyme composites displayed stable redox peaks at a potential close to that of the FAD/FADH₂ cofactor of immobilized GOx. The immobilized enzyme was stable for one month and retained catalytic activity for the oxidation of glucose [254].

Direct electrochemistry of GOx immobilized on a hexagonal mesoporous silicamodified glassy carbon electrode was investigated. A pair of redox peaks at a potential of -417 mV was obtained and a diffusion-controlled electrode process with a two-electron transfer coupled with a two-proton transfer reaction process was postulated [255]. However, despite of the well-defined FAD/FADH₂ redox process, biocatalytic oxidation of glucose was only possible in the presence of a free-diffusing redox mediator such as ferrocene monocarboxylic acid. This behavior is quite common and supports the assumption that the FAD causing the redox process may be free or surface-adsorbed FAD, which is no longer bound to the enzyme. There are a number of similar studies in which the first criterion, namely the visible voltammogram of the cofactor, seems to be met; however, no electrocatalytic current could be obtained upon addition of glucose. Another series of publications propose direct ET based on a decrease of the electrocatalytic response of the reduced form of GOx to dissolved oxygen [256, 257] or using complex multicomponent immobilization layers with integrated nanomaterials and binders such as GOx-graphene-chitosan [258], dispersed multiwalled carbon nanotubes in a gold nanoparticle colloid stabilized by chitosan and an ionic liquid [259], a carbon nanotube-modified glassy carbon electrode with GOx immobilized within a chitosan film containing gold nanoparticles [260, 261], CdTe quantum dotcarbon nanotube-Nafion films [262], a conductive cellulose-multiwalled carbon nanotube matrix with a porous structure using a room temperature ionic liquid as solvent and encapsulating GOx within this matrix [263], or carbon nanotubes in combination with platinum nanoparticles and chitosan [264].

In all the attempts mentioned above the enzyme was not modified, and hence its size and the large ET distance from GOx to the (nanostructured and highsurface-area) electrode remained constant. Despite the FAD/FADH2 redox wave often being visible in the related cyclic voltammogram, the measured redox potentials varied largely between about -0.49 and -0.41 V which remained without large changes upon addition of glucose. Due to the limitations for direct ET as derived from Marcus theory, these observations are most likely not caused by a true direct ET process but alternative explanations have to be considered despite the observed and repeatedly obtained effects. Alternatively, the ET distance may be decreased by the formation of enzyme-nanoparticle hybrids in which the nanoparticle penetrates into the protein shell [265]. However, in these cases the catalytic current for glucose oxidation is often obtained at high overpotentials. Recently, a more rational approach aimed at decreasing the size of GOx either by preparing genetically modified GOx [266] or by wrapping off the glycosylation shell of the enzyme [267–269]. However, even then it is very hard to distinguish if the catalytic reaction is at the potential of the functional enzyme-integrated FAD or of FAD which may have been released from enzyme molecules.

Thus, despite the large number of publications and the steep increase in the number of publications about direct ET between GOx and modified electrode surfaces, one has to be extremely careful with the possible over-interpretation of the observed effects. The proposed sensors may work fine in dedicated applications; however, it is a fundamental difference if a sensor concept can be applied

and glucose concentrations can be reliably determined or if a basic physicochemical claim about a potential direct ET pathway is suggested.

1.4.3 Mediated ET Employed for Biosensors and Biofuel Cells

As already described in Sections 1.1.4 and 1.1.6 (mobile or immobilized) mediators and/or conducting polymers can also be employed to shuttle electrons between a redox enzyme and an electrode surface [11, 12, 16, 20, 21, 24, 25, 28, 270–273]. This approach is called mediated ET (see also Figure 1.6). Efficient ET throughout the entire sensing layer is envisaged in order to avoid only the enzyme layer in close vicinity to the electrode surface contributing to the overall current signal.

For many applications, soluble mediators are not suitable. Thus, redox hydrogels (hydrogels covalently modified with a redox-active mediator) are increasingly being used for reagentless biosensor structures and more recently also for biofuel cells. Heller and coworkers introduced osmium complex-modified redox hydrogels as matrices for biosensors [47, 137–139]. It was determined that the linker length between the osmium complex and the polymer backbone has an impact on the sensor response [274]. Most likely, the mobility of the osmium complex is affected by the length and, hence, flexibility of the polymer backbone and results in a higher efficiency of ET if optimized.

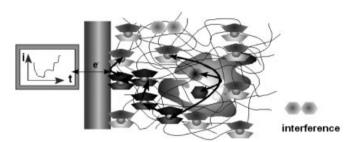
The polymer backbone of typical redox hydrogels is highly hydrophilic and is based on, for example, poly(vinyl pyridine) [48, 275–279], poly(vinyl imidazole) [280, 281], poly(acrylic acid) [282], or poly(allyl amine) [283]. Onto these backbones redox mediators, for example, osmium complexes or ferrocene derivatives, are covalently attached. The biosensors are typically realized by dropping a mixture of the redox hydrogel, a bifunctional linker, and the biological recognition element on the electrode surface. The obtained sensing film adheres well on the electrode surface in most cases and swells in aqueous solutions. Thus, the polymer is rather flexible which promotes the ET rate, the mobility of the counter-ions, and the diffusion of the substrate of the enzyme and the resulting reaction products within the sensing layer [284, 285]. The properties of a hydrogel may also provide an enzyme-friendly microenvironment, and even extend the lifetime of the involved biological recognition elements.

Electron hopping between redox relays covalently incorporated at the polymer backbone dominates the ET. Note, however, that often the first ET between the active site of the redox enzyme and the polymer-bound redox relay represents the rate-limiting step of the entire ET reaction. Biosensors have been miniaturized on the basis of redox hydrogels by employing manual dropping or dipping procedures and, for example, needle-type implantable glucose sensors have been fabricated [137, 286–289]. Properties of electron-conducting redox hydrogels were reviewed most recently in 2006 [272]. Figure 1.14 highlights the analytical task and the challenges involved for mediated ET-based devices.

The approach of employing redox hydrogels helps one to obtain higher current densities which are not only advantageous for, for example, long-term glucose

analytical task

mediated electron transfer (ET)



key features & challenges for mediated ET

- · efficiency of the overall ET process
- enzyme-friendly formation of the sensing layer
- enzyme-friendly microenvironment (important for biocatalysis and stability)
- pore size of the polymer film should securely incorporate the enzyme
- mobility of substrate and products of the biocatalytic reaction, and counter ions
- incorporation of a suitable amount of biological recognition elements for amplification
- selection of a suitable redox relay (refer to ideal characteristics of a redox mediator) that is preferably covalently attached to the polymer
- The number of redox relays within the polymer chains should facilitate fast ET.
- The contribution of current signal obtained by plain adsorption of the enzyme and/or polymer directly on the electrode surface needs to be significantly smaller than the current signal generated by mediated ET
- reduction of the influence of potential interferences or biofouling processes by selecting suitable film compositions and redox mediators

Figure 1.14 Analytical task of developing and optimizing biosensors based on mediated ET.

determinations but also for biofuel cell applications. Higher current densities compared to those of conventional biosensors are a prerequisite for bringing fundamental studies on biofuel cells closer to real-world applications. Osmium complexes exhibit many properties of an ideal mediator as outlined in Section 1.1.4. For example, their coordination structure is not very much impacted by the oxidation or reduction of the complexes. By modifying the ligand structure the redox potential can be fine tuned to the desired range [272]. The same principle is true for redox-modified electrodeposition paints (EDPs) [146], which were introduced by our group in 2002 [17, 290, 291].

For a rational design of biosensor devices, it is advantageous to aim for nonmanual fabrication processes. Electrochemical techniques provide advantages as many polymers can be electrochemically formed or deposited such as conducting polymers and EDPs, for example. The sensing layer is formed by applying potential cycles or sequences of suitable potential pulses while the biological recognition element is present in the solution [17, 292]. The advantage of this approach is that the films are formed exclusively on the electrode surfaces due to the electrochemical initiation of the deposition process. Thus, miniaturization of model biosensor architectures is straightforward and mass production of devices even at small dimensions is feasible. In addition, by automating the fabrication process, the reproducibility of the obtained biosensors should be improved.

One successful strategy to improve ET rates between enzyme and electrode is the modification of conducting polymers with redox mediators in order to obtain reagentless biosensors [11, 270, 271, 292–299]. The drawback of electropolymerization of conducting polymers is that the reaction is sensitive to oxygen, which complicates fabrication at the industrial scale.

Mediated enzyme electrodes were also realized on combined microscale and nanoscale supports [300]. Bioelectrocatalytic hydrogels have also been realized by co-assembling electron-conducting metallopolypeptides with bifunctional building blocks [301]. More recently, redox-modified polymers have been employed to build biofuel cells [25, 70, 302, 303]. In 2003, an enzymatic glucose/O₂ fuel cell which was implanted in a living plant was introduced [147].

The main potential of mediated ET lies in the increase of current densities, as the essential challenge of designing biofuel cells is to increase the biocatalytic power of these devices. Biofuel cells presently reach a power output in the range of about 10^{-6} to $10^{-3} \rm W \, cm^{-2}$. Practical conventional fuel cells operate in the range of about 1 to $10^{8} \rm W \, cm^{-2}$ [303]. Taking the calculations from Barton and coworkers into consideration [70], in which, as mentioned above, the theoretical current density of a monolayer was estimated to be about $80 \, \mu \rm A \, cm^{-2}$, one would require thousands of layers to obtain a current density above $10 \, \rm mA \, cm^{-2}$.

To summarize, the advent of redox-relay modified polymers, such as redox hydrogels, conducting polymers, or EDPs, enabled the development of biosensors that even made it to commercial applications such as implantable glucose sensors. In addition, this approach is now increasingly used for the development of biofuel cells.

1.4.4

Nanomaterials and Biosensors

Without any doubt, nanotechnology has had and is still having an enormous impact on science. When speaking of nanotechnology one typically assumes that structures are used with at least one dimension being in the sub-100 nm range. The advantages of and new possibilities offered by nanotechnology are manifold. Materials exhibit new properties when scaled down from bulk material to nanometric dimensions. These properties can be precisely fine tuned, thus allowing for the fabrication of defined structures and materials optimized for a certain purpose. Consequently, nanomaterials and concepts from nanotechnology have been much employed in biosensor development. Several reviews on the topic [182,

304-306] provide a detailed overview of the possibilities of nanotechnology in the field of biosensor research. The following summarizes the most important trends.

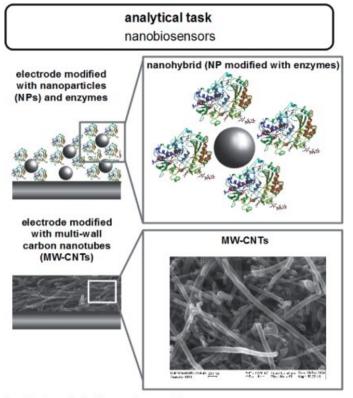
The main challenges in the application of nanomaterials for biosensor designs are the definition of the material properties, the reproducible synthesis of materials with suitable properties, and the meaningful application of nanotechnological concepts to biosensors. Definition of material properties and, thus, the choice of materials are common to other areas of biosensor research and have been discussed earlier in more general terms. The question of how to synthesize or otherwise access these materials will not be answered exclusively by the biosensor expert. Instead, multidisciplinary effort will be necessary to obtain nanomaterials with properties as required for a novel biosensor design. The seemingly most challenging task of applying nanotechnology to biosensors is to really make use of "nano features" and not simply using nanomaterials without them adding value to the biosensor architecture. In the area of biosensor research some features of nanostructures become important in addition to pure material properties. For instance, in nanometric structures diffusion lengths become very short and hence mass transport is highly efficient. Since mass transport is crucial in many biosensor designs, an increase or at least a change in sensor performance can be expected from using nanometric structures.

There are basically three broad categories of approaches towards nanobiosensors and in particular in electrochemical nanobiosensor development. The modification of a (macroscopic) transducer with nanomaterials is the first of these approaches. In electrochemical biosensors, this would translate into large electrodes modified with nanomaterials. The second approach is the miniaturization of the transducer, namely the use of nanoelectrodes [307] or other miniaturized circuitry of nanometric dimensions. The modification of biomolecules with nanomaterials or coupling of biomolecules and nanomaterials is the third category of approach towards nanobiosensors. Of course the lines between these approaches are blurred and some sensor designs may draw from more than one of these concepts.

1.4.4.1 Modification of Macroscopic Transducers with Nanomaterials

There is an enormous variety of nanomaterials that can potentially be employed in biosensor architectures. The most prominent among them are metal nanoparticles [304], quantum dots [308], and carbon nanotubes [309-311]. All of them have been employed in biosensors though not necessarily exclusively electrochemical biosensors. Quantum dots (QDs) offer unique absorption properties making them highly suitable for the construction of biosensors with optical readout. The most diverse electrochemical nanobiosensors are, however, obtained from carbon nanotubes (CNTs) which offer a wide range of different applications.

CNTs were discovered in the early 1990s [312]. CNTs have a tubular structure of closed topology and consist of hexagonal honeycomb lattices made up of sp² carbon units. A schematic of the structure of CNTs is shown in Figure 1.15. The diameters of CNTs are typically several nanometers. The length of CNTs can be up to several micrometers. Two basic forms are distinguished, single-walled



key features & challenges for nanobiosensors

- novel material properties due to nanometric dimensions
- tunable properties of nanomaterials
- cost and time effectiveness of nanobiosensors (small amount of material, small dimensions)
- comparable sizes of nanomaterials and biological molecules
- potential harm of nanomaterials

Figure 1.15 Analytical task of nanobiosensors.

carbon nanotubes (SWCNTs) and multiwalled carbon nanotubes (MWCNTs). Besides their chemical stability [313], one of the most interesting characteristics of CNTs for electrochemical biosensors are their ET properties [314]. The ET properties of CNTs can be modified by surface groups such as oxygen, NO₂, or amino groups. Electrodeposition and other forms of growth of metal nanoparticles on CNTs result in another class of nanomaterials with high application potential in electrochemical biosensors [315]. The suitability of CNTs as immobilization matrices retaining or even enhancing the activity of the respective biomolecule has been discussed [316]. In addition, the large surface area of CNTs results in a large active electrode area and CNTs can prevent electrode fouling such as caused by NADH oxidation [317].

The fabrication of electrodes (often glassy carbon or gold electrodes) modified with CNTs typically suffers from the low solubility of CNTs in most commonly used solvents. Hence, CNTs are in many cases dispersed within solvents or polyelectrolytes and drop-coated onto the electrode to be modified. Alternatively, CNTs are incorporated within composite binding materials such as Teflon [318]. Another route to CNT-modified electrodes is the direct growth of CNTs on the electrode material [319]. Electrochemical biosensors based on CNTs have been used in the determination of a wide variety of analytes including glucose, fructose, cholesterol, lactate, catechols, hydrogen peroxide, alcohols, cholines, and organophosphates, as recently reviewed in [310], as well as DNA and proteins [320]. With the level of pioneering work left behind, the powerful combination of biorecognition and extraordinary ET properties and material properties of CNTs can be expected to yield even more high-performance electrochemical biosensors in the near future.

1.4.4.2 Nanometric Transducers

This section highlights two trends in nanobiosensing. First, the use of nanofluidics [321, 322] in biosensing and, second, the use of nanoelectrodes [307] and nanoelectrode arrays [323] will be briefly discussed.

Nanofluidics is part of the field of so-called lab-on-a-chip analytical devices which integrate all essential tasks of an analytical problem into a chip-based format [45]. Lab-on-a-chip devices originate from microsystems technology and have several of advantages over conventional instrumental analysis, such as cost-effectiveness due to small material amounts used, time-effectiveness due to small diffusion lengths and therefore extremely efficient mixing of reagents, and other transport phenomena that can be employed to efficiently separate reagents. Furthermore, lab-on-a-chip devices are ideally suited for automated analysis allowing for high-throughput screening. These advantages become even more pronounced when the devices are of nanometric dimensions. In nanofluidic devices, at least one dimension of the device is close to the Debye length and hence transport phenomena not known at the macroscale and even microscale predominate. Biosensing employing nanofluidics includes immunoassays [324] among other analytical schemes such as reviewed in [322].

The electrochemical properties of nanoelectrodes differ significantly from those of macroelectrodes. Like microelectrodes [325], nanoelectrodes are characterized by a hemispherical diffusion field (whereas at macroelectrodes, linear diffusion dominates in amperometric measurements and voltammetric experiments at slow and moderate sweep rates). Consequences of the hemispherical diffusion field are the fast establishment of a stationary diffusion current, high current densities, and a favorable signal-to-noise ratio. Hence, nanoelectrodes have proven to be highly sensitive probes of biorecognition reactions. Often nanoelectrodes are employed as nanoelectrode arrays. Interdigitated nanoelectrode arrays allowed for the label-free detection of DNA using redox mediators [326]. Just as microelectrodes were much employed as probes in scanning electrochemical microscopy [327, 328] to study the immobilization processes of biomolecules on surfaces [329, 330],

nanoelectrodes have the potential to allow for an even more detailed mapping of biological activity.

1.4.4.3 Modification of Biomolecules with Nanomaterials

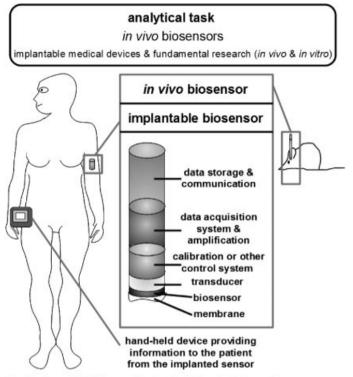
The direct modification of biomolecules with nanomaterials resulting in biomolecule—nanomaterial hybrids offers interesting possibilities for biosensing. Gold nanoparticles can be used to immobilize enzymes or other biorecognition elements on electrodes or other supports. However, in this case the nanoparticles often just function as a linker and the sensor architecture does not benefit from a unique property due to nanometric dimensions. In contrast, the already mentioned unique optical properties of QDs make these materials well suited as fluorescent labels in optical sensors [331, 332] really taking advantage of a nanofeature.

The use of nanoparticle–enzyme hybrids has been recently reviewed [333, 334] as has been the use of nanotechnology in the manipulation of redox systems at an earlier stage [335]. The wiring of enzymes by redox hydrogels or osmium-modified EDPs is the subject to another part of this chapter. Such an establishment of electrical contact between a redox enzyme and an electrode can also be achieved by nanoparticles. Standing out in this field of research is the wiring of redox enzymes by gold nanoparticles [265]. In this work, GOx was reconstituted with a gold nanoparticle (1.4 nm in diameter, corresponding to the size of the redox center of the enzyme) that was functionalized with the enzyme's cofactor FAD. Such enzyme–nanoparticle hybrids were assembled on gold electrodes leading to exceptionally good electrical contact between the enzyme redox center and the macroscopic electrode.

In conclusion, nanotechnology has contributed significantly to recent developments in biosensor research. Modification of macroelectrodes with nanomaterials has resulted in new exciting ET properties and biocompatibility. Nanometric transducers have been used to obtain new classes of biosensor devices. Finally, biomolecule—nanoparticle conjugates show a promising application potential in biosensor development. An aspect of nanotechnology that is rarely mentioned is the potential harm of nanomaterials towards health. Though a significant effort has been put into the research of this field [265, 336, 337], the consequences of the use of nanomaterials in everyday life are not yet fully understood. It seems, however, that the potential risks of nanomaterials are by far outweighed by the possibilities offered by nanotechnology.

1.4.5 Implanted Biosensors for Medical Research and Health Check Applications

As depicted in Figure 1.16, (electrochemical) biosensors are either placed in laboratory animals for fundamental (patho-)physiological and neurochemical *in vivo* measurements or implanted in the human body for health check purposes and metabolite monitoring. In the field of *in vivo* medical research, enzyme-based



key features & challenges for in vivo & in vitro biosensing

- stability over time
- selectivity with respect to the complex biological matrix (interferences)
- robustness towards temperature, electrical or environmental interferences
- immune reactions towards the device
- reliability, self-referencing system, physical robustness
- device safety (sterilization, biocompatibilty)

Figure 1.16 Analytical task of developing and optimizing in vivo biosensors.

analytical tools are often used for spatially confined measurements of their corresponding target species in preselected regions of living test subjects.

The main challenges in the field of implantable sensors are the stability of the sensor, the selectivity of the sensor, and the biocompatibility of the sensor. First of all, the sensor must not be rejected by the body. When implanted, the sensor should operate for a prolonged time to justify any surgical procedure necessary for the introduction of the sensor into the body. Even when these two challenges are met, the sensor has typically to deal with a very complex sample matrix, most commonly blood. *In vitro* sensors have to cope with the same demands in terms

of their selectivity while long-term stability is usually not such a critical issue. However, *in vitro* sensors also have to be biocompatible in such a way that their presence should not influence the biological environment in which they operate.

At present, the highly heterogeneous rat or mouse brain environment is probably most prominently addressed by in vivo and in vitro biosensors. Nevertheless, other parts of the rodent central nervous system, the many secretory glands of the regulatory endocrine system, or the tissue of muscles are sites of interest for implantable biosensors. Fixed in a particular brain region for fundamental cognitive, pathological, and pharmacological investigations, the sensing tips of, for instance, tapered voltammetric enzyme microbiosensors have demonstrated their ability to directly record up- and down-regulations of neurochemicals that may appear in response to premeditated external stimuli such as feeding, drug administration, or gratification at the local level with high time resolution. Biosensors that are implanted in humans, on the other hand, are supposed to report the dynamics of the levels of metabolites related to mental or physical disease states. To reach this challenging goal they may be placed subcutaneously, just beneath the carrier's skin, or at deeper body locations close to target organs such as the kidney, liver, and pancreas or the muscles of the extremities. Specific applications of biosensors in the human body include clinical point-of-care testing in hospital settings and personal diabetes management using handheld monitoring devices. However, the main focus in this success story of biosensor development is not on the description of a number of examples from in vivo and healthcare measurements but advancements that were reported in the last five years in terms of the design and quality of electrochemical biosensors for successful analysis in a firmly implanted configuration, be it in animals or humans. For in-depth information on specific examples of both classical in vivo (neurochemical) studies and standard human metabolite monitoring with implanted glucose, glutamate, lactate, acetylcholine, or peroxide biosensors the reader is referred to several recently published comprehensive review articles on the two subjects [65, 163, 289, 338–351].

The clever involvement of new enzymes or adapted enzyme blends in the design of implantable biosensors was used to detect physiologically or pathologically relevant biochemical compounds other than the five conventional ones already mentioned. The release of the well-known purine ATP as potent extracellular signaling molecule was, for instance, demonstrated in vivo for the Xenopus tadpole spinal cord during motor activity with implanted biosensors that had coimmobilized glycerol kinase, glycerol-3-phosphate oxidase, and phosphocreatine kinase [352]. Miniaturized carbon fiber-based biosensors for in vivo measurements of acetylcholine and choline have been prepared by means of a co-immobilization of acetylcholine esterase and choline oxidase [353]. The gliotransmitter D-serine, well-known for a long time to modulate neurotransmission at the glutamatergic synapse, has been monitored in the rat brain striatal extracellular fluid with implanted biosensors employing mammalian D-amino acid oxidase as the indicating biological recognition element [354]. The common neurotransmitter dopamine is typically measured in vivo in particular brain sections with direct fast CV at the solid graphite disc of polished glass-epoxy insulated carbon-fiber microelectrodes [355, 356]. In an attempt to improve the selectivity of local dopamine measurements in the complex extracellular matrix of brain fluid, an implantable enzyme-based dopamine microbiosensor has been constructed based on the immobilization of tyrosinase in a thin-film chitosan coating of carbon-fiber disc microelectrodes [357]. *o*-Dopaquinone, which is the product of the tyrosinase reaction with dopamine, was monitored via its reduction at the modified microelectrode surface. The application of these cathodic tyrosinase dopamine microbiosensors was reported for the continuous real-time *in vivo* visualization of electrically stimulated dopamine release in the brain of anesthetized laboratory rats. Remarkably, due to the cathodic potential the sensor response was not significantly disturbed by the presence of typical interferences such as ascorbic and uric acid, serotonin, norepinephrine, and epinephrine.

As with any conventional electrochemical biosensor, an implanted biosensor should also have an exceptional selectivity and sensitivity for the target compound, a low detection limit, and a fast response time that is well tailored to the time course of the expected dynamic changes in the concentration of the target analyte in the surrounding tissue. There are, however, important additional properties to look for when the ambition is for long-term stable electrochemical biosensor performance in the complex matrix of the bodies of animals or humans. It is very important for in vivo brain biosensor analysis, but also valid for other situations, to obtain a sensitive acquisition of a strongly localized signal from the molecule in question. In this case, a sensor design is needed that offers a positionable tapered sensor of small total tip dimension which often equals the diameter of an electroactive disc plus twice the thickness of its insulating sheath. Small sensor tip size will of course also be beneficial for placement with minimal (brain) cell and surrounding tissue damage. The exploration of glass- or polymer-insulated needle-type carbon or metal microelectrodes as diminutive precursor structures for biosensors offered an appropriate solution for this problem and no real innovation in this aspect arose in the period under consideration.

The second relevant issue for success with biosensors in the chronically implanted configuration is sufficient sensor stability over the extended time of data acquisition throughout a trial. For a lot of significant behavioral studies but basically in the general implantation case the desired period is days if not weeks of measuring time. The long-term quality of the sensor performance is of course impeded by the gradual loss of proper signal generation caused by the foreign body response and contaminating contact of functional sensor entities on the electrode surface, the immobilization matrix, and the immobilized biological recognition element with protein and lipid contents of the immediate physiological measuring environment [358-362]. Among the issues that can be adverse to long-lasting sensor functioning are (i) the fouling of the immobilization layer in the form of a delamination or loss of porosity which is essential for substrate (analyte) diffusion, (ii) the degradation or denaturation of the biological recognition element, (iii) the passivation of the electrode surface by nonspecific adsorption of proteins and lipids, and (iv) the slow formation of a barrier for substrate diffusion through an ongoing fibrous encapsulation of the biosensor tip. In view of these considerations, both the optimization of immobilizing top coat and an advanced morphological and chemical design of the transducing electrode surface of enzyme-based in vivo biosensors set the scope for the development of new concepts that offer a well-thought-out prevention against the listed set of detrimental effects of sensor tip implantation and the preservation of the analytical response. Several reports in this context have dealt with the adaptation of redox hydrogels employed for the entrapment of the enzymes used via either a mild nondegrading biocompatible environment for the active macromolecule or the creation of hydrogel surfaces that are less prone to the adsorption of contaminating (protein) species. Suggestions include a self-cleaning nanocomposite hydrogel membrane [363], biomimetic hydrogels [364], the involvement of surfactants in the formation procedure of redox hydrogels [365], and hydrogels with optimized type and ratios of individual polymerizing components [366, 367]. Taking advantage of the fact that nitric oxide effectively inhibits platelet and bacterial surface adhesion, Shin and Schoenfisch proposed advanced biosensor interfaces with a high potential to resist biofouling via the implementation of an additional nitric oxide-releasing top coating made of N-diazeniumdiolate-modified polymers [368]. Self-assembling polyelectrolyte-poly(ethylene glycol)-based nanofilm multilayers have been demonstrated on porous alumina supports as effective diffusion-controlling and protein adsorption-resistant coatings and were reported as optimized dual-function immobilization matrices for implanted biosensors [369]. Also recommended as surface modifications with promising biocompatibility properties were apparently low-fouling zwitterionic carboxybetaine methacrylate coatings [370], microporous collagen scaffolds that minimized unfavorable tissue reactions while stimulating angiogenesis in the vicinity of biosensor tips [371], porous poly(1-lactic acid) coatings to reduce fibrosis and promote new blood microvessel formation in the tissue surrounding the implanted biosensor surface [372], intentionally pre-adsorbed coatings of constructive proteins capable of inhibiting bad foreign body responses [373], new hydrophilic poly(ethylene glycol)-based redox copolymers bearing electrochemically active ferrocene and thiol/disulfide functionalities for anchoring to a gold electrode surface [374], and special nanoporous membranes [375, 376].

At present, roughly a quarter of a billion (!!!), a still steeply increasing number, worldwide cases of diabetes are reported. Situations of hypo- and hyperglycemia in patients have to be avoided and thus effective blood glucose measurement and control is a top analytical task in medical diagnostics and healthcare, respectively. Already prior to the period covered by this section, personal self-monitoring of internal glucose levels became routine in small-volume blood samples obtained, for instance, by piercing the fingertip or arm. The required commercial tools and information on both their technology and on the glucose meter marketplace is available, for instance, in [59, 65, 163, 343]. Glucose meters typically take advantage of sophisticated single-use screen-printed arrays of electrodes one of which is designed as the glucose sensor via specific immobilization of mostly GOx as the biological recognition element and the involvement of artificial free-diffusing redox mediators. Upon placement of a microliter droplet of whole blood, the electronics of the glucose measuring device assesses and digitally displays a

glucose equivalent in reasonably short time. Dependent on the quality of the measured value in relation to the accepted normal level, insulin injection or dietary carbohydrate uptake should be performed. Even if carried out several times a day, timed glucose monitoring with external sensors activated at user-chosen intervals obviously has the shortcomings that it fails to report irregular up and downs in between assessments and cannot utilize trends associated with daily habits of diabetics for an instant therapeutic action. In this context a better, albeit more challenging, route of blood glucose analysis and management is the operation of permanently implanted glucose sensors for a continuous direct detection of the analyte either in the bloodstream or the interstitial fluid of the subcutaneous tissue. The advantages and disadvantages of continuous glucose sensing as well as the difficulties in and steps forward to the establishment of the approach have been discussed in depth elsewhere [65, 338, 340, 341, 344, 346, 348, 377, 378].

Worth mentioning here as an excellent example of the remarkable achievements of focused joint academic and industrial glucose biosensor R&D is the appearance of the Freestyle Navigator® continuous glucose monitoring system from Abbot Diabetes Care/TheraSense, which recently got approval by the US Department of Health and Human Service, Section Food and Drug Administration (FDA) and became commercially available for diabetics in 2008 [65, 341, 379]. Other similar systems seem to be on their way, and include, as an example, a device that has already been tested successfully for stable long-term glucose monitoring in diabetic and nondiabetic animal models [380] and currently is awaiting FDA approval for GlySens Incorporation.

Electrochemical enzyme biosensors for in vivo studies and human body metabolite monitoring have in recent years been brought to quite an advanced level. A clear proof of the achievements is the good number of successful biosensor recordings of brain activity and the enterprise of marketable continuous blood glucose monitoring. Further improvements in the spatial and time resolution of in vivo measurements would need further sensor miniaturization and tapered nanobiosensors that should be similar to their microelectrode analogues in terms of the proper conductor embedment and resistance against sensor fouling. However, they should be equipped with a reduced total tip dimension for better positioning and nanometric sensing areas for fast and highly localized recording. In vivo biosensor measurements at the single-cell level or at least a very small cell cluster level could then add novel information to the fundamental insights that were and still are gained through in vitro single-cell electrochemistry with isolated cells out of their native matrix [381]. Though tapered enzyme-based nanobiosensors with small total tip radii have already been reported [382], these fragile needle-like versions of biosensors have, to the best knowledge of the authors, not yet been successfully operated in vivo. Here, there is definite room for future innovative research activity. Another area worth working on is the further extension of the lifetime of sensors for continuous blood glucose monitoring and the transfer of the principles of well-working GOx-based implanted biosensors to those incorporating other enzyme systems for broadening the scope of target analytes. The

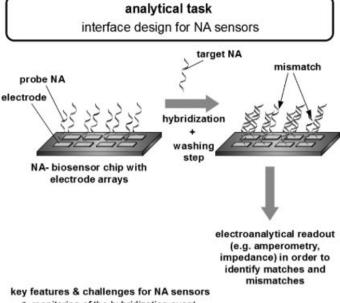
related possible enhancements and expansion of vital health and disease marker monitoring could open up the individualized and portable medication and care plan that is envisioned by clinicians and patients around the globe.

1.4.6 Nucleic Acid-Based Biosensors: Nucleic Acid Chips, Arrays, and Microarrays

Biosensors and high-throughput electrochemical screening devices based on DNA, ribonucleic acid (RNA), and peptide nucleic acid (PNA) gain their outstanding sensitivity and selectivity from the very strong base pair affinity between complementary sections of lined-up nucleotide strands, which are the evolutionary genetic code maps of living beings. In fact, all of the genetic information required for body development and functioning as well as details on disease prevalence and states of an organism are smartly made available in the programmed sequence of the nitrogenous bases adenine (A), cytosine (C), guanine (G), thymine (T; in DNA), and uracil (U; in RNA). To create so-called nucleic acid (NA; or gene) chips (or (micro-)arrays), physicochemical transducers (e.g., the surface of an electrode, a microscope glass slide, or a quartz crystal microbalance) are decorated with regular patterns of spots of synthetic single-stranded oligonucleotides, each of them being associated with an intentionally designed A, C, G, and T/U order. Subsequent to the exposure of the sensor surface to a (clinical) sample the remaining analytical task is the identification of all NA spots that underwent hybridization. Definite localization of the collection of immobilized "probe" NA strands that undoubtedly found their matching binding partner (the "target") in a pre-prepared complex blend of genetic material is at the center of the analysis of an apparent gene expression.

The major challenge for an (electrochemical) NA biosensor is the full exploration of the massive amount of information that is buried in the totality of the genomes of mammalian species. Currently, only the very tip of the "genetic iceberg" is revealed and a vast amount of effort has to be invested to finally make the best out of the technology for health science and clinical diagnosis and medication. Figure 1.17 visualizes the analytical task for electrochemical NA sensors and assays.

The main challenge for designing NA-based biosensors is that the hybridization event needs to be monitored correctly. Mismatches need to be distinguished from matches. The assay procedure needs to be compatible with the conditions of hybridization and to allow tuning of the binding specificity. For this, assay strategies employing labels such as redox-active dyes or intercalators as well as enzymes or nonlabeled detection schemes can be used. Each approach has its own benefits and limitations. The design of the capture probe is critical for overall assay performance. If one is employing electrodes, one always has to take into account the impact of the electric field on the orientation of the NA molecules, because NAs are highly negatively charged molecules. In cases where the biosensor design should be compatible with commercialization of the device, the NA chip fabrication procedure needs to take production effort and costs into account.



- · monitoring of the hybridization event
- mismatches need to be detected
- develop either labeled (e.g. dyes, intercalators, enzymes) or nonlabeled detection schemes
- usage of assay procedures that are compatible to the hybridization event and allow to tune the binding specificity (annealing temperature, ionic strength, ...)
- influence of the electric field to the orientation of NA molecules
- design of a suitable capture probe (length and nucleic base composition, linker type & length for coupling to the electrode)
- strategy for chip fabrication which allows later commercialization

Figure 1.17 Analytical task of NA-based biosensors.

The most common electrode material for electrochemical NA chips is gold; however, other metals, carbon, and certain semiconductors have been used as well. The immobilization of NA probe strands can be achieved, for instance, via simple physisorption, the chemisorption of thiol-modified NA (when gold is the transducer material), the covalent binding of, for example, biotinylated NA to (strept)avidin-modified surfaces, and NA fixation into ultrathin polymeric surface coatings. For ultrasensitive electrochemical recognition of hybridization on NA chips, reagentless, labeled, and label-free schemes have been reported. Enzymes, for instance, may be attached to the endings of target NA fragments and trace detection of double-strand formation can be established via adapted electroreduction or electrooxidation of a product of an enzyme interaction with intentionally added substrate. Other successful schemes exploit redox-active intercalating or groove-binding mediator molecules that can enter the tubular and twisted

structure of the double helix of fused probe and target NA strands to become available for the generation of electrical current. Hybridization detection strategies may use redox mediators that are tethered to either probe or target strands for the creation of configurations favorable for the induction of a Faradaic sensor response, or utilize the distinguished electrostatic properties of immobilized single- and double-stranded NA for impedimetric or voltammetric hybridization detection. An already quite matured NA chip technology in its various facets has entered almost all fields of biology and medicine and a thorough analysis of gene screening data is currently an extensively explored specialty for genotyping, pharmacogenomics, pathogen classification, gene expression profiling, drug discovery, and molecular medical diagnostics. The very obvious indication of the prominent role of state-of-the-art NA biosensing is the noticeable explosion in the number of publications on the subject. A scientific literature screening with the search terms "DNA chip, DNA array, DNA microarray, gene chip, gene array, gene microarray genosensor, DNA biosensor," all combined with the Boolean operator OR and tested for topic or title appearance gave 19227 and 1148 hits, respectively, for the five-year period 2006–2010. A search refinement with the phrase combination "electrochemical OR voltammetry" reveals close to 1000 topical publications, which, of course, are still far too many to have all been summarized in the constraint of the NA subsection of an overview article on electrochemical biosensors. Hence, the focus was placed on a selection of issues that - from the viewpoint of the authors - provide promising threads for practical advancements of NAbased diagnostics and reflect emerging trends in the field. Several recently published methodical review articles are, however, recommended as excellent additional sources of information about the basic concepts of NA immobilization on single or arrayed electrodes, for the fundamental details of the many existing analytical schemes for electrochemical detection of hybridization, and for the particulars of the variety of published designs of (electrochemical) NA chips [36, 38, 80, 95, 109, 128, 140, 145, 153, 207, 302, 383-386]. Good summaries on the state of the art of the specialties of aptamer-based NA sensing [70, 96, 106, 144, 300, 387] and on impedance/capacitance-based hybridization detection [388-390] are also available in the literature.

Aiming at an improvement of the analytical performance of NA chips/arrays in general and targeting a more sensitive detection of clinically relevant point mutations in particular, immobilized probes made of synthetic PNA have been brought into play as promising alternative biological key components of gene assays. In contrast to "normal" biological NA strands with negatively charged phosphate groups contained in their filamentary polymeric structure, PNA has an entirely uncharged backbone. In contact with single-stranded DNA or RNA target strands of a sample, PNA probe strands obey the rules of NA hybridization. However, as there is no negative polarity in PNA backbones, electrostatic repelling forces between hybridizing PNA and complementary DNA or RNA pieces are absent and the strand bonds in PNA–DNA/RNA hybrids are thus stronger than in the conventional case [391–393]. Benefits related to this effect are improved hybridization properties in terms of affinity, specificity, and sensitivity against single-base mis-

matches, a better chemical stability of the obtained duplex structures and resistance against enzyme cleavage by, for example, nucleases and proteases, and, last but not least, a reduced impact of the ionic strength of the measuring buffer on the outcome of a hybridization screening. An early demonstration of the potential of PNA probes was their successful immobilization onto a quartz crystal microbalance transducer and the subsequent application of the construct in hybridization experiments for the discrimination of perfect matches and single-base mismatches [394]. Sensitive single nucleotide mismatch detection with PNA-modified electrode transducers was the subject of a number of studies in the past five years. Representative examples are the exploitation of the impedance characteristics of a PNA-metal ion interaction for the identification of an individual C-T mismatch in a 15-mer PNA-DNA hybrid [40], the discrimination of completely complementary from mismatched double helices via specifically acting redox-active diviologen indicator molecules [49], the application of osmium mediator end-labeled PNA and stripping voltammetry for single-base mismatch detection [110], the use of ferrocene-labeled PNA for full match/mismatch identification [94], a single nucleotide polymorphism detection via joint employment of electroactive chitosan nanoparticles and PNA strings [97], and hybridization and mismatch recognition via the diverse electrostatic interaction of a cationic ruthenium(III) mediator with neutral PNA capture probes and anionic target DNA backbones [301]. Involvement of PNA probes in work on samples with clinical relevance includes, for instance, the detection of short sequences of the hepatitis C 3a virus [35], implementation in silicon nanowire biosensors for highly sensitive and rapid detection of Dengue virus [395], the application with a PNA-modified electrode for Mycobacterium tuberculosis pathogen detection [93], and the development of a PNA array for the identification and quantification of the cancer gene c-Ki-ras [396]. Another form of uncharged synthetic DNA is the morpholinos. With a comparable motivation as for PNA they have been explored as components of (label-free) surface hybridization assays [34, 50, 101].

Common electrochemical DNA chips/arrays are fabricated either with automated spotting procedures and controlled dispensing of small-volume droplets of NA solutions into a regular pattern or via a microlithography-controlled patterned (bio-)chemical on-chip synthesis. Both methodologies work well but they are associated with expensive apparatus and have their practical limits when the desired characteristic array dimensions such as probe spot diameters and distances approach the nanometric level. In view of that, cheap and spatially more accurate NA patterning techniques are sought after. In this context NA origami, the folding of longer filamentary NA structures into nanoscale two- and three-dimensional surface features [37, 51, 104, 397-400], has recently been suggested in the field of molecular NA sensing technology as a tactic for manufacturing nanoscopic NA chips [39, 98, 169, 401]. The potential of a combination of NA origami with existing nanoelectrode array fabrication technology has not yet been explored; however, this is an attractive theme that could lead to the development of powerful novel electrochemical tools for scaled-down NA hybridization screening in ultrasmall sample volumes.

Usually the electrode surface in an electrochemical NA biosensor acts as the physicochemical transducer that responds to exposure to target NA-containing sample solution and the formation of probe–target hybrids with a change in current flow, capacitance, or impedance. An interesting atypical configuration of a NA probe-carrying electrode is to use alternating electrode potential variations for controlled modulation of the structural conformation of the surface-tethered probe and fused probe–target strands and establish what is named a switchable DNA interface [402–404]. Hybridization with this method is not determined as usual via the acquisition of electrode properties but by means of an extra optical scheme that is susceptible to situations in which NA strands extend away from or fold onto the electrified sensor surface. At frequencies in the kilohertz region, the combination of electrode potential variation and an optical readout revealed distinct switching kinetics for changes from the upright to the flat surface position of NA strands, and the sensitivity was reported good enough to allow single-mismatch detection.

Without a doubt, NA-based biosensing and the application of the related arrayed gene chips for studying mRNA levels and examining gene expression profiles in human, animal, and plant samples are among the hot topics in current analytical chemistry. The prominent role is of course strongly related to the accumulated success of the many running genome projects. In particular the previous disclosure of the human genome [405] and the expectations from specific gene identification for disease diagnostics are fostering the exclusive position of hybridization assays. Electrochemical hybridization detection has been proven competent and competitive enough to be an attractive alternative to the more expensive and often technically more complex optical options in certain circumstances. However, handling the massive genetic data material of the human genome is a huge challenge for NA chip technology no matter what detection scheme is employed. So far, only an extremely small percentage of the wide range of possibilities has been accomplished. Significant advancements in both NA array fabrication and electrochemical, mass, or optical readout strategies and equipment are indispensable to enable personalized medical care based on individual gene profiles become part of daily life.

1.4.7

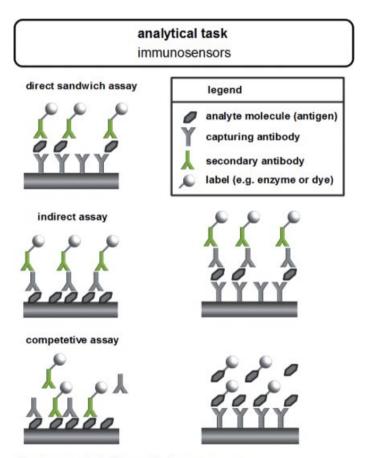
Immunosensors

Immunosensors rely on the extremely high binding affinity of antibodies towards their respective antigens. Apart from the very specific biorecognition reaction, antibodies and antigens can be produced to obtain a specific binding partner for a target of interest. A comprehensive overview of antibody production is given in [406]. However, the challenge of producing antibodies should not be underestimated. Apart from the production of the actual biorecognition element, the challenge in the design of (electrochemical) immunosensors lies in the development of a suitable detection scheme. Basically two types of immunosensor approaches can be distinguished: labeled and nonlabeled approaches.

1.4.7.1 Labeled Approaches

Labeled immunosensors typically make use of a reporter molecule attached to the respective antibody. Apart from the electrochemical detection schemes discussed below, immunosensors employ fluorescent labels, radioactive labels, or nanoparticles, among other reporter systems [407–410]. Labeled immunosensors normally operate using either a direct or indirect sandwich procedure or a competitive format as depicted in Figure 1.18.

Sandwich assays rely on secondary antibodies binding to the target antigen or antibody after the primary biorecognition reaction. In a competitive assay format, labeled and nonlabeled antibodies (or antigens) compete for the binding sites of



key features & challenges for immunosensors

- high binding affinity between antigen and antibodies
- fabrication of antigens against any antibody/hapten and vice versa
- nonlabeled approaches are a challenge for instrumental electroanalytical chemistry

Figure 1.18 Analytical task of immunosensors.

the biorecognition layer. In direct assay formats basically all binding events of the target analyte can be addressed by the secondary antibody leading to typically higher sensor signals than in competitive assays. Competitive assays, however, require less working steps leading to potentially more cost- and time-effective sensors.

Electrochemical immunosensors have been the subject of a large number of reviews [409, 411–413]. By far the most prominent electrochemical immunosensor is the enzyme-linked immunosorbent assay (ELISA) [98, 99]. This class of sensor uses enzyme labels that produce or consume an electroactive substrate or cofactor which can be monitored at an electrode interface. The most common enzyme labels are horseradish peroxidase, GOx, and alkaline phosphatase. The latter opens up a tremendous potential for powerful immunosensors. Alkaline phosphatase cleaves the non-electroactive p-aminophenylphosphate into p-aminophenol. p-Aminophenol undergoes a reversible oxidation at moderate working potentials of 150 mV (vs. Ag/AgCl) [414]. This feature does allow for the amplification of the obtained signal. If an electrode system is used comprising two closely spaced electrodes such as interdigitated electrodes (IDEs) [415, 416], p-aminophenol can be recycled (reduction of p-benzoquinonimine) at the second electrode after primary detection at the first electrode (oxidation of *p*-aminophenol). The reporter molecule is cycled between the two electrodes leading to a significant amplification of the sensor signal. This detection scheme has hence not only introduced an alternative enzyme as reporter system but has inspired a whole class of amplified electrochemical sensors including but not limited to immunosensors. Analytes that have been detected using this detection scheme as reviewed in [409, 411–413] include a wide range of bacteria, viruses, tumor markers, and others.

1.4.7.2 Nonlabeled Approaches

Detection of biological recognition reactions between antibodies and antigens omitting labels and reporter systems typically relies on sophisticated instrumental analysis. Among others, mass spectrometry and chromatographic approaches can be used to detect immunoreactions. Among optical methods, surface plasmon resonance (SPR) spectroscopy is an interesting way to detect antibody–antigen binding events [58]. Mass-sensitive biosensors based on vibrating cantilevers [417] or the quartz crystal microbalance [418, 419] offer a straightforward way to detect antibody–antigen binding.

Instrumental electroanalysis offers some alternatives for the detection of biological recognition events in immunosensors such as capacitive immunosensors and sensors based on electrochemical impedance spectroscopy. Both techniques are universal platforms for the detection of immunoreactions and hence bacteria, viruses, tumor markers, and more, as described above, have been detected with these methods.

Capacitive biosensors [390] detect changes in the capacitance of an electrode upon the occurrence of a binding event. The capacitive structure comprises a series of components such as the electrochemical double layer including the diffuse layer from ions in solution, the grafting layer, and the biorecognition layer. Since the contribution of the biorecognition layer to the overall capacitance is typi-

cally large compared to that of the other components, changes in the biorecognition layer upon the binding of antibodies or antigens can be probed by measuring the changes in the capacitance of the biosensor. This is often accomplished by potential-step experiments that require relatively cost-effective electrochemical equipment.

Electrochemical impedance spectroscopy (EIS) [420, 421] measures the complex resistance of an electrochemical system. An electrochemical system in equilibrium is perturbed by a low-amplitude sinusoidal perturbation, typically in the range 5-10 mV around the equilibrium potential. The different components of the electrochemical system will react at different speeds resulting in a frequency-dependent shift in magnitude and phase between perturbation and response of the electrochemical system. A detailed analysis of the response of the electrochemical system and its components such as double-layer capacitance, charge-transfer resistance, or even diffusion coefficients of the molecules involved can be quantified separately. The components of an electrochemical system including a biorecognition layer are altered upon a biorecognition event and can thus be probed by EIS. The occurrence of an immune recognition reaction typically results in an increase of the charge-transfer resistance of an electrochemical system. Consequently, EIS detection schemes have been widely employed in biosensors and in immunosensors in particular, for the quantification of target analytes [422, 423].

In conclusion, electrochemical immunosensors are a useful class of biosensors that have taken advantage of some major developments during the past decades. The use of enzyme labels in ELISA-type immunosensors and simple amperometric detection schemes resulted in simple and cost-effective alternatives to fluorescence immunosensors. In particular, the use of alkaline phosphatase as enzyme label allowed for the fabrication of advanced immunosensors with signal amplification by means of redox cycling, which has been a success story of its own. This detection scheme has been used in immunosensors and other biosensors and has stimulated significant developments in electrode fabrication. Instrumental electroanalysis, namely capacitance measurements and EIS allow for label-free detection of immunoreactions.

1.5 Conclusion

The full potential of amperometric biosensors has not yet been tapped, especially with respect to the versatile and broad range of applications for which biosensors can be used. Many contributions to the field of biosensors and biofuel cells still are at the "proof-of-concept" stage. Thus, the authors hope that this chapter will promote lively and valuable discussions in order to generate new ideas and approaches towards the development and optimization of biosensor architectures.

This field of research has a true multidisciplinary, "boundary-crossing" nature which is actually one of the driving forces in science in general and a great impetus for the biosensor field in particular. From the perspective of the later application of a biosensor design, for example, physicians, biologists, or engineers are involved in order to provide the specifications for a certain analyte of interest and to define the analytical challenge. Chemists and engineers are involved in the packaging of the biosensor device and additional instrumentation for readout of the signals. With respect to the actual sensing layer on the transducer, not only analytical chemists such as electrochemists but also material chemists, organic chemists, polymer chemists, biochemists, and biologists may well work hand in hand in order to achieve the desired performance of the biosensor.

There has been more than four decades of developing amperometric biosensors. Of the many approaches towards biosensor architectures reported in the literature, a vast majority of the possibilities introduced still remain restricted to the realm of academic papers. This is mainly due to the still unsolved problem of unspecific adsorption which affects biosensor performance in many cases, degradation of the sensing layer in in vivo applications, limited long-term stability of biological recognition elements, etc. In addition, in complex biosensor architectures comprising a large number of different components for immobilization, for a designed ET pathway, anti-interference layers, amplification systems, etc., the extremely complex multiparameter optimization procedure still does not provide sufficient information for a rational design of sensors. Thus, many papers provide recipes leading to a concentration-dependent change in a sensor signal without a full understanding of the underlying physicochemical properties of the sensor components and the interplay of the most significant influencing parameters. Thus, especially the knowledge gained for a particular sensor design cannot be easily transferred to other biological recognition elements, other transducer materials, etc. Having this in mind, we feel that cooking-book-type mixing of many components on a transducer surface, recording of a calibration graph, and using the same sensor for the quantification of an artificial sample do not provide substantial progress in the field. Therefore, we would like to propose, on the one hand, an application-driven approach in which a real analytical task is finally accomplished taking also into account the repeatable and reproducible fabrication of the developed sensor concepts and statistical evidence for the applicability of the biosensor. On the other hand, a more in-depth understanding of the complex influence of the different parameters has to be acquired and a general understanding about the functioning of sensors at a molecular level is required to allow for a rational optimization of the sensors. Many routes are still open to novel approaches in biosensor design and to bridge the gap between academic model studies and real-world applications.

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Abbreviations

CVcyclic voltammetry DNA deoxyribonucleic acid

DPV differential pulse voltammetry

electrode potential E electron transfer ET

FAD flavin adenine dinucleotide

GOx glucose oxidase turnover rate k_{cat} $K_{\rm M}$ Michaelis constant

microarray MA NA nucleic acid

 NAD^{+} nicotinamide adenine dinucleotide

NP nanoparticle

pyrroloquinoline quinone PQQ self-assembled monolayer SAM standard calomel electrode SCE

SECM scanning electrochemical microscope

Glossary

This glossary explains many of the technical terms relevant to amperometric biosensors. For additional terms not listed here, the reader is referred to, for example, the *Electrochemical Dictionary* [1].

bioelectrochemistry a scientific discipline describing ET reactions between bio-

> logical redox-active entities such as enzymes, redox-labeled proteins, living cells, etc., and suitable modified electrode

surfaces.

biofuel cell a device that converts chemical energy to electrical energy

by means of biocatalysis.

biosensor an analytical device that consists of a biological recognition

> element for the analyte of interest either integrated within or in close proximity to a transducer. The transducer enables the transformation of selective and quantitative information on the presence of the analyte into a quantifiable signal

[2-7].

- first generation direct oxidation or reduction of natural and freely diffusing

electroactive reactants, cofactors, or products of a biological

recognition reaction at an underlying electrode.

oxidation or reduction of an artificial electroactive mediator second that transfers electrons between the enzyme and the elecgeneration

trode surface.

- third generation

direct ET between the enzyme and the electrode surface excluding any intermediate ET reactions with redox mediators.

chemical sensor

a device that identifies and/or quantifies an analyte-of-interest as a result of a chemical reaction or interaction of the analyte with a sensor components. This chemical information is transduced into a physical signal which can be amplified.

cofactor

the terms "prosthetic group" and "coenzyme" are used for tightly bound, specific non-polypeptide units required for the biological activities of proteins. The term "cofactor" is often used synonymously with "prosthetic group" or "coenzyme." However, it is important to note that enzymes containing more than one prosthetic group are usually called multi-cofactor enzymes. Here, the term "cofactor" is exclusively used for redox-active non-polypeptide substructures in an enzyme which are tightly but not necessarily covalently bound within the protein. Free-diffusing compounds such as NAD+/NADH are not included in this meaning of the term "cofactor." They are called "coenzymes" or "mediators."

conducting polymer

polymers or "synthetic metals" that are intrinsically conducting or semiconducting. Examples are, among many others, polypyrrole, polyaniline, and polythiophenes [8–16].

diffusion

random movement of a species under the influence of a concentration gradient.

- linear

diffusion that is found at electrodes with an electroactive surface that is large in comparison to the thickness of the diffusion layer at the electrode surface. Commonly associated with macroelectrodes.

- hemispherical

diffusion that is found at electrodes with an electroactive surface that has a similar dimension to the thickness of the diffusion layer. Commonly associated with disc-shaped microelectrodes.

electrodeposition

a process in which a film of metal, polymer, oxide, or another composite is formed on an electrode surface by electrochemically induced oxidation or reduction of a precursor reagent.

electrodeposition polymer

a polymer that can be electrodeposited. For example, protons or hydroxide ions can be electrochemically generated within the diffusion zone of an electrode. This enables a pH-induced modulation of the solubility of the electrodeposited polymer, leading to a precipitation of the polymer and the formation of a polymer film on the electrode surface [17, 18].

electron transfer (ET)

a process in which an electron is transferred from one redox center to a second. ET according to Marcus theory is mainly dependent on the distance between the electron donor and acceptor, the driving force for the overall process (i.e., the potential difference), the reorganization energy of the involved redox centers, and the intervening medium (i.e., the nature of the protein matrix) [19, 20].

- direct ET in biosensors

ET from the active center of the biological recognition element to an electrode without the involvement of a mediator or cofactor [20-27].

- mediated ET in biosensors

ET transfer from the active center of the biological recognition element to an electrode via intermediate ET processes with a redox mediator [20, 21, 24, 25, 28].

enzyme electrodes flow injection analysis

electrodes modified with an immobilized enzyme. an analysis technique which is based on the injection of a defined volume of a liquid sample into a continuous flowing

carrier stream that at one point passes a detector.

immobilization, electrochemically induced

a process in which a composite or reagent is altered from a free-moving state to a fixed state by means of an electrochemically induced modulation of molecular properties of the compound that has to be deposited.

impedance interdigitated array complex resistance of an electrochemical system.

interdigitated arrays normally consist of two electrodes with finger-like structures which intertwine like a zipper structure.

interferences

substances that can electrochemically react with transducer surfaces or redox centers, which are not the substrate of the biological recognition element in a biosensor. In addition, compounds can interfere with the biological recognition reaction (e.g., inhibitors) or substrates other than the analyte of interest if the enzyme does not exhibit narrow substrate selectivity.

ion-sensitive field effect transistor (ISFET)

a semiconductor device in which the current between two electrodes (source and drain) is controlled by a third electrode (gate) which is ion-sensitive. Changes in ion activity in the surrounding electrolyte result in a change of the potential at the gate and hence in a modulation of the current between source and drain.

label-free

an assay strategy or a biosensor architecture in which no artificial reporter molecules have to be attached to one of the assay components.

Marcus theory microarray (MA) theory for single-electron transfer reactions [19, 29, 30]. electrochemical MAs are electrode structures that consist of ensembles of at least four or more microelectrodes that are typically arranged in an orthogonal grid. Electrodes can be either interconnected or individually addressable [31-33].

monolayer

microfabrication techniques that lead to well-defined structures and patterns in the micrometer range or below. Includes techniques such as physical or chemical vapor deposition and photolithography [34-40]. microfluidics discipline that deals with devices in which fluids are confined and moved in micrometer-sized spaces and channels [41-46]. multi-electrode electrode structure that consists of four or more electrodes array that are arranged in an orthogonal grid. Electrodes are typically individually addressable and miniaturized (see microarray). multiplexing the signals from individual sensors are sequentially read out by a single measuring device. materials in which at least one of the spatial dimensions of nanomaterials the material is in the sub-100 nm range. nanosensors a sensor that makes use of either nanomaterials or transducers with at least one spatial dimension in the sub-100 nm protein conjugate a synthetic molecule linking a protein with a partner molecule. The linked molecule can be either a small molecule such as a linker, a dye, or biotin or larger molecules such as antibodies or enzymes or other materials such as nanoparticles or quantum dots. The resulting hybrid molecules combine properties of both of the linked molecules. protein chips analytical devices to probe protein-protein interactions. A protein chip typically comprises an array of sites capable of carrying out an analytical task. chemical or genetic modification of a protein to adapt its protein modification properties to a certain task. reagentless a biosensor which comprises all components required for biosensor the analytical reaction securely fixed on the transducer surface. The only free-diffusing component of the sensing process is the analyte. redox hydrogel a hydrogel consists of a polymer that is highly dispersed in water and hence provides fast diffusion for molecules through the polymer network. In addition, a redox hydrogel is modified with redox-active groups such as redox mediators [47, 48]. redox mediator a redox mediator shuttles electrons between two molecules or between a molecule and an electrode (homogeneous or heterogeneous charge transfer). redox relay see redox mediator. selectivity the ability to discriminate between different substances. self-assembled molecular monolayers spontaneously formed at interfaces

(e.g., electrode-solution) by self-assembly. The monolayer is

characterized by a high degree of orientation, molecular order, and packing [49–53].

scanning electrochemical microscope (SECM) a tool for imaging of local electrochemical activity and topography of a sample by means of an ultramicroelectrode (tip) with the tip scanning either at constant height or constant distance. A variety of detection modes, such as feedback, generator–collector, direct, alternating current, and redox competition modes, are available [54–57].

surface plasmon resonance (SPR)

surface-sensitive optical technique in which the changes in the oscillation of surface electromagnetic waves are employed to detect changes on the surface of a metallic substrate [58].

thin-film electrode

an electrode that is modified with a thin layer of a substance that is not the basic electrode material. Includes metal films, polymer films, and self-assembled monolayers.

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2 Imaging of Single Biomolecules by Scanning Tunneling Microscopy

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2.1 Introduction

Interfacial electrochemistry of biological molecules such as redox metalloproteins and their constituent amino acid building blocks, DNA components, biomimetic lipid membranes, and bioinorganic "hybrids" of metallic nanoparticles and metalloproteins is moving towards new levels of understanding. Key notions are structural imaging of biomolecules at the electrochemical interface, electrochemical adsorption, electron transfer, and even enzyme processes followed and controlled towards the resolution of the single molecule. Scanning tunneling microscopy (STM) and electrical conductivity of biological (macro)molecules directly in action in aqueous biological media under full electrochemical potential control (*in situ* STM) have been powerful tools in this exciting development. Atomic force microscopy (AFM) of biomolecules in aqueous biological environments has opened other routes to single-(macro)molecular biological function, particularly protein unfolding and DNA unzipping.

The remarkable evolution of physical electrochemistry from the late 1970s, almost a renaissance of the electrochemical sciences [1], prompted by novel close interaction between electrochemistry and both solid-state physics and surface science is well recognized. The introduction of single-crystal, atomically planar electrode surfaces [2-6] sometimes with quite simple preparation procedures [4-6] was a major breakthrough. This also laid the foundation for other crucial electrochemical technologies, not the least being the scanning probe microscopies. At the same time a range of other surface techniques and theories was introduced. These included spectroscopy (UV/visible [7, 8], infrared [9, 10], Raman [11, 12], and X-ray photoelectron spectroscopy [13]), quartz crystal microbalance [14], and other physical techniques. To these were added statistical mechanical [15, 16] and electronic structural theories and computations [17-19], warranted by the new electrochemistry. Only slightly later STM [20, 21] and AFM [22, 23] signaled a new lift of surface science and interfacial electrochemistry to an unprecedented level of structural resolution [24]. Atomic resolution of pure metal and semiconductor electrode surfaces and molecular or sub-molecular resolution of electrochemical

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adsorbates could now be achieved. This opened a new world of microscopic structures and processes and a whole new understanding of electrochemical nanotechnology.

In addition to the new physical electrochemistry, imaging of single biomolecules has come to rest on the development of new biotechnology by use of mutant proteins, DNA-base variability, and *de novo* synthetic metalloproteins such as the 4α -helix heme proteins (synthetic biology). Combination of the new physical electrochemistry with new biotechnology has led interfacial bioelectrochemistry of redox metalloproteins and DNA-based molecules to a level where similar boundary-traversing efforts as in physical electrochemistry are being seen. This has been an essential prerequisite for structural mapping of the bioelectrochemical solid—liquid interface to the level of the *single* molecule.

Imaging of single biomolecules by in situ STM and mapping of the electrochemical interface have rested on pioneering efforts by Nichols and coworkers [25-27], Kolb [24], Wandlowski [28], Tao and coworkers [29], Itaya and coworkers [30, 31], and others. These efforts have been paralleled by other pioneering efforts in the electrochemistry of redox metalloproteins by Yeh and Kuwana [32], Niki et al. [33], Eddowes and Hill [34, 35], and others. Essential prerequisites in new protein film voltammetry and in data interpretation relating to single-biomolecule mapping are an underlying fundamental understanding of molecular electron transfer (ET) (and proton transfer) processes in homogeneous and interfacial environments. Molecular charge transport theory has continued to develop over the last few decades and more recently as a basis for in situ STM and other condensed matter single-molecule conductivity phenomena. Single-molecule mapping by (in situ) STM rests on molecular tunneling conductivity, rather than topographic shape. Theoretical support is therefore needed from the very beginning in order to translate, for example, STM contrasts into molecular structure and single-molecule function. The observation that molecules as large and as fragile as redox metalloproteins on atomically planar electrode surfaces can be mapped to single-molecule resolution in their natural functional state by a physical phenomenon as subtle as the quantum mechanical tunneling effect is in fact remarkable.

Theoretical support for single-molecule conductivity has assumed at least a twofold appearance. Electronic structure computations have disclosed, sometimes unexpected, details in the STM contrasts of "small" molecules such as single amino acids or DNA bases. Solvation may be of minor importance for nonpolar molecules such as straight and branched alkanethiols [36, 37]. On the other hand, static and dynamic solvation effects are crucial and computationally much more demanding for *in situ* STM of electrostatically charged molecules, for example functionalized alkanethiol self-assembled monolayers (SAMs), but can be accommodated within evolving computational schemes [38]. Other new challenges arise for redox molecules, not to say biological (macro)molecules such as redox metalloproteins. Molecular interfacial ET theory [39–41] has here emerged as a rigorous, powerful tool at first in a parameterized form due to the complexity of the systems

but all parameters are now being addressed by large-scale computational schemes [42–52]. By their analytical form, the "phenomenological" theories have offered immediate insight into current/overpotential and other central *in situ* tunneling spectroscopic correlations and also revealed new phenomena specific to *in situ* STM [45, 51–53]. These include coherent multi-electron tunneling through single (biological) macromolecules, the role of the metal centers in single-molecule conductivity of transition metal complexes and metalloproteins, features specific to limited-size molecular assemblies (ultimately a single molecule), etc. [54–56].

The combination of protein and DNA biotechnology with well-defined (single-crystal) electrochemical interfaces offers other perspectives towards biosensing and bioelectrochemical communication for electrical signal transfer between target molecules and external electrochemical circuitry. Strategic surface preparation, functional linker molecules, and, last but not least, imaging of metalloproteins and metalloenzymes directly in electron transporting or enzyme action are all part of this.

2.2 Interfacial Electron Transfer in Molecular and Protein Film Voltammetry

Imaging of single biomolecules by (*in situ*) STM rests on interfacial ET phenomena and must be supported by notions from this broad class of processes. As a quantum mechanical tunneling phenomenon, *in situ* STM image interpretation and tunneling current/overpotential and bias voltage spectroscopy must also be based on theoretical support. Two kinds of electrochemical support are important. First, electrochemical determination of molecular monolayer coverage offers important "boundary conditions" for STM image contrast interpretation. The sharp reductive desorption signal of pure and functionalized alkanethiol SAMs is a key example. Second, protein film voltammetry (PFV) is crucial to characterize and control conditions for optimal immobilized protein function. The protein molecules must, thus, be immobilized firmly enough to stay on the surface but gently enough that the molecules retain their ET, enzyme, or other function, to be carried over to *in situ* STM.

Condensed matter molecular charge transfer theory, now developed to high sophistication [39–61], offers comprehensive support both for evolving bioelectrochemistry of redox proteins and DNA-based molecules and for *in situ* STM. The theory addresses, in these contexts, *two* fundamental features. One is the aspect of electron tunneling between the electrode and donor or acceptor groups in the molecules through intermediate protein or other molecular "matter." The other one is the nuclear environmental effects from local modes, collective protein and DNA nuclear dynamics, and external solvent. The strong interaction between the transferring electrons and the environment is thus crucial. Both need attention to be paid to the inhomogeneous, anisotropic environment of the electrode–solution interfacial region.

The high spatial in situ STM resolution of biological macromolecules offers new theoretical challenges. Even novel charge transfer phenomena have been revealed as noted. We therefore first overview a few conceptual notions of the fundamental electrochemical ET process, with emphasis on ET phenomena in nanogap electrode systems and in situ STM.

2.2.1

Theoretical Notions of Interfacial Chemical and Bioelectrochemical **Electron Transfer**

Views of the electrochemical ET process as an electronic transition composed of contributions from all the electronic levels of the enclosing electrodes carry over to (bio)electrochemical nanoscale systems. Notions in focus are illuminated by the following cathodic current density form (with an analogous form for the anodic process) [39-41]:

$$j(\eta) = \int d\varepsilon f(\varepsilon) \rho(\varepsilon) j(\varepsilon; \eta); \quad j(\varepsilon; \eta) = e \Gamma_{\text{ox}}^{(1-\alpha)} \Gamma_{\text{red}}^{\alpha} W(\varepsilon; \eta)$$
 (2.1)

where $W(\varepsilon; \eta)$ is the rate constant, $j(\varepsilon; \eta)$ is the (infinitesimal) current density from a given electronic energy level in the metal electrode ε at the overpotential η , $f(\varepsilon)$ is the Fermi function, and ho(arepsilon) is the electronic level density. $\Gamma_{
m ox}$ and $\Gamma_{
m red}$ are the populations of the oxidized and reduced state, respectively, of the redox (bio)molecule close to the electrode surface, e is the electronic charge, and α is the electrochemical transfer coefficient given by

$$\alpha = -k_{\rm B}T \frac{\mathrm{d} \ln j(\eta)}{\mathrm{d}(e\eta)} \tag{2.2}$$

 $W(\varepsilon; \eta)$ holds all the information about electron tunneling, overpotential, and environmental nuclear reorganization. The following equations apply broadly [39-41, 60]:

$$j = j(\eta) = e\Gamma_{\text{ox}}^{(1-\alpha)}\Gamma_{\text{red}}^{\alpha}k^{\text{o/r}}$$

$$k^{\text{o/r}} \approx \kappa_{\text{eff}} \frac{\omega_{\text{eff}}}{2\pi} \exp\left[-\frac{(E_{\text{R}} + e\eta)^{2}}{4E_{\text{R}}k_{\text{B}}T}\right]$$

$$\kappa_{\text{eff}} = 4\pi\kappa_{\text{el}}(\varepsilon_{\text{F}}; \eta)\rho(\varepsilon_{\text{F}})k_{\text{B}}T \quad \text{if } \kappa_{\text{eff}} \ll 1, \quad \text{otherwise } \kappa_{\text{eff}} \to 1$$

$$(2.3)$$

where $E_{
m R}$ is the nuclear reorganization free energy, $\omega_{
m eff}$ the effective nuclear vibrational frequency of all the nuclear modes reorganized, $\varepsilon_{\rm F}$ the Fermi energy of the electrode, $k_{\rm B}$ Boltzmann's constant, and T the temperature. The parameter $\kappa_{\rm el}(\varepsilon_{\rm F})$ η) is the microscopic electronic transmission coefficient, the most important part of which is the electron exchange energy, $T_{\epsilon A}(\varepsilon_F; \eta)$, which couples the molecular acceptor level (A) with the metallic electronic level $\varepsilon_{\scriptscriptstyle F}$.

This reduction does not apply immediately to in situ STM of redox molecules. Equations (2.1)–(2.3), however, represent broadly interfacial (bio)electrochemical ET processes. The transparency of Eqs. (2.1)–(2.4) prompts some observations:

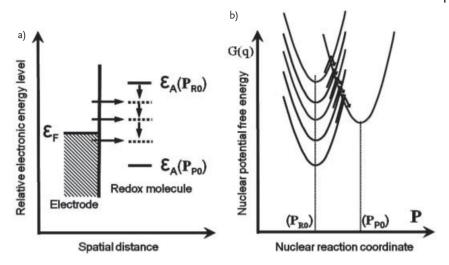


Figure 2.1 (a) Electronic energies of electrode in the reactant state (electron in the and redox molecule in (cathodic) electrochemical ET at different solvent polarization ${\bf P}.~{\bf P}_{R0}$ and ${\bf P}_{P0}$ are the equilibrium values of ${\bf P}$

electrode) and product state (electron on the molecule). (b) Corresponding potential free energy surfaces.

- The current form contains an electronic tunneling factor, $\kappa_{\text{eff}}(\eta)$, and a nuclear activation factor. Nuclear activation in all the vibrational, protein conformational, and external solvent polarization modes, along with driving force effects, thus "precedes" the electronic transition, which occurs at the crossing between the potential surfaces of reactants and products (Figure 2.1).
- The character of the electronic factor differs from ET in homogeneous solution, being composed of an infinitesimally small single-level transmission coefficient, $\kappa_{el}(\varepsilon; \eta)$, but a macroscopically large number of contributing levels, $4\pi\rho(\varepsilon_{\rm F})k_{\rm B}T$. The electronic factor is therefore the "effective" transmission coefficient, $\kappa_{\text{eff}} = 4\pi \kappa_{\text{e1}}(\varepsilon_{\text{F}}; \eta) \rho(\varepsilon_{\text{F}}) k_{\text{B}} T$. The process belongs to the weak-coupling diabatic limit when $\kappa_{\rm eff} \ll 1$, while the adiabatic limit $\kappa_{\rm eff} \to 1$ prevails when the coupling is strong.
- In addition to the driving force or overpotential, the activation free energy contains the nuclear reorganization free energy, ER, addressed further in Section 2.2.2.

Molecular charge transfer theory has developed into much more powerful frameworks than implied by this simple formalism. Features that can be incorporated include [39, 40]:

- Anharmonic nuclear motion, frequency changes, mode mixing, etc.
- Dynamic effects in the electrochemical double layer.

- 3) Nuclear tunneling, important for proton and hydrogen atom transfer.
- 4) Details of the electronic structure of the metal electrode surface.
- 5) Interfacial electrochemical ET via adsorbate states and electrocatalysis.
- 6) Relaxation of common schemes of separating electronic and nuclear dynamics, important for long-range ET.
- Stochastic theory and multistep ET via dynamically populated intermediate states.

We address briefly the two central quantities, the nuclear reorganization free energy and the electronic tunneling factor.

2.2.2

Nuclear Reorganization Free Energy

The reorganization free energy E_R contains an intramolecular and an environmental contribution. Modified forms of the simple quadratic form of Eq. (2.4) and other procedures are needed when vibrational frequency changes, nuclear mode anharmonicity, etc., are important [61]. This would apply when major local nuclear reorganization, proton or hydrogen atom transfer, or solvent dielectric saturation accompanies the ET process. Strong interaction between localized electronic states and the nuclear environment is crucial in chemical and biological charge transfer processes. At the same time the complexity of the inhomogeneous, anisotropic electrochemical interface is a theoretical challenge. Comprehensive descriptions of solvation phenomena are integrated in molecular charge transfer theory. These mostly rest on "structure-less" dielectric continuum theory [62–66], but extend to include vibrational and spatial dispersion [67, 68], dielectric interfaces [69], statistical mechanical views [70], and other reorganization phenomena, for example dynamic ionic strength features [71, 72].

The quadratic form, however, rests on the very general assumption that whatever structural features characterize the environment, conformational, polarization, density, or other physical properties respond *linearly* to the field changes, that is, changes in electric field, pressure, or other forces. The quadratic form of the activation free energy is thus broadly valid independently of the nature of the solvation and reorganization.

2.2.3

Electronic Tunneling Factor in Long-Range Interfacial (Bio)electrochemical Electron Transfer

Nuclear reorganization to a nonequilibrium configuration precedes electron tunneling, and is followed by electron (hole) tunneling along electronically facile route(s) in this configuration. The strong electronic–vibrational interaction thus leads to pre-organization prior to the electronic transition and nuclear relaxation subsequent to the transition.

Simple analytical forms illustrate the fundamental features of the (diabatic) tunneling factor. The tunneling process at the crossing of the potential surfaces (Figure 2.1) can be viewed as tunneling "percolation" through a network of intermediate groups between the electrode and the molecular redox group(s). LUMO or HOMO levels of amino acid residues, single DNA bases, or linker groups in molecular monolayers used to immobilize the proteins constitute the tunneling barrier [73–77]. The electron exchange factor, $T_{\varepsilon_{EA}}$ (Eq. (2.2)), can be given the following approximate form [40, 73-75, 78]:

$$T_{\varepsilon A} = \sum_{j} \frac{\beta_{\text{electr},1} \beta_{1,2} \dots \beta_{j,A}}{(\varepsilon - \varepsilon_{1 \text{mol}} - \Delta_{1})(\varepsilon - \varepsilon_{2 \text{mol}} - \Delta_{2}) \dots (\varepsilon - \varepsilon_{j \text{mol}} - \Delta_{j})}$$
(2.4)

which attaches closely to Green's function formalism broadly used to describe the conductivity of small molecules [79-82]. This form, however, also dates back to early approaches to long-range molecular ET [73, 83–89]. In Eq. (2.4), $\beta_{k,k+1}$ is the electron exchange factor for coupling between the kth and (k + 1)th intermediate group, $\beta_{\text{electr},1}$ couples the electrode with the nearest molecular group, and $\beta_{i,A}$ the terminal bridge group with the acceptor group. The energy denominators are the energy gaps between the level ε in the electrode and the nearest intermediate molecular level, ε_k . Δ_k ($\approx \beta_{k-1,k} + \beta_{k,k+1}$) represents all other exchange couplings in which the kth group is engaged.

Equation (2.4) can be simplified and the tunneling feature directly disclosed if (i) only the shortest tunneling path through nearest-neighbor interactions is important and (ii) all the intermediate group exchange interactions and energy gaps take the same values, β and $\Delta\varepsilon$, respectively. Equation (2.4) then reduces to

$$T_{\varepsilon A} = \frac{\beta_{\text{electr},1} \beta_{\text{NA}}}{\beta} \left(\frac{\beta}{\Delta \varepsilon(\eta)} \right)^{N} = \frac{\beta_{\text{electr},1} \beta_{\text{NA}}}{\beta} \exp \left[-\frac{1}{a} \ln \left(\frac{\Delta \varepsilon(\eta)}{\beta} \right) R \right]$$
(2.5)

where N is the number of intermediate groups, R the distance between the electrode and the molecular acceptor along the particular nearest-neighbor route, and a the average structural extension of each group. $\beta_{\text{electr},1}$ and β_{NA} are the exchange coupling between the electrode and the nearest intermediate group and between the terminal group and the acceptor molecule, respectively. Equation (2.5) shows the directional exponential distance decay associated with tunneling, with faster decay the larger the energy gap and the weaker the coupling.

The following other form is also convenient. By introducing the energy "broadening" of the group linked to the electrode [51, 52, 90]

$$\Delta_1 = \pi (\beta_{\text{electr},1})^2 \rho(\varepsilon_F)$$
 (2.6)

combination of Eqs. (2.1)-(2.3) with Eq. (2.5) gives

$$j(\eta) = 8e\Gamma_{\text{ox}}^{(1-\alpha)}\Gamma_{\text{red}}^{\alpha}\sqrt{\frac{\pi^{3}k_{B}T}{E_{r}\hbar^{2}\omega_{\text{eff}}^{2}}}\frac{\omega_{\text{eff}}}{2\pi}\left(\frac{\beta_{\text{NA}}}{\beta}\right)^{2}\Delta_{1}\exp\left[-\frac{2}{a}\ln\left(\frac{\Delta\varepsilon(\eta)}{\beta}\right)R\right]\exp\left[-\frac{(E_{R}+e\eta)^{2}}{4E_{R}k_{B}T}\right]$$
(2.7)

This form will be used in Section 2.4.

2.3

Theoretical Notions in Bioelectrochemistry towards the Single-Molecule Level

2.3.1

Biomolecules in Nanoscale Electrochemical Environment

Interfacial bioelectrochemistry of monolayers of proteins, DNA-based molecules as well as of amino acids and DNA bases as their building blocks has come to include a range of electrochemical techniques. Linear and cyclic voltammetry remain central [91–93]. Other electrochemical techniques include impedance [94, 95] and electroreflectance spectroscopy [96, 97], ultramicroelectrodes [98], and chronoamperometry [94, 99]. To these can be added spectroscopic techniques such as infrared [100], surface-enhanced Raman and resonance Raman [101, 102], second harmonic generation [103], surface plasmon [104, 105], and X-ray photoelectron spectroscopy [94, 106, 107]. A second line has been to combine state-of-the-art physical electrochemistry with corresponding state-of-the-art microbiology and chemical synthesis, relating to the use of a wide range of designed mutant proteins [108–111], and to *de novo* designed synthetic redox metalloproteins [112–114].

This broad interdisciplinary approach to the new bioelectrochemistry testifies to increasing and multifarious detail in both structural and functional (mechanistic) mapping and control at the monolayer level. Introduction of single-crystal, atomically planar electrode surfaces [94, 115] has opened a basis for the use of the scanning probe microscopies, STM and AFM, also for biological macromolecules. Importantly this extends to the *electrochemical* STM mode where now also biological macromolecules can be mapped directly in their natural aqueous environment (*in situ* STM and AFM) to single-molecule resolution [94, 116–122]. Reactive (bio) molecules combined with *in situ* STM and nanoparticle configurations also show new interfacial electron transfer phenomena, which pose new theoretical challenges. We first give an overview of some of these and then discuss some systems that have been analyzed within these new conceptual frameworks.

2.3.2

Theoretical Frameworks and Interfacial Electron Transfer Phenomena

We focus on the electronic conductivity of molecular monolayers and on single molecules enclosed between a pair of metallic electrodes. We address specifically *in situ* STM of redox (bio)molecules but concepts and formalisms carry over to other metallic nanogap configurations. Importantly, in addition to the substrate and tip, a third electrode serves as reference electrode [42–44] (Figure 2.2). This allows electrochemical potential control of both substrate and tip. The three-electrode configuration is the basis for *two* kinds of tunneling "spectroscopy" unique to electrochemical *in situ* STM. One is the current–bias voltage relation as for STM in air or vacuum, but with the notion that the substrate (over)potential is kept constant. The other is the current–overpotential relation at constant bias

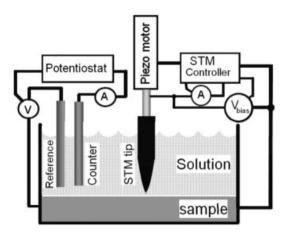


Figure 2.2 Schematic of *in situ* STM. In this four-electrode configuration the working electrode potential is controlled relative to the electrochemical reference electrode ("gate") and the (coated) tip potential relative to the

working electrode potential. The counter electrode enables recording of electrochemical processes by the independently controlled tip electrode.

voltage, that is, the electrochemical substrate and tip potentials are varied in parallel relative to the common reference electrode. The two correlations correspond to the current—bias and current—gate voltage relations in solid-state nanoscale transistors. Analogous dual-type correlations in other molecular three-level systems such as molecular Raman spectroscopy have been noted [122].

2.3.2.1 Redox (Bio)molecules in Electrochemical STM and Other Nanogap Configurations

Figure 2.3 shows schematic views of a redox molecule enclosed in a nanoscale or molecular-scale gap between two electrochemically controlled metallic electrodes. In contrast to reported systems that operate in ultrahigh vacuum at cryogenic temperatures [123–127], those represented by Figure 2.3 and addressed below apply to room temperature and condensed matter environments. Figure 2.3 also shows an extension in which a metallic nanoparticle is combined with a redox molecule and the combined supermolecule is inserted in the gap. Configurations corresponding closely to both schemes in Figure 2.3 have been characterized recently [94, 116–119, 121, 126, 128] and prompt the following observations:

1) Figures 2.3a and b extend the scheme for electrochemical ET at a single metallic electrode surface. Two electrode surfaces are now present. Their Fermi levels are separated by the bias voltage, eV_{bias} , at given overpotential η . The overpotential is the substrate potential, E_{s} , relative to the substrate equilibrium potential, E_{s}^0 , that is, $\eta = E_{\text{s}} - E_{\text{s}}^0$. The bias voltage is viewed as the tip potential relative to the substrate potential, $V_{\text{bias}} = E_{\text{t}} - E_{\text{s}}$.

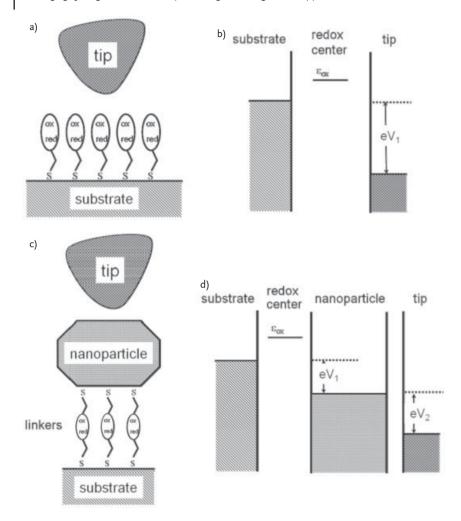


Figure 2.3 Schematic view of a redox molecule in a STM gap or enclosed between two nanogap electrodes. (a) Redox molecule in tunneling gap and (b) electronic energy

scheme. (c) Tunneling junction with a redox molecule-metallic nanoparticle hybrid and (d) corresponding electronic energy scheme, with two potential drops.

- 2) The molecular redox level, say the oxidized, electronically "empty" level at electrochemical equilibrium, is located well above the substrate Fermi energy, that is, by the reorganization free energy E_R . Figure 2.3 shows explicitly the configuration for *positive* bias voltage, so that the tip Fermi level is located *below* the substrate Fermi level.
- 3) The energies can be controlled by three external factors. The equilibrium locations can be shifted by the electrochemical (over)potential. This raises the Fermi energy of the substrate electrode, $\varepsilon_{\text{Fsubstr}}$, relative to the redox level. If

all energies are counted from $\varepsilon_{\text{Fsubstr}}$, the empty redox level, ε_{ox} , is lowered relative to $\varepsilon_{\text{Esubstr}}$ with increasing negative η . The bias voltage is a second controlling factor. As the redox site is exposed to part of the bias voltage, the redox level is also shifted relative to $\varepsilon_{\text{Fsubstr}}$ on bias voltage variation. Conformational and solvent polarization fluctuations as in molecular ET in homogeneous solution and electrochemical ET are, finally, crucial in the (bio)molecular conduction process. These novel interfacial bioelectrochemical phenomena are now reaching the level of single-molecule resolution.

The steady-state current through the nanoparticle configuration in Figures 2.3c and d offers similar dual-type current-bias voltage and currentoverpotential relations. When the nanoparticle is large enough to behave as bulk metal (diameter ≥ 5 nm) the correlations display "spectroscopic" features resembling those of the redox group alone [128, 129]. The correlations of very small particles, however, display electronic fine structure. This is caused by quantum size effects, that is, discrete electronic structural effects of the particle, which now is more like a molecular structure. These points can be incorporated in a formalism [45, 46, 48-52], a few elements of which we now overview.

New Interfacial (Bio)electrochemical Electron Transfer Phenomena

The clearest results in electrochemical in situ STM of redox molecules and biomolecules are obtained when the bias voltage and the electronic broadenings are small, that is

$$\gamma | eV_{\text{bias}}|, \Delta_{\text{substr}}, \Delta_{\text{tip}} < E_{\text{R}} - e\eta$$
 (2.8)

where γ is the fraction of the bias voltage at the site of the redox center. The bias voltage is thus in a sense a "probing energy tip." The "spectral resolution" is better, the narrower the probing tip. As the overpotential is raised, at fixed bias voltage the cathodic current first increases due to more favorable driving force, but decreases as the overpotential is increased further since practically all active species are converted to reduced form with the energy level trapped below the Fermi levels. A different character of the tunneling process arises when the bias voltage is large and the opposite inequality of Eq. (2.8) applies. Either the reduced level or, at even larger bias voltage, both the oxidized and reduced redox level forms are then trapped between the Fermi levels. Multi-electron transport (coherent or stepwise) then continues until the overpotential is raised still further and Eq. (2.8) again applies.

An additional distinction between "weak" and "strong" interactions between the redox level and the electrodes is important. In the former limit the steady-state STM process can be viewed as two consecutive, environmentally relaxed interfacial single-ET steps, each analogous to electrochemical ET (Section 2.2). This gives the steady-state tunneling current

$$i_{\text{turn}}^{\text{weak}} = e \frac{k^{\circ/r} k^{r/\circ}}{k^{\circ/r} + k^{r/\circ}}$$
 (2.9a)

$$k^{\circ/r} \approx 8\kappa_{\rm tip} \rho_{\rm tip} k_{\rm B} T \frac{\omega_{\rm eff}}{2\pi} \exp \left[-\frac{(E_{\rm R} - \xi e \eta - e \gamma V_{\rm bias})^2}{4E_{\rm R} k_{\rm B} T} \right]$$
 (2.9b)

$$k^{\rm r/o} \approx 8\kappa_{\rm subst}\rho_{\rm subst}k_{\rm B}T\frac{\omega_{\rm eff}}{2\pi}\exp\left[-\frac{\left[E_{\rm R}+\xi e\eta-(1-\gamma)eV_{\rm bias}\right]^2}{4E_{\rm R}k_{\rm B}T}\right] \tag{2.9c}$$

These forms are equivalent to the interfacial electrochemical ET rate constants (Eqs. (2.2) and (2.3)). Thus $k^{o/r}$ is the rate constant for ET from the reduced molecule to the substrate and $k^{r/o}$ the rate constant for ET from the tip to the oxidized molecule. The parameter ξ represents the potential drop between the electrode and the solution. In the tunneling gap with approximately the same spatial extension as the electrochemical double layer(s), ξ and γ cannot, however, be regarded as independent but are correlated as [48–50, 60]

$$\xi(z) = 1 - \gamma(z) - \gamma(L - z) \tag{2.10}$$

where L is the tunneling gap width and z the distance of the redox center from the substrate electrode surface. This is important in real data analysis.

A different scenario emerges in the opposite limit of strong electronic interactions. Equations (2.9a)–(2.9c) are then replaced by

$$i_{\text{tunn}}^{\text{strong}} = 2en_{o/r} \frac{k^{o/r} k^{r/o}}{k^{o/r} + k^{r/o}}$$
 (2.11a)

$$n_{\text{o/r}} = \frac{|eV_{\text{bias}}|}{\Delta \varepsilon}; \quad \Delta \varepsilon = \frac{1}{\kappa_{\text{substr}} \rho_{\text{substr}}} + \frac{1}{\kappa_{\text{tip}} \rho_{\text{tip}}} \ll k_{\text{B}} T$$
 (2.11b)

$$k^{\circ/r} \approx \frac{\omega_{\text{eff}}}{2\pi} \exp \left[-\frac{(E_{\text{R}} - e\eta - e\gamma V_{\text{bias}})^2}{4E_{\text{P}}k_{\text{R}}T} \right]$$
 (2.11c)

$$k^{\rm r/o} \approx \frac{\omega_{\rm eff}}{2\pi} \exp \left[-\frac{\left[E_{\rm R} + e\eta - (1-\gamma)eV_{\rm bias} \right]^2}{4E_{\rm R}k_{\rm B}T} \right] \tag{2.11d}$$

The difference from Eqs. (2.9a)-(2.9c) is that the electronic transmission coefficients now appear in the quantity $n_{o/r}$ in Eq. (2.11a) but the physical meaning is different from the meaning in Eqs. (2.9a)-(2.9c). Due to the strong electronic molecule-electrode interactions, the first single-ET event is followed by a large number (up to or exceeding a hundred or so) of subsequent events, while the occupied/reduced redox level relaxes through the energy window between the two Fermi levels. This is a novel ET phenomenon associated with the three-electrode in situ STM process and may be a reason for the frequently observed large tunneling current densities (per molecule), rooted in the large number of electrons transferred.

Equations (2.9) and (2.11) are a useful basis for experimental data analysis. The following simple form derives directly from Eqs. (2.9) and (2.11) for a symmetric contact, $\kappa_{\text{substr}} \rho_{\text{substr}} = \kappa_{\text{tip}} \rho_{\text{tip}}$:

$$i_{\text{tunn}}^{\text{symm}} = \frac{1}{2}en_{\text{o/r}}\frac{\omega_{\text{eff}}}{2\pi}\exp\left(-\frac{E_{\text{R}} - eV_{\text{bias}}}{4k_{\text{B}}T}\right)\left\{\cosh\left[-\frac{\left(\frac{1}{2} - \gamma\right)eV_{\text{bias}} - \xi e\eta}{2k_{\text{B}}T}\right]\right\}^{-1}$$
(2.12)

This form discloses an immediate general implication in Eqs. (2.9)–(2.12), namely a pronounced ("spectroscopic") maximum in the tunneling current–overpotential relation (at given bias voltage). For the symmetric configuration of Eq. (2.12), the maximum appears at the overpotential

$$\eta = \eta_{\text{max}} = \frac{1}{\xi} \left(\frac{1}{2} - \gamma \right) V_{\text{bias}} \tag{2.13}$$

If the redox center in the gap is exposed to half of the bias voltage drop, $\gamma = \frac{1}{2}$, and the maximum is at the equilibrium redox potential, $\eta_{\rm max} = 0$. This holds a diagnostic clue regarding the mechanism of single-molecule electronic tunneling. We shall return later to data analysis based on this view. The precise location of the maximum depends, however, rather sensitively on the potential distribution in the tunneling gap, as reflected in the correlation between the parameters ξ and γ (Eq. (2.10)).

Similar "spectroscopic" features are associated with the current-bias voltage relations, that is, new electronic conduction channels open when the redox level is brought to cross into the energy region between the Fermi levels. The "spectroscopic" current-bias voltage peaks are reflected in the *conductivity*, that is, the tunneling current derivative with respect to the bias voltage, rather than in the tunneling current itself. The tunneling current-bias voltage correlations, however, raise other issues as the conduction mechanism changes at large bias voltage where the equilibrated reduced level is trapped between the two Fermi levels [48–50, 60]. We proceed instead to some recent data of single-molecule *in situ* STM imaging and image interpretation based on both computational support and the formalism above.

2.4 In Situ Imaging of Bio-related Molecules and Linker Molecules for Protein Voltammetry with Single-Molecule and Sub-molecular Resolution

The two fundamental classes of chemical building blocks of biological macromolecules, proteins and DNA/RNA, have been prominent targets in efforts towards single-biomolecular imaging. The former group includes the amino acids among which thiol-derived cysteine and homocysteine have been particularly important. The latter group is represented by the nucleobases adenine, thymine, guanine, cytosine, and uracil. Extensive recent reviews are available elsewhere [130].

2.4.1 Imaging of Nucleobases and Electronic Conductivity of Short Oligonucleotides

Early in situ STM (and AFM) of nucleobases focused on adenine and guanine on molybdenum disulfide and highly oriented pyrolytic graphite [131–133]. Other early studies addressed all the nucleobases and Au(111) electrode surfaces. The bases were concluded to form planar hydrogen-bonded networks on the surfaces

and to display a composite pH dependence. *In situ* STM using several low-index Au and Ag surfaces (including also Au(210)) have provided both high-resolution images of thymine and uracil in particular and an interesting surface potential-dependent phase behavior [134–141]. Different chemisorbed and physisorbed phases appear in different potential ranges. The molecules in the former mode are in an upright orientation but lying flat on the surface in the latter. Notably, the phase transitions between highly ordered chemisorbed and physisorbed adlayers could be followed by cyclic voltammetry and directly visualized by *in situ* STM.

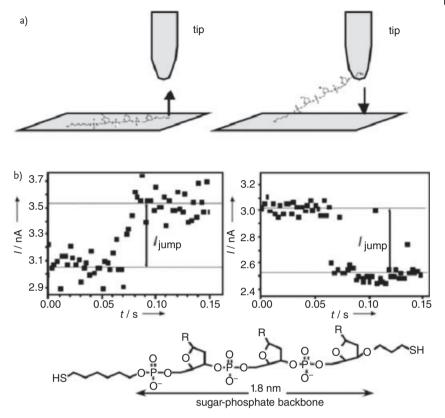
High-resolution *in situ* STM as well as phase transition dynamics of nucleobases on Au(111) and other low-index electrode surfaces supported by infrared spectroscopy have been reviewed recently by Nichols and coworkers [142] and Wandlowski and coworkers [143]. We refer to these reviews for details and note instead another aspect of single-molecule dynamics of DNA-based molecules. The observed electronic conductivity of oligonucleotides of variable length and variable base composition has opened almost a "Pandora's box" of novel DNA-based electronic properties. These include particularly photochemical and interfacial electrochemical ET. We refer to other recent reviews [144, 145] for this, still far from settled, issue but note the following STM-based studies that illuminate the conductivity issue at the single-molecule level (Figure 2.4).

Tao and associates [147] and Nichols and associates [146, 148] studied single-molecule conductivity of variable-length single-base short oligonucleotides. The studies of Nichols and associates illuminate both important issues and some puzzles. Short single- and double-strand repetitive oligonucleotides thiolated at both ends were adsorbed on a Au(111) electrode surface. A Au tip was then brought to approach the surface and capture one thiol end of the molecule while the other end kept the molecule linked to the surface. This was followed by conductivity recording of each of the all-A, -T, -G, and -C single-strand bases and of a single GC double-strand.

Hopping of electronic holes has been a strongly advocated DNA conduction mechanism as a rationale for the weak distance dependence of the ET rate along the DNA strands [149]. Hole hopping would, however, imply a strong correlation between the conductivity and the oxidation potential of the bases. The observed single-molecule conductivity displayed in fact a very weak dependence on either distance or nature of the base, say a factor of two or so higher for oligoguanine compared with the other bases, whereas orders of magnitude might have been expected. Local strand conformational fluctuations with accompanying large fluctuations in the hopping electronic energy levels have been proposed as a rationale for the robust thermal conduction [150]. This strongly gated charge transport mechanism seems, however, also to remain a puzzle.

2.4.2 Functionalized Alkanethiols and the Amino Acids Cysteine and Homocysteine

The class of straight or branched, variable-length, and pure or functionalized alkanethiols have been core molecular targets in the development of *in situ* STM



technique for single-molecule single- and double-strand oligonucleotide conductivity. The arrows indicate the direction of motion of by R and the length of the fully extended the molecule. (b) I(t) curves for molecular

Figure 2.4 (a) Schematic of the measurement attachment/detachment to the STM gold tip. Experiments in air. (c) Molecular structure of the oligonucleotides. The bases are denoted sugar-phosphate indicated. From [146].

towards single-molecule or even sub-molecular resolution (Figure 2.5) [60, 151]. Imaging has been closely tied to electrochemistry of highly ordered SAMs of alkanethiol-based molecules at single-crystal electrode surfaces where the reductive desorption process

$$Au-S-R+e^- \to Au+S-R \tag{2.14}$$

has been crucial. Alkanethiol-based SAMs have been targets in efforts towards imaging optimization regarding, for example, electrolyte composition or anaerobic versus aerobic environments. They have further been subject to state-of-the-art large-scale electronic structure computations that have disentangled the STM contrast to minute detail in both the ex situ vacuum environment and in the in situ aqueous environment [151].

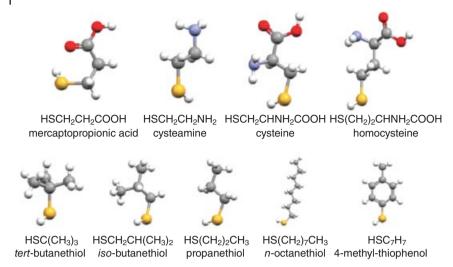


Figure 2.5 Overview of molecular structures of alkanethiol-based molecules imaged by in situ STM in various aqueous buffer solutions. For a literature overview, see [151].

Electrochemically controlled SAMs of the alkanethiol class characterized to high voltammetric resolution and to molecular and sub-molecular structural *in situ* STM resolution have been reviewed recently [60, 151]. We note here first some issues of importance to functionalized alkanethiols as linker molecules for gentle immobilization of fully functional redox metalloprotein monolayers on single-crystal Au(111) electrode surfaces. We discuss next specifically the functionalized alkanethiols cysteine (Cys) and homocysteine (Hcy). These two molecules represent a core protein building block and a core metabolite, respectively. The former has been used to display unique sub-molecular *in situ* STM resolution [152]. The latter shows a unique dual surface dynamics pattern that could be followed both by single-molecule *in situ* STM and by high-resolution capacitive voltammetry.

2.4.2.1 Functionalized Alkanethiols as Linkers in Metalloprotein Film Voltammetry With a few exceptions, redox metalloprotein voltammetry is unstable or not possible unless the electrode is modified by chemisorbed monolayers of linker molecules, or the protein modified by insertion of non-native amino acid residues. The linker molecules are mostly thiol-containing molecules which adsorb strongly on the Au surface. The opposite end holds a functional group that interacts "gently" with the protein, ensuring that the latter is immobilized and retains full functional integrity. Strategies for linker–protein interactions are broadly available but the interactions are often subtle. Closely related linker molecules can induce widely different voltammetric responses, and linker groups with no immediate expectable protein compatibility can cause strong voltammetric signals.

The use of single-crystal electrodes both offers significantly improved voltammetric resolution compared to polycrystalline electrodes and enables surface struc-

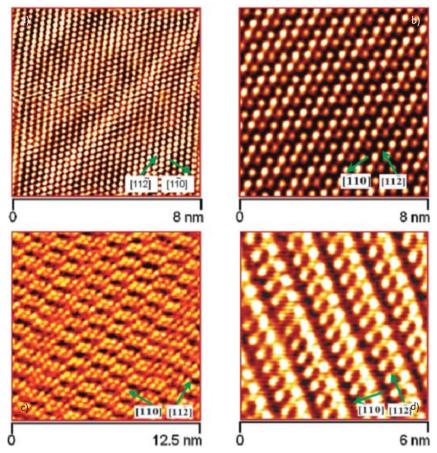


Figure 2.6 High-resolution in situ STM images of some pure and functionalized alkanethiols and protein film voltammetry linkers on Au(111) electrode surfaces in various aqueous buffers. (a) Bare reconstructed Au(111) surface;

(b) $(2\sqrt{3}\times\sqrt{3})R30^{\circ}-4$ butanethiol monolayer (hydrophobic) [153]; (c) $(2\sqrt{3}\times6)R30^{\circ}-6$ monolayer of mercaptopropionic acid (negatively charged) [154]; (d) $(\sqrt{3}\times4)R30^{\circ}-2$ monolayer of cysteamine (positively charged) [38].

tural characterization at the atomic level for pure electrodes and at the molecular level for modified electrodes. Figure 2.6 shows *in situ* STM images of a reconstructed Au(111) surface and Au(111) surfaces modified by various thiol-based linker molecules. All of them form highly ordered monolayers in aqueous buffer. By virtue of their different terminal groups, the linker molecules are efficient promoters in protein and enzyme voltammetry. Figure 2.6b shows a $(2\sqrt{3} \times \sqrt{3})R30^{\circ}-4$ monolayer of butanethiol with a hydrophobic surface [153]. Figure 2.6c shows a $(2\sqrt{3} \times 5)R30^{\circ}-6$ monolayer of mercaptopropionic acid [154] which is a highly efficient voltammetric promoter of the strongly negatively charged iron–sulfur protein

Pyrococcus furiosus ferredoxin (see Section 2.6.2.1). Mercaptopropionic acid gives a negatively charged, hydrophilic surface with a lattice structure of clusters of six mercaptopropionic acid molecules. Cysteamine with a terminal ammonium group and an efficient promoter for copper nitrite reductase electrocatalytic voltammetry gives a highly ordered positively charged $(\sqrt{3} \times 4)R30^{\circ}-2$ monolayer (Figure 2.6d) [38]. The unit cell contains two molecules giving two different in situ STM contrasts. These have been addressed comprehensively using molecular dynamics and density functional computations [38]. The two contrasts are assigned to two differently tilted molecular orientations where the strongest contrast arises from the most upright orientation. I-Cysteine, also an efficient promoter molecule for protein voltammetry, is a natural amino acid and protein building block. The zwitterionic nature of the molecule gives highly ordered domains of a more specific $(3\sqrt{3}\times6)R30^{\circ}$ -6 structure, controlled by subtle electrostatic and hydrogen bond networks on the surface [153]. Each of the clusters contains six Cys molecules but the surface structure is specific to the buffer medium used (50 mM ammonium acetate, pH = 4.6), and quite different in other media [151].

Most of the linker molecular monolayers are stable over broad potential ranges, limited by reductive and oxidative desorption and cleavage of the Au–S bond. The adsorption process can be followed in real time through several intermediate phases, such as reported for cysteamine [38] (and 1-propanethiol [155]). The *in situ* STM images shown in Figure 2.6 thus offer an overall impression of the microenvironment for immobilized redox proteins "in voltammetric action."

2.4.2.2 In Situ STM of Cysteine and Homocysteine

As the only natural amino acid building block in proteins containing a thiol group, Cys in the form of SAMs on various substrates in different environments [152, 153, 156–161] (Figure 2.7) has attracted great attention, and a number of *in situ* STM studies of Cys adsorption on Au(111) electrode surfaces have been reported [60, 151]. Notably Cys packing appears to depend sensitively on the buffer and electrolyte environment with widely different packing modes in different electrolyte solutions and at different pH values. Cystine is the dimer of Cys, with a disulfide bridge, -S-S-, rather than the -SH group. The cystine disulfide bridge appears to break and single S-Au bonds to form on SAM formation, giving virtually identical *in situ* STM images and reductive desorption peaks for Cys and cystine [153]. Figure 2.7 shows *in situ* STM images of SAMs of Cys/cystine on a Au(111) electrode surface in ammonium acetate (pH = 4.6), showing highly ordered clusters that include six Cys molecules with the same $(3\sqrt{3} \times 6)R30^{\circ}$ cell.

The dependence of the molecular SAM organization on the interaction with substrate is strikingly illustrated by the organization of the same or similar alkanethiol-based molecules on different low-index Au electrode surfaces. Cys SAMs on Au(111) and Au(110) in both aqueous solution and ultrahigh vacuum (UHV) have been studied in particular detail and illustrate the strong effects of the atomic Au substrate structures on the SAM structures.

Figures 2.7e and f compare in situ STM images of Cys SAMs on Au(111) and Au(110) surfaces in ammonium acetate (pH = 4.6). Highly ordered lattices are

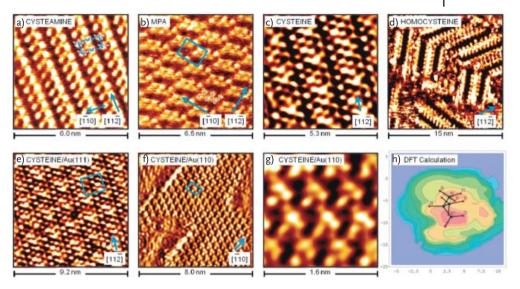


Figure 2.7 High-resolution *in situ* STM images of the amino acid cysteine in different buffers compared with other alkanethiol-based molecules on different low-index Au electrode surfaces. Overview images of (a) cysteamine [38], (b) mercaptopropionic acid (MPA) [154], (c) cysteine [153, 158], and (d) homocysteine [162], all on a Au(111) electrode surface. Cysteine on (e) a Au(111) electrode surface [153, 158] and (f) a Au(110) electrode surface

[152]. (g) Zoom-in on cysteine on the Au(110) electrode surface with sub-molecular resolution [152]. The three lobes show the three functional cysteine groups, the AuS unit, the carboxyl group, and the ammonium group. (h) Density functional theory (DFT) calculation of optimized solute cysteine structure of the Au(110) electrode surface [152].

present on both substrate surfaces. The $(\sqrt{3} \times 22)R30^{\circ}$ reconstruction on Au(111) and the (1×3) reconstruction on Au(110) are lifted in the presence of Cys, in contrast to observations in UHV, where both Cys adlayer and Au(111) herringbone reconstruction lines are visible [151, 159, 160]. Cluster structures are found on Au(111) in both UHV [160] and liquid environment [153] but the unit cell and the cluster size are different, with six and four Cys molecules assigned to each cluster in liquid and UHV, respectively. No Cys cluster structure is found in Cys SAMs in liquid environment on Au(110) [152], while clusters with eight Cys molecules are observed in UHV [161]. Both solvation and crystal orientation of the substrate are therefore undoubtedly important in controlling the molecular arrangement in the SAMs. Figure 2.7 also shows a $c(2 \times 2)$ lattice of I-Cys monolayers on Au(110). Combined with voltammetric surface coverage analysis, each unit cell is found to contain two molecules. Notably, each molecule gives three spots in the in situ STM image [152] (Figure 2.7g). Sub-molecular in situ STM resolution has thus been reached in this case. First-principles computational support has further led to the assignment of each spot to a particular chemical group (-COOH, -NH₂, and -SH), that is, to detailed image interpretation of the origin of the STM contrasts.

The higher Cys homolog Hcy with an additional –CH₂ link (Figures 2.5 and 2.7) is also a central metabolite associated with the metabolism of the amino acid methionine and other metabolic processes. In spite of the structural similarity with Cys, Hcy SAMs are packed in a quite different $(\sqrt{3} \times 5)R30^{\circ}$ structure, even in the same buffer medium, with three Hcy molecules per unit cell (Figure 2.7d). Very notably, in spite of the absence of a redox group, the voltammetry of Hcy monolayers gives a pair of well-defined sharp (24 mV) pH-dependent peaks (-0.06 V vs. SCE, pH = 7.7 [162]). The origin of the peaks, as strongly supported by in situ STM, is capacitive and caused by structural reorganization of the Hcy molecules in a narrow potential range around the potential of zero charge [162]. The voltammetric scans could in fact be followed by in situ STM all the way across the capacitive peak potential. Highly ordered domains are observed only around the peak potential, disorder appearing reversibly on either side of this potential. A molecular mechanism where the -COO- and -NH₃ groups at neutral pH can approach the Au(111) surface around the fixed anchor Au-S during the potential sweep, at potentials positive or negative of the peak potential, respectively, has been proposed [151, 162].

2.4.2.3 Theoretical Computations and STM Image Simulations

Multifarious patterns of differently functionalized alkanethiol SAMs have been mapped to single-molecule and sub-molecular resolution by in situ STM in aqueous electrolyte, strongly supported by electrochemical studies of reductive desorption in particular. In situ STM is, however, rooted in electronic conductivity and quantum mechanical tunneling. Theoretical support is therefore needed in detailed image interpretation of all the many facets of alkanethiol-based SAM packing and in situ STM contrasts [163].

The variety of straight versus branched, otherwise nonfunctionalized alkanethiols constitutes one system class where computational support has been decisive. This support has clarified the subtle interplay between Au-S binding sites (hollow, bridge, and a-top sites, or intermediates in between), composite lateral interactions, and Au atom "mining" out of planar Au electrode surfaces [36, 37, 164, 165]. A second class of functionalized alkanethiol SAMs where theoretical and computational support has provided new insight is the strongly solvated 1-Cys [158] and cysteamine [38] SAMs. Solvation has been included in different ways. Both continuum models and large-scale molecular dynamics have been combined with quantum chemical computations at the density functional theory (DFT) level, adding immensely to the understanding of the molecular packing and of the in situ STM contrasts. The DFT computations of in situ STM L-Cys on Au(111) surface models included, for example, a dielectric solvent and representation of the STM contrasts directly in the form of the commonly applied constant current mode with electronic coupling to a model tungsten tip [158]. Image contrasts could be reproduced but not to the same resolution as the data. Maximum electrostatic stability of clusters of exactly six zwitterionic Cys molecules as observed in ammonium acetate solution (pH = 4.6) was also found. This kind of computational support applies better in the sub-molecular image interpretation of in situ STM of L-Cys on a Au(110) electrode surface [152]. As observed also for cysteamine [38], the computed lobe positions do not, however, directly accord with the atomic surface structure of L-Cys but rather reflect the dominating electronic densities, and the molecular orbital contributions from the three groups closest to the Fermi levels.

2.4.3 Single-Molecule Imaging of Bio-related Small Redox Molecules

Tao reported the first case of *in situ* STM spectroscopic features using Feprotoporphyrin IX on highly oriented pyrolytic graphite as target system [166]. Although sophisticated and far from straightforward, single-molecule *in situ* STM imaging and electrochemical scanning tunneling spectroscopy (STS) of redox molecules is now an expanding area of single-molecule science and a wider range of target molecules have been addressed and characterized. Figure 2.8 shows selected examples of redox molecules both imaged to single-molecule resolution and displaying single-molecule tunneling spectroscopy features. These and other reported cases include (i) organic redox molecules (viologens, perylene tetracarboxylic diimide, oligoanilines, tetrathiafulvalenes, and quinones/hydroquinones); (ii) transition metal complexes (metalloporphyrins and metallophthalocyanines, bipyridine and terpyridine complexes of osmium and cobalt); and (iii) molecular-scale metallic nanoparticles in the size range of single-electron charging [60].

In situ STS (Figure 2.8) follows broadly the pattern of sequential two-step interfacial ET. In addition to the expected spectroscopic STS feature, other observations have included (i) conspicuous in situ STS resonance features ("molecular transistor" function, "on–off" ratio > 50) and single-molecule rectification ("molecular diode" function, rectification ratio > 20) [167]; (ii) systematic variation of the peak potential with the bias voltage; (iii) comparative in situ STS involving different metals and ligands with widely different interfacial electrochemical ET rate constants; (iv) stochastic features in which distributions of single, double, triple, etc., molecular conductivity have been observed; and (v) current–distance correlations from which the notion of coherent multi-ET in a single-molecule in situ STS event has received substance. Phenomenological theoretical approaches have been used for successful framing of the observations including the dynamic solvent aspect of the process. New theoretical efforts towards a description of the solvent fluctuational dynamics on a molecular basis are now warranted. We shall readdress some of these aspects below.

The bio-related redox molecular entities shown in Figure 2.8 thus display a pattern which follows consistently the concepts of two-step electrochemical tunneling and the formalism discussed above. Working principles of redox switching, rectification, and amplification at the single-molecule level of interfacial electrochemical ET have thus been achieved. This can be compared with biological redox macromolecules addressed below. We consider first briefly two cases of biomolecules or bio-related molecules intermediate in size between

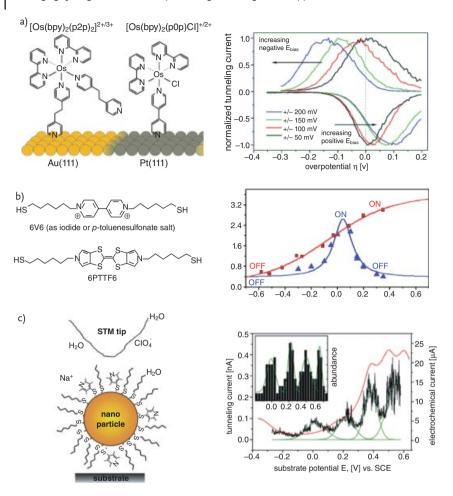


Figure 2.8 Three examples of small redox molecules for which *in situ* STM and single-molecule *in situ* STS have been recorded. Molecular structures and tunneling current–overpotential correlations are shown.
(a) Two osmium polypyridine complexes on a Pt(111) electrode surface and tunneling current–overpotential correlations at different bias voltages [55]. (b) Hexanethiol 4,4'-substituted viologen (6V6) and 6-*p*-tetrathiafulvalene

(6PTTF6). These two molecules display rather different looking current—overpotential correlations assigned to a softer molecular structure for 6V6 leading to "gated" tunneling, whereas 6PTTF6 has a much more rigid structure [171, 172]. (c) Coated Au₁₄₅ nanoparticle displaying single-electron charging in both differential pulse voltammetry and single-particle *in situ* STS [173]. Inset: histogram showing the peak abundance.

the biological building blocks and the larger "working" metalloproteins and metalloenzymes. The first are membranes immobilized on single-crystal electrode surfaces; the second is human insulin monomer or dimer. *In situ* STM imaging of both, to single-molecule structural resolution, has been accomplished recently [168–170].

2.5

Imaging of Intermediate-Size Biological Structures: Lipid Membranes and Insulin

We discuss here two types of intermediate-size biomolecules imaged in considerable detail by *in situ* STM. The first type consists of mono- and bilayer lipid molecules assembled on Au(111) electrode surfaces. These molecular assemblies resemble biological membranes and offer insight into biomimetic membrane structure and activity. The other system is the protein hormone insulin immobilized on low-index Au(111), Au(100), and Au(110) electrode surfaces. Although a true (but small) protein (molecular mass of 5800 Da), the molecule undergoes drastic deformation and perhaps even decomposition on Au electrode surfaces classifying in a certain sense insulin as an "intermediate-size" biological molecular target.

2.5.1

Biomimetic Mono- and Bilayer Membranes on Au(111) Electrode Surfaces

Phospholipid bilayers assembled on atomically planar Au surfaces rather than on mica or glass surfaces were introduced by Lipkowski and associates [168, 169] as a novel target class of biological molecules for which in situ STM has offered single-molecule insight. These molecules, represented by 1,2-dimyristoyl-snglycero-3-phosphocholine (DMPC) (Figure 2.9) are intermediate in size between amino acids and nucleobases on the one hand and oligonucleotides and proteins on the other hand. Monolayers and bilayers can be assembled by fusion of unilamellar vesicles or using the Langmuir-Blodgett-Schaefer technique but specially prepared ultrasmooth Au surfaces are essential for defect-free adlayers. However, once this is achieved the assembled monolayers or bilayers have offered notable insight regarding structure, assembly dynamics, and biomimetic membrane composition. The latter includes the presence of cholesterol, or insertion of proteinbased transport channels and antibiotics such as the short helical protein gramicidin, all at the molecule scale in aqueous biological environment. To this is added the powerful electrochemical aspect by which structural transitions, transmembrane transport, and other biomimetic membrane activity can be addressed.

Figure 2.9a shows the lipid molecule DMPC. Two layers contacted via the hydrophobic tails lead to spontaneous formation of a double-layer biomimetic membrane that can be transferred to a single-crystal ultraplanar electrochemical Au(111) surface. The hydrophilic head groups contact the electrode surface via an intermediate water film. Due to the structurally very well-defined assembly, not only AFM and *in situ* STM but also neutron reflectivity, X-ray diffraction, and infrared reflection absorption spectroscopy (IRRAS) have been employed to support the direct visual *in situ* STM. Electrochemically controlled structural changes, phase transitions, and the effects of the common membrane component cholesterol (Figure 2.9b) and peptide drugs have been investigated in this way.

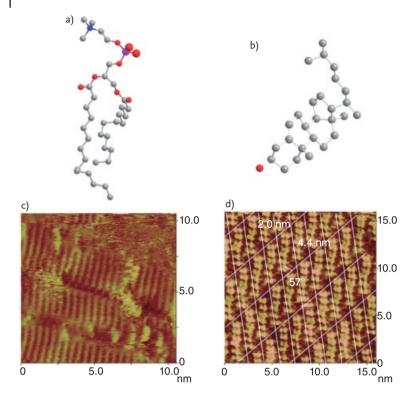


Figure 2.9 *In situ* STM imaging of 1,2-dimyristoyl-*sn*-glycero-3-phosphocholine (DMPC) lipid membranes. Molecular structures of the membrane constituents: (a) DMPC; (b) cholesterol. (c) DMPC

monolayer in early phase of immobilization and (d) segregated cholesterol monolayer component in 7:3 mixed DMPC/cholesterol monolayer. STM images adapted from [168, 169].

As illustrations of the DMPC/cholesterol-based biomimetic membranes on electrochemical Au(111) electrode surfaces, Figures 2.9c and d show *in situ* STM images just after exposure to a 7:3 DMPC/cholesterol solution. Segregated ordered monolayer domains with flat-lying DMPC and cholesterol molecules are clearly seen. After prolonged exposure, bilayers with the molecules in upright orientation form with featureless *in situ* STM images. In comparison, Figure 2.10 shows *in situ* STM images of a differently prepared 1:9 molar mixture of (helical) gramicidin and DMPC. Gramicidin appears as dark spots or cavities randomly dispersed inside the highly ordered DMPC monolayer. Together with support from AFM, neutron reflectivity, and IRRAS, images such as those in Figure 2.10 therefore offer powerful information regarding single-molecule membrane dynamic events such as phase transitions, transmembrane transport, and the action of antibiotic peptides.

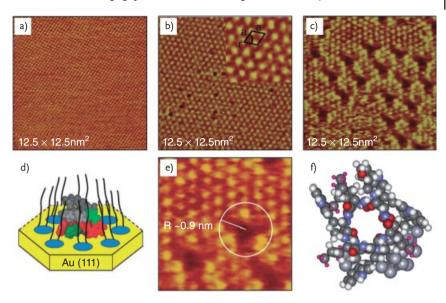


Figure 2.10 In situ STM images of mixed gramicidine/DMPC film: (a) pure Au(111) electrode surface; (b) DMPC monolayer on the surface (prepared differently from the layer shown in Figure 2.9); (c) mixed gramicidine/DMPC monolayer. The dark cavities are gramicidine molecules embedded

in the membrane matrix. (d) Schematic of gramicidine embedded in the DMPC matrix on the Au(111) surface; (e) zoomed *in situ* STM image of the membrane with gramicidine (dark holes) inserted; (f) structural representation of the gramicidine molecule (PDB 1JNO). Images adapted from [168, 169].

2.5.2 Monolayers of Human Insulin on Different Low-Index Au Electrode Surfaces Mapped to Single-Molecule Resolution by *In Situ* STM

Insulin is one of the most important hormones in the cellular uptake of glucose followed by conversion of glucose to glycogen in the liver [174–176]. The active form is the insulin monomer, a small protein (molecular mass of 5800 Da) with 51 amino acids (human insulin) in two strands, the A- and the B-strands, folded into a globular tertiary structure assisted by three disulfide groups [177–182]. Stimulated by glucose uptake, insulin is released to the blood as the physiologically active monomeric form (Figure 2.11) [177, 178]. Dimer insulin is formed by hydrophobic monomer—monomer interactions at concentrations above $1 \, \mathrm{ng} \, \mathrm{ml}^{-1}$ [183].

The insulin molecule is exposed to multiple interactions with interfaces *in vitro* and *in vivo*. *In vivo* surface interactions are with protein and lipid serum components, membrane-bound receptors, or membrane surfaces [176, 184–186]. Surface interactions strongly affect the folding of insulin and the interconversion between

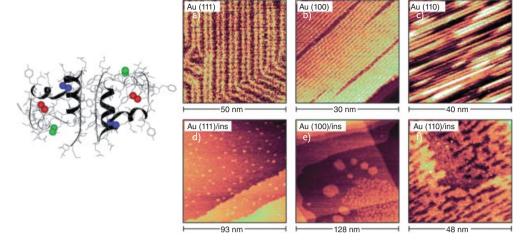


Figure 2.11 In situ STM of human insulin on single-crystal Au electrode surfaces [170]. At left is a structural representation of the expected dominating insulin dimer form (PDB 1B9E). The A- and B-chains and the three disulfide groups in each monomer (blue, red,

and green) are indicated. *In situ* STM images of the three low-index reconstructed Au surfaces: (a) Au(111); (b) Au(100); (c) Au(110). *In situ* STM images of molecular-scale insulin structures on the three surfaces: (d) Au(111); (e) Au(100); (f) Au(110).

insulin monomer and dimer. Surface structure and dynamics are therefore crucial in physiological action and in insulin therapy.

Insulin monolayers on surfaces (mica, silica, mercury, gold, platinum) have been investigated using spectroscopy (ellipsometry [187, 188], reflectometry [189], mass spectrometry [190], surface plasmon resonance spectroscopy [191]), using electrochemical techniques such as cyclic voltammetry [192-197], electrochemical impedance spectroscopy [196], and electrochemical quartz crystal microbalance techniques [195], and using AFM (insulin fibrils) [198-200]. Following recent progress in redox metalloprotein film voltammetry and single-molecule in situ STM [60, 106, 130, 154, 201, 202], Welinder et al. [170] studied voltammetry and in situ STM of dimeric human insulin using single-crystal Au(111), Au(100), and Au(110) electrode surfaces. The data indicated that structural single-molecule mapping and highly surface-specific packing of insulin dimer had been achieved (Figures 2.11d-f). These observations hold prospects for following insulin selfassociation such as the interconversion between dimeric and hexameric forms in real time at the single-molecule level. This could be compared with cysteamine and 1-propanethiol monolayer formation and the Hcy orientational interconversion noted above [38, 155, 162].

Figures 2.11a–c show high-resolution in situ STM images of the three bare (reconstructed) electrode surfaces. The reconstruction lines and herringbone organization of the Au(111)-(22 $\times\sqrt{3}$) surface structure are clearly visible (Figure 2.11a). Figure 2.11b shows the Au(100)-hex reconstructed electrode surface, domi-

nated by the corrugation lines caused by the incommensurability of the quasi-hexagonal surface layer with the square arrangement of the underlying Au atoms. Figure 2.11c shows the (1×3) missing row reconstructed Au(110) electrode surface. These images represent the microscopic environment for insulin adsorption [152, 203, 204].

The surface reconstructions are lifted when insulin is adsorbed leading to patterns that show highly specific organization on the different surfaces. Figures 2.11d–f show insulin adsorption on the Au(111), Au(100), and Au(110) electrode surfaces at low insulin concentration (0.01 mg ml⁻¹). *In situ* STM has clearly reached the level of resolution of the single molecule. The data show distinct surface-specific adsorption on the three electrode surfaces and point to the anticipated role of disulfides in both adsorption and *in situ* STM contrast. A common feature is a high surface density of molecular-scale (single-molecule) structures. These represent different insulin adsorbate features on the three surfaces.

Insulin monomer/dimer structures on Au(111) are evenly scattered over the wide terraces. The different sizes reflect different binding modes and Au-S binding units as well as concentration-dependent monomer/dimer distribution in the adsorption of presumably heavily unfolded protein. Different molecular-scale structures appear on Au(100) electrode surfaces. A dense adlayer covers most of the surface. Molecular-scale structures appear both at the Au(100) terraces and close to the terrace edges, where the layers extend across the steps. Larger structures are also scattered over the surface. Their rectangular shape, monatomic height, and alignment along the crystallographic axes of the substrate suggest that these structures are Au(100) islands from the lift of the surface reconstruction. The more open Au(110) surface structure induces still another adsorption mode. Like small thiol-based molecules such as Cys [94, 152, 153, 205], insulin displays much higher adsorption reactivity on Au(110) than on Au(111) and Au(100). Even the smallest insulin concentration used, which leads to low or intermediate coverage on Au(111) and Au(100) electrode surfaces, gives almost complete coverage on Au(110) electrode surfaces. The adsorbate organization follows the Au(110) electrode surface topology with insulin molecules aligned in the Au(110) surface grooves, at some places spilling over and merging into larger structures.

Insulin adsorption on the three low-index Au electrode surfaces thus displays a diversity of patterns resolved to the single-molecule level by *in situ* STM. Together with the voltammetric diversity of insulin monomer/dimer monolayers on the same surfaces, multifarious molecular-scale scenarios emerge but with some common notions. Au–S bond formation and disulfide bond splitting are first key determinants in the adsorption of S-rich, structurally "soft" protein molecules. The adsorption of such soft molecules is therefore invariably accompanied by extensive protein unfolding. More rigid molecules such as the blue copper protein azurin retain functional ET integrity in spite of the structural perturbation caused by disulfide binding to Au(111) electrode surfaces [60, 94, 115, 130]. Solvation in the bulk and adsorbed states can, further, vary widely. As noted, the amino acid Cys is, for example, organized in quite different patterns in different electrolyte solutions [152, 153, 158]. Adsorption of insulin would be accompanied both by strong

solvation changes associated with hydrophilic amino acids in folded, unfolded, and adsorbed states, and with competing hydrophobic forces. Hydrolytic or other protein degradation caused by the strong chemical interaction between insulin and kink or other local metallic structures may accompany the adsorption process. Other, smaller molecular fragments can therefore contribute to the adsorption patterns.

In addition to the chemical diversity of insulin on the different Au electrode surfaces, the *in situ* STM contrasts reflect different electronic conductivity of the different insulin molecular fragments. The electronic density of the Au–S unit at the site of the tip thus exceeds significantly the contributions from the rest of the protein [37, 38, 158, 164]. *In situ* STM contrasts can therefore be dominated by the Au–S units while the molecular-scale contrast *distribution* is of course also determined by the rest of the protein.

2.6 Interfacial Electrochemistry and *In Situ* Imaging of Redox Metalloproteins and Metalloenzymes at the Single-Molecule Level

2.6.1 Metalloprotein Voltammetry at Bare and Modified Electrodes

PFV, reviewed extensively elsewhere [60, 92, 93, 206-208], is established as a powerful tool in protein science, including mapping molecular mechanisms of intramolecular and interfacial ET processes in surface-bound protein systems. PFV methodologies include linear, cyclic, fast-scan, square wave and differential pulse voltammetry, electrochemical impedance, X-ray photoelectron spectroscopy, and microcantilever sensor technology. Other approaches are represented by artificial interfacial biological ET chains [109], the use of ultramicroelectrodes [98], and the use of gold nanoparticles [209]. A range of redox metalloproteins including representatives of small ET proteins, the blue copper, heme group, and iron-sulfur proteins, and a range of redox metalloenzymes at the level of protein monolayers have been addressed [210]. The enzymes include glucose oxidase [211, 212], fumarate oxidase [213], succinate dehydrogenase [214], nitrate [215] and nitrite reductases [216, 217], peroxidases [206, 218], DMSO reductase [219], hydrogenases [220], and cytochrome c oxidase [221]. These represent complex interfacial electrocatalytic functions. Novel detail can be expected from nanoscale and single-molecule imaging such as for the smaller ET metalloproteins.

2.6.2 Single-Molecule Imaging of Functional Electron Transfer Metalloproteins by *In Situ* STM

Imaging of single protein molecules on conducting surfaces by STM and AFM was reported early on, but mostly in ambient air environment and on bare surfaces [45]. *In situ* STM of proteins in their natural aqueous biological media under

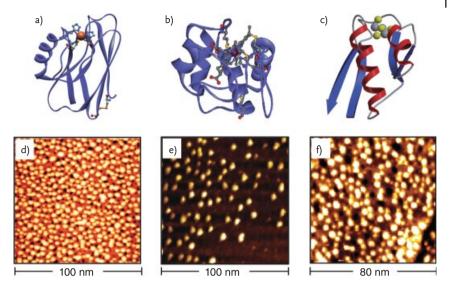


Figure 2.12 Overview of three-dimensional structures and *in situ* STM images of metalloproteins representative of the three ET protein classes characterized by single-crystal PFV and *in situ* STM to single-molecule resolution. (a) Blue copper protein *P. aeruginosa* azurin (PDB 4AZU) [94];

(b) heme protein *S. cerevisiae* cytochrome c (PDB 1YCC) [106]; (c) iron-sulfur protein *P. furiosus* ferredoxin (PDB 1SJ1) [154]. (d) *P. aeruginosa* azurin and (e) *S. cerevisiae* cytochrome c on bare Au(111); (f) *P. furiosus* ferredoxin on Au(111) modified by a mercaptopropionic acid SAM (cf. Figure 2.7).

electrochemical control and in combination with SAM-modified single-crystal electrode surfaces, on which the proteins retain their ET or enzyme function, is much more recent. Figures 2.12-2.14 show overviews of redox metalloproteins on surfaces such as those shown in Figure 2.6 imaged to single-molecule resolution by in situ STM. The molecules include representatives of the three major classes of ET metalloproteins, the blue copper proteins [54, 94, 118, 119, 202] (azurin in particular), heme group proteins [106], and iron-sulfur proteins [154] (Figure 2.12). Other single-molecule studies have addressed the multicenter redox metalloenzymes copper and decaheme nitrite reductases [54, 217, 222] (Figure 2.13). To these can be added *de novo* designed 4α-helix proteins without [223, 224] and with a heme group inserted [223, 224]. Single-crystal, bare, and modified electrodes were used, and the proteins in their functional states were mapped to singlemolecule resolution. Other studies of azurin [114, 126, 127, 225], plastocyanine [226], and yeast cytochrome c [121] have also been reported. We discuss first briefly in situ STM studies of these different system classes [60] which illuminate both the powerful potential and some limitations of in situ STM imaging of single functional biological macromolecules. We then proceed to three new cases not reviewed before in the context of single-molecule imaging. These are the wild-type and mutant 4α -helix bundle heme protein cytochrome b_{562} , the bacterial diheme protein cytochrome c4, and surface-immobilized metalloprotein-nanoparticle hybrid entities.

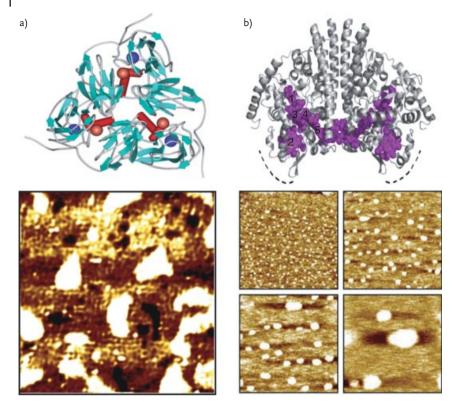


Figure 2.13 Three-dimensional structures and *in situ* STM images of the multicenter metalloenzymes (a) *A. xylosoxidans* copper nitrite reductase (PDB 1HAU) [217] and (b) *E. coli* decaheme nitrite reductase (PDB 1GU6) [227].

2.6.2.1 Small Redox Metalloproteins: Blue Copper, Heme, and Iron-Sulfur Proteins

Pseudomonas aeruginosa azurin (Az), horse heart (HHC) and yeast (Saccharomyces cerevisiae) cytochrome c (YCC), and Pyrococcus furiosus ferredoxin (PfFd), as representatives of the three classes of small ET metalloproteins (Figure 2.12), have been single-molecule in situ STM targets supported by single-crystal voltammetry and other methods. Focus was on conditions where the molecules retain their ET function in the immobilized state on bare (YCC, Az) Au(111) electrode surfaces or on Au(111) electrode surfaces modified by thiol-based SAMs. In separate ways HHC and P. aeruginosa Az have emerged as electrochemical paradigms for protein interfacial electrochemical ET [60].

2.6.2.2 Single-Molecule Tunneling Spectroscopy of Wild-Type and Cys Mutant Cytochrome b₅₆₂

 4α -Helix heme proteins of approximately the same size as HHC and YCC constitute a second class of redox metalloproteins recently introduced as single-molecule

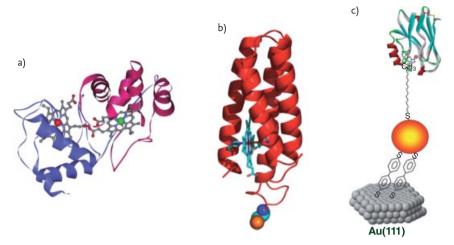


Figure 2.14 Three new target metalloproteins in single-molecule combined voltammetric and in situ STM studies recently reported:

(a) P. stutzeri cytochrome c₄ (PDB 1EPT)

[228]; (b) D50C mutant cytochrome b₅₆₂ [229]; (c) P. aeruginosa azurin to 3 nm coated Au nano metalloprotein–AuNP has defined as a surin coated Au nano metalloprotein–AuNP has defined as a surin coated Au nano metalloprotein and in situ STM studies recently reported:

(a) P. aeruginosa azurin to 3 nm coated Au nano metalloprotein–AuNP has defined as a surin coated Au nano metalloprotein and in situ STM studies recently reported:

(b) P. aeruginosa azurin to 3 nm coated Au nano metalloprotein and in situ STM studies recently reported:

(c) P. aeruginosa azurin to 3 nm coated Au nano metalloprotein and in situ STM studies recently reported:

(a) P. stutzeri cytochrome c₄ (PDB 1EPT)

(b) D50C mutant cytochrome b₅₆₂ [229]; electrode surface [230].

(c) P. aeruginosa azurin hydrophobically linked to 3 nm coated Au nanoparticle in a metalloprotein—AuNP hybrid immobilized in a 4,4'-biphenyldithiol matrix on a Au(111) electrode surface [230].

in situ STM and STS targets [229]. A totally synthetic 4α -helix heme-free carboprotein with a thiol-based linker group [223] and another totally synthetic MOP-C (modular organized protein; "C" represents Cys) linked via maleimidopropionic acid to Cys, in turn linked to a Au(111) electrode surface [224], are previously reported in situ STM cases of 4α -helix proteins. Although feasible, there were limitations in both cases. Molecular resolution could only be obtained for the molecular linker group in the former case probably due to high internal conformational flexibility of the heme-free ("apo-") protein. Notably, single-molecule structures did appear close to the potential of reductive desorption of the S-linked molecules. Single-molecule resolution of the MOP-C protein was achieved but it was inconclusive as to the role of the heme group in the single-molecule conduction process.

The recent study of Della Pia *et al.* [229] offered another, more successful strategy addressing the "natural" wild-type and mutant 4α -helix heme protein class cytochrome b_{562} (Figure 2.15). Cytochrome b_{562} is a class of bacterial (*Escherichia coli*) periplasmic respiratory ET proteins and contains a single heme group attached parallel to the α -helices by axial coordination via Met7 and His102. The reduction potential is slightly lower than those of the c-type cytochromes that also link the heme groups to the protein by thioether groups. The mutant protein obtained by replacing aspartic acid (D) with Cys at position 50, that is, the D50C mutant protein (Figure 2.15), has proved highly suitable for direct linking of the protein to Au(111) electrode surfaces in sub-monolayers diluted in a 1,4-dithiothreitol coadsorbed matrix. Robust voltammetry (in contrast to the wild-type protein), stable

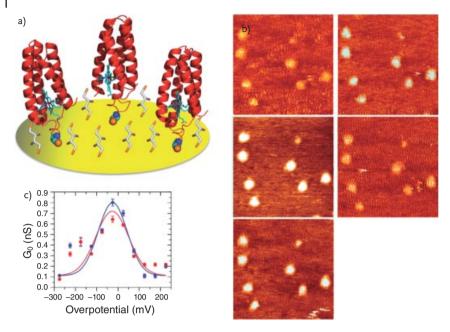


Figure 2.15 (a) D50C cytochrome b₅₆₂ mutant (c) Conductivity-overpotential correlation of linked directly to a Au(111) electrode surface via the inserted Cys residue [229]. (b) In situ STM images at different overpotentials.

the D50C cytochrome b_{562} mutant linked to the Au(111) electrode surface [229].

single-molecule in situ STM imaging, and single-molecule in situ STS with a conspicuous spectroscopic feature were obtained (Figure 2.15).

Figure 2.15a shows a schematic of the adsorbed D50C protein and Figure 2.15b the apparent height variation of a small assembly of adsorbed molecules with the electrochemical overpotential at constant bias voltage. The data are recast as a single-molecule normalized in situ STS plot in Figure 2.15c. A spectroscopic feature around the equilibrium redox potential with an "on-off" ratio of about five is observed (cf. the azurin case and a range of small molecules). By its gene technological diversity, this redox protein class thus offers a new type of singlemolecule in situ STM targets.

2.6.2.3 Cytochrome c4: A Prototype for Microscopic Electronic Mapping of Multicenter Redox Metalloproteins

Core functions in biological ET and redox enzyme function are commonly controlled by large metalloproteins with several transition metal centers. The photosynthetic reaction centers, redox protein complexes such as cytochrome c oxidase, the nitrogen-fixing enzyme nitrogenase, and other large redox enzyme complexes are examples. Intramolecular ET between the metal centers is a key operational function but mutual "cooperativity" among the centers is another key molecular function. This notion refers to the fact that ET to or from a given center affects the microscopic, as opposed to the macroscopic redox potentials and ET rate constants of all the other centers in overall cooperative charge transport. The number of electronic interactions is mostly prohibitive for microscopic mapping, but *two-center* metalloproteins offer the merits of representing prototype multicenter redox metalloproteins, at the same time with a simple enough electronic communication network such that complete microscopic thermodynamic and kinetic ET mapping is within reach. The bacterial respiratory two-heme protein *Pseudomonas stutzeri* cytochrome c_4 has been investigated most comprehensively in the context of cooperative intramolecular, and "gated" interfacial electrochemical ET. *Pseudomonas stutzeri* cytochrome c_4 has also acquired the recent status of a single-molecule *in situ* STM target [228].

Pseudomonas stutzeri cytochrome c₄ is organized in two globular domains, each with a single heme group, and connected with a 12-residue peptide chain (Figure 2.16). The protein is strongly dipolar, with excess positive and negative charge

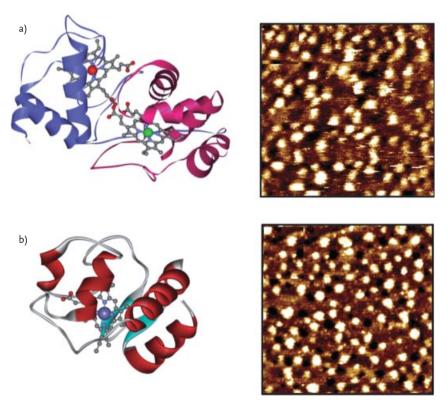


Figure 2.16 Three-dimensional structures with corresponding *in situ* STM images of molecules immobilized on Au(111) electrode surfaces modified by a mercaptodecanoic acid

SAM (negatively charged): (a) *P. stutzeri* cytochrome c₄ (PDB 1EPT); (b) HHC (PDB 1HRC) [228].

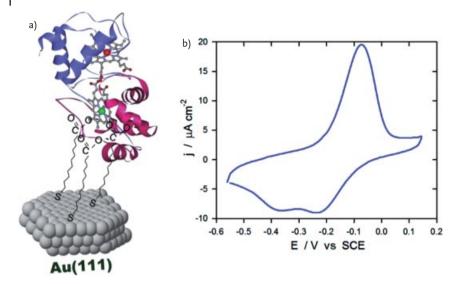


Figure 2.17 (a) *P. stutzeri* cytochrome c₄ immobilized on a Au(111) electrode modified by a mercaptodecanoic acid SAM. The positively charged high-potential domain is marked in red, the negatively charged low-potential domain in blue. (b) Cyclic

voltammogram of *P. stutzeri* cytochrome c₄. The asymmetric appearance reflects the orientation of the molecule and intramolecular ET between the heme groups as a key feature [228].

(pH = 7) on the C- and N-terminal domains, respectively. The excess charges are reflected in a redox potential difference of about $100\,\mathrm{mV}$, with the higher potential domain associated with the C-terminal domain and the lower potential with the N-terminal domain. These properties are crucial for electrostatic protein immobilization in specific orientations on SAM-modified Au(111) electrode surfaces and corresponding voltammetric patterns. Spectroscopic and other data are also available [231–233].

The heme groups are strongly hydrogen bonded via two propionates. The 19 Fe–Fe equilibrium distance implies that electronic contact is established but bimolecular ET with inorganic reaction partners discloses no evidence of intramolecular ET in time ranges up to 10– $100\,\mathrm{s}$. In contrast, the notably asymmetric cyclic voltammograms (Figure 2.17) indicate a pattern only compatible with fast intramolecular ET (10– $100\,\mu\mathrm{s}$) in the electrochemical two-ET process. Cytochrome c_4 is here oriented with the positively charged, high-potential C-terminal domain adjacent to the negatively charged SAM-modified Au(111) electrode surface used to record the data. This domain is reduced first in a cathodic scan followed by reduction of the remote N-terminal domain at the lower potential of this group, via intramolecular ET through the adjacent high-potential C-terminal domain. Reoxidation of the remote low-potential heme group in the anodic scan, however, only begins when the higher potential of the adjacent high-potential heme group is reached, again via fast intramolecular ET through the latter. Gated intramolecular

lar ET triggered by protein binding to the electrode surface thus appears as a new *P. stutzeri* cytochrome c₄ feature.

The mechanistic view involving upright protein orientation and fast intramolecular ET is strikingly supported by *in situ* STM imaging of the molecule directly on the electrochemical surface (Figure 2.16). The individual molecular structures correspond in size closely to those of a single cytochrome c₄ domain or of the single-heme HHC. This accords with upright rather than horizontal or recumbent cytochrome c₄ orientation. *In situ* STM has therefore provided both single-molecule structural and ET mechanistic support for the interfacial behavior of this protein. The strikingly different intramolecular ET rates of *P. stutzeri* cytochrome c₄ in free solute and surface-bound states suggest further that surface binding triggers an efficient intramolecular ET channel. The latter could involve breaking of the hydrogen bond between the two propionates and translational relocation of the heme groups towards more efficient direct electronic overlap between the heme groups.

2.6.2.4 Redox Metalloenzymes in Electrocatalytic Action Imaged at the Single-Molecule Level: Multicopper and Multiheme Nitrite Reductases

Imaging of redox metalloenzymes directly in enzyme action offers fascinating novel *in situ* STM perspectives. Such target molecules pose greater challenges than the smaller ET proteins due to their larger size, fragile nature, and more sophisticated conductivity mechanisms. Recent efforts have illuminated prospects and limitations of imaging single-molecule operational enzymes by *in situ* STM. *In situ* STM of single-molecule electrochemical enzyme dynamics follows more established, optically based stochastic single-molecule enzyme dynamics [234–236]. The–so far very few–*in situ* STM studies of single-molecule electrochemical surface enzyme dynamics illustrate the challenges of this area of single-molecule biological surface science.

The redox nitrite reductase enzymes are central in the bacterially controlled global biological nitrogen cycle where they catalyze the reduction of nitrite to lower oxidation states of nitrogen [237–242]. There are three classes of nitrite reductases: the copper nitrite reductases (CuNiRs), the multiheme-based class, and the twoheme cytochrome cd₁-type nitrite reductases [241, 242]. There are several rationales for the trimeric CuNiRs (each monomer molecular mass about 36kDa, here represented by Achromobacter xylosoxidans CuNiR) as single-molecule in situ STM targets. Each monomer contains a type I blue copper center for electron inlet, here from the substrate electrode, and a type II center for catalytic NO₂ reduction. The two centers are directly linked, offering facile intramolecular ET (cf. patterns of cytochrome c₄ in Section 2.6.2.3). The substrate nitrite is a small molecule structurally not detectable by in situ STM on the enzyme background. As frequently observed in enzyme voltammetry, binding of substrate induces, however, significant electronic changes in the enzyme, notably in the contact between the electron acceptance center and the electrode surface [217, 243]. This holds prospects for electronic mapping of the enzyme in action at the single-molecule level.

Achromobacter xylosoxidans CuNiR as a representative of the CuNiR class is electrocatalytically active on modified Au(111) electrode surfaces [217]. The

voltammetric patterns are controlled by subtle combinations of hydrophilic and hydrophobic surface properties of many surface linker molecules tested [243]. A notable outcome of these studies [217, 243] is that the enzyme on cysteamine- and benzylthiol-modified Au(111) electrode surfaces can be directly imaged in action at the level of the single molecule (Figure 2.13). Enzyme molecules even with the triangular crystallographic CuNiR substructure are observed. These structures *only* appear when nitrite is present. Single-molecule CuNiR thus offers a case for following electrochemical redox metalloenzyme activity at the single-molecule level.

Butt and coworkers studied the voltammetry of the (*Escherichia coli*) decaheme class of nitrite reductases [244] (Figure 2.13). Well-defined although unstable catalytic multi-electron voltammetric reduction of nitrite by the enzyme immobilized on bare Au(111) electrode surfaces is notable. The decaheme nitrite reductase is a second case for single-molecule *in situ* STM of a redox metalloenzyme, but image interpretation is presently not at the level for CuNiR [227]. Molecular-scale structures can be observed on the Au(111) electrode surface under conditions where the enzyme is electrocatalytically active, with both the natural dimer and surface-dissociated monomer enzyme structures identified. Molecular conductivities (*in situ* STM contrasts) of the enzyme and the active enzyme–substrate states are, however, not very distinctive.

2.6.2.5 Au-Nanoparticle Hybrids of Horse Heart Cytochrome c and *P. aeruginosa* Azurin

Inorganic particle, tube, and other structures of controlled size and shape have reached the size range of biomolecules such as proteins. This is also the size range of nanotechnology where electronic structures of the objects gradually transform from macroscopic to single-molecule behavior. The combination of inorganic metallic or semiconductor structures with comparable-size (bio)molecules into biological—inorganic hybrid structures is a novel core notion [245–256]. Nanoparticle- and nanowire-based electroanalytical chemistry and biological diagnostics have presently gone beyond proof of principle levels [245, 246]. We first address some issues of variable-size gold nanoparticles (AuNPs) at electrochemical surfaces. This is followed by a discussion of electrochemical properties and single-molecule *in situ* STM of AuNP–redox metalloprotein hybrids [230, 257].

AuNPs in Liquid-State Environment Solute pure and monolayer-coated ("capped") AuNPs are central targets in colloid and surface science also with a historical dimension [258–262]. Facile chemical syntheses introduced by Schmid *et al.* [260] and by Brust *et al.* [263] have boosted AuNP and other metal nanoparticle science towards characterization of the physical properties and use of these nanoscale metallic entities by multifarious techniques and in a variety of environments. Physical properties in focus have been the surface plasmon optical extinction band [264–269], scanning and transmission electron microscopy properties, and electrochemical properties of surface-immobilized coated AuNPs [173, 268–276]. To this can be added a variety of AuNP crosslinked molecular and biomolecular

structures with applications in (electro)analytical chemistry, biological diagnostics, and others [245, 246, 258, 259]. The smallest (≤1 nm) particles behave like a similar-sized molecule with a discrete electronic spectrum or a wide HOMO/ LUMO gap [277–279]. Intermediate-sized AuNPs, say 1.6 nm (Au₁₄₅) to 2.5 nm (already many hundreds of Au atoms), display electrochemically detectable Coulomb charging effects at room temperature [173, 272-275, 280-283] while the Coulomb energy spacings in larger AuNPs (≥3–5 nm) are too close for discretenessof-charge effects. Successive Coulomb charging is associated with the electrostatic charging energy increments [280]

$$E_{\text{el.stat.}} = \left(z - \frac{1}{2}\right) \frac{e^2}{C_{\text{NP}}}; \quad C_{\text{NP}} = \varepsilon_{\text{s}} R_{\text{NP}}$$
 (2.14)

where z is the number of electronic charges (e) already on the particle, C_{NP} the capacitance of a spherical particle of radius $R_{\rm NP}$, and $\varepsilon_{\rm s}$ the dielectric constant of the surrounding medium. The energy separation between successive charge states exceeds the thermal energy, k_BT , if

$$\frac{e^2}{C_{\rm NP}} > k_{\rm B}T; \quad R_{\rm NP} < \frac{e^2}{\varepsilon_{\rm s}k_{\rm B}T} \tag{2.15}$$

At room temperature this condition accords with $R_{\rm NP}$ < 33 nm in vacuum ($\varepsilon_{\rm s}$ = 1) and $R_{\rm NP}$ < 6–7 nm or $R_{\rm NP}$ < 0.5 nm if the short-range ($\varepsilon_{\rm s}$ = 5) and bulk static dielectric constant of water ($\varepsilon_s = 80$) applies, respectively. These radii should be corrected by the coating monolayer. AuNPs of 1.6 nm, Au₁₄₅, are observed to give electrochemical energy spacings of about 0.17 eV [173, 272-274, 283], corresponding to an "effective" dielectric constant of $\varepsilon_{\rm eff} \approx 13-20$.

Quantized electrochemical capacitive charging of variable-size AuNP monolayers with protective monolayer coatings in aqueous and organic solvents has been reported [272, 277-279, 283]. A recent study reported in situ STM and Coulomb charging of water-soluble protected 1.6 nm AuNPs by both differential pulse voltammetry (DPV) and electrochemical in situ STM under the same aqueous electrolyte and room temperature conditions (Figure 2.18) [173]. Multiple 0.15 V spaced peaks were observed on the anodic side of the potential of zero charge both in DPV and in situ STM current-overpotential correlation (at fixed bias voltage) (cf. Section 2.3). The peak heights corresponded to 1.1 and 0.9×10^{-18} F per particle from DPV and in situ STM, respectively. This discrepancy was addressed in [272, 273, 283]. A recent report by Wandlowski and associates [284] has followed up these observations and reported what appears to be single-ET charging in Au(111) electrode-ferrocene interfacial electrochemical ET. This could reflect ET via molecular-scale roughened Au surface structures caused by chemical interaction of the thiol-modified ferrocene [284]. Observations such as these raise the issue of whether a molecular-sized AuNP in these contexts is appropriately regarded as a ("quasi-")molecule or as a small metallic entity. The environmental dynamics would strongly affect the ET processes in the former case but be of minor importance in the latter. This issue is presently unsettled.

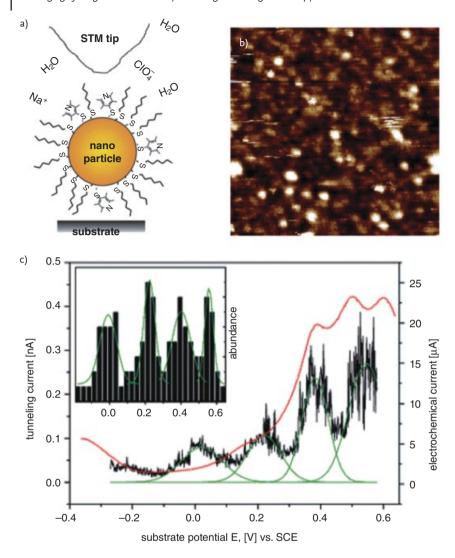


Figure 2.18 Electrochemical single-electron charging of coated 1.5 nm Au₁₄₅ nanoparticle on a Au(111) electrode surface. (a) Schematic view of the AuNP in the *in situ* STM environment. (b) *In situ* STM image. (c) Differential pulse voltammogram of AuNP

monolayer (red line) and *in situ* STM tunneling current—overpotential correlation (dark line). The green lines show Gaussian resolution but this is only to guide the eye. Inset: abundance histogram of the observed peak currents. For details, see [173].

Electrocatalysis by AuNPs and *In Situ* **STM of AuNP–Metalloprotein Hybrids** Protected AuNPs have been combined with redox metalloproteins [230, 257] into covalently, chemisorptively, or electrostatically linked hybrids of artificial supramolecular biostructures. AuNP–protein hybrids are important in electrocatalysis and bioelectrochemistry extending to large protein complexes such as cytochrome c oxidase. Biological recognition and diagnostics [245, 246] with links even to protein-based "nanocircuitry" are other areas [285]. We note here a few issues of the fundamental protein–nanoparticle interactions as reflected in interfacial electrochemical ET and *in situ* STM of protein–nanoparticle hybrids. These issues are illuminated by a recent study of a hybrid between a 3 nm coated AuNP linked to *P. aeruginosa* azurin [230]. An analogous study of a AuNP–cytochrome c hybrid was also reported [257].

Pseudomonas aeruginosa azurin (cf. Section 2.6.2.1) can be linked by strong hydrophobic forces to alkanethiol-protected 3 nm AuNPs in turn immobilized on a Au(111) electrode surface via an aromatic 4,4'-biphenyldithiol linker (Figure 2.19) [230]. Three observations of importance for AuNP-induced electrocatalysis emerge. The first is that the AuNP hydrophobically linked to azurin evokes an electrochemical ET rate enhancement of at least an order of magnitude compared to azurin alone on similar alkanethiol-modified surfaces. Notably, both AuNPenhanced and direct interfacial ET voltammetric peaks are apparent at high scan rates (Figure 2.19). The second observation is that a two-step, azurin-AuNP and AuNP-electrode ET mechanism accords with the data. Interfacial AuNP-electrode ET appears as virtually activationless compared to protein-AuNP ET. The third observation is that the dual voltammetric pattern accords with dual in situ STM contrasts of sub-monolayers of the AuNP-linked azurin molecules (Figure 2.19). The weaker contrast which also displayed contrast fluctuations in time was assigned to the AuNP-azurin hybrid, the stronger robust contrast to individual AuNPs or azurin molecules.

AuNPs inserted between the electrode surface and redox metalloproteins therefore both work as effective molecular linkers and exert efficient electrocatalysis. Recent considerations based on resonance tunneling between the electrode and the molecule via the AuNP as a mechanism for enhanced interfacial ET rates suggest that electronic spillover rather than energetic resonance is a likely origin of the effects (J. Kleis *et al.*, work in progress). Even slightly enhanced spillover compared with a planar Au(111) surface is enough to enhance the ET rate by the observed amount over a 10–15 Å ET distance.

2.7 Some Concluding Observations and Outlooks

Single-crystal, atomically planar electrode surfaces have paved the way for introducing the scanning probe microscopies of STM and AFM in the bioelectrochemical sciences. The powerful *in situ* STM technology has increased the structural resolution of (bio)electrochemical electrode surfaces to the molecular and

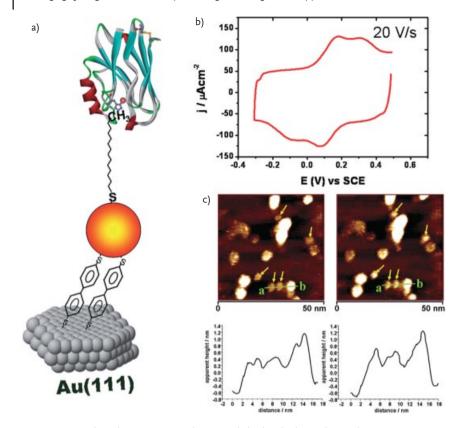


Figure 2.19 Cyclic voltammogram and *in situ* STM images of *P. aeruginosa* azurin–AuNP hybrid. (a) Schematic of the hybrid and of *P. aeruginosa azurin* alone on 4,4'-biphenyldithiolmodified Au(111) electrode. (b) Cyclic voltammogram showing the duality caused by the presence of both hybrid and azurin alone on the surface. The outer pair of voltammetric peaks represents azurin, the inner pair the

hybrid with electrochemical ET rate constants higher by up to two orders of magnitude. (c) In situ STM images of azurin–AuNP hybrid in the 4,4'-biphenyldithiol matrix with height profiles indicated. The weak and temporally fluctuating contrasts were assigned to the hybrid, the strong and robust contrasts either to azurin or to AuNPs [230].

sometimes sub-molecular levels. High-resolution images have been achieved for the building blocks of biological macromolecules, that is, amino acids and nucleo-bases, and for lipid monolayers as building blocks of membrane structures. All of these form highly ordered, two-dimensional monolayers on Au(111) or Pt(111) electrode surfaces. Dynamic surface phenomena such as phase transitions in the adlayers and the monolayer formation process can also be followed. The degree of image detail in both the individual adsorbate molecules and in their lateral organization holds clear perspectives for the understanding of the interaction of "biological liquids" broadly with solid surfaces. It is in fact notable that these large and fragile metalloproteins can now be controlled towards the level of resolution

of the single molecule and retain close to full enzyme functionality. This offers other technological perspectives for metalloenzyme-based biosensor function where functional units that respond to optical or magnetic signals can be inserted between the electrode and the reactive protein/enzyme.

We have overviewed cases of imaging of redox metalloproteins and metalloenzymes, of their amino acid building blocks, and of lipid membrane structures by in situ STM. As in situ STM is essentially based on molecular electronic conductivity, we have included some discussion of single-molecule electronic properties and interfacial ET processes. Focus has been on "natural" biomolecular aqueous solution environment and on the target molecules in monolayers on metallic electrode surfaces, where (bio) molecular function is controlled by the electrochemical potential. Redox metalloproteins have been identified as biomolecular target systems, but novel hybrid configurations where proteins are combined with metallic nanoparticles have also been addressed. Single-molecule resolution has been achieved under conditions where the molecules (i.e., particularly proteins) are fully active in ET or enzyme function. This opens the possibility that structural mapping of immobilized redox metalloproteins, DNA-based molecules, and membrane structures can be achieved. Given adequate theoretical support, ET and redox enzyme function as well as cooperative phenomena such as molecular monolayer and membrane formation can also be addressed, at the level of the single molecule. Pseudomonas aeruginosa azurin and cytochrome b₅₆₂ have illuminated these perspectives. These could be extended to exploit surface mapping of biological sensors and other bioelectrochemical systems with potential device function.

STM and in situ STM are theoretically demanding because the properties recorded, that is, electrical currents through molecules, do not translate directly into molecular shape or topography. Long-range off-resonance molecular electronic conductivity is broadly understood in terms of electron exchange and energy gaps of the atomic or molecular orbitals involved and energy broadening of the orbitals closest to the substrate and tip. The "percolation" of electron density through the protein structure in interfacial protein ET processes, however, still constitutes a challenge. Exponential distance dependence of the tunneling current or conductivity is broadly expected (Section 2.3). This is sometimes observed but unexpected very weak current attenuation emerges in other cases such as for single- and double-strand oligonucleotides linked to enclosing Au electrodes [146, 147]. Hopping of excess electrons or holes by temporary particle accommodation in intermediate molecular orbitals does not immediately account for this due to unfavorable energetics. Configurational and energetic fluctuational effects [145, 286], formation of, so far elusive, electronic surface states [287], and trapping in extended polaron states have been forwarded as possible physical origins [288, 289].

Electronic conductivity of molecules including redox metalloproteins with accessible low-lying redox states in nanogap electrode configurations or in situ STM displays quite different patterns. These are dominated by sequential two-step (or multiple-step) hopping through the redox center, induced both by potential variation and environmental configurational fluctuations. Both redox molecules and metalloproteins have been brought to display interfacial electrochemistry at the single-molecule level along these lines. Theoretical notions rest on theory of interfacial electrochemical ET, but the nanogap environments or association with metallic nanoparticles have revealed new ET phenomena. "Switching" or "negative differential resistance," quite different from electrochemical ET at single, semi-infinite electrode surfaces, is an immediate example. "Coherent" multi-ET in a single *in situ* STM event when the redox level is strongly coupled to the electrodes is another novel ET phenomenon. We have discussed these phenomena, and new nanoscale electrochemical and bioelectrochemical systems have been shown to accord with these views.

(Bio)molecular electronics, enzyme electrochemistry, oligonucleotide monolayer organization, and DNA-based biological screening towards the single-molecule level are attractive applied perspectives of the new bioelectrochemistry and *in situ* STM. Networks of hybrid biomolecular structures of biomolecules with nanoparticles and nanowires in electrochemical nanogaps and *in situ* STM could here be novel targets. From a biotechnological perspective, *fundamental* bioelectrochemical innovation including new interfacial ET phenomena and theoretical support remains, however, a prerequisite.

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3

Applications of Neutron Reflectivity in Bioelectrochemistry

Ian J. Burgess

3.1 Introduction

The assembly of biomimetic phospholipid layers on solid substrates is a means to create sensors and platforms to study the biophysics of membrane processes. If the solid substrate upon which the film is assembled is an electronic conductor then careful manipulation of the surface charge density allows the physical and chemical properties of the adsorbed organic film to be altered in a controlled fashion. Interest in biomimetic-based research has led to the need to extract more detailed information pertaining to adsorbed organic films and their electrical variable-dependent molecular structure. Electrochemical measurements alone are often insufficient to extract detailed information with molecular-scale resolution of the components of the adsorbed films. To achieve this level of understanding, electrochemists have long recognized that traditional experimental tools need to be coupled with in situ experimental methods. For example, infrared and Raman spectroscopy [1–3], scanning probe microscopy [4], surface X-ray scattering [5], and surface plasmon resonance [6] have all been adopted by electrochemists in the pursuit of finding new surface-sensitive means to probe the structure and composition of adsorbed organic films in situ.

The specular reflection of neutrons is a surface technique that, unlike X-ray methods, is sensitive to low atomic number elements and can distinguish between isotopes of the same element. This makes neutron reflectivity (NR) especially effective for the study of biologically relevant material due to the ubiquitous nature of carbon and hydrogen in these systems. However, until relatively recently the technique had evaded the attention of the electrochemical community. The reasons for this stem largely from the limited number of facilities where neutron scattering measurements can be made and the unfamiliarity of the technique to scientists outside the field of physics. The creation of dedicated research facilities and the encouragement of external, user-oriented beam time allocation have led to increasing interest in the field of slow and thermal neutron scattering. In the last decade, NR has begun to attract increasing attention from electrochemical researchers. The intent of this contribution is to provide an introduction to the field of NR

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studies of biomimetic thin organic films adsorbed on electrode surfaces both in theoretical and practical terms. The scope and context are addressed to future possible users of NR in the bioelectrochemistry community particularly those interested in studying biomimetic lipid layers supported on electrode surfaces. There exist several excellent reviews in the literature concerning the concepts of NR [7-11]; however, the starting level and the complexity of these papers makes them difficult reading for non-physicists. It is hoped that while being slightly less rigorous, this contribution will provide electrochemists with an accessible introduction to the theoretical and experimental applications of NR.

3.2 Theoretical Aspects of Neutron Scattering

3.2.1

Why Use Neutrons?

The purpose of a scattering-based experiment is to determine the spatial and/or temporal correlation of the atomic or molecular components of the object under study. If one wishes to extract spatial information pertaining to collections of atoms or molecules it is expected that the probe chosen for a scattering experiment must have wavelengths comparable in magnitude to the distance between the constituent units. Similarly, in order to study temporal correlations, the incident radiation should have comparable energies to those associated with atomic or molecular motions. Beyond these criteria, there are several other requirements of the probe which are particularly relevant to the study of thin organic films such as biomimetic membranes.

- The probe must be capable of penetrating condensed matter if the sample of interest is not at an accessible interface.
- 2) The probe should respond equally to light atoms as well as heavier ones.
- The probe should not deposit destructive levels of energy in the sample when absorbed.

Two types of scattering probes that have wavelengths small enough to provide information on the atomic or molecular length scale are electromagnetic radiation in the form of X-rays and de Broglie waves in the form of neutrons. It should be stated that although these two probes are often considered complementary, only neutrons satisfy all the above listed desired criteria. This is shown in Figure 3.1, where the energies of photons and neutrons of the same wavelength are displayed. In the range of wavelengths useful for studying interatomic or intramolecular spacings in condensed matter, photons have energy roughly five orders of magnitude higher than that of neutrons and are usually far too energetic to probe temporal correlation functions in condensed matter. In addition, upon absorption, the keV range of energy carried by a photon is sufficient to damage the material under study, particularly when soft materials are involved. Another distinction between

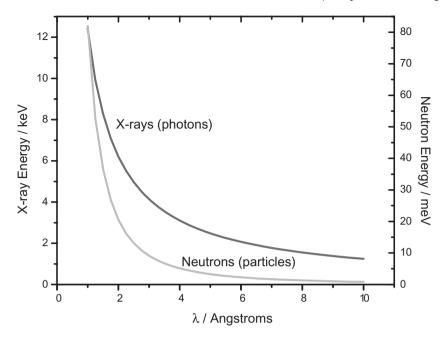


Figure 3.1 Kinetic energies of X-rays (photons) and neutrons as a function of wavelength.

neutrons and X-rays involves the fundamental difference between the interactions of these two forms of radiation with matter. X-rays are scattered by electron density, and thus the probability for a scattering event to occur is directly proportional to the atomic number of the matter causing the scattering. This means that light atoms such as hydrogen are barely visible to X-ray radiation. Whereas X-rays probe matter via an electromagnetic interaction, the next section describes how the primary interaction in a neutron–matter scattering event is via the strong nuclear force. Two important consequences follow from this fact. First, unlike X-rays, neutrons are strongly scattered from low atomic number atoms; second, a neutron has a different scattering potential with isotopes of the same element.

3.2.2 Scattering from a Single Nucleus

The essence of NR is the scattering of incident neutrons from the nuclei of the material being studied due to the strong nuclear force. As the wavelength of the incident neutron (about $1-10\,\text{Å}$) is much greater than the range of the strong nuclear force (about $1-10\times10^{-5}\,\text{Å}$) the result of a single scattering event can be envisioned as a spherical wave analogous to the scattering of water waves from a pole projecting from the surface of water. An analytical form of the scattered wave requires an adequate description of the strong nuclear potential energy function. For the moment, assume that the strong nuclear interaction potential for a given

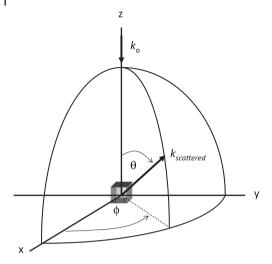


Figure 3.2 Coordinate system used to describe a single scattering event between a neutron of initial wave vector (k_0) and a bound nucleus positioned at the origin.

nucleus can be defined by a square well potential of known barrier height and interaction radius, R. The coordinate system, as shown in Figure 3.2, is defined such that the nucleus lies at the origin and the z-axis is along the direction of k_0 , the wave vector of the incident neutron. It is assumed that the nucleus is fixed at the origin and has no freedom to recoil from the collision with the impinging neutron. In the absence of any potential energy, the incident free neutron can be described by a traveling plane wave propagating along the z-coordinate:

$$\psi_{\rm in} = \exp(ik_0 z) \tag{3.1}$$

The wave scattered from the nucleus is spherically symmetric and the wavefunction of the scattered neutrons at any point *r* can be written in the form

$$\psi_{\rm sc} = -\frac{b}{r} \exp(ikr) \tag{3.2}$$

where b is a constant, independent of the angles θ and ϕ . The minus sign in Eq. (3.2) was originally chosen by Fermi and Marshall [12] so that a repulsive potential would correspond to a positive value of b. The quantity b is known as the scattering length and along with the closely related scattering length density parameter, ρ , is found ubiquitously in all forms of neutron scattering (see below). Strictly speaking the scattering length is a complex parameter. In physical terms, the imaginary contribution to the scattering length represents a measure of neutron capture via a resonance phenomenon and is highly dependent on the incident neutron's energy, whereas the real component is energy independent. The majority of nuclei have relatively small capture cross sections, and for these nuclei the scattering length can be treated as a real-valued quantity. There are some notable exceptions such as 103 Rh, 113 Cd, and 157 Gd which are strong neutron absorbers but the remain-

ing discussion will be limited to nuclei of the former description. As there is no proper theory to describe the strong nuclear interaction, it is impossible to predict values of b from other properties of the nucleus and consequently the scattering length parameters need to be determined experimentally. Empirically it is seen that the value of b depends on the spin state and the mass of the combined neutron-nucleus system. This opens up the field of isotopic substitution as a method of improving a great number of neutron scattering experiments.

The Fermi Pseudo Potential

It was Fermi who realized that it was possible to invoke an equivalent potential, which can be used to calculate the changes in the wavefunction outside the interaction by perturbation theory [13]. The unknown form of the strong nuclear interaction can be replaced by a new potential, which gives the same scattered wavefunction as the square well potential. In the derivation of Fermi's equivalent or pseudo potential [14] it is seen that the magnitude of the scattering potential depends on the scattering length of the nucleus and the mass of the neutron, m:

$$V_{\rm f}(r) = \frac{2\pi\hbar^2}{m}b\delta(r) \tag{3.3}$$

where the delta function is defined to be $\delta(r) = 1$ for r = b and $\delta(r) = 0$ for all other values of r. It should be clearly stated that the pseudo potential does not describe the true strong nuclear-nucleus interaction but rather is a mathematical construct that produces the correct form of the scattered wavefunction at large distances from the nucleus and depends entirely on the scattering length parameter of the nucleus from which the neutron scatters.

3.2.3

Scattering from a Collection of Nuclei

3.2.3.1 Neutron Scattering Cross Sections

If a collimated beam of neutrons (having initial energy E) is directed along the z-axis onto a target composed of a collection of nuclei then it is possible to measure the number of scattered neutrons in a given direction using a detector that is also capable of discriminating the energies, E', of the scattered neutrons. If Ω represents a solid angle along the direction defined by θ and ϕ and σ represents the total number of scattered neutrons in all directions then the partial differential scattering cross section is

$$\frac{\mathrm{d}^2\sigma}{\mathrm{d}\Omega\mathrm{d}E'} =$$

No. of neutrons per second scattered into $d\Omega$ with energy between E' and E' + dE' $\Phi d\Omega dE'$

(3.4)

where Φ is the flux of incident neutrons.

3.2.3.2 Coherent and Incoherent Scattering

In the multinuclear target system described above, the relative occurrence of scattering length b_i is defined to be f_i such that

$$\sum_{i} f_i = 1 \tag{3.5}$$

and the average value for the scattering length of the system is

$$\overline{b} = \sum_{i} f_i b_i \tag{3.6}$$

Note that the scattering length has been treated as a purely real-valued number in Eqs. (3.5) and (3.6). If the number of scattering centers in the target system is sufficiently large, then the partial differential cross section is the cross section averaged over all the scattering centers:

$$\frac{\mathrm{d}^2 \sigma}{\mathrm{d}\Omega \mathrm{d}E'} = \frac{k'}{k} \frac{1}{2\pi\hbar} \sum_{jj'} \overline{b_{j'} b_{j}} \int \langle j', j \rangle \exp(-\mathrm{i}\omega t) \mathrm{d}t$$
(3.7)

where $\omega = (E - E')/h$ and the matrix element $\langle j', j \rangle$ represents a complex function relating the positional vectors of the nuclei (j and j') before and after the scattering event. The derivation of Eq. (3.7) is given in [14].

Noting that

$$\overline{b_{j'}b_{j}} = (\overline{b})^{2}, \quad j \neq j'
\overline{b_{j'}b_{j}} = (\overline{b^{2}}), \quad j = j'$$
(3.8)

it is possible to rewrite Eq. (3.7) as follows:

$$\frac{\mathrm{d}^{2}\sigma}{\mathrm{d}\Omega\mathrm{d}E'} = \frac{k'}{k} \frac{1}{2\pi\hbar} (\overline{b})^{2} \sum_{jj',j\neq j'} \int \langle j',j \rangle \exp(-\mathrm{i}\omega t) \mathrm{d}t + \frac{k'}{k} \frac{1}{2\pi\hbar} (\overline{b^{2}}) \sum_{jj',j=j'} \int \langle j',j \rangle \exp(-\mathrm{i}\omega t) \mathrm{d}t$$
(3.9)

The first term in Eq. (3.9) defines what is known as the coherent scattering cross section whereas the second term defines the incoherent scattering cross section. It is seen that coherent scattering arises from the positional correlation of different nuclei at any given time. Physically, this means coherent scattering results in interference effects. In contrast, incoherent scattering depends only on the correlation between the positions of the same nucleus at different times. It does not give interference effects and its contribution to the scattered neutron intensity should be removed when attempting to perform structural studies.

3.2.3.3 Effective Potential and Scattering Length Density

According to Eq. (3.3), a neutron incident on a solid or liquid would see an array of δ -function potentials

$$V_f(r) = \frac{2\pi\hbar^2}{m} \sum_{i} b_i \delta(r - r_i)$$
(3.10)

where r_i is the position of the *i*th nucleus and the sum over *i* includes all nuclei in the system. Thus the total wavefunction of the neutron consists of the incident wavefunction plus the sum of all the spherical waves scattered by each nucleus. It is important to realize that the wave incident on each nucleus is not the same as the incident wave on the entire sample. A good discussion on treating the problem of multiple scattering has been given by Sears [15] but falls beyond the scope of this introduction to neutron scattering theory. It can be shown [16] that the effective potential for a neutron scattered by a collection of nuclei is simply the volume average of the Fermi potentials

$$V = \frac{2\pi\hbar^2 bN}{m} = \frac{2\pi\hbar^2 \rho}{m} \tag{3.11}$$

where b is the effective scattering length of the medium and N is the effective number density of scattering nuclei. The product bN is known as the scattering length density (SLD), ρ . Knowledge of the SLD provides a direct measure of the nuclear composition of the scattering medium.

3.2.4

Theoretical Expressions for Specular Reflectivity

3.2.4.1 The Continuum Limit

Consider the reflection of neutrons elastically scattered from the interface between vacuum (the incident medium) and a layer of condensed matter. The interface is chosen to lie in the xy plane and is assumed to be perfectly flat such that the only variation in the SLD occurs along the z-direction. The problem to address is the formulation of a description of the reflected wave as a function of its initial wave vector. The scattering is considered to be specular in the sense that the angle of the incident wave vector, $k_{\rm in}$, is equal to the angle of the reflected wave vector, $k_{\rm f}$. The momentum transfer wave vector Q is perpendicular to the interface, that is, the z-direction, and is given as the difference between the wave vectors of the outgoing and incoming plane waves:

$$Q_z = k_{\rm f,z} - k_{\rm in,z} \tag{3.12}$$

If Q_z is sufficiently small such that $2\pi/Q_z$ is much larger than interatomic distances in the condensed phase, then the medium can be treated as a continuum, though its nuclear density need be neither constant nor homogeneous. In vacuum, the total energy of the neutron outside the medium is simply its kinetic energy

$$E = \frac{\hbar^2 k_0^2}{2m} \tag{3.13}$$

In the continuum limit, the effective potential energy in the condensed phase is given by Eq. (3.11) and the total neutron energy is

$$E = \frac{\hbar^2 k^2}{2m} + \frac{2\pi\hbar^2 \rho}{m} \tag{3.14}$$

where k is the wave vector of the neutron inside the non-vacuum medium. By rearranging Eqs. (3.13) and (3.14) for the wave vectors k_0 and k, respectively, one can determine the refractive index in the direction perpendicular to the interfacial plane:

$$n(z) = \frac{k_z}{k_{0,z}} = \sqrt{1 - \frac{4\pi\rho(z)}{k_{0,z}^2}}$$
(3.15)

The one-dimensional wave equation describing the propagation of the neutron wave inside the medium arises from the time-independent Schrödinger equation

$$\frac{d^2\psi}{dz^2} + \frac{2m}{h^2} [E - V(z)] \psi = 0$$
 (3.16)

Substituting in Eqs. (3.11) and (3.15) yields the following differential equation:

$$\psi'' + k_z^2 \psi = 0 (3.17)$$

with the general solution

$$\psi(z) = a_1 \exp(ik_z z) + a_2 \exp(-ik_z z)$$
 (3.18)

where the coefficients a_1 and a_2 describe the amplitudes of the forward going and scattered wave, respectively. The reflection amplitude (r) and transmission amplitude (t) are defined in terms of the limiting forms of Eq. (3.18):

$$\psi(z) = \exp(ik_{0,z}z) + r\exp(-ik_{0,z}z)$$
(3.19)

$$\psi(z) = t \exp(ik_z z) \tag{3.20}$$

The procedure can be expanded in a piecewise fashion [17, 18] to obtain the reflection and transmission amplitudes arising from the reflection of neutrons from an arbitrary potential or SLD profile if the potential is divided into a discrete number (i) of rectangular lamellae. The reflection and transmission amplitudes are obtained from a pair of simultaneous equations which, when written in matrix notation, define the transfer matrix:

where $\theta_i(Q) = (Q/2)n_id_i$ and $n(Q) = \sqrt{1 - 16\pi\rho/Q^2}$ for the fronting medium (n_i) , the backing medium (n_i) , and the interstitial layers (n_i) . The transmission and reflection amplitudes are seen to be complex numbers and consequently the transmissivity and reflectivity are defined as $t * t = |t|^2 = T$ and $r * r = |r|^2 = R$, respectively. In a NR experiment, it is the complex conjugate term that is measurable and all information concerning the imaginary component (or phase) of the reflectivity is lost. It is apparent from Eq. (3.21) that the reflectivity for any known SLD profile can be calculated directly. However, as will be discussed later, the converse of this statement is not necessarily true.

3.2.4.2 The Kinematic Approach

The previous derivation of the reflection and transmission coefficients correctly describes the intensity of reflected neutrons at any value of momentum transfer vector. However, there is a useful alternative derivation, which gives a highly analytical function describing the reflectivity. This derivation is based on the Born approximation and is often referred to as reflectivity in the kinematic limit. Suppose there are two arbitrary but different SLD profiles $\rho_1(z)$ and $\rho_2(z)$ and one wishes to determine the separate reflectivities $R_1(Q_z)$ and $R_2(Q_z)$ for the two scattering potentials. The solution to the problem is described by combining Eqs. (3.15) and (3.17)

$$\psi_j'' + (k_0^2 - 4\pi\rho_j)\psi_j = 0, \quad j = 1, 2$$
 (3.22)

from which the Wronskian function

$$W(z) \equiv W[\psi_1(z), \psi_2(z)] = \psi_1(z)\psi_2'(z) - \psi_2(z)\psi_1'(z)$$
(3.23)

can be constructed. Differentiating both sides of Eq. (3.23) with respect to z and applying Eq. (3.22) yields

$$W'(z) = -\psi_1(z)\psi_2(z)4\pi[\rho_1(z) - \rho_2(z)]$$
(3.24)

Inspection of Eq. (3.24) reveals that the original Wronskian function, W(z), must be a constant whenever $\rho_1 - \rho_2 = 0$ (i.e., whenever the two SLD profiles are equal). If the argument is extended to a real situation, where the two different SLD profiles share a common fronting medium and a common backing medium, then it is evident that $\rho_1 - \rho_2 = 0$ for $z < \tau_1$ and $\rho_1 - \rho_2 = 0$ for $z > \tau_2$, where τ_1 and τ_2 mark the end of the fronting medium and the beginning of the backing medium, respectively. For simplicity the derivation continues for a vacuum fronting $(\rho_1 = \rho_2 = 0)$ although this restriction is not necessary. The wavefunctions for $z < \tau_1$ can be written

$$\psi_j(z) = \exp(ik_0z) + r_j \exp(-ik_0z), \quad j = 1, 2$$
 (3.25)

where r_i is the reflection coefficient for each SLD profile. As the two profiles share a common backing such that $\rho_1 = \rho_2 = \rho_{\text{backing}}$ for $z > \tau_2$, then the wavefunctions in this region are

$$\psi_{i}(z) = t_{i} \exp(iKz), \quad j = 1, 2$$
 (3.26)

where t_i is the transmission coefficient for each problem and K is the wave vector after the transmission through the scattering system:

$$K^2 = k_0^2 - 4\pi \rho_{\text{backing}} \tag{3.27}$$

By substituting Eq. (3.26) into the Wronskian function (Eq. (3.23)) one obtains

$$W(z) = 0; \quad z \ge \tau_2 \tag{3.28}$$

which arises from the fact that the wavefunctions are linearly dependent for $z \ge \tau_2$. However, substitution of Eq. (3.25) into Eq. (3.23) leads to

$$W(z) = 2ik_0[r_1 - r_2]; \quad z \le \tau_1 \tag{3.29}$$

which is a complex constant and applies for all $z < \tau_1$. Finally, for $\tau_1 < z < \tau_2$ integration of Eq. (3.24) yields

$$\int_{\tau_1}^{\tau_2} W' dz = W(\tau_2) - W(\tau_1) = -\int_{\tau_1}^{\tau_2} \psi_1(z) \psi_2(z) 4\pi [\rho_1(z) - \rho_2(z)] dz$$
 (3.30)

and as $W(\tau_2) = 0$ (via Eq. (3.28)) and $W(\tau_1) = 2ik_0[r_1 - r_2]$ (via Eq. (3.29)), Eq. (3.30) can be rewritten as

$$r_1 = r_2 + \frac{1}{iQ_z} \int_{\tau_1}^{\tau_2} \psi_1(z) \psi_2(z) 4\pi [\rho_1(z) - \rho_2(z)] dz$$
(3.31)

where Q_z is equal to $2k_0$, as k_0 has been previously defined to be the *z*-component of the incident wave vector. By choosing to set $\rho_2(z) = 0$ for all values of z, then $\psi_2 = \exp(ik_0z)$ and $r_2 = 0$. Accordingly,

$$r_1 = \frac{4\pi}{\mathrm{i}Q_z} \int_{-\infty}^{\infty} \psi(z) \exp(\mathrm{i}k_0 z) \rho(z) dz$$
(3.32)

where the subscript has been omitted and the integration has been formally extended over all values of z even though it is evident that the only contribution to the integral exists for the interval $\tau_1 < z < \tau_2$. Note that Eq. (3.32) is written in terms of the wavefunction $\psi(z)$ that describes the neutron *inside* the scattering medium. From Eq. (3.32) the expression for the reflectivity of the neutrons is

$$R = |r|^2 = \frac{16\pi^2}{Q_z^2} \left| \int_{-\infty}^{\infty} \psi(z) \exp(ik_0 z) \rho(z) dz \right|^2$$
 (3.33)

The difficulty in applying Eq. (3.33) lies in finding an appropriate expression for $\psi(z)$. An important simplification which has been widely applied in X-ray and neutron reflectivity is obtained if $\psi(z)$ is replaced with the free space wavefunction $\exp(ik_0z)$. This is known as the Born approximation and provides

$$R = \frac{16\pi^2}{Q_z^2} \left| \int_{-\infty}^{\infty} \exp(iQ_z z) \rho(z) dz \right|^2$$
 (3.34)

This equation is commonly referred to as the "kinematic" expression and is valid whenever the perturbation of the incident free space wavefunction by the scattering medium is sufficiently small. A highly reflective scattering event greatly perturbs the incident wavefunction and consequently the kinematic approximation is inadequate for such interactions.

To evaluate the adequacy of the kinematic approximation one can compare the reflectivity as calculated via Eq. (3.34) and compare it to the output from Eq. (3.21) for a simple SLD profile. The SLD profile for a smooth interface between air and quartz is shown in Figure 3.3a where the function $\rho(z)$ is seen to be a Heaviside function

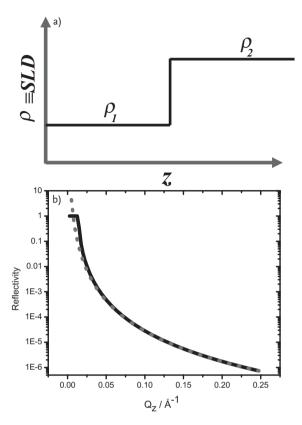


Figure 3.3 (a) SLD profile for a smooth interface between air and a single-crystal substrate. As the interface is stated to be infinitely sharp, the function $\rho(z)$ is a

Heaviside function. (b) Comparison of reflectivity curves for the SLD profile using the matrix method (solid curve) and the kinematic approximation (dotted curve).

$$\rho(z) = \begin{cases} \rho_{\text{air}} & z < 0 \\ \rho_{\text{ouartz}} & z > 0 \end{cases}$$
(3.35)

Substitution of Eq. (3.35) into Eq. (3.34) yields the Fourier integral for the Heaviside function H(z)

$$\int_{-\infty}^{\infty} \exp(iQ_z z) H(z) dz = \left(\pi \delta(Q_z) + \frac{1}{Q_z}\right) \Delta \rho$$
(3.36)

where $\delta(Q_z)$ is the Dirac delta function, which is equal to zero for all values of Q_z except for $Q_z = 0$ where $\delta(Q_z) = 1$. From Eq. (3.36), the reflectivity for any Q_z greater than zero is

$$R = \frac{16\pi^2}{Q_z^4} (\rho_{\text{quartz}} - \rho_{\text{air}})^2$$
 (3.37)

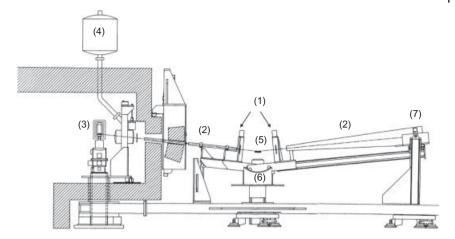
Figure 3.3b plots $R(Q_z)$ calculated from Eq. (3.37) and $R(Q_z)$ calculated from Eq. (3.21). At values of Q_z greater than about 0.03 Å⁻¹ the two curves almost perfectly coincide. For smaller momentum transfer vectors the kinematic approximation significantly diverges from the correct reflectivity for reasons explained above. To reduce this discrepancy a distorted wave Born approximation (DWBA) has been applied to high-reflection domains of Q space. As the DWBA only modifies the low-Q range of the reflectivity curve it does not change the analytical usefulness of the kinematic formalism. Therefore, the DWBA will not be discussed herein but the interested reader is directed elsewhere for details [19, 20].

3.3 **Experimental Aspects**

Four main aspects are addressed in turn: (i) neutron sources and reflectometer operation; (ii) choice of substrate, substrate preparation, and characterization; (iii) cell design and assembly; and (iv) data acquisition and analysis.

3.3.1 **Experimental Aspects of Reflectometer Operation**

Neutron intensities sufficiently large to perform scattering experiments are only available at large-scale research facilities. These facilities are either reactor- or spallation-based neutron sources. A recent review provides a comprehensive description of new neutron sources with particular emphasis on instruments dedicated for biological studies [21]. Reactor-sourced reflectometers such as the NG-7 and NG-1 instruments at the NIST Centre for Neutron Research (NCNR) at the National Institute of Standards and Technology operate at a fixed neutron wavelength and usually rely on variation of the incident angle to access momentum transfer space. Time-of-flight instruments are usually found at spallation sources such as the SPEAR reflectometer at the Los Alamos Neutron Science Centre (LANSCE) and make use of pulsed polychromatic neutron radiation. The simultaneous acquisition of a wide range of Q_z has been shown to be very advantageous for studying kinetic processes in thin films [22-25]. The horizontal NG-7 reflectometer at NCNR is shown schematically in Figure 3.4. The intensity of neutrons is determined using two ³He scintillators, one placed before the sample (denoted the monitor) and one placed after the sample (the detector). In some instances a multichannel detector can be used to collect off-specular as well as specular reflected neutrons. In the simplest applications the off-specular signal can be used for background correction but this signal also carries information about the SLD profile in the plane of the substrate and has been used to measure in-plane order [26-28] but will not be discussed further. The ratio of the specular detector count to the monitor count gives the measured specular reflectivity. As the wavelength is fixed for reactor-source neutrons, the momentum transfer vector is systematically changed by varying the angle of the incident (and, consequently,



reflectometer used for NR studies at NCNR. 1. Collimating slits-these are located on either side of the sample. A beam monitor on the precollimating slit allows the primary detector to be scaled to absolute reflectivites. 2. Pre- and post-sample flights-both have LiF slits for collimation of the incident or reflected beam. Flight tubes have $2\theta_c$ (58Ni) supermirrors and are evacuated to minimize losses. 3. Tilting pyrolitic graphite monochromator-provides four possible wavelengths: 2.35,

Figure 3.4 Schematic of the NG-7, horizontal, 4.10, 4.75, and 5.50 Å. 4. Thermal neutron filter-a liquid nitrogen-cooled beryllium filter is placed after the monochromator to remove thermal neutrons. 5. Sample stage-available space is approximately 30 cm between the collimating slits. 6. Translation stage-computer-driven stepper motors provide precise alignment of the sample with the beam. 7. Detector-a 2.5 cm diameter, cylindrical ³He proportional counter is used as the detector.

the specularly reflected) neutrons ($Q_z = 4\pi \sin \theta/\lambda$). This requires concerted movement of the main sample elevator, the pre- and post-sample flights, and the collimating slits. Slits are required to minimize the angular divergence in the scattering plane of the incident neutrons. Angular distribution leads to multiple angles of incidence, and consequently a distribution of momentum transfer vectors rather than a single Q_z value. The angular divergence in Q_z space defines the instrumental resolution dQ_z/Q_z of the reflectometer. When the neutron source is monochromatic (as opposed to a "white" source used with a time-of-flight instrument) the slit apertures are increased with increasing incident angle in order to maintain a constant instrumental resolution. If the substrate supporting the film of interest is curved rather than perfectly flat then the effective angular divergence of the incident beam will be increased leading to reduced instrumental resolution. Thus low curvature, also referred to as bow or warp, is an additional mandatory feature for a suitable substrate (see Section 3.3.2). In a typical experiment the angle of incidence can be varied such that the Qz range can extend beyond about 0.5 Å⁻¹. However, for most experiments the loss of measurable signal above background at $R \approx 10^{-8} - 10^{-7}$ limits $Q_{z,\text{max}}$ to the range of about 0.20 to about

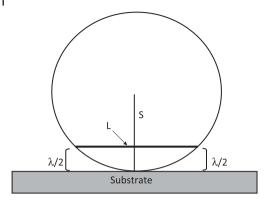


Figure 3.5 A spherical wave of radius *S*, impinging on the interface of interest. The line of length *L* lies parallel to the substrate and defines the length scale where the distance between the wave front and the

substrate remains less than or equal to half the wavelength of the incident neutron. This defines the lateral coherence length of the incident neutron.

 $0.25\,\text{Å}^{-1}$. To improve the statistical quality of data, typical counting times can exceed 8–10 h which requires that systems being studied are sufficiently stable.

The neutron beam incident upon the sample is an incoherent collection of individual neutrons whose lateral coherence length is determined by the method of preparation (how it was emitted by the fissile material, its collision with atoms in the cold source, diffraction by the collimating slits, etc.). The lateral coherence length is defined as the spatial extent over which an incident neutron wave front has a constant phase in its interaction with the sample. Geometrically, this is only achieved over the dimension of the wave front which can be considered sufficiently flat. A simplified explanation follows. Imagine that a thermal neutron is generated in a reactor core and its final interaction before proceeding along the guide tube was a collision with a single hydrogen nucleus in the cold source moderator. The scattered wave from this interaction is spherical with a wavelength λ . Subsequently, after traveling along the guide tube, it reaches the sample at a distance S away from the hydrogen atom of origin. Referring to Figure 3.5, a line segment can be inscribed in the sphere, which is parallel to the interface, but whose terminal points are still a distance $\lambda/2$ from the interface. The length L of such a line is given by

$$\left(\frac{L}{2}\right)^2 = \lambda \left(S - \frac{\lambda^2}{4}\right) \tag{3.38}$$

The end points define the lateral distance in the sample plane across which a given spherical wave front becomes out of phase by π degrees. For a spherical wavelength of 5 Å emanating from a source 5 m away, the length L is of the order of 100 μ m. In practice, the lateral coherence length depends on the energy spread and the wave vector spread of the incident neutron wave front. If the instrumental

resolution is kept constant during the scan this means that the lateral coherence length varies with Q_z but in all instances it is a macroscopic distance of the order of micrometers to hundreds of micrometers. The finite coherence length can influence the interpretation of NR data. There are two limiting cases to be addressed. If the lateral inhomogeneities, or deviations from perfect flatness, are on length scales significantly smaller than the coherence length of the incident neutron, this scale of roughness gives rise to diffuse reflection, but as long as the contribution to the total reflection by this off-specular component is significantly smaller than the specular signal then a one-dimensional approach as described in Section 3.2 remains valid. The in-plane roughness will be manifest in the SLD profile by slabs of thickness dz which represent average values over the area that the incident neutron wave is effectively coherent or in-phase. In the other limiting scenario, the in-plane inhomogeneities are on a length scale much greater than the incident neutron coherence length scale. In such a case the observed reflectivity will be proportional to a sum of independent, area-weighted reflected intensities for each different region. The analysis of data of this type is very complex and usually cannot be successfully treated with simple one-dimensional approaches.

3.3.2 Substrate Preparation and Characterization

There are several parameters that dictate which materials are viable substrates for NR studies of the solid-liquid interface. If the momentum transfer vector is being varied by adjustment of angle, neutron beam footprints are of the order of several square centimeters at lowest grazing angles. At larger angles (higher Q_z), the dramatic decrease in reflectivity requires wider slits (maintaining comparable sample footprint areas) to maintain a measurable intensity of reflected neutrons. Generally speaking, a large-dimension surface is required for NR experiments due to the comparatively low incident neutron flux. For example, the neutron flux on the NG-1 beamline at the NCNR is $1-5 \times 10^4$ neutrons cm⁻² s⁻¹ [29] compared to the X-ray flux of roughly 10¹⁷ photons cm⁻² s⁻¹ at the Brookhaven National Light Source Synchrotron [30]. New and proposed next-generation neutron sources have increased time-averaged fluxes to more than 1013 neutrons cm-2 s-1 [21].

Apart from the special case of large losses due to incoherent scattering from proton-containing materials such as water, the main loss of neutron intensity in bulk media is due to scattering from diffraction. It is impossible to avoid large diffraction losses if the neutron beam passes through an appreciable thickness of liquid. For supported films in aqueous electrolytes, the incoherent scattering of neutrons by water also greatly attenuates intensity such that experiments are almost always run in an inverted configuration. In this mode, neutrons are incident on the interface through the supporting medium rather than through the electrolyte. As the neutrons must travel through a sizeable path length in the supporting material (on the centimeter scale), one may also expect high losses due to multiple diffraction processes. This can be avoided if the substrate is an appropriately oriented single crystal and/or suitable neutron wavelengths are chosen that

do not lead to Bragg diffraction. These arguments lead to the criteria that the substrate material should be single-crystalline and transparent to neutrons. In principle a wide range of solid crystals would be suitable, but in practice there are few materials that are available as large single crystals (typical dimensions of suitable quartz blocks are of the order of $10\,\mathrm{cm} \times 5\,\mathrm{cm} \times 1\,\mathrm{cm}$ thickness) for an accessible price. Three readily available substrates suitable for neutron reflectivity are silicon, quartz (SiO₂), and sapphire (Al₂O₃). Fortunately, it is possible to chemically modify these surfaces to change their surface properties or one can coat them with thin metallic films in order to use them as electrodes.

Proper characterization of the substrate is imperative when data analysis is based upon fitting analyses (see Section 3.3.4). As most neutron facilities have ready access to off-line X-ray reflectivity infrastructure, it is possible to characterize the roughness of the surface prior to any chemical and/or physical modifications. The typical roughness of highly polished substrates such as silicon and quartz should be less than 5 Å (root-mean-square). A very smooth substrate is critical for reflectivity experiments because roughness greatly attenuates specular reflection causing the measured reflectivity curves to drop to background levels at prematurely low momentum transfer vectors. For a detailed description of the effect of roughness on specular reflectivity, see [8]. To convert single-crystal substrates into electrodes, thin layers of chromium and gold can be sequentially sputtered on the substrates. A chromium layer has to be present in order to ensure adhesion of gold to quartz. Other means to achieve the adherence of smooth gold films on inorganic substrates include chemically modifying quartz with mercaptopropylsiloxane [31] and this has been successfully used to adhere gold onto quartz substrates for electrochemical NR studies [32]. Before using chemically or physically modified substrates it is imperative to once again characterize them in air with neutron and/or X-ray reflectivity. These pre-measurements provide valuable information on the roughness and SLDs of chromium and gold layers. Cyclic voltammetry (CV) curves recorded on these thin-film gold electrodes often resemble CV profiles recorded at the Au(111) surface suggesting that the sputtered gold layer is usually preferentially (111) oriented. Substrates can be reused by stripping the chromium and gold layers in a heated mixture of H₂SO₄ and HNO₃ (1:1 ratio); however, it is desirable to measure a final X-ray reflectivity curve after electrochemical NR experiments and before stripping. Experimentally, it has been observed that the gold and chromium layers can sometimes be affected structurally when a negative potential is applied to the substrate. In general, this potentialinduced alteration of modified quartz substrates has a much greater rate of occurrence when the quartz block is being reused (i.e., it had been previously coated with chromium and gold). It is hypothesized that the stripping procedure may not lead to complete removal of the old metal films and the reused quartz substrate may leave patches of chromium oxide on its surface when it is recoated. The chromium oxide is then reduced when the electrode is biased negatively and consequently the observed parameters for the substrate differ for the before and after measurements. Figure 3.6 shows a comparison of the before and after reflectivity measurements for a virgin coated quartz substrate and a reused substrate. Figure

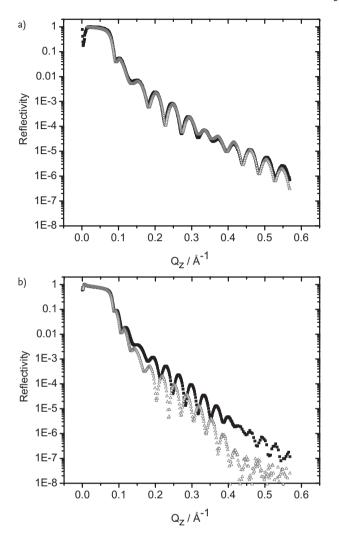


Figure 3.6 Experimental X-ray reflectivity curves in air for gold- and chromium-modified quartz substrates measured before (filled squares) and after (open triangles) electrochemical experiments. (a) Nearly identical

curves for a virgin-coated substrate indicative of a metal layer that is not perturbed by the potential bias. (b) Recycled substrates lead to metal layers that are significantly altered by cathodic biases.

3.6a shows that the X-ray reflectivity curves for a pristine quartz crystal coated with chromium and gold measured before and after electrochemical experiments are almost superimposable. For comparison, Figure 3.6b plots the X-ray reflectivity curves, measured before and after electrochemical experiments, for a quartz block recoated with chromium and gold. Note that the "before" curve in Figure 3.6b is marked by a pronounced loss in the oscillation amplitude at momentum transfer

vectors larger than $0.45\,\text{Å}^{-1}$ which is an indication of a rough interface. The curve for the same crystal after electrochemical experiments shows marked changes and indicates that the structure of the gold and chromium films has been altered. In such a case, the in situ NR data cannot be properly analyzed and experiments need to be repeated on a stable substrate.

Thin silicon wafers appear to be an attractive alternative to thicker quartz blocks for several reasons. Due to their prominence in the semiconductor industry, largediameter, thin (500 µm) silicon wafers are commercially available from a large number of sources. Wafers, 10cm in diameter, easily meeting the demanding flatness criteria required for NR experiments, can be purchased for as little as US\$15-20 per wafer, which is roughly 20 times lower in price compared to polished quartz blocks. Such silicon wafers could be treated as disposable samples whereby each wafer is coated only once and not stripped and recoated. An additional advantage of circular silicon wafers over rectangular quartz blocks is an increased neutron footprint area. The larger the area illuminated by the incident neutron beam the better the signal-to-noise ratio for a given count time duration. As a proviso it should be stated that this area advantage only occurs if the illuminated surface is uniformly coated. Quartz blocks have a maximum working surface of approximately 32 cm² compared to 79 cm² in the case of a 10 cm diameter silicon wafer. In theory, switching from quartz blocks to silicon wafers could result in a nearly 2.5 times increase in the surface area and, as the signal-to-noise ratio is proportional to the square root of the area, a more than 50% increase in signal-tonoise ratio could be achieved. Disadvantages to using silicon wafers include the presence of the native oxide film inherently found on all untreated silicon surfaces. This oxide film is usually between 10 and 50Å thick depending on the precise method used to grow the single-crystal wafer [33]. For X-rays, the difference in the SLD of silicon and its oxide is very small, but for neutrons the SLD difference is more appreciable (3.5 \times 10⁻⁶ Å⁻² for the oxide versus 2.1 \times 10⁻⁶ Å⁻² for silicon). As a result, the native oxide film on silicon contributes to the NR response in an electrochemical experiment and the measured reflectivity would have contributions from three layers (SiO2, chromium, gold) from the substrate alone. Fortunately, the oxide layer is very stable to electrochemical perturbation and should not be influenced when a potential is applied to the modified substrate. Most commercially available silicon is doped with group III or group V elements. As long as the level of doping is not extremely high, then the scattering from elements such as antimony and phosphorus is insignificant. However, silicon doped with boron should be avoided due to its strong scattering cross section.

3.3.3 Cell Design and Assembly

Like many other in situ coupled techniques, the ideal experimental conditions for the collection of specularly reflected neutrons are not ideal for maintaining electrochemical control. This mandates the improvisation of new cell designs. As detailed above, measurements performed in the presence of electrolyte should use an inverted cell geometry. The cell needs to be constructed so that the neutrons are incident on a transparent substrate, which has been modified in such a way to fashion a working electrode upon which the films of interest can be formed. The major difficulty in cell design for in situ electrochemistry-NR studies is that the two techniques have conflicting requirements. An optimal electrochemical cell for studying thin-film adsorption would have the following characteristics:

- A small, ideally polarized working electrode.
- The working electrode should be constructed out of a pure material, ideally a single crystal.
- 3) The reference electrode should be electrically connected to the cell via high ionic conductivity.
- A large-surface-area counter electrode.
- 5) An inert atmosphere to remove faradaic current arising from dissolved oxygen in the electrolyte.

On the other hand, an ideal cell for studying NR would have the following features:

- A large-area working electrode to increase the number of reflected neutrons.
- The working electrode should be derived from a material optically transparent to neutrons (e.g., quartz, silicon, sapphire).
- The amount of solvent in the cell should be minimized to reduce incoherent scattering and lower the background level.
- 4) A quiescent solution to minimize incoherent scattering with no void volume or trapped air bubbles.

Comparing the two lists, the greatest challenge in cell design is the electrolyte volume. From a NR perspective, a great enhancement in the signal above background is achieved by reducing the thickness of the electrolyte to less than 1 mm. However, too thin a layer of electrolyte provides large ohmic impedance and may have disastrous electrochemical consequences if it completely prevents conductivity between the working and reference electrodes. Often a thin film of electrolyte leads to the formation of bubbles in the cell. Achieving an inert atmosphere for long-duration NR experiments can be problematic. Electrolyte introduced into the cell can be pre-purged with argon but once the cell is filled there is no headspace, preventing an inert atmosphere from being maintained over the electrolyte. CV curves obtained immediately after filling a cell typically show very little oxygen contamination but over the course of 30 min or so a progressively larger oxygen reduction wave is often observed in the CV curves as oxygen permeates the Teflon components of the cell. Depending on the electrochemical application and techniques employed, the presence of oxygen in the cell may not greatly affect the experiment.

Two types of cells designed and used for inverted-geometry electrochemistry-NR experiments are shown in Figure 3.7. Figure 3.7a shows a large-cavity cell used in early applications of NR in bioelectrochemistry. The advantages of such a cell are its ease of assembly, reliable electrical contact to the working and counter electrodes, and its overall electrochemical stability due to the large electrolyte reservoir. The disadvantages of this cell are the large amounts of Teflon and

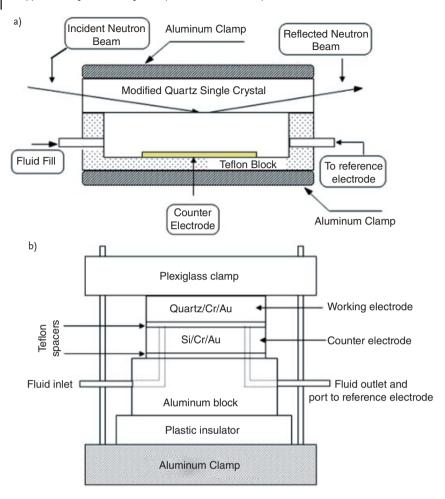


Figure 3.7 Schematics of two electrochemical cell configurations used in NR experiments: (a) larger electrolyte cavity design; (b) thin electrolyte layer configuration. Figure adapted from [79, 84].

aqueous solvent which yield large levels of incoherent scattering. This cell configuration is limited to only being capable of measuring reflectivities greater than 10^{-6} . Figure 3.7b shows a thin-layer cell which extends the lower limit of measurable reflectivities by almost an order of magnitude due to the removal of the Teflon and the reduction of incoherent scattering from the aqueous solvent.

3.3.4 Data Acquisition and Analysis

In measured neutron reflectivity curves, the error points in the experimentally measured data represent the statistical errors in the measurements (standard

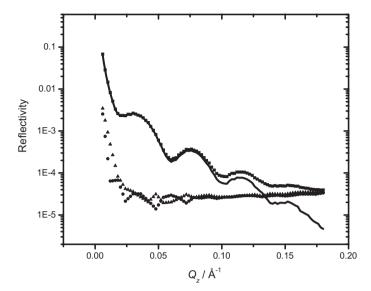


Figure 3.8 Influence of the background correction on the specular reflectivity data. The uncorrected specular data are shown as the square data points, while the triangles and circles show the background reflectivity

determined with two off-specular measurements. The solid line is the result of subtracting the averaged, off-specular data from the specular measurement.

deviation, σ). The standard deviation scales inversely with the square root of the number of neutron counts and consequently lower reflectivities carry a larger degree of uncertainty than high reflectivities. Data reduction requires accounting for neutron beam transmission losses through the substrate and correction for the background (off-specular) signal. In the absence of a multichannel detector, the background measurement can be made by deliberately misaligning the crystal so that the off-specular reflection is measured on both sides of the specular angle. The averaged off-specular signal is then subtracted from the measured specular signal. To illustrate the background correction procedure, Figure 3.8 shows a representative reflectivity curve, two background curves, and the background-corrected curve. The SLD profile is then usually determined by fitting a model to the measured reflectivity data.

A more detailed description of the analysis of neutron reflectivity data is provided with the aid of hypothetical examples. The reflection of neutrons from a multilaminar interface can be calculated in an analogous fashion to the optical matrix method for electromagnetic radiation [34, 35], provided the SLD, roughness, and thickness of each layer are known. For neutrons, the refractive index of a given phase may be calculated from the knowledge of the SLD, ρ , using Eq. (3.15). The dependence of the SLD on the composition of a given phase is described by the formula

$$\rho = \sum_{j} b_{j} n_{j} \tag{3.39}$$

where b_j is the scattering length and n_j is the number density of atomic species j. Neutron scattering lengths are empirically determined and the values for elements and their isotopes are tabulated.

When the SLD profile of an interface is known, a matrix method or a recursion method can easily be used to calculate reflectivity curves. However, for pedagogical purposes the relationship between the reflectivity and the structure of the interface is better revealed by the analytical expressions derived with the help of the kinematic approximation. The kinematic approximation has been shown to describe the reflection of neutrons from stratified media very well when the reflectivity is significantly less than unity. When a film of SLD ρ_1 and thickness τ_1 is sandwiched between two phases of identical SLD ρ , the expression for the reflectivity derived from a kinematic approach is [36]

$$R = \frac{16\pi^2}{Q_z^4} \left[4(\rho_1 - \rho)^2 \sin^2\left(\frac{Q_z \tau}{2}\right) \right]$$
 (3.40)

and the reflectivity is seen to be proportional to the square of the difference between the SLD of the film and the backing and fronting phases and inversely proportional to the fourth power of the momentum transfer vector.

In an electrochemistry–NR experiment, the reflection of neutrons takes place at an interface consisting of five parallel phases as schematically shown in Figure 3.9. Table 3.1 lists the numerical values of the theoretical SLDs of the materials used in typical electrochemical studies. Each phase contributes to the overall measured NR, and to understand the shape of the experimental reflectivity curves it is instructive to examine the contribution of each individual lamina. To do this, a recursion scheme for stratified media described by Parratt [18] can be used to calculate the reflectivity of a simulated interface. These calculated reflectivities are then compared to the reflectivities predicted by the kinematic approximation. Consider a 20 Å thick film of a hydrocarbon-based surfactant deposited on a gold/chromium-modified quartz sample. To simplify the analysis, a mixed D₂O/H₂O

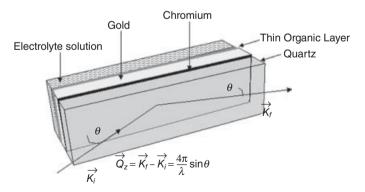
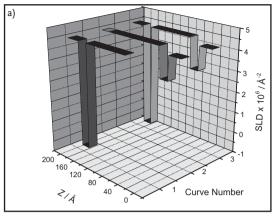


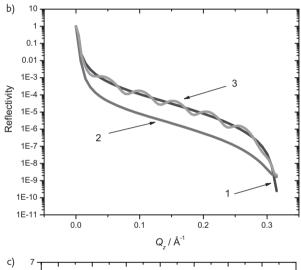
Figure 3.9 Schematic of an interface consisting of five parallel phases employed in electrochemistry–NR studies. Figure taken from [80].

Table 3.1 Calculated SLD values for the materials used in electrochemistry-NR studies of thin organic films.

Material/substance	$ ho imes 10^6$ (Å $^{-2}$)
Quartz	4.18
Chromium	3.03
Gold	4.50
-CH ₂ -	-0.49 (determined for $C_{10}H_{22}$)
-CD ₂ -	6.60 (determined for C ₁₀ D ₂₂)
D_2O	6.33
H_2O	-0.56

solvent is simulated such that the SLD for the solvent is contrast-matched to the gold (gold contrast-matched water). The presence of the quartz (SiO₂) and the thin film of chromium is temporarily neglected. Curve 1 of Figure 3.10a plots the SLD profile in the direction normal to the interface. The reflectivity curve calculated for this profile is presented in Figure 3.10b, which shows that the reflectivity decreases quickly with Q_z . This decay is modulated by the interference between neutrons reflected from the front and the back sides of the organic film lamina. In Figure 3.10c the same set of data are plotted as $Q_z^2 \sqrt{R} / 8\pi$ against Q_z . This presentation eliminates the Q_z^{-4} dependence of the reflectivity and allows a better analysis of the interference fringes. Consistent with Eq. (3.40), the period of the oscillation is $\Delta Q_z = 2\pi/\tau_1$ which allows the thickness of the film to be readily determined. The amplitude of the oscillation gives the absolute value of the difference between SLDs, $|\rho_1 - \rho|$, from which information on the composition of the film can be determined, with the help of Eq. (3.40). The presence of the chromium layer can also be examined. Curve 2 in Figure 3.10a plots the SLD profile for an interface that consists of quartz, 20 Å chromium, gold, and gold-matched solvent in the absence of the organic film. The SLD contrast between chromium and either adjacent layer is very weak (gold and quartz have very similar SLDs) and, as Figure 3.10b shows, the reflectivity is much lower in this case. The effect of the SLD contrast on the amplitude can be conveniently seen when the data are plotted as $Q_z^2 \sqrt{R} / 8\pi$ in Figure 3.10c. Curves 1 and 2 in Figure 3.10c have the same periodicity; however, curve 2 has much lower amplitude consistent with the lower contrast shown in the SLD profile in Figure 3.10a. To simulate an actual biomimetic membrane, a film of organic molecules is added to a 100 Å thick gold film placed on top of the chromium layer. The SLD profile at the interface is now represented by curve 3 in Figure 3.10a. The reflectivity curve corresponding to this profile shows a new interference fringe pattern with higher frequency due to the thick gold layer. When the gold is sandwiched between the different SLDs of the chromium and organic film layer its presence becomes visible in the reflectivity profile. Curve 3 in Figure 3.10c demonstrates that this pattern results from a sum of two periodic functions: a low-frequency component whose periodicity is determined by the





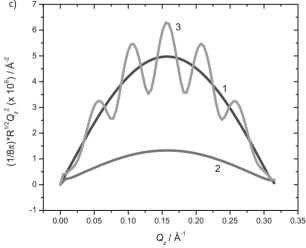


Figure 3.10 (a) SLD profiles for components of the interface shown in Figure 3.9 assuming that the SLD for the solution phase is contrast-matched to gold. Curve 1 represents a hydrocarbon film adsorbed on a gold film and purposely omits the chromium layer and the quartz substrate. Curve 2

includes the chromium and quartz but omits the organic film. Curve 3 shows the SLD profile for the entire interface. (b) Reflectivity curves calculated from the SLD profiles in (a). (c) Plot of $Q_z^2 \sqrt{R} / 8\pi$ versus Q_z calculated from the reflectivity curves shown in (b). Figure taken from [80].

thickness of the organic film and to a lesser extent by the chromium (which has the same thickness but much lower amplitude); and a high-frequency component determined by the thickness of the gold film.

The above discussion reveals how NR can be used to provide direct information concerning both the thickness and the composition via the SLD of the various layers in the interface. This constitutes an advantage over ellipsometry or surface plasmon resonance methods where the value of the refractive index (composition) of the film is usually assumed in order to determine its thickness. However, to use the kinematic approach to directly determine these quantities, the reflectivity has to be measured in a sufficiently large range of Q_z so that at least one-half of the longest period interference fringe, corresponding to the thinnest layer in the interface, is observed on the reflectivity curve. The spatial resolution for a film of thickness τ is defined as $\tau = \pi/Q_{z,\text{max}}$, where $Q_{z,\text{max}}$ is the maximum momentum transfer vector accessible to the experiment [37]. Thus, access to reliable reflectivities at increasingly larger momentum transfer vectors is required to probe the finer details of thin films. The NG-7 line at the NCNR allows reflectivity to be measured up to $Q_{z,max} = 0.25 \,\text{Å}^{-1}$. At the NG-1 reflectometer at the NCNR, experiments can be run routinely up to 0.3 Å⁻¹ [38, 39] and in an exceptionally welldesigned experiment even up to $0.7 \,\text{Å}^{-1}$ [40]. For $Q_{z,\text{max}} = 0.25 \,\text{Å}^{-1}$, the spatial resolution is about 12 Å and for $Q_{z,max} = 0.15 \,\text{Å}^{-1}$ it is about 20 Å. This resolution is sufficient to study polymer films or corrosion layers with thickness of the order of 100 Å. However, it imposes a limitation on studies of monolayers and bilayers of surfactants and phospholipids where the thickness of the adsorbed film is comparable to the spatial resolution. This does not prevent one from studying aspects of the organic film less than the spatial resolution as, although a pure kinematic-based analysis of the data is insufficient to resolve a very thin component of the film, each discrete layer will contribute to the overall measured reflectivity. If the contrast is high enough, even a very thin layer (about 5 Å) can have a pronounced effect on the entire reflectivity curve which then can be accounted for in a fitting procedure.

The common procedure used to calculate the SLD profile from the reflectivity curve is to assume a model profile, calculate the theoretical reflectivity curve using the optical matrix or recursion method, and compare calculated and experimental curves. A least-squares iterative procedure is then used to vary the parameters of the SLD profile until a good fit between the calculated curve and the experimental data is achieved. Although the inversion of the reflectivity data is not unique and

the SLD profile calculated from the reflectivity curve is not the only mathematical function that fits the reflectivity curve, it usually gives a good physical model of the interface. In order to reduce the ambiguity and the potential non-uniqueness of the fitting procedure, one can measure a system of interest with several different contrasts by varying the nuclear composition (isotopic variation) of either the organic or solvent phase. This is usually done through the use of H_2O/D_2O but many perdeuterated phospholipids and surfactants are commercially available. To further reduce the number of adjustable parameters it is critical to determine a portion of the structural information from independent experiment. The thickness, the roughness, and the SLD of chromium and gold layers sputtered onto quartz crystals can be independently determined by measuring the NR and/or X-ray reflectivity of the crystals in air. These parameters can then be fixed when the substrates are used for *in situ* electrochemistry–NR experiments.

3.4 Selected Examples

Although Thomas and coworkers proposed NR as a sensitive technique for investigating surface structure in 1981 [41], the first papers describing successful application of the methodology did not appear until the late 1980s [42–44]. Since these early papers describing applications of NR to surface and interfacial problems, the technique has slowly started to blossom as more dedicated instruments are brought on-line throughout the world. The increased popularity of the technique is manifest in a recent special issue of *Langmuir* (the American Chemical Society's journal for interfacial science) dedicated to applications of NR [45]. In recent years there have been many new NR studies of soft biological samples supported on solid supports but only a limited number of these were performed with potential control. Some instances of the former are reviewed in Section 3.4.1 as these systems remain of interest to the bioelectrochemical community. A case study of electric field-driven transformations in a model biomembrane is discussed in Section 3.4.2.

3.4.1

Supported Proteins, Peptides, and Membranes without Potential Control

3.4.1.1 Quartz- and Silicon-Supported Bilayers

In 1991, Johnson *et al.* reported one of the first NR studies of phospholipid bilayers at the solid–solution interface [46]. Although these measurements were not the first to employ NR to study molecules adsorbed at the solid–liquid interface, they did constitute the first measurements of a supported bilayer using NR. A bilayer of dimyristoylphosphatidylcholine (DMPC) was spread on a quartz surface by the fusion and rupturing of small unilamellar vesicles. The very smooth, single-component substrate allowed a complex model of the interface to be constructed from layers corresponding to (i) the quartz, (ii) a thin film of water on the quartz

surface, (iii) the lipid head group directed toward the quartz, (iv) the hydrocarbon core of the bilayer, (v) the lipid head group directed toward the solvent, and (vi) the pure solvent itself. With access to perdeuterated DMPC and by varying the ratio of D₂O and H₂O in the subphase several different neutron-contrast measurements could be performed to increase the reliability of their fitting analyses. Information concerning the extent of head group hydration and the presence of solvent cushion layers was extracted from the NR data but a major limitation of this early effort was the limited accessible Q space ($Q_{\text{max}} \approx 0.11 \,\text{Å}^{-1}$). Koenig et al. [38] also studied the spontaneous assembly of lipid bilayers on a solid support using NR. They studied dipalmitoyl glycero-3-phosphocholine (DPPC) and distearoyl glycero-3-phosphocholine on silicon single crystals. Their novel cell design allowed access to much higher momentum vector space and consequently their model-fitting parameters carry much less uncertainty. Koenig et al. found a much thinner water layer separating the native surface silicon oxide from the head group region of the lipid molecules and reported the thickness of the water layer to be $6-12\,\text{Å}$ compared to the $30\,\text{Å}$ found in [46]. The inner core of the DPPC lipid bilayer was altered by 4-6 Å by crossing the phase transition temperature which is consistent with increased gauche conformations in the melted hydrocarbon tails. Analysis of the lipid layer's SLD indicated a patchy bilayer structure was formed on the silicon-silicon oxide surface. An assessment of the kinetics of bilayer film formation showed that the patchy bilayer is rapidly formed on the large substrate (about 3-5 min) and that a small apparent increase in the surface coverage was the result of vesicles loosely associating with the bilayer film. Washing the substrate with vesicle-free buffer returned the surface coverage to the values observed at short times (3–5 min). After removing adsorbed vesicles, the surface coverage was found to be invariant with time in the aqueous environment.

For supported membranes to maintain the structural and dynamic properties of free biomembranes, the interaction between the membrane and the substrate should be minimized. One strategy for decoupling the biomembrane from the underlying surface is to support the biomembrane upon a soft hydrated polymer or polyelectrolyte film. Various approaches to achieve this have been reviewed by Sackmann [47]. Majewski and coworkers [48-50] have employed NR to perform structural studies of polyethyleneimine (PEI)-cushioned lipid bilayers. As characterized by NR, DMPC vesicles added to PEI-coated quartz substrates led to an inhomogeneous structure consisting of a mixture of vesicles and multilayers of lipids. The lipid multilayer was dramatically illustrated by high-frequency oscillations in the reflectivity curve at low momentum transfer vectors. It was postulated that, as opposed to quartz, glass, or mica, the fusion of the DMPC lipids on a polymer surface is hindered by the absence of a strong van der Waals interaction between the phospholipid molecules and the substrate. As a result, the interface becomes a complicated structure consisting of multilayer complexes of PEI with bilayers as well as intact bilayers adsorbed to the surface. In contrast, Majewski and coworkers were able to successfully spread DMPC bilayer films on the unmodified quartz substrate from vesicles and observed qualitatively similar results to those of Johnson et al. [46]. Remarkably, when PEI was added to the cell it was

noted that the PEI polymer appeared underneath the previously deposited DMPC bilayer. The authors explained their observation thermodynamically by noting that the positively charged PEI is drawn to the negatively charged quartz substrate.

Recently, Brzozowska *et al.* used NR and *ex situ* electrochemical techniques to characterize an innovative type of monolayer system intended to serve as a support for a bilayer lipid membrane on a gold electrode surface [51]. Zr⁴⁺ ions were used to noncovalently couple a phosphate-terminated self-assembled monolayer (SAM) formed on a gold surface to the carboxylate groups of negatively charged phosphatidylserine (PS). This tethered surface was then used for the formation of a PS lipid bilayer structure formed by vesicle fusion and spreading. NR studies revealed the presence of an aqueous environment associated with the tether layer which arises from nonstoichiometric water associated with the zirconium phosphate moieties [52].

3.4.1.2 Hybrid Bilayers on Solid Supports

Another method used to create biomimetic membranes on solid supports involves the use of so-called hybrid bilayers. This method was first reported by Tamm and McConnell [53] and involves a monolayer-coated substrate being immersed through a film of lipid spread at the air-solution interface. Kuhl et al. employed such a grafting procedure to study poly(ethylene glycol) lipids on octadecyltrichlorosilane (OTS)-modified substrates [54] with NR. The procedure involved silanizing the quartz substrates and then, using Langmuir-Blodgett deposition, adding a single layer of distearoylphosphatidylcholine with poly(ethylene glycol) covalently bound to the head group. After passing the silane-modified substrate through the air-water interface the cell was assembled under water. From NR measurements it was determined that this procedure led to continuous and tightly packed lipid monolayers on the substrates. Meuse et al. slightly modified the above procedure in their infrared spectroscopy and NR studies of hybrid bilayer membranes (HBMs) [55]. In this experimental approach, the lipid layer was added to the monolayer-covered substrate with a single horizontal touch and was subsequently withdrawn without passing the substrate entirely through the interface. This eliminated the need to assemble the cell under water. Meuse et al. were able to access momentum transfer vectors as high as 0.25 Å⁻¹ allowing sufficient resolution to observe differences in the thickness of the single phospholipid layer above and below the phase transition temperature. A thiol layer self-assembled on goldcoated silicon was used rather than an OTS layer chemically grafted to silicon. The presence of the 50 Å thick layer of gold boosted the overall reflectivity of the interface helping access higher Q_z values. Meuse et al. were able to demonstrate that a hybrid bilayer approach leads to nearly 100% surface coverage by the membrane. In comparison, earlier work by the same group estimated the surface coverage of pure lipids to be between 70 and 95% depending on the sample [38]. From a biophysical point of view, the extended spatial resolution of these measurements led to much greater detail in the description of the lipid head group. The head group of a single bilayer is hydrated to a much greater extent as compared to the average degree of hydration for multilamellar systems. Hybrid bilayers composed

of covalently bound thiol-modified lipid inner layers and physisorbed distal leaflets have been used in several NR studies [56–59] with particular interest devoted to designing hydrophilic tethering groups that hydrate the head groups of both sides of the model bilayer.

Krueger and coworkers worked to further improve the resolution of NR measurements and their work culminated in one of the most elegant applications of NR to the study of biological layers [40]. They investigated the interaction of the peptide toxin melittin with the head group of a phospholipid. The main purpose of this work was to determine the effect of melittin insertion into a hybrid bilayer membrane. Melittin is believed to enhance ion transfer across biological membranes which, for example, can lead to lysis of red blood cells. However, the precise structure of the channel-forming pores is the subject of some debate. Krueger and coworkers were able to perform NR experiments up to an unprecedented momentum transfer vector maximum and were therefore able to measure very small changes in the hybrid bilayer (down to 2-3 Å) due to the high resolution of their experiment. This work is impressive from both a technical aspect and the scientific information garnered. Background scattering that arises due to scattering from the substrate, the air environment, and the aqueous backing was greatly reduced with the use of very thin (0.5 mm) silicon wafers and the placement of the entire cell assembly in a helium-filled chamber. Importantly, a very thin (about 15-20 µm) aqueous reservoir of D₂O was used which greatly attenuated the incoherent scattering arising from the much thicker aqueous environments commonly used in NR studies of aqueous systems. Using this experimental set-up, data were obtained out to a maximum Q_z value of 0.7 Å⁻¹ and reflectivities as low as 10^{-8} . A schematic of the cell design used in [40] is shown in Figure 3.11.

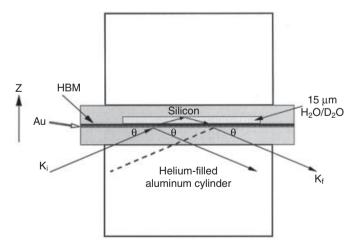


Figure 3.11 Schematic of the novel cell design used to measured the reflectivity of a hybrid bilayer system down to the level of about 10^{-7} . Figure taken from [40].

The hybrid bilayer was built by first derivatizing the silicon substrate with a thin gold film (50 Å) layered on top of a 25 Å thick layer of chromium. A SAM of octadecanethiol or a thiahexa(ethylene oxide) alkane (THEO) was formed on the gold surface followed by the transfer of a monolayer of deuterated DMPC. In the absence of the melittin protein, NR measurements indicated that the roughness of the gold layer sputtered onto the substrate is not transferred across the SAM to the phospholipid leaflet. Additionally NR measurements indicated that the DMPC molecules comprising the outer half of the hybrid bilayer are sufficiently fluid to effectively anneal the roughness of the inner layer which is conformal to the roughness of the deposited gold film. In the presence of melittin, the region of the head group of the HBM's deuterated DMPC leaflet showed the greatest changes. The SLD profiles were determined from a model independent fitting routine and are reproduced in Figure 3.12. Figure 3.12 clearly demonstrates that the addition of melittin displaces a large quantity of solvent (D2O in this instance) from the lipid head group. Although the SLD profiles do not offer strong evidence of the insertion of the melittin into the deuterated DMPC hydrocarbon core, this occurrence cannot be ruled out because the SLD of the hydrated melittin pore is expected to be similar to the SLD of the -CD₂- tails. Furthermore, it was noted that the overall thickness of the -CH₂- layer of the thiol becomes thicker by about 3 Å. This is consistent with molecular dynamics

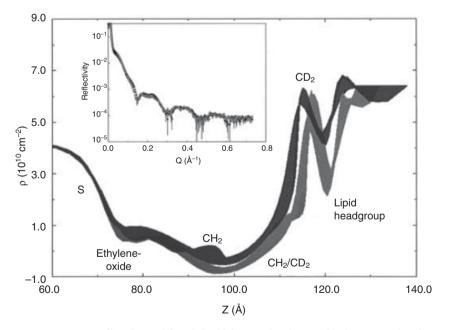


Figure 3.12 SLD profiles obtained for a hybrid bilayer in the absence (black curve) and in the presence (grey curve) of the protein melittin. The experimental NR curves used to generate the SLD profiles are shown in the inset (taken from [53]). Figure taken from [40].

simulations of a single melittin molecule in a DMPC bilayer and it implies that the insertion of melittin not only affects the lipid leaflet in which it is present but also the hydrocarbon tail of the innermost leaflet of the HBM. However, the SLD profiles also clearly demonstrate that the presence of melittin does not lead to hydration of the ethylene oxide region in the THEO-based HBM. This surprising result indicates that melittin does not alter the HBM in such a way as to allow D₂O to fully penetrate the membrane. It was proposed that alternate tethering chemistries need to be considered to lead to more biomimetic hybrid bilayers (see above). In any event, this very impressive paper deserves emphasis as it is one of the most methodologically advanced NR studies of biological systems reported in the literature. The small changes in the inner layer thickness of the HBM upon insertion of melittin would not be measurable using "traditional" NR measurements.

Valincius et al. used NR and ex situ electrochemical impedance spectroscopy to study the effect of amyloid β-peptide insertion on supported lipid bilayers [60]. It has been speculated that the interaction between these oligomeric proteins and bilayer membranes plays an important role in the aggregation and protein misfolding that leads to various neurodegenerative diseases including Alzehimer's and Parkinson's diseases [61]. Using a variety of solvent and phospholipid contrasts, including the use of perdeuterated phospholipids, NR demonstrated that amyloid β-peptides inserted into lipid bilayers tethered to a gold-modified silicon substrate. The NR data revealed that the protein fully inserts into the hydrophobic core, disrupting both lipid leaflets but does not lead to significant disruption of the head group region as determined by the absence of increased water content in the SLD profiles. This information was vital for explaining the results of ex situ impedance spectroscopy and modeling the increased ion transport capabilities of the supported membranes in the presence of amyloid β -peptides.

3.4.1.3 Protein Adsorption and DNA Monolayers

Many recent NR studies of protein and peptide assembly on various solid and liquid interfaces have been reviewed by Zhao et al. [62]. The interested reader is directed to that paper for a more comprehensive outline of these NR studies. The first NR study of protein-membrane interactions was reported by Schmidt et al. in the early 1990s [63]. The interfacial binding reaction between streptavidin and a solid-supported lipid monolayer partly functionalized by biotin moieties was evaluated. NR was sensitive enough to detect the presence of biotin proteins attached to the membrane, but again a poor range of accessible data prevented high-resolution details pertaining to protein adsorption. Fragneto et al. were able to provide much greater detail in their analysis of bovine β -casein adsorbed on a silicon surface derivatized with deuterated octadecyltrimethylsiloxane [64]. A hydrophobic surface was chosen as the conformation of proteins is in most cases determined by the hydrophobic interactions in the nonpolar residues of the peptide chains. A deuterated OTS SAM was chosen to enhance the contrast between the solvent and the nondeuterated protein. Measurements were repeated using three different solvent contrasts, pure D₂O, pure H₂O, and silicon contrast-matched water, and a two-layer model of the protein layer was proposed to fit the reflectivity data for all three contrasts. The inner layer of adsorbed β-casein was relatively protein dense (about 60% by volume fraction) and was $23 \pm 1 \text{ Å}$ thick. The outer layer was thicker (35 \pm 1 Å) and was only 12% protein by volume fraction. The results from the reflectivity experiments can be explained in terms of the charge distribution of the protein molecule. β-Casein resembles a surfactant with a polar head and a nonpolar tail. The high concentration of charged amino acids in the head group makes it essentially negatively charged whereas the remainder of the protein is hydrophobic. The inner layer observed by Fragneto et al. is consistent with a hydrophobic layer, containing little water, whereas the outer layer corresponds to the hydrophilic head group which is heavily solvated and protrudes into the bulk of the aqueous phase. Fragneto et al. continued their studies of protein adsorption on hydrophobic surfaces by investigating the influence of pH on the adsorption of β-casein and β-lactoglobulin on silicon [65]. It was shown that at pH greater than 7, the two-layer structure of β-casein applied but β-lactoglobulin formed a single uniform layer. At lower pH the adsorption of both proteins was greatly enhanced but the β-lactoglobulin could no longer be described as a single uniform layer. As before, the observed behavior was readily explained in terms of the degree of ionization of the proteins.

Levicky et al. [66] used NR to investigate how self-assembly can control the structure of DNA monolayers adsorbed onto gold surfaces. This report was an extension of an earlier effort to use single-stranded DNA monolayers as model DNA chips for diagnostic applications. The strategy employed by Levicky et al. was to prepare mixed monolayers of 6-mercapto-1-hexanol (MCH) and thiol-terminated single-stranded DNA (HS-ssDNA) on gold. NR clearly demonstrated that the oligomeric DNA is oriented parallel with the gold-coated substrate in the absence of MCH. However, upon introduction of MCH, the HS-ssDNA is displaced from the surface in favor of the MCH. Furthermore, the replacement reaction leaves the HS-ssDNA directed to the aqueous solution and oriented perpendicular to the substrate. This orientation is greatly favored for hybridization of the ssDNA with its complementary sequence. Similar work with thiolated hairpin DNA strands immobilized on gold-modified substrates was performed by Steichen et al. [67]. NR confirmed the opening of the stem-loop configuration of the ssDNA upon hybridization associated with helix formation. More recent work [68-70] has demonstrated how NR is ideally suited to follow transfection-related interactions between lipoplexes (vesicles of cationic lipid-DNA) and model membranes. A primarily DMPC bilayer was deposited on a silicon substrate and lipoplexes containing 1:1 complexes of dimethyldioctadecylammonium bromide and DNA were used as transfection mimics. With clever use of contrast variation and selective deuteration of lipid components it was shown that the rate of lipid exchange between the cell membrane and the lipid-DNA complex is vital in determining transfection efficiency. All of the studies described above are well suited for electrochemical measurements as protein binding, DNA hybridization, and drug transfection would be expected to be highly dependent on the electrical state of the supporting substrate.

Electric Field-Driven Transformations in Supported Model Membranes

Until about 10 years ago, electrochemical applications of NR had largely been directed toward hard material problems such as corrosion [71, 72] and oxide formation [5, 73-75]. More recently electrochemical NR has been used to study the hydration of Nafion in proton-exchange membranes for fuel cell applications [76]. The characterization of soft materials at electrode surfaces began in the late 1990s when Lipkowski and coworkers began reporting NR studies of electric field-driven transformations in several model surfactant monolayer and bilayer systems including dodecylsulfate [77, 78], pentadecylpyridine [79, 80], and octadecanol [81]. This was concurrent with Hillman and coworkers using NR to profile the composition (particularly solvent permeation) in electropolymerized films including dynamic measurements of polymer swelling upon redox cycling [22, 25, 32, 82]. The technical development of these works has laid the foundation for bioelectrochemical applications, particularly the study of supported phospholipid films under the influence of applied AC [83] and static [84-86] electric fields. In the latter case, both Burgess et al. [84, 85] and Hillman et al. [86] have characterized changes in biomimetic layers upon variation of the electrical state of the supporting electrode. A case study of the former is provided below.

Lipids and proteins in natural biological membranes are frequently exposed to static electric fields of the order 10⁷ to 10⁸ V m⁻¹ [87] and a model membrane supported on a conductive substrate may be used to study voltage-gated membrane proteins and lipid-lipid and lipid-protein interactions [88-91]. A model lipid system consisting of a mixed DMPC-cholesterol bilayer (70:30 mol% ratio) was formed on a gold surface by fusion and spreading of small unilaminar vesicles and electrochemically characterized as shown in Figure 3.13. The membrane is stable at the electrode when the surface charge density is in the range $-8\mu\text{C}\,\text{cm}^{-2} < \sigma_{\text{m}} < 8\mu\text{C}\,\text{cm}^{-2}$. Outside this region, the charge density curve for the membrane-covered electrode abruptly rises or falls to approach the corresponding curve of a lipid-free interface. This behavior indicates that the membrane becomes detached from the gold surface at sufficiently large polarizations. An interesting feature is evident in the charge density measurements at applied potentials of about -0.8V where the charge of the electrode in the presence of the detached film approaches but does not quite merge with the curve for the film-free surface. It was inferred that this was caused by either incomplete desorption or the possible formation of intact vesicles that remain in very close proximity to the electrode surface. It is impossible from the electrochemical experiments alone to adequately describe the film structure and water distribution at the interface for either the detached or contact adsorbed layer. However, this problem is ideally suited for electrochemistry-NR studies as described below.

The substrates were formed from virgin quartz blocks modified with nominally 3 nm thick chromium and 12 nm thick gold layers. The thickness, roughness, and SLD of chromium and gold layers sputtered onto the quartz crystals were determined independently by measuring the X-ray reflectivity of the crystals in air

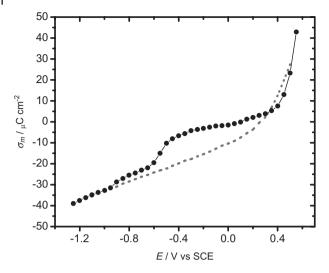


Figure 3.13 Surface charge density on a gold electrode surface plotted versus the electrode potential for (dotted curve) 50 mM NaF supporting electrolyte and (circles) mixed 7:3 DMPC–cholesterol bilayer spread from vesicle solution. Figure taken from [84].

before and after the electrochemical NR experiments. Figure 3.14 shows both the experimental and fitted X-ray reflectivity curves for the experiments before the electrochemical experiments. The curve obtained for the crystal after the *in situ* electrochemical measurements was nearly identical demonstrating that the chromium and gold layers remained essentially unaffected by the electrochemistry. The inset to Figure 3.14 shows the electron density (as determined from the real component of the SLDs) as a function of the distance normal to the electrode surface. From these data the parameters for the chromium and gold layers were determined and, as expected, the differences between the fit parameters for the chromium and gold layers determined before and after the experiment are small.

Curve 2 in Figure 3.15 shows the NR curve for the membrane formed by fusing mixed DMPC–cholesterol vesicles at the gold electrode at $E=50\,\mathrm{mV}$ ($\sigma_{\mathrm{m}}\approx-1\,\mu\mathrm{C\,cm^{-2}}$). For comparison, curve 1 shows the reflectivity curve recorded at the gold-coated quartz electrode in D_2O in the absence of DMPC. The reflectivity is significantly higher in the presence of vesicles, indicating that vesicles fuse and form a membrane at the gold surface. The data show that reflectivities could be measured with good statistics down to about 2×10^{-6} and $Q_z=0.18\,\mathrm{\AA^{-1}}$. Figure 3.16 shows reflectivity plots for different applied potentials as well as the SLD profiles determined from the best-fit models of the reflectivity data. The reflectivity was best described by a single layer of proton-containing molecules deposited directly on the metal surface. This fit requires a limited number of adjustable parameters but averages the SLDs for the phosphatidyl head groups and the hydrocarbon tail regions. In order to extract more detailed information concerning the membrane, the same reflectivity data were fitted to a three-layer model assum-

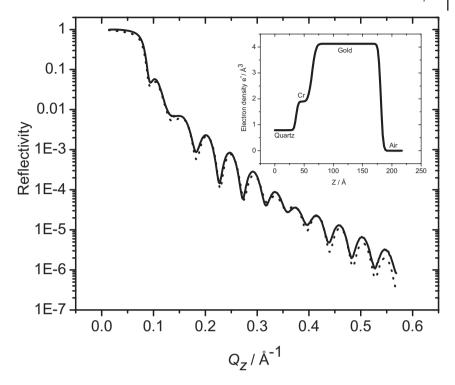


Figure 3.14 X-ray reflectivity as a function of momentum transfer vector for a gold—chromium-covered quartz substrate. The curve corresponds to the measurement made before electrochemistry. The curve for the post-electrochemical measurement is

essentially identical and is omitted for clarity. The dotted curve represents the results of modeling. The inset shows the electron density profile for the model that best fits the data. Figure taken from [84].

ing that the membrane consists of two 8 Å thick polar head regions and a middle section composed of hydrocarbon acyl chains. While the three-layer model is physically more realistic and statistically gives a slightly better fit to the experimental data, it involves at least four additional adjustable parameters. The contrast for the polar head region and the backing gold or D_2O phases is weak. In addition, the SLD profiles calculated for a single-layer film allowing for adjustable roughness of the film were very close to the SLD profiles calculated for the three-layer model of the film. For these reasons a single layer was considered to be a sufficient approximation. The thickness and the SLD values for the acyl chain region of the membrane, determined from the fit to the single-layer model, are compiled in Table 3.2. The SLD for the organic layer was then used to determine the occupancy of phospholipid and cholesterol at the electrode surface assuming that the modeled layer contains only the deuterated solvent and the hydrocarbon adsorbate. The total SLD of the organic layer (ρ_{total}) is, therefore, given by the equation $\rho_{\text{total}} = x \rho_{\text{org}} + (1-x) \rho_{\text{D}_2\text{O}}$, where ρ_{org} and $\rho_{\text{D}_2\text{O}}$ are the SLDs of the organic and

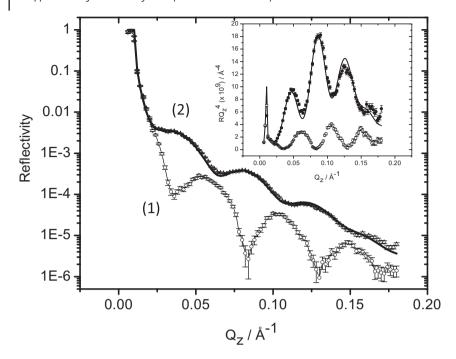


Figure 3.15 Experimentally determined reflectivity curves (points with associated error bars) for $E = 50 \, \text{mV}$ in $50 \, \text{mM}$ NaF in D_2O : curve 1, the film-free electrode surface; curve 2, the electrode covered by a bilayer of a 7:3 mixture of h-DMPC and cholesterol. The

inset shows plots of RQ^4 versus Q_2 calculated from the data presented in the main part of the figure. The solid curve shows the calculated reflectivity curves from the best-fit model whose parameters are given in Table 3.2. Figure taken from [84].

solvent material, respectively, and x is the volume fraction of hydrocarbon in the film. The value for $\rho_{D_{20}}$ is known (6.33 × 10⁻⁶ Å⁻²) and the value for ρ_{org} was estimated using the densities and molecular formulae of cholesterol and hydrocarbons. The values for the best-fit parameters of the NR data obtained for potentials between +50 mV and -600 mV exhibit an interesting trend. As the potential applied to working electrode becomes increasingly negative the phospholipid-cholesterol layer thickness increases by nearly 28% with a significant concomitant increase in the SLD. This thickening of the phospholipid layer can be attributed to swelling of the organic layer by ingress of progressively larger amounts of solvent molecules with decreasing electrode potential. Figure 3.17 plots the RQ_z^4 versus Q_z curves for E equal to -700, -800, and -950 mV and the SLD profiles determined from the best-fit analysis. The reflectivity of the electrode surface remains very high indicating that the bilayer remains near the electrode surface even though the charge density measurements indicate the film has delaminated. These curves could only be adequately fitted to a two-box model with the box directly adjacent to the gold surface consisting of a film a few angstroms thick having a very high SLD. Moving

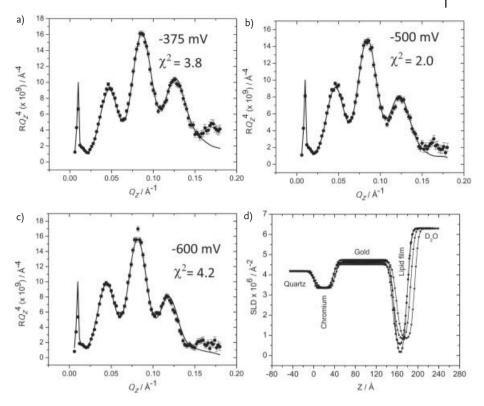


Figure 3.16 Plots of RQ^4 versus Q_z for a bilayer of a 7:3 mixture of h-DMPC and cholesterol in 50 mM NaF in D₂O: (a) E = -375 mV; (b) E = -500 mV; (c) E = -600 mV. Points with associated error bars show the experimental data. Solid curves show the reflectivity calculated from the parameters obtained from the fitting

procedure. (d) SLD profiles for the interface for $E = 50 \,\text{mV}$ (squares), $E = -375 \,\text{mV}$ (circles), $E = -500 \,\text{mV}$ (up triangles), and $E = -600 \,\text{mV}$ (down triangles). The best-fit model parameters corresponding to the SLD profiles are listed in Table 3.2. Figure taken from [84].

the potential in the negative direction, the thickness of this box increases and its SLD reaches a limiting value very close to that of pure D_2O . Apparently, the bilayer becomes separated from the metal surface by a thin layer of solvent at negative potentials. In an effort to further verify the validity of the observation of a solvent cushion layer the NR measurements for $E = -800\,\mathrm{mV}$ were repeated using a different isotopic contrast. The bilayer was formed by fusion of vesicles prepared from perdeuterated DMPC mixed with nondeuterated cholesterol and the solvent was H_2O . Figure 3.18 shows the RQ_z^4 versus Q_z curve and SLD profile for this system. The result confirms that the bilayer is separated from the gold surface by an approximately $10\,\mathrm{\mathring{A}}$ film of water at this negative potential. The bilayer is $38\pm1\,\mathrm{\mathring{A}}$ thick and the SLD for the bilayer is within the experimental error equal

Table 3.2 Best-fit results describing a mixed DMPC–cholesterol bilayer deposited at the electrode surface as a function of the applied potential. The results are obtained from the best-fit results of the measured neutron reflectivity curves in Figures 3.16–3.18.

System (E (mV) vs. SCE)	Organic layers							
	Water layer				Lipid-cholesterol layer			
	τ ^{a)} (Å)	SLD ^{b)} (×10 ⁶ Å ⁻²)	Volume fraction of solvent	σ ^{c)} (Å)	τ ^{a)} (Å)	SLD ^{b)} (×10 ⁶ Å ⁻²)	Volume fraction of solvent	σ ^{c)} (Å)
h-DMPC+ chol (50)	N/A	N/A	N/A	N/A	25.6 ± 0.9	0.03 ± 0.22	0.08	5.6
h-DMPC+ chol (-375)	N/A	N/A	N/A	N/A	29.3 ± 1.0	0.51 ± 0.12	0.15	5.9
h-DMPC+ chol (-500)	N/A	N/A	N/A	N/A	32.8 ± 1.0	0.82 ± 0.18	0.19	6.3
h-DMPC+ chol (-600)	N/A	N/A	N/A	N/A	35.3 ± 0.7	0.85 ± 0.17	0.20	6.8
h-DMPC+ chol (-700)	4.1 ± 0.2	5.38 ± 0.40	0.86	6.1	35.8 ± 0.7	0.05 ± 0.20	0.08	5.4
h-DMPC+ chol (-800)	6.0 ± 1.4	6.33 ± 0.41	1.00	5.5	37.0 ± 0.9	-0.43 ± 0.32	0.01	5.6
h-DMPC+ chol (–950)	10.1 ± 1.0	6.28 ± 0.32	1.00	5.3	36.8 ± 0.8	-0.51 ± 0.19	0.00	5.8
d-DMPC+ chol (–800)	10.4 ± 1.3	-0.56 ± 0.21	1.00	2.1	37.2 ± 1.0	5.22 ± 0.28	0.05	4.1

a) Thickness of layer.

Source: Taken from [40].

to the theoretical value of the SLD calculated for the water-free film estimated to be $5.5 \times 10^{-6}\,\text{Å}^{-2}$. The good agreement between the results of NR experiments performed using hydrogenated and deuterated DMPC gives credibility to the SLD profiles calculated from the reflectivity curves. The main results of this work are summarized pictorially in Figure 3.19. It should be emphasized that Figure 3.19 is not a model but rather a schematic representation of the DMPC–cholesterol film consistent with the analysis of the specular NR data. The NR analysis does not provide information about the lateral distribution of individual solvent molecules nor does it provide sufficient resolution to give details concerning the fine structure of the film normal to the gold surface due to a limited maximum Q_z .

b) Scattering length density.

c) Roughness.

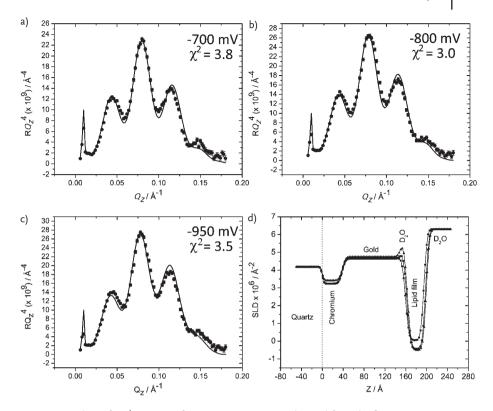


Figure 3.17 Plots of RQ^4 versus Q_z for a bilayer of a 7:3 mixture of h-DMPC and cholesterol in 50 mM NaF in D_2O : (a) E = -700 mV; (b) E = -800 mV; (c) E = -950 mV. Points with associated error bars show the experimental data. Solid curves show the reflectivity calculated from the

parameters obtained from the fitting procedure. (d) SLD profiles for the interface for $E = -700 \, \text{mV}$ (squares), $E = -800 \, \text{mV}$ (circles), and $E = -950 \, \text{mV}$ (triangles). The best-fit model parameters corresponding to the SLD profiles are listed in Table 3.2. Figure taken from [84].

The analysis of specularly reflected neutrons provides a compositional profile, in the directional normal to the film, averaged over the transverse coherence length of the incident neutrons, which for the range of momentum transfer vectors accessed in these measurements spans roughly between 10 and $100\,\mu m$. As long as the dimension of the lateral inhomogeneities is less than the coherence length of the incident neutrons then the measured reflectivity curve accurately probes the area-averaged interface [39]. As such, the water molecules shown incorporated in the biological films in Figure 3.19 qualitatively represent the volume fraction of water inside the lipid film but they do not represent the precise location or distribution of the solvent molecules. On the other hand, when water exists as a discrete, uniform layer, as is the case for negative electrode potentials, NR analysis can accurately place the slab of water within the overall SLD profile of the entire

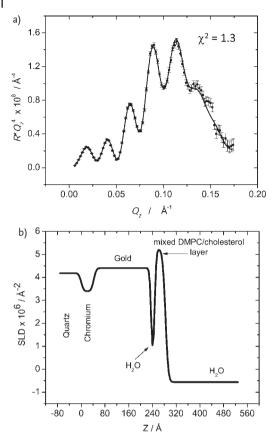


Figure 3.18 (a) Plot of RQ^4 versus Q_z for a bilayer of a 7:3 mixture of deuterated DMPC and cholesterol in 50 mM NaF in H_2O at E=-800 mV. Points with associated error bars show the experimental data. Solid curve

shows the reflectivity calculated from the SLD profile presented in (b). The best-fit model parameters corresponding to the SLD profile are listed in Table 3.2. Figure taken from [84].

interface. This is a unique advantage of NR as no other technique can probe buried structures in soft materials.

3.5 Summary and Future Aspects

NR is a tool that has yet to capture the imaginations of bioelectrochemists despite several examples of successful implementation of the technique. NR provides a unique means to study soft interfaces under potential control and its remarkable sensitivity to water is ideally suited for structural studies of supported phospholi-

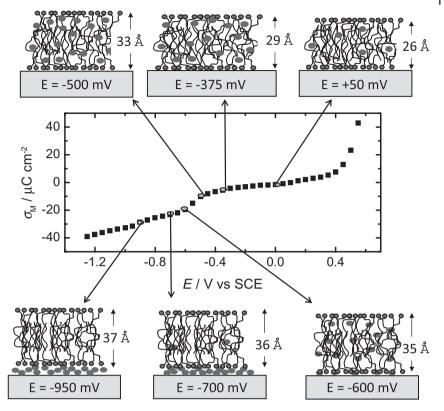


Figure 3.19 Pictorial description of the changes in the structure of a mixed DMPC– cholesterol bilayer deposited at the electrode surface as a function of the applied potential.

Small circles depict the lipid head groups, whereas the large circles represent solvent molecules. Figure adapted from [84].

pid layers. Many of the technical difficulties associated with interfacing potential control with a suitable cell design for NR measurements have been outlined and discussed herein. Several seminal papers emphasizing the ability of neutrons to probe electrified interfaces would establish the field as a unique tool to investigate aspects of adsorbed molecular films. The particular area where this breakthrough will most likely occur is the field of biomimetic research such as membrane bilayers supported on electrode surfaces where many opportunities remain unrealized. For example, isotopic variation of the inner and outer layers of a membrane would allow a user to discriminate between field-driven processes occurring in the proximal and distal leaflets of the bilayer. Probing the structure of buried components of biomimetic films such as solvent cushions and interstitial layers as a function of the potential drop across the membrane seems particularly well suited for NR studies. However, even among proponents of NR the reliance on model-dependent fitting is seen as a potential weakness of the technique, especially if there is no

pre-existing, physical intuition of the system under study. One can greatly reduce the uncertainties in fitting-based analysis by careful characterization of the substrate prior to NR measurements to limit the number of fitting parameters and by measuring multiple contrast systems. Model-independent means of extracting SLD profiles rely on obtaining phase information which is lost in a single measurement of reflectivity. Several methodologies to obtain phase information have been proposed including direct inversion [92, 93]; however, these are technically demanding and have yet to be implemented for electrochemical NR. A more fundamental problem is the truncation effects imposed by limited accessible momentum transfer space. The consequences of limited Q_z space measurements have been discussed in terms of the kinematic and model-fitting approaches but also lead to oscillations appearing in SLD profiles obtained through direct inversion analysis.

The growth of electrochemical NR will most likely develop along two fronts: (i) methodological advances made by physicists who work at dedicated neutron science facilities and (ii) increased exposure from users who wish to apply the method to their areas of expertise. These two avenues of development are discussed in turn below. Expanding the range of accessible momentum vectors is the critical element in terms of increasing the resolution of the method. In the last 15 years the lower limit of reflectivity has already been reduced from about 10⁻⁵ to about 10⁻⁷. This has been achieved by implementing higher cold neutron fluxes and better experimental sample design. Next-generation neutron reflectometers such as the Liquid Reflectometer at Oak Ridge National Laboratories in Tennessee, USA, and the advanced neutron diffractometer/reflectometer [94] at the NCNR can lower the level of measured reflectivites by a further one to two orders of magnitude. Lower measurable reflectivities lead to larger accessible Q_z domains and ultimately to more unique fits and higher spatial resolution. NR studies of electrified interfaces pose an exciting and challenging direction in the field of interfacial science. The method requires a great deal of patience and high levels of effective collaboration between physicists, physical chemists, synthetic chemists, and materials scientists. Though conducting experiments is often taxing because it requires "suitcase-science" it is hoped that this work will persuade both electrochemists and those interested in interfacial architecture that NR at electrified interfaces should be added to their repertoire of surface-sensitive techniques.

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4

Model Lipid Bilayers at Electrode Surfaces

Rolando Guidelli and Lucia Becucci

4.1

Introduction

Biological membranes are by far the most important electrified interfaces in living systems. They consist of a bimolecular layer of lipids (the lipid bilayer) incorporating proteins. Lipid molecules are "amphiphilic," that is, they consist of a hydrophobic section (the hydrocarbon tail) and a hydrophilic section (the polar head). In biological membranes the two lipid monolayers are oriented with the hydrocarbon tails directed toward each other and the polar heads turned toward the aqueous solutions that are in contact with the two sides of the membrane. The resulting lipid bilayer is a matrix that incorporates different proteins performing a variety of functions.

Biomembranes form a highly selective barrier between the inside and the outside of living cells. They are highly insulating to inorganic ions, and large electrochemical potential gradients can be maintained across them. The permeability and structural properties of biological membranes are sensitive to the chemical nature of the membrane components and to events that occur at the interface or within the bilayer. For example, biomembranes provide the environmental matrix for proteins that specifically transport certain ions and other molecules, for receptor proteins, and for signal transduction molecules.

4.2

Biomimetic Membranes: Scope and Requirements

In view of the complexity and diversity of the functions performed by the various proteins embedded in a biomembrane (the integral proteins), it has been found convenient to incorporate single integral proteins or smaller lipophilic biomolecules into experimental models of biological membranes, so as to isolate and investigate their functions. This serves to reduce complex membrane processes to well-defined interactions between selected proteins, lipids, and ligands. There is

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great potential for application of experimental models of biomembranes (so-called biomimetic membranes) for the elucidation of structure–function relationships of many biologically important membrane proteins. These proteins are the key factors in cell metabolism, for example, in cell–cell interactions, signal transduction, and transport of ions and nutrients. Because of this important function, membrane proteins are a preferred target for pharmaceuticals. Biomimetic membranes are also useful for the investigation of phase stability (e.g., lipid–lipid phase separation, lipid raft formation, lateral diffusion), protein–membrane interactions (e.g., receptor clustering and co-localization), and membrane–membrane processes such as fusion, electroporation, and intercellular recognitions. They are also relevant to the design of membrane-based biosensors and devices, and to analytical platforms for assaying membrane-based processes.

With only a few exceptions, metal-supported biomimetic membranes consist of a more or less complex architecture that includes a lipid bilayer. In order of increasing complexity, they can be classified into: solid-supported bilayer lipid membranes (sBLMs), tethered bilayer lipid membranes (tBLMs), polymer-cushioned bilayer lipid membranes (pBLMs), S-layer stabilized bilayer lipid membranes (ssBLMs), and protein-tethered bilayer lipid membranes (ptBLMs).

To incorporate integral proteins in a functionally active state, biomembrane models consisting of lipid bilayers should meet a number of requirements: (i) they should be robust enough for long-term stability, and be easily and reproducibly prepared; (ii) they should have the lipid bilayer in the liquid crystalline state, and such as to allow lateral mobility; (iii) they should have water (or, at least, a highly hydrated hydrophilic region) on both sides of the lipid bilayer; and (iv) they should be sufficiently free from pinholes and other defects that might provide preferential pathways for electron and ion transport across the lipid bilayer. Requirements (ii) and (iii) are necessary for the incorporation of integral proteins into the lipid bilayer in a functionally active state. In fact, integral proteins have a hydrophobic domain buried inside the biomimetic membrane, which must be sufficiently fluid to accommodate this domain. Often, they also have hydrophilic domains protruding by several nanometers outside the lipid bilayer. To avoid their denaturation and to promote their function, incorporation of integral proteins into biomimetic membranes must ensure that their extramembrane hydrophilic domains are accommodated in a hydrophilic medium on both sides of the lipid bilayer. Moreover, the transport of hydrophilic ions across a solid-supported lipid bilayer via ion channels or ion pumps is only possible if an aqueous or hydrophilic layer is interposed between the bilayer and the support. Requirement (iv) is needed to make the biomembrane model sufficiently blocking so as to enable characterization of ion channel or ion pump activity by electrochemical means without the disturbing presence of stray currents due to defects.

Apart from lipid molecules, the molecules that are most commonly employed for the fabrication of biomimetic membranes are "hydrophilic spacers" and "thiolipids." Hydrophilic spacers consist of a hydrophilic chain (e.g., a polyethyleneoxy or oligopeptide chain) terminated at one end with an anchor group for tethering to a support and, at the other end, with a hydrophilic functional group (e.g., a

Figure 4.1 (a) Structure of a widely adopted thiolipid, called DPTL [1]. (b) Structure of the corresponding hydrophilic spacer (TEGL), in which the two phytanyl chains are replaced by a hydroxyl group.

hydroxyl group). Sulfhydryl or disulfide groups are employed as anchor groups for tethering to metals such as gold, silver, or mercury (Figure 4.1b); methyl-, methyloxy- or chloride-substituted silane groups are used for tethering to glass, quartz, silica, or mica. The latter supports are nonconducting and cannot be investigated by electrochemical techniques. Hydrophilic spacers serve to separate the lipid bilayer from a solid support, to compensate for surface roughness effects, to prevent any incorporated peptides or proteins from touching the support surface (thus avoiding loss of their functionality due to denaturation), and to provide an ionic reservoir underneath the lipid bilayer.

Thiolipids differ from hydrophilic spacers in that the hydrophilic chain is covalently linked to one or, more frequently, two alkyl chains at the opposite end with respect to the anchor group. A convenient and widely used thiolipid, first employed by Schiller *et al.* [1] and denoted by the abbreviation DPTL, consists of a tetraethyleneoxy hydrophilic chain covalently linked at one end to a lipoic acid residue, for anchoring to a metal via a disulfide group, and bound at the other end via ether linkages to two phytanyl chains (Figure 4.1a). The alkyl chains simulate the hydrocarbon tails of a lipid molecule and provide one half of the lipid bilayer to the biomimetic membrane. When tethered to a support, hydrophilic spacers expose to the bulk aqueous phase a hydrophilic surface, while thiolipids expose a hydrophobic surface. Clearly, lipid bilayers formed on top of hydrophilic spacers are noncovalently linked to them and can be regarded as "freely suspended." Conversely, lipid monolayers self-assembled on top of thiolipid monolayers form lipid bilayers that are tethered to the support.

Before discussing advantages and disadvantages of these systems, we will briefly describe the electrochemical technique that is commonly employed for their investigation, namely electrochemical impedance spectroscopy (EIS), as well as some fabrication methodologies. In fact, attention is focused on those biomimetic membranes that are amenable to investigation by electrochemical methods. Biomimetic membranes that are investigated exclusively by nonelectrochemical surface-sensitive techniques, such as those formed on insulating supports (e.g., glass, mica, quartz, silica, etc.), are outside the scope of this chapter.

4.3 Electrochemical Impedance Spectroscopy

Many membrane proteins are "electrogenic," that is, translocate a net charge across a membrane. Consequently, it is possible to monitor their function directly by measuring the current flowing along an external electrical circuit upon their activation. The techniques of choice for these measurements are EIS and potential-step chronoamperometry or chronocoulometry, because the limited volume of the ionic reservoir created by a hydrophilic spacer in solid-supported biomimetic membranes cannot sustain a steady-state current.

EIS applies an AC voltage of given frequency to the system under study and measures the resulting current that flows with the same frequency. Both the amplitude of the AC current and its phase shift with respect to the AC voltage are measured. The frequency is normally varied gradually from 10⁻³ to 10⁵ Hz. To interpret measured impedance spectra, it is necessary to compare them with the electrical response of an "equivalent circuit" assembled from resistors and capacitors, capable of simulating the system under investigation. In general, a metalsupported self-assembled mono- or multilayer can be regarded as consisting of a series of slabs with different dielectric properties. When ions flow across each slab, they give rise to an ionic current $J_{\text{ion}} = \sigma E$, where E is the electric field and σ is the conductivity. Ions may also accumulate at the boundary between contiguous dielectric slabs, causing a discontinuity in the electric displacement vector $D = \varepsilon E$, where ε is the dielectric constant. Under AC conditions, the accumulation of ions at the boundary of the dielectric slabs varies in time, and so does the electric displacement vector, giving rise to a capacitive current $I_c = dD/dt$. The total current is, therefore, given by the sum of the ionic current and of the capacitive current. In this respect, each dielectric slab can be simulated by a parallel combination of a resistance, accounting for the ionic current, and of a capacitance, accounting for the capacitive current, namely by an "RC mesh." Accordingly, the impedance spectrum of a self-assembled layer can be simulated by a series of RC meshes.

Application of an AC voltage of amplitude V and frequency f to a pure resistor of resistance R yields a current of equal frequency f and of amplitude V/R, in phase with the voltage. Conversely, application of the AC voltage to a pure capacitor of capacitance C yields a current of frequency f and amplitude $2\pi fC$, out of phase by $-\pi/2$ with respect to the voltage, that is, in quadrature with it. This state of affairs can be expressed by stating that the admittance Y of a resistance element equals 1/R, while that of a capacitance element equals $-i\omega C$, where $\omega = 2\pi f$ is the angular frequency and i is the imaginary unit. More generally, in an equivalent circuit consisting of resistances and capacitances, Y is a complex quantity, and the impedance Z is equal to 1/Y, by definition. Hence, Z equals R for a resistance element, and $i/\omega C$ for a capacitance element. Impedance spectra are often displayed on a Bode plot, namely a plot of $\log |Z|$ and phase angle Φ against $\log f$, where |Z| is the magnitude of the impedance. This plot is relatively featureless. Over the frequency range where the equivalent circuit is exclusively controlled by a capacitance, the $\log |Z|$ versus $\log f$ plot is a linear segment of slope -1, and the phase angle equals

 -90° . Conversely, in the case of pure control by a resistance, the $\log |Z|$ versus $\log f$ plot is horizontal and Φ equals zero. Another commonly used plot, called a Nyquist plot, reports the quadrature component, Z'', of the impedance Z against the corresponding in-phase component, Z'. In the case of a single RC mesh, the Nyquist plot yields a semicircle of diameter R and center of coordinates (R/2, 0). The angular frequency, ω , at the maximum of this semicircle equals the reciprocal of the time constant RC of the mesh. In the presence of a series of RC meshes, their time constants are often close enough to cause the corresponding semicircles to overlap partially. In this case, if the mesh of highest time constant has also the highest resistance, R_1 , as is often the case, then the Nyquist plot of the whole impedance spectrum exhibits a single well-formed semicircle, R_1 in diameter. The semicircles of the remaining meshes are compressed in a very narrow area close to the origin of the Z'' versus Z' plot, and can be visualized only by enlarging this area. Therefore, the Nyquist plot of the whole spectrum is conveniently employed if one is interested in pointing out the resistance R_1 of the dielectric slab of highest resistance.

To better visualize all semicircles, we have found it convenient to represent impedance spectra on a $\omega Z'$ versus $\omega Z''$ plot [2]. Henceforth, this plot will be briefly referred to as an "M plot," since $\omega Z'$ and $\omega Z''$ are the components of the modulus function M. It is possible to demonstrate that a single RC mesh yields a semicircle even on a $\omega Z'$ versus $\omega Z''$ plot; its diameter equals C^{-1} and its center has coordinates $(C^{-1}/2, 0)$. Moreover, ω at the maximum of the semicircle is again equal to the reciprocal of the time constant RC of the mesh. While ω decreases along the positive direction of the abscissas on a Nyquist plot, it increases on an M plot. Therefore, for a series of RC meshes, the last semicircle on the M plot is characterized by the lowest time constant. This is, unavoidably, the semicircle simulating the solution that bathes the self-assembled film, due to its very low capacitance. Figure 4.2 shows the M plot for a biomimetic membrane consisting of a DPTL monolayer anchored to a mercury electrode, with a diphytanoylphosphatidylcholine (DPhyPC) monolayer on top of it; the tBLM incorporates the ion carrier valinomycin, a hydrophobic depsipeptide that cages a desolvated potassium ion, shuttling it across the lipid bilayer [2]. The plot shows four partially overlapping semicircles. The solid curve is the best fit of the plot by an equivalent circuit consisting of four RC meshes in series. The semicircles overlap only to a moderate extent, thus allowing their straightforward deconvolution. This is due to an appreciable difference between the time constants of the four RC meshes, which are evenly distributed over a frequency range covering seven orders of magnitude. Proceeding along the positive direction of the abscissas, the four semicircles are ascribable to the lipoid acid residue, the tetraethyleneoxy hydrophilic spacer, the lipid bilayer moiety, and the aqueous solution bathing the lipid bilayer.

A plot that has been frequently adopted in the literature to display an impedance spectrum as rich in features as the M plot is the Y'/ω versus Y''/ω plot, sometimes called a Cole–Cole plot [3–5]. Here Y' and Y'' are the in-phase and quadrature component of the electrode admittance. However, it can be shown that this plot yields a semicircle for a series combination of a resistance and a capacitance, and

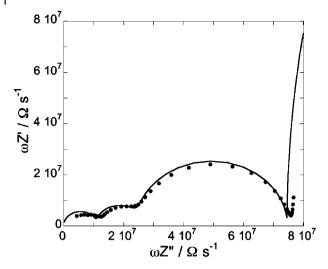


Figure 4.2 Plot of $\omega Z'$ against $\omega Z''$ (M plot) for a mercury-supported DPTL-DPhyPC bilayer incorporating gramicidin from its 1×10^{-7} M solution in aqueous 0.1 M KCl at

-0.600 V vs. Ag/AgCl (0.1 M KCl) [2]. The solid curve is the best fit of the impedance spectrum by four *RC* meshes in series.

not for their parallel combination. For such a series combination, the Cole–Cole plot yields a semicircle of diameter C and center of coordinates (C/2, 0). Here too, ω at the maximum of the semicircle equals 1/RC. Strictly speaking, a Cole–Cole plot is not suitable for verifying the fitting of an impedance spectrum by a series of RC meshes. Thus, if we force a Y'/ω versus Y''/ω plot to fit a single RC mesh, we obtain "formally" a radius of the semicircle that is given by (C/2)(1 + $1/\omega^2R^2C^2$). Consequently, the Cole–Cole plot for a single RC mesh yields a semicircle of diameter C only for ω values high enough to make $\omega^2R^2C^2\gg 1$. Figure 4.3 shows the Cole–Cole plot for the same impedance spectrum displayed on the M plot of Figure 4.2. It is apparent that the fitting of the experimental spectrum by a series of four RC meshes shows appreciable deviations at the lower frequencies, which correspond to the higher values of Y''/ω .

4.4 Formation of Lipid Films in Biomimetic Membranes

Methodologies for the fabrication of biomimetic membranes vary somewhat from one biomimetic membrane to another. However, a number of experimental procedures for the formation of lipid monolayers and bilayers on solid supports are common to several biomimetic membranes. The most popular procedures are vesicle fusion, Langmuir–Blodgett and Langmuir–Schaefer transfers, and rapid solvent exchange. The formation of lipid monolayers and bilayers on gold and

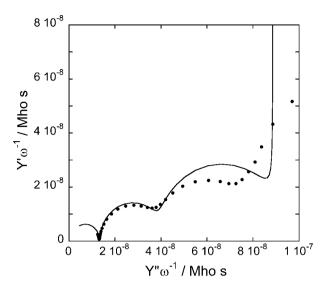


Figure 4.3 Plot of Y'/ω against Y''/ω (Cole–Cole plot) for the same tBLM as in Figure 4.2. The solid curve is the best fit of the impedance spectrum by four *RC* meshes in series.

silver substrates is commonly monitored by surface plasmon resonance (SPR). Surface plasmons are collective electronic oscillations in a metal layer, about 50 nm in thickness, excited by photons from a laser beam [6]. The beam is reflected by the back surface of the metal layer, while its front surface supports a dielectric film (e.g., a lipid bilayer), usually in contact with an aqueous solution. The evanescent electromagnetic field thereby generated in the metal layer can couple with the electronic motions in the dielectric film. The intensity of the electromagnetic field associated with surface plasmons has its maximum at the metal-dielectric film surface and decays exponentially into the space perpendicular to it, extending into the metal and the dielectric. This makes SPR a surface-sensitive technique particularly suitable for the measurement of the optical thickness of ultrathin films adsorbed on metals. As a rule, the incident angle of the laser beam is varied with respect to the back surface of the metal layer, and the reflected light intensity is measured. At a certain angle of incidence there exists a resonance condition for the excitation of surface plasmons, which causes the energy of the incident laser light to be absorbed by the surface plasmon modes and the reflectivity to attain a minimum. The angular position of the minimum of the SPR reflectivity curves (i.e., the curves of reflectivity versus incident angle) is critically dependent on the thickness of the layer adsorbed on the support surface, and is used to estimate such a thickness. Both SPR and EIS allow an evaluation of film thickness, based on a reasonable estimate of the refractive index of the film in the case of SPR, or of its dielectric constant in the case of EIS. However, it must be borne in mind that the two techniques are sensitive to different features of a film.

4.4.1

Vesicle Fusion

Vesicles (or, more precisely, unilamellar vesicles) are spherical lipid bilayers that enclose an aqueous solution. The procedure for vesicle fusion consists of adsorbing and fusing small unilamellar vesicles (SUVs; 20 to 50 nm in diameter) on a suitable substrate from their aqueous dispersion. If the substrate is hydrophilic, vesicle fusion gives rise to a lipid bilayer by rupture of the vesicles and their "unrolling" and spreading onto the substrate, as shown in Figure 4.4b. Conversely, if the substrate is hydrophobic, a lipid monolayer with the hydrocarbon tails directed toward the substrate is formed by rupture of the vesicles, splitting of the vesicular membrane into its two monomolecular leaflets and their spreading [7], as shown in Figure 4.4a. This is confirmed by the different increase in thickness following vesicle fusion on a hydrophobic substrate (2–2.5 nm) with respect to a hydrophilic substrate (4.5–5 nm), as estimated by SPR [8].

The kinetics of vesicle fusion, followed by monitoring the position of the minimum of the SPR reflectivity curves, depends on the composition and molecular shape of the vesicular lipids and on the nature of the substrate. As a rule, bilayer formation by vesicle unrolling onto a hydrophilic surface is faster than monolayer formation by vesicle fusion onto a hydrophobic surface. This is probably due to the fact that the processes involved in forming a planar bilayer starting from a vesicular bilayer are considerably less complex than those involved in forming a planar monolayer [8, 9].

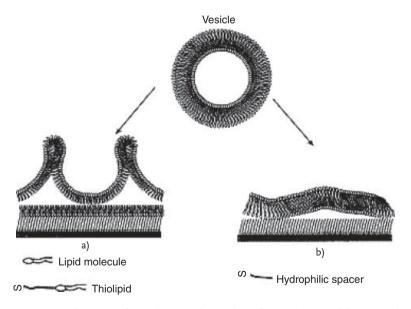


Figure 4.4 Schematics of (a) splitting and spreading of a vesicle on a solid-supported thiolipid monolayer and (b) unrolling and spreading of a vesicle on a solid-supported hydrophilic spacer.

It has been suggested that the initial rapid stage of vesicle adsorption on hydrophilic surfaces is controlled by vesicle adsorption at free sites of the surface, according to a Langmuirian-type behavior. A second, later stage is ascribed to vesicle unrolling and spreading processes. Among hydrophilic substrates, those allowing the formation of lipid bilayers by vesicle fusion more easily are freshly oxidized surfaces of silica, glass, quartz, and mica [10]. However, hydrophilicity is a necessary but not a sufficient condition to promote vesicle fusion. Surfaces of oxidized metals and metal oxides (e.g., TiO2, platinum, and gold) allow adsorption of intact vesicles but resist the formation of bilayers, presumably due to weak surface interactions [11]. Electrostatic, van der Waals, hydration, and steric forces cause a noncovalently supported lipid bilayer to be separated from a solid surface by a nanometer layer of water [12]. This water layer prevents the support from interfering with the lipid bilayer structure, thus preserving its physical attributes, such as lateral mobility of the lipid molecules. The quartz crystal microbalance with dissipation monitoring (QCM-D) has proved quite valuable for monitoring the macroscopic features of vesicle deposition [13]. Use of the QCM-D permitted confirmation of the formation of lipid bilayers on silica and the adsorption of intact vesicles on oxidized gold [14].

Initial adsorption of vesicles on hydrophobic surfaces is energetically disfavored, due to the presence of the hydrophilic polar heads on the outer surface of the vesicular membrane. Therefore, the vesicular membrane must split to allow its inner hydrophobic tails to get in contact with the hydrophobic surface. A possible pathway for vesicle fusion involves vesicle splitting, unrolling, and spreading on the hydrophobic surface, as shown in Figure 4.4a. The kinetics of vesicle fusion on the hydrophobic surface of gold-supported alkanethiol self-assembled monolayers was followed by SPR [15]. In the initial stage, the adsorbed layer thickness d increases linearly with the square root of time t, denoting control by vesicle diffusion to the surface according to Fick's first law. In a second stage, d increases roughly linearly with $\log t$. Finally, the time dependence of d becomes typical of an adsorption process on an almost fully occupied surface. The curve of the surface coverage by vesicles against time was also monitored by SPR at different vesicle concentrations [7]; it was fitted to an equation practically identical to that derived for an electrode process controlled by diffusion and by a heterogeneous electron transfer step [16]. The resulting kinetic constant was ascribed to some surface reorganization of the vesicles.

When vesicles fuse spontaneously on a support, some of them may remain only partially fused, or even intact in the adsorbed state. When the surface density of vesicles is sufficiently high, their presence is revealed by an anomalously high thickness attained by the lipid film after vesicle fusion, as monitored by SPR. This phenomenon is particularly evident with supports consisting of mixtures, with comparable molar ratios, of two different molecules exposing to the bulk aqueous phase hydrophobic alkyl chains and hydrophilic functional groups, respectively [5, 17]. With very high vesicle concentrations (5 mg ml⁻¹), monolayer thicknesses greater than 8 nm were reported on hydrophobic surfaces [9]. Thus, fusion of SUVs onto a binary mixture of hydrophobic cholesteryl-terminated molecules and

hydrophilic 6-mercaptohexanol molecules yields atomic force microscopy (AFM) images showing heightened areas; their diameter being close to that of the SUVs denotes the presence of adsorbed vesicles [3]. The presence of a membrane protein, such as cytochrome bo3, in these vesicles increases the number density and the size of the heightened structures ascribable to adsorbed vesicles. Analogously, fusion of large unilamellar vesicles (LUVs) on a hydrophobic support exposing dipalmitoylphosphatidylethanolamine alkyl chains to the aqueous phase yields tapping-mode AFM images with a number of dome-shaped structures [18]. Some of these structures, whose diameter is close to that of the LUVs, tend to disappear after an hour, due to complete vesicle fusion. However, if the vesicles contain the membrane protein Na,K-ATPase, all the domelike structures are stable even after three hours. In general, adsorption of proteoliposomes (namely, vesicles incorporating membrane proteins) on hydrophobic surfaces prevents their complete spreading and fusion, due to the presence of protein molecules with extramembrane domains and to the hydrophilicity of the outer polar heads of the proteoliposomes [19, 20].

The presence of heightened areas in metal-supported lipid films formed by fusion of vesicles labeled with a fluorophore can also be monitored by fluorescence microscopy. In fact, the energy of a photoexcited fluorophore is transferred non-radiatively to the metal support (quenching) if the fluorophore is at a distance from the metal surface shorter than a critical transfer distance (the Foerster radius). This critical distance amounts to about 20–30 nm on gold. Therefore, unquenched bright spots in the fluorescence microscopy images of gold-supported lipid films obtained by fusion of fluorophore-labeled vesicles mark the presence of heightened areas ascribable to adsorbed or hemifused vesicles [21].

4.4.2 Langmuir-Blodgett and Langmuir-Schaefer Transfer

Another procedure for forming a lipid monolayer on a hydrophobic substrate or a lipid bilayer on a hydrophilic substrate makes use of a Langmuir trough equipped with a movable barrier. By spreading a lipid dissolved in an organic solvent on the surface of the aqueous electrolyte contained in the trough and by allowing the solvent to evaporate, a lipid film is formed at the air-water interface. This film is compressed with the movable barrier until it is brought close to the liquid crystalline state by adjusting the surface pressure. To form a lipid monolayer on a hydrophobic slab, the slab is immersed vertically through the lipid monolayer. This brings the hydrocarbon tails of the lipid monolayer, turned toward the air, in direct contact with the hydrophobic surface of the slab, which remains coated by a lipid monolayer (Figure 4.5a). This technique, called Langmuir-Blodgett (L-B) transfer, is also used to form a lipid bilayer on a hydrophilic slab [22, 23]. In this case, the slab is initially immersed vertically through the lipid monolayer into the trough. No significant change in surface pressure is observed at this stage. The slab is then withdrawn at a speed slow enough to permit water to drain from the surface. During the withdrawal, the polar heads of the lipid monolayer are turned

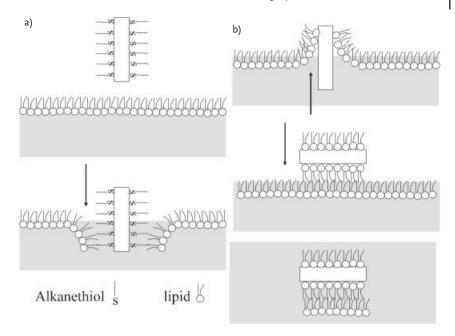


Figure 4.5 (a) Formation of a lipid monolayer on a hydrophobic alkanethiol-coated slab by L-B transfer. (b) Formation of a lipid bilayer on a hydrophilic slab by L-B transfer and subsequent L-S transfer.

toward the surface of the hydrophilic slab, giving rise to noncovalent self-assembly (Figure 4.5b). A resulting decrease in the surface pressure of the lipid monolayer at the air—water interface is prevented by reducing its area with the moving barrier. A further lipid monolayer is then self-assembled on top of the first one by Langmuir—Schaefer (L-S) transfer. In practice, the lipid-coated slab is brought into horizontal contact with the lipid monolayer at the air—water interface, after compressing it at a preset surface pressure. Finally, the slab covered by the lipid bilayer is detached from the aqueous subphase. The combination of L-B and L-S transfers is used principally for lipid bilayer deposition on hydrophilic surfaces of glass, mica, quartz, polymeric materials, graphite, and, to a minor extent, metals such as gold [24].

An advantage of L-B and L-S transfers over vesicle fusion is that the strict composition of a mixed lipid monolayer on the trough surface is maintained after its transfer. Conversely, the composition of a mixed lipid layer on a substrate does not necessarily correspond to that of the vesicles used to form it, and varies strongly with the history of the sample preparation. On the other hand, an advantage of vesicle fusion over L-B and L-S transfers is that vesicles may easily incorporate membrane proteins, forming proteoliposomes. To this end, membrane proteins are usually stabilized in detergent micelles and then incorporated into the lipid vesicles upon removing the detergent molecules by dialysis. Fusion of

proteoliposomes on a metal-supported thiolipid monolayer may then cause the insertion of the membrane proteins in the resulting lipid bilayer.

4.4.3

Rapid Solvent Exchange

A further procedure for depositing a lipid monolayer is rapid solvent exchange. This method is usually employed to self-assemble a lipid monolayer onto a hydrophobic surface exposing alkyl chains to the bulk aqueous phase. The method involves placing a small amount of lipid dissolved in a water-miscible solvent, such as ethanol, onto the hydrophobic substrate and incubating it for a few minutes [25, 26]. The ethanol solution is then vigorously displaced in a few seconds by a large excess of an aqueous buffer solution, taking care to avoid the formation of air bubbles at the surface. During the lipid addition and the subsequent rinsing with the aqueous solution, the lipid self-assembles to form a lipid monolayer and the excess lipid is rinsed away.

4.4.4 Fluidity in Biomimetic Membranes

A fundamental property of biological membranes is the long-range lateral mobility of the lipid molecules. The fluidity of the plasma membrane should be preserved in supported biomimetic membranes. Free movement of lipid molecules enables the biomimetic membrane to react to the presence of proteins, charges, and physical forces in a dynamic and responsive manner. A satisfactory fluidity allows biomimetic membranes to reorganize upon interaction with external perturbations, mimicking the functionality of living cell membranes. In particular, lateral mobility enables a biomimetic membrane to incorporate bulky membrane proteins from their detergent solutions by making space for them; it also determines the spontaneous separation of the components of a lipid mixture (demixing), giving rise to the formation of important lipid microdomains, called lipid rafts.

The fluidity and lateral mobility of biomimetic membranes can be characterized quantitatively by AFM [27] and fluorescence microscopy. Fluorescence recovery after photobleaching (FRAP) is one of the most popular ways of measuring molecular diffusion in membranes [28]. It relies on introducing a small amount of fluorescent probe molecules, usually covalently bound to lipids, into the membrane. A short burst of intense excitation light is projected onto the membrane, destroying the fluorescence of the fluorophore molecules in a well-defined spot, a photochemical process called photobleaching. The gradual fluorescence recovery within the given spot is followed as a function of time, thus permitting an estimate of the lipid diffusion coefficient. In supported membranes, this coefficient typically ranges from 1 to $10\,\mu\text{m}^2\,\text{s}^{-1}$. When applied to lipid bilayers self-assembled on smooth supports such as silica, glass, quartz, mica, or indium tin oxide (ITO), FRAP usually confirms a satisfactory lateral mobility of lipid molecules [10, 29]. Conversely, biomimetic membranes consisting of a thiolipid monolayer tethered

to a gold electrode, with a self-assembled lipid monolayer on top of it, do not exhibit lateral mobility. This is also true for the distal lipid monolayer noncovalently linked to the thiolipid monolayer, no matter if obtained by vesicle fusion or by L-B transfer [5, 21]. A biomimetic membrane consisting of a hydrophilic spacer tethered to gold, with a lipid bilayer formed on top of it, was reported to exhibit FRAP, if the noncovalently bound lipid bilayer was formed by L-B and L-S transfers [5]. Fluorescence recovery in the gold-supported biomimetic membrane could only be observed for a short period of time, due to the gradual energy transfer from the fluorophore molecules to gold (quenching). No fluorescence recovery could be observed on forming the lipid bilayer on top of the hydrophilic spacer by vesicle fusion. Evidently, the unavoidable presence of adsorbed and hemifused vesicles prevents the lateral mobility of the lipid molecules of the distal monolayer.

As distinct from solid supports such as gold or silver, mercury imparts lateral mobility to lipid monolayers directly self-assembled on its surface, because of its liquid state. This is demonstrated by rapid spontaneous phase separation, with microdomain formation, in a lipid mixture monolayer self-assembled on top of a DPTL thiolipid monolayer tethered to a mercury microelectrode [30]. The presence of microdomains was directly verified from the images of the distal lipid monolayer obtained using two-photon fluorescence lifetime imaging microscopy.

4.5 Various Types of Biomimetic Membranes

Biomimetic membranes comprising a lipid bilayer are discussed here in the order of approximately increasing complexity.

4.5.1

Solid-Supported Bilayer Lipid Membranes

The term sBLM or, simply, "solid-supported membrane," is commonly used to denote a biomimetic membrane consisting of a lipid bilayer in direct contact with a solid support. These biomimatic membranes are typically formed on a hydrophilic solid support by immersing the support in an aqueous dispersion of SUVs, which slowly rupture and spread on the surface of the support. Alternatively, they can be formed by L-B and L-S transfers. When appropriately formed, these sBLMs are separated from the support surface through a hydration layer of water, estimated to be between 6 and 15 Å thick [31, 32]. Several theoretical and experimental investigations of water near polar hydrophilic surfaces suggest that it is more ordered than bulk water, with higher viscosity and lower dielectric constant [33, 34]. In accordance with these predictions, the characteristics of the lipid bilayer have been found to be structurally coupled with the support properties, such as its charge [35], wettability [36], and topography [37]. The lubrification effect of the water layer imparts a significant long-range lateral mobility to the lipid bilayer. However, significant frictional coupling between the bilayer and the underlying substrate

slows down lateral diffusion, which may be accompanied by a breakdown of the two-dimensional fluid nature of the membrane.

The above biomembrane models are not suitable for studying the function of integral proteins. In fact, these proteins have hydrophilic domains protruding outside the lipid bilayer. To avoid their denaturation and to promote their function, the incorporation of integral proteins into biomembrane models must ensure that their protruding hydrophilic domains are accommodated in a hydrophilic medium on both sides of the lipid bilayer. Not surprisingly, embedded membrane-spanning proteins usually show no lateral diffusion, because of their interaction with the substrate [38], even though some of them maintain their function if their active site is far from the solid substrate.

The majority of sBLMs are formed on nonconducting supports such as silica, glass, or mica [10, 23, 29, 38, 39]. In what follows, a few examples of sBLMs investigated by electrochemical techniques will be examined. Phospholipid bilayers on Au(111) single-crystal faces have been prepared by Lipkowski and coworkers either by vesicle fusion [40] or by L-B and L-S transfers [24]. The latter procedure yields bilayers with a higher packing density and with smaller tilt angles of the alkyl chains with respect to the surface normal. These bilayers have been characterized by charge density measurements, photon polarization modulation infrared reflection absorption spectroscopy (PM-IRRAS) [24, 40], and neutron reflectivity (NR) [41]. The minimum differential capacitance of these bilayers is about 2µF cm⁻², and is attained at charge densities $\sigma_{\rm M}$ on the metal higher than $-8\,\mu{\rm C\,cm}^{-2}$. The differential capacitance being greater than that of a solvent-free BLM denotes the presence of a number of defects, while NR reveals the presence of water molecules within the lipid bilayer. As $\sigma_{\rm M}$ becomes more negative than $-8\,\mu{\rm C\,cm}^{-2}$, the lipid bilayer starts to detach from the electrode, but remains in close proximity to the electrode surface. In fact, NR indicates that the lipid bilayer is suspended on a thin cushion of the aqueous electrolyte, which screens the metal charge and contains a large fraction of the overall interfacial potential difference; consequently, the electric field across the lipid bilayer becomes weak. This suspended lipid bilayer is essentially defect-free and its structure resembles that of lipid bilayers supported by a quartz surface. The PM-IRRAS data demonstrate that the carbonyl and phosphate groups are more hydrated when the bilayer is adsorbed on the electrode surface at $\sigma_{\rm M} > -8 \,\mu{\rm C\,cm}^{-2}$ than when it is detached from the electrode at more negative charge densities [40]. NR shows that a significant amount of water enters the polar head region of the lipid bilayer when it is in contact with the metal surface. The detachment of the lipid bilayer from the electrode surface is accompanied by a decrease in the tilt angle of the alkyl chains with respect to the surface normal from about 55° to 35°, with a resulting increase in bilayer thickness. The PM-IRRAS measurements indicate that, as long as the lipid bilayer remains attached to the metal surface, changes in the local electric field by several orders of magnitude have only a small effect on the orientation of the phospholipid molecules [42]. Electrochemical scanning tunneling microscopy images of adsorption of dimyristoylphosphatidylcholine (DMPC) on Au(111) show that the lipid molecules are initially adsorbed flat, with the alkyl chains oriented parallel to the surface [43]; the resulting ordered monolayer resembles that formed by alkanes. With time, the molecules reorient and the monolayer is transformed into a hemimicellar film. In the presence of a high vesicle concentration in solution, the hemimicellar state is transformed further into a bilayer.

A sBLM formed by fusing vesicles consisting of a mixture of 50 mol% cholesterol and 50 mol% dihexadecyldimethylammonium bromide on boron-doped (p-type) silicon covered with native oxide was investigated by EIS [44]. The positively charged lipid was chosen to favor vesicle spreading on the negatively charged silicon/SiO₂ surface. The impedance spectrum was simulated by an equivalent circuit consisting of two RC meshes in series, with the resistance R_{Ω} of the aqueous solution in series with them. One RC mesh simulates the electrode, regarded as a combination of the silicon space-charge region, the SiO₂, and the thin water layer interposed between the semiconductor surface and the lipid bilayer; the other RC mesh simulates the lipid bilayer. The resistance and capacitance of the electrode amount to about $19 \,\mathrm{M}\Omega\,\mathrm{cm}^2$ and $2.2 \,\mu\mathrm{F}\,\mathrm{cm}^{-2}$, respectively; those of the lipid bilayer to $0.98\,\mathrm{M}\Omega\,\mathrm{cm}^2$ and $0.75\,\mu\mathrm{F}\,\mathrm{cm}^{-2}$, respectively. Incorporation of gramicidin from an aqueous solution decreases the resistance of the lipid bilayer by more than one order of magnitude in the presence of Na⁺ ions, and even more in the presence of K⁺ ions. This agrees with the cation selectivity scale of gramicidin. A sBLM formed by fusing DMPC-cholesterol vesicles on the optically transparent semiconductor ITO was investigated by EIS [45]. The vesicles also contained 10 mol% of a positively charged lipid to favor their interaction with the negatively charged ITO surface. The impedance spectrum was analyzed by the same equivalent circuit used for the silicon/SiO₂ substrate. Upon incorporating gramicidin, the resistance $R_{\rm m}$ of the lipid bilayer was strongly reduced in the presence of a Na⁺ salt with an organic anion not permeating membranes, whereas its capacitance did not change markedly. Conversely, $R_{\rm m}$ was practically unaffected in the presence of a Cl⁻ salt with an organic cation not permeating membranes. This experiment confirms the functional activity and ion selectivity of gramicidin, which is known to be highly selective toward monovalent inorganic cations. The outer membrane proteins OmpF and OmpA from Escherichia coli were reconstituted in vesicles that were fused on ITO [45]. In an aqueous solution of NaCl, the resistance of the ITOsupported lipid bilayer was strongly reduced by incorporating OmpF, whereas it was only slightly decreased by incorporating OmpA. Such a behavior was explained by the fact that only OmpF is a pore-forming protein, whereas OmpF does not form pores. Note that the functional activity of OmpF was preserved because, on the cytosolic side of the bacterial membrane, it has no extramembrane domain that might be endangered by direct contact with the ITO surface.

4.5.2 Tethered Bilayer Lipid Membranes

As a rule, tBLMs refer to architectures in which the lipid bilayer is separated from the support through a monomolecular layer tethered to the support via a sulfhydryl or disulfide group (for gold, silver, or mercury supports) or via a silane group (for silica and glass supports). The monolayer interposed between the support surface and the lipid bilayer should have a well-defined composition and geometrical arrangement, as distinct from pBLMs. In general, tBLMs can be classified on the basis of the nature of the molecules composing the tethered monolayer, as follows: (i) spacer-based tBLMs; (ii) thiolipid-based tBLMs; and (iii) thiolipid-spacer-based tBLMs. We will examine these three types separately.

4.5.2.1 Spacer-Based tBLMs

The spacer may consist of an alkanethiol, functionalized with a hydrophilic group (e.g., hydroxyl, carboxyl, or amino group) at the opposite end of the alkyl chain with respect to the sulfhydryl group. More frequently, it consists of a thiolated or disulfidated hydrophilic chain. The spacer is tethered to a support via its sulfhydryl or disulfide group and a lipid bilayer is self-assembled on top of it, often by vesicle fusion. The interior of ω -functionalized alkanethiol monolayers is hydrophobic, as opposed to that of thiolated hydrophilic chains. However, even the latter spacers do not necessarily ensure a satisfactory hydration if they are too closed packed. In both cases, a thin water layer, about 1 nm thick, is normally interposed between the spacer monolayer and the lipid bilayer, as in the case of sBLMs; this provides a small ionic reservoir for channel-forming peptides incorporated in the lipid bilayer moiety. Since the lipid bilayer is not covalently linked to the support, it is expected to be sufficiently fluid to accommodate relatively bulky membrane proteins, unless they have large extramembrane domains on both the cytosolic and the extracellular side. The lateral mobility of lipid molecules on a hydrophilic spacer was verified by FRAP in about 10 min on smooth gold surfaces [5], but a sufficient fluidity may be verified at longer times even on rougher surfaces (Becucci et al., unpublished data). The capacitance of the lipid bilayer in these tBLMs is close to 1 µF cm⁻², but its resistance assumes relatively low values of the order of $0.5 \,\mathrm{M}\Omega\,\mathrm{cm}^2$.

A tBLM consisting of 3-mercaptopropionic acid (MPA) tethered to gold, with a bilayer of the positively charged lipid dimethyldioctadecylammonium bromide on top, was used to incorporate gramicidin [46]. The lipid bilayer was stabilized by electrostatic interactions with the spacer. To this end, a pH of 8.6 was used to completely deprotonate the MPA monolayer, and the ionic strength of the solution was kept sufficiently low by using 1,1-valent electrolytes of concentration less than 50 mM. The ion-channel activity of gramicidin was verified by EIS. The same tBLM was used to incorporate the 25 kDa *Clavibacter* ion channel, which exhibits anion selectivity [47].

Another procedure for preparing spacer—bilayer assemblies on electrodes consists in anchoring a polyethyleneoxy hydrophilic spacer to a hanging mercury drop electrode via a terminal sulfhydryl group by immersing the mercury drop in an ethanol solution of the spacer for about 20 min [48]. After extracting the spacer-coated mercury drop from the ethanol solution, it is slowly brought into contact with a lipid film previously spread on the surface of an aqueous electrolyte, by taking care to keep the drop neck in contact with the lipid reservoir. This disposition allows a free exchange of lipid material between the lipid reservoir on the

surface of the aqueous electrolyte and the spacer-coated drop. This procedure gives rise to the formation of a lipid bilayer in contact with the spacer-coated drop, by exploiting the spontaneous tendency of a lipid film to form a bilayer when interposed between two hydrophilic phases. The fluidity of the tBLM was tested by recording the cyclic voltammogram for the electroreduction of ubiquinone-10 incorporated in the lipid bilayer. The voltage-dependent ion channel activity of melittin incorporated in the tBLM was verified by EIS. A similar mercury-supported tBLM was fabricated by using a thiolated hexapeptide molecule with a high tendency to form a 3_{10} -helical structure as a spacer [49].

4.5.2.2 Thiolipid-Based tBLMs

A thiolipid molecule consists of a hydrophilic polyethyleneoxy or oligopeptide hydrophilic chain terminated at one end with a sulfhydryl or disulfide group for anchoring to the support and covalently linked at the other end to two alkyl chains simulating the hydrocarbon tails of a lipid (Figure 4.1a).

Gold-Supported Thiolipid-Based tBLMs Polyethyleneoxy-based thiolipid monolayers and (polyethyleneoxy-based thiolipid)-phospholipid bilayers tethered to gold have been characterized by SPR, EIS, and cyclic voltammetry, and the synthesis of the thiolipids has been described [50-52]. The cross-sectional area of a hydrophilic polyethyleneoxy chain is smaller than that of the two alkyl chains, if it is in its fully extended conformation, but not if it is coiled. In the former case it is sufficiently hydrated to provide a satisfactory ionic reservoir; conversely, in the latter case, it may accommodate only a limited number of water molecules. Whether the conformation is extended or coiled depends both on the interfacial electric field and on the nature of the metal support, tBLMs fabricated with polyethyleneoxy-based thiolipids exhibit capacitances in the range from 0.5 to $0.7\,\mu F\,cm^{-2}$ and resistances in the range from 5 to $10 \,\mathrm{M}\Omega\,\mathrm{cm}^2$, which are comparable with the values found for conventional BLMs interposed between two aqueous phases; however, somewhat higher capacitance values (1 µF cm⁻²) have been reported for long spacers [53]. A particularly convenient tBLM of this type is DPTL (see Figure 4.1a). Its cross-sectional area is of about 55 Å².

"Thiolipopeptides" consist of an oligopeptide chain terminated at one end with a sulfhydryl group and covalently linked at the other end to the polar head of a phospholipid. They are considered to assume a helical structure [54]. In this case, their cross-sectional area amounts to about 75 Ų. They are usually obtained by tethering to a gold electrode a "thiopeptide" consisting of an oligopeptide chain terminated with a sulfhydryl group at one end and with a carboxyl group at the other end. This thiopeptide monolayer is then coupled *in situ* with dimyristoyl-phosphatidylethanolamine [55]. tBLMs fabricated with thiolipopeptides have been employed for the incorporation of a number of integral proteins; they exhibit high capacitances (from 2 to $10\,\mu\text{F}\,\text{cm}^{-2}$) and low resistances, of the order of $10^4\,\Omega\,\text{cm}^2$, which are three orders of magnitude lower than those of conventional BLMs [21, 56–58]. The low resistance is ascribed to a not perfectly homogeneous coverage of the thiolipid monolayer by the distal lipid monolayer, which is estimated at

about 70% of full coverage and makes the tBLMs insufficiently insulating [55]. This is indicative of the presence of an appreciable number of pinholes and other defects in the bilayer, as well as of a certain disorder of the hydrocarbon chains. It was suggested that the thiolipopeptide monolayer actually consists of a mixture of the thiolipopeptide and of the corresponding uncoupled thiopeptide. This conclusion was based on the observation that gold-supported tBLMs formed from a pre-synthesized thiolipopeptide did not permit the functional activity of incorporated proteins [21]. Such a functionality was recovered only by mixing the thiolipopeptide with a hydrophilic thiol, such as mercaptohexanol.

A structural and functional characterization of a DPTL monolayer tethered to gold was recently reported by Vockenroth et al. [59] using NR and EIS. The tetraethyleneoxy moiety was found to be only partly hydrated at the more positive potentials. However, at -0.600 V vs. Ag/AgCl (0.1 M KCl) a pronounced increase in the neutron scattering length density of the spacer was observed, denoting an increased amount of water transferred into this region. Leitch et al. [60] drew similar conclusions using PM-IRRAS. Thus, the fraction of nonhydrated C=O of the lipoic acid ester group was found to be about 50% at the more positive potentials and to reach a value of about 30% at -0.600 V, which denotes an increasing hydration of the spacer at these negative potentials. Analogous conclusions where also drawn by McGillivray et al. [26] by using NR, EIS, and Fourier-transform IRRAS (FT-IRRAS) to investigate a gold-supported thiolipid monolayer similar to DPTL, with a hydrophilic spacer moiety consisting of a hexaethyleneoxy chain directly bound to a sulfhydryl group. FT-IRRAS revealed a significant disorder in the spacer region and a substantial order in the hydrocarbon tail region. Moreover, NR showed that the spacer region had a thickness smaller than its fully extended length and only 5 vol% exchangeable water, despite its significant disorder. Since the incorporation of proteins with extramembrane domains requires a significant hydration of the spacer, the thiolipid monolayer was then diluted with short β-mercaptoethanol molecules. This permitted water molecules to be accommodated in the more spacious thiolipid–β-mercaptoethanol mixture. By self-assembling a lipid monolayer on top of this mixed monolayer, McGillivray et al. [26] obtained a tBLM with a differential capacitance comparable to that of conventional BLMs. Moreover, NR data revealed the presence of an appreciable amount of exchangeable water in the spacer moiety of this tBLM.

Thiolipid-based tBLMs, when anchored to solid supports such as gold or silver, do not meet the requirement of fluidity and lateral mobility. The thiolipid molecules are rigidly bound to the metal surface atoms. In principle, the lipid molecules on top of the thiolipid monolayer might be free to move laterally. In practice, however, their lateral mobility is hindered by the presence of adsorbed or hemifused vesicles and by the roughness of the metal support (see Section 4.4.4). Moreover, the hydration of the polyethyleneoxy moiety of thiolipids anchored to gold is low, while the incorporation of proteins with extramembrane domains requires a significant hydration of the spacer. Only small channel-forming peptides and ion carriers can be accommodated in the lipid bilayer moiety of polyethyleneoxy-based tBLMs, via incorporation from their aqueous solutions.

Typical examples are the depsipeptide valinomicin, an ion carrier that complexes a potassium ion with its carbonyl groups and shuttles it across the bilayer, and gramicidin, which spans the bilayer by forming a channel-forming dimer. The resulting impedance spectra are usually fitted by an equivalent circuit consisting of three circuit elements in series: an $R_{\rm m}C_{\rm m}$ mesh simulating the lipid bilayer, a capacitance C_s simulating the spacer [53, 61] or (more generally) an RC mesh simulating the remaining part of the tBLM [62], and a resistance R_{Ω} simulating the aqueous solution. In a KCl solution, the incorporation of these ionophores decreases the resistance, R_m, of the lipid bilayer moiety, while, as a rule, the corresponding conductance, C_m, remains practically constant [53, 61]. A goldsupported DPTL-DPhyPC tBLM was also used to incorporate the M2 peptide, a segment of the nicotinic acetylcholine receptor, which spans the lipid bilayer by forming a hydrophilic pore consisting of a bundle of M2 α-helixes [63]. In spite of the lack of lateral mobility and low hydration of these tBLMs, attempts have been made to incorporate the exotoxin α-hemolysin from *Staphylococcus aureus*, a watersoluble, monomeric, 293-residue polypeptide that forms heptameric pores in lipid bilayers [64, 65]. Upon incorporating α-hemolysin, the resistance of the lipid bilayer moiety of a DPTL-DPhyPC tBLM decreases by one order of magnitude [65]. In view of the well-known lysing effect of α -hemolysin on biomembranes, we cannot exclude a similar effect on these tBLMs, with membrane breakdown.

By replacing the lipoic acid residue of DPTL with a trichloropropylsilane group, a supramolecule (DPTTC) was obtained, which was self-assembled on a SiO_2 surface via the silane tether [62]. The SiO_2 layer was very thin (about 0.2 nm), being the native oxide layer of a highly p-doped silicon wafer. Consequently, a tBLM obtained by forming a lipid monolayer on top of the DPTTC monolayer tethered to the silicon wafer could be investigated by EIS. A tBLM was also formed on AlO_x sputtered on template-stripped gold [66]. An "anchor lipid" was immobilized on the AlO_x film, about 2.7 nm thick, via L-B transfer. This anchor lipid differs from DPTL by the replacement of the lipoic acid residue with a $-P=O(OEt)_2$ group for anchoring to the AlO_x surface. A DPhyPC monolayer was deposited on top of the anchor-lipid monolayer by vesicle fusion. The thin AlO_x film on gold allowed EIS and SPR characterizations of the system. The high capacitance, $10\,\mu\text{F}\,\text{cm}^{-2}$, of the lipid bilayer suggests that the tBLM adheres to some extent to the pores of the AlO_x layer.

Gold-supported thiolipopeptide-based tBLMs have been reported to incorporate a few bulky proton pumps. Thus, Naumann $et\ al.$ incorporated the proton pumps ATPase CF_0F_1 from chloroplasts and ATPase EF_0F_1 from *Escherichia coli* in a tBLM fabricated with a hydrophilic oligopeptide spacer [21, 56, 57]. Activation of these ATPases by ATP causes an increase in the reduction peak for hydrogen evolution, which takes place at $-0.7\,\mathrm{V}$ vs. Ag/AgCl (sat. KCl) on gold. This increase was ascribed to proton pumping into the hydrophilic oligopeptide spacer; it is suppressed by tentoxin, a specific inhibitor of the ATPase from chloroplasts. Another proton pump, cytochrome c oxidase (COX), was incorporated in an analogous gold-supported, thiolipopeptide-based tBLM [58]. COX is the terminal component of the respiratory electron transport chain and spans the inner mitochondrial

membrane. It catalyzes the redox reaction between the peripheral protein ferrocytochrome c and oxygen, with formation of ferricytochrome c and water; it also pumps protons from the mitochondrion matrix to the intermembrane space. Increasing additions of ferrocytochrome c cause a progressive decrease of the reduction peak for hydrogen evolution; this decrease was ascribed to a decrease in the proton concentration within the thiopeptide spacer, due to proton pumping from the spacer to the aqueous solution. Addition of the inhibitor cyanide eliminates the effect of ferrocytochrome c. The above conclusions as to proton pump incorporation in thiolipopeptide-based tBLMs should be examined in the light of the observation that vesicles have a low propensity to fuse on the hydrophobic surface exposed to the aqueous solution by a thiolipid monolayer, especially if they incorporate an integral protein; rather, they are adsorbed or partially fused (see Section 4.4.1). Incorporation of the above proton pumps from their solutions in detergent may easily take place in the membrane of adsorbed or partially fused vesicles, since the vesicular membrane is clearly interposed between two aqueous phases. In this respect, the functional activity of the above proton pumps might well be due to vesicles or proteoliposomes adsorbed or partially fused on a thiolipid monolayer, without the need of protein incorporation in the lipid bilayer moiety of the tBLM. In fact, their activation would cause an increase in the proton concentration on top of the thiolipid monolayer (in the case of F₀F₁ ATPase activated by ATP) or its decrease (in the case of COX activated by ferrocytochrome c). In view of the relative permeability of the leaky thiolipopeptide monolayers to protons, this would determine an increase or a decrease in the proton electroreduction current on gold, as actually observed.

A tBLM consisting of a gold-supported thiolipopeptide, with a soybean-PC monolayer on top, was used to incorporate the odorant receptor OR5 from *Rattus norvegicus* during its *in vitro* synthesis [67, 68]. The vectorial insertion of the protein into the tBLM in a functional and oriented form was verified. A gold-supported thiolipopeptide-based tBLM was also employed to incorporate the acetylcholine receptor (AChR), a ligand-gated channel protein present in the postsynaptic membrane of muscle cells [69]. Addition of the neurotoxin α -bungarotoxin peptide causes a slow but appreciable increase in the thickness of the tBLM incorporating AChR, thus denoting the binding of this toxin to the binding sites of AChR. The inhibitory effect of α -bungarotoxin on the ion-channel activity of AchR was not verified.

Gold-supported DPTL–DPhyPC tBLMs on microchips have been employed to measure single-channel currents of peptides and proteins. To this end, a microelectrode array device consisting of many ($100 \times 100 \, \mu \text{m}^2$) "sensor" pads was employed (Figure 4.6) [70–72]. A DPTL monolayer was tethered to the gold-coated sensor pads from a DPTL solution in ethanol; then, a lipid monolayer was formed on top of it by vesicle fusion. In view of the very small surface area of the pad, the resistance of the resulting tBLM ranged from 1.5 to $17 \, \text{G}\Omega$. This resistance was high enough to reduce the background electrical noise to the low level required for the use of the patch-clamp technique. This device allowed the recording of single-channel currents of gramicidin A [70], the high-conducting Ca²+-

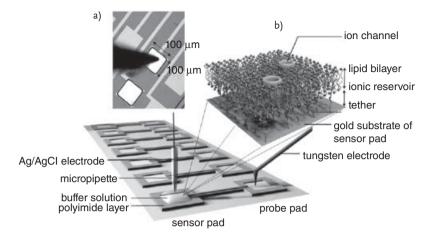


Figure 4.6 Tethered bilayer membrane array. (a) An optical microscope image of the probe pad and the tungsten electrode tip. (b) Schematic of the tethered bilayer membrane array. The lower left corner shows the gold sensor pad covered with a tBLM that incorporates ion channels. An Ag/AgCl electrode situated in a patch micropipette was permission from Elsevier.

used as an electrode and was inserted into the buffer drop. The gold substrate of the sensor pad was connected by a thin gold line to the probe pad onto which a tungsten tip was lowered as a counter electrode. The inset shows the tBLM formed at the gold surface of the sensor pad. Reprinted from [72] with kind

activated K⁺ (BK or Maxi-K) channel, the synthetic M2δ ion channel [71], and the mechanosensitive channel of large conductance from Escherichia coli [72]. All these peptides and proteins were incorporated into lipid vesicles, before fusing them onto the DPTL-coated sensor pad. With the exclusion of the gramicidin channel, the unitary conductance of the remaining ion channels was found to be from one-third to one-tenth that obtained with conventional BLMs. In this connection one cannot exclude, even in this case, the possibility that the channels responsible for single-channel currents be located in the membrane of adsorbed or partially fused vesicles. In fact, in this case, the capacitive coupling between the vesicular membrane and the tBLM is expected to decrease the unitary conductance of the channels. The capacitance and resistance of DPTL-DPhyPC tBLMs on gold microelectrodes of circular shape, with diameters ranging from 4000 to 8 µm, were examined as a function of the electrode size [73]. For the larger electrodes, the capacitance is directly proportional to the electrode area, while the resistance is inversely proportional to it. For the smaller electrodes, the capacitance decreases linearly and the resistance increases linearly with a decrease in the electrode diameter. This indicates that the capacitance and resistance of the larger electrodes are dominated by the electrode area, while the circumference seems to have the dominant role at smaller electrodes. Consequently, for small electrode sizes, a slight disorder of the bilayer structure at the edge of the electrode has a major influence on the electrical properties of the membrane. This

conclusion is supported by the consideration that the micro-tBLMs used for the recording of single-channel currents have resistances ranging from 1.5 to 15 G Ω . These resistances are high enough to reduce the level of the background noise down to the range of a few picoamps. However, the corresponding specific resistances are much less than those attained with identical tBLMs anchored to macroscopic gold electrodes of areas of the order of 10⁻² cm². In fact, the specific resistance of the latter tBLMs amounts to about $10 \,\mathrm{M}\Omega\,\mathrm{cm}^2$. If a $100 \times 100 \,\mathrm{\mu m}^2$ micro-tBLM had such a specific resistance, it should have a resistance of $100\,\mathrm{G}\Omega$. The loose packing of micro-tBLMs may explain why they may incorporate relatively bulky proteins much more easily than macro-tBLMs, without having to dilute the thiolipid with a short spacer.

Mercury-Supported Thiolipid-Based tBLMs As distinct from gold-supported thiolipid-based tBLMs, mercury-supported thiolipid-based tBLMs do not require the use of thiolipid-spacer mixtures to incorporate channel-forming proteins. Because of the fluidity imparted to the thiolipid monolayer by the liquid mercury surface, these tBLMs may incorporate bulky proteins, such as OmpF porin from Escherichia coli [74] and the HERG potassium channel [75], in a functionally active state. Upon incorporating gramicidin [76] or valinomicin [2], the tetraethyleneoxy (TEO) moiety of DPTL in aqueous KCl solution undergoes a conformational change ascribable to its elongation, as the applied potential is stepped from a fixed initial value of -0.200 V vs. SCE to a final value of -0.500 V vs. SCE [2]. As the final value of this potential step becomes progressively more negative, the charge of K⁺ ions accommodated in the TEO spacer increases rapidly, attaining a maximum limiting value of about $45 \,\mu\text{C}\,\text{cm}^{-2}$ at $-0.8 \,\text{V}$ vs. SCE [77]. This corresponds to three potassium ions per DPTL molecule, denoting an appreciable hydration of the spacer. Moreover, EIS measurements of the surface dipole potential of the TEO spacer tethered to mercury yield values that compare favorably with the dipole moment of TEO molecules measured in organic solvents [78]; this suggests a substantial ordering of the TEO chains in mercury-supported tBLMs. Incidentally, tBLMs supported by gold [21, 57] or silver [79] do not allow ionic charge measurements carried out by stepping the applied potential to final values negative of about -0.650 V. In fact, the resulting charge versus time curves show a linear section with a relatively high and constant slope that is maintained for an indefinitely long time [21, 57, 79]. The constant current responsible for this linear increase in charge is ascribed to a slight water electroreduction with hydrogen formation. The high hydrogen overpotential of mercury avoids this inconvenience. When comparing interfacial phenomena on different metals, rational potentials should be used, namely potentials referred to the potential of zero charge (PZC) of the given metal in contact with a nonspecifically adsorbed 1,1-valent electrolyte. The PZC equals -0.435 V vs. SCE for mercury [80] and -0.040 V vs. SCE for polycrystalline gold [81]. Therefore, an appreciable hydration of the TEO moiety, possibly accompanied by its elongation [2], takes place in the proximity of a rational potential of about zero on mercury, but at a much more negative rational potential of about -0.600 V on gold [59, 60], close to the DPTL desorption from this metal. A drawback of the use of mercury-supported tBLMs is represented by the notable difficulty in using surface-sensitive techniques for their structural characterization.

With respect to solid metal supports, mercury has the advantage of providing a defect-free, fluid, and readily renewable surface to the self-assembling thiolipid—lipid bilayer. Moreover, it imparts lateral mobility to the whole mixed bilayer. In addition, the self-assembly of a lipid monolayer on top of a thiolipid monolayer is readily carried out by simply immersing a thiolipid-coated mercury drop in an aqueous electrolyte on whose surface a lipid film has been previously spread [2]. Because of the hydrophobic interactions between the alkyl chains of the thiolipid and those of the lipid, this simple procedure gives rise to a lipid bilayer anchored to the mercury surface via the hydrophilic spacer moiety of the thiolipid. By avoiding the use of vesicles, this procedure excludes any artifacts due to partially fused vesicles. These advantageous features make the incorporation of membrane proteins in mercury-supported thiolipid-based tBLMs easier and safer than in solid-supported tBLMs.

The impedance spectra of mercury-supported DPTL-DPhyPC tBLMs incorporating the ion carrier valinomycin [2], the channel-forming peptides melittin [82] and gramicidin [76], and the channel protein OmpF porin [74] were fitted by four RC meshes in series. Fitting was particularly straightforward for the spectrum of the tBLM incorporating valinomicin, where four partially overlapping semicircles are clearly distinguishable in the M plot (see Figure 4.2). The R and C values relative to the four RC meshes vary appreciably with the applied potential. To this end, a generic approximate approach was developed, which applies the concepts of impedance spectroscopy to a model of the electrified interface and to the kinetics of potassium ion transport assisted by valinomycin across the tBLM [2]. This permits the four RC meshes to be ascribed to four different slabs composing the tBLM. The first semicircle has the highest resistance and was ascribed to the lipoic acid residue, in direct contact with the electrode surface. The second semicircle has a capacitance of about 7 µF cm⁻², close to that of a monolayer of tetraethyleneoxythiol self-assembled on mercury, and was ascribed to the TEO moiety. The third semicircle in the absence of valinomycin has a capacitance close to 1 µF cm⁻², a value typical of a conventional solvent-free BLM, and was reasonably ascribed to the lipid bilayer moiety. Finally, the last semicircle has the same resistance as the electrolyte solution and a very low capacitance of the order of 1 nF cm⁻², and was ascribed to the electrolyte solution adjacent to the tBLM.

The sigmoidal charge versus time curves following a potential step from a value at which melittin channels are not formed to one at which they are formed were interpreted on the basis of a generic kinetic model [83]. This model accounts for the potential-independent disruption of melittin clusters adsorbed flat on the lipid bilayer, induced by the potential-dependent penetration of the resulting monomers into the lipid bilayer; the potential-independent aggregation of the monomers inside the bilayer with channel formation is then treated on the basis of a mechanism of nucleation and growth. This explains the initial induction period responsible for the sigmoidal shape of charge versus time curves. Gramicidin is a peculiar channel-forming peptide whose helical structure differs from the α -helix

of common peptides and membrane proteins by the fact that its lumen is large enough to allow the passage of simple desolvated monovalent cations. Since its length is about one-half that of a biomembrane, it spans it by forming a dimeric channel. The N-terminuses of the two monomeric units are directed toward each other, in the middle of the lipid bilayer, just as their dipole moments. A transmembrane potential different from zero is, therefore, expected to favor electrostatically one monomeric orientation at the expense of the other, thus destabilizing the dimer. Nonetheless, the stationary current due to the flow of potassium ions along gramicidin channels incorporated in a mercury-supported tBLM increases with an increase in the transmembrane potential, exhibiting an almost quadratic dependence [76]. To explain this behavior, it was assumed that the rate constant for dimer formation increases in parallel with an increase in the ionic flux. In fact, when the time elapsed between the passage of two consecutive cations through the junction between the two monomers forming the conducting dimer starts to become comparable with, and ultimately shorter than, the time required for the dissociation of the two monomers, such a dissociation becomes increasingly less probable.

The HERG K^+ channel is present in the plasma membrane and consists of four identical subunits, each spanning the membrane with six α -helixes [84]. Upon reconstituting the HERG K^+ channel in the tBLM, the potential difference across the lipid bilayer moiety (i.e., the transmembrane potential) was caused to pass from a value more positive than the resting potential of a plasma membrane to a value more negative, by choosing the appropriate potential step [75]. This step causes the channel to pass from an inactive to an open state, which lasts for about 100 ms, followed by a closed state. To separate the "inward" current, which flows during the open state, from the very high capacitive current, the current recorded after blocking the HERG channel with a specific inhibitor was subtracted from that previously recorded in the absence of the inhibitor. The inward current decays monoexponentially in time, with a time constant in good agreement with that obtained by the patch-clamp technique with conventional BLMs.

Sarcolipin (SLN) and phospholamban (PLN) are two small membrane proteins that modulate the function of Ca-ATPase of the sarcoplasmic reticulum. They were incorporated in a mercury-supported DPTL–DPhyPC tBLM to verify whether they may form ion-selective pores in biomembranes [85, 86]. SLN consists of a single α -helix that spans the whole membrane. By measuring the conductance of a tBLM incorporating SLN by EIS, it was shown that SLN forms channels highly selective toward small inorganic anions, such as chloride ions [85]. Incorporation of SLN in a tBLM immersed in an aqueous solution (pH = 5.3) of 0.05 M NaH₂PO₄ causes only a slight decrease in the resistance of the lipid bilayer, as measured by the diameter of a Z'' versus Z' plot (Nyquist plot; Section 4.3); however, submicromolar additions of ATP decrease the resistance of the tBLM to an appreciable extent, as shown in Figure 4.7 [86]. Plotting the conductance against the ATP concentration yields a curve that tends asymptotically to a limiting value and can be satisfactorily fitted by the Michaelis–Menten equation, with an association constant for the SLN–ATP complex of about 0.1 μ M (inset of Figure 4.7). This behavior was

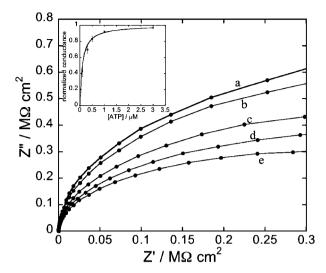


Figure 4.7 Filled circles are experimental points on a Z'' versus Z' plot for a tBLM in an aqueous solution of 0.05 M NaH₂PO₄ (pH = 5.3) at $-0.500\,\text{V}$ vs. Ag/AgCl (0.1 M KCl) (a) in the absence of SLN, (b) after incorporation of SLN from its 0.7 μM solution, and after subsequent additions of (c) 0.1, (d) 0.3, and (e) 3 μM ATP. The solid curves are fits of the solid circles by a series of four *RC* meshes. The values of the resistance, R_m , of the lipid bilayer moiety resulting from the

fits are (a) 0.90, (b) 0.70, (c) 0.53, (d) 0.42, and (e) 0.35 $M\Omega cm^2$. The corresponding C_m values are all close to $1 \mu F cm^{-2}$. The inset shows conductance, $1/R_m$, normalized to its maximum value equated to unity, as a function of the ATP concentration. The error bars denote standard deviations. The solid curve is a fit of the experimental points by the Michaelis–Menten equation. Reprinted from [86] with kind permission from Elsevier.

explained by assuming that SLN forms a hydrophilic pore consisting of a bundle of four or five SLN α -helixes that turn their hydrophilic side, containing two hydrophilic threonine residues, toward the interior of the bundle. On the basis of this and other features of SLN, it was proposed that SLN is just the "P_i transporter" described by Lee and coworkers [87] in 1991 and whose nature was not known up to now. PLN exists in equilibrium between the pentameric and monomeric forms. The pentamer releases to Ca-APTase a monomer, which forms a 1:1 complex inhibiting the affinity of Ca-ATPase for Ca²⁺ [88]. Phosphorylation of the monomer by ATP removes the inhibition. Both SLN and the PLN monomer have a single α -helix that spans the membrane. The main difference between the SLN and PLN helixes is represented by the absence of two hydrophilic threonines in the PLN helix. Incorporation of PLN in a mercury-supported DPTL–DPhyPC tBLM does not affect its resistance [89]. It was concluded that the absence of the two threonine residues prevents the PLN pentamer from forming a hydrophilic pore.

A mercury-supported tBLM was formed at the tip of a microelectrode for measuring single-channel activity [90]. To this end, use was made of a platinum wire embedded in a thin glass capillary and terminated with a platinum microdisc,

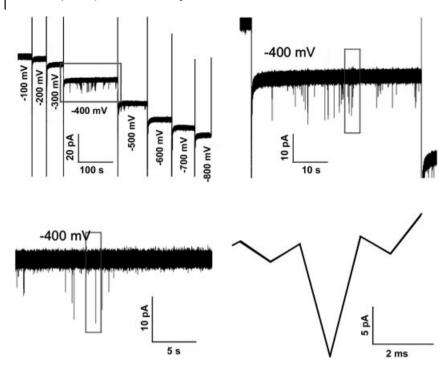


Figure 4.8 Single-channel traces of OmpF porin from *Escherichia coli* incorporated in a mercury-supported tethered bilayer lipid micromembrane immersed in aqueous 0.1 M KCl, at different applied potentials measured

vs. a Ag/AgCl (3 M KCl) reference electrode. Potentials must be decreased by 50 mV to refer them to the SCE. Reprinted from [90] with kind permission from Elsevier.

about 20 μ m in diameter. Mercury was electrodeposited on this microdisc from a mercurous nitrate aqueous solution (pH = 1), giving rise to a mercury spherical cap that was coated with a DPTL–DPhyPC bilayer. The high resistance of this "tethered bilayer lipid micromembrane," about $5\,G\Omega$, allows the recording of single-channel currents by the patch-clamp technique. Figure 4.8 shows the single-channel currents due to the opening of the OmpF porin channel from *Escherichia coli*, reconstituted in this micromembrane, as a function of time at different applied potentials. A burst of single-channel currents is observed at $-400\,\text{mV}$ vs. SCE, and to a minor extent at $-500\,\text{mV}$, which corresponds to a zero transmembrane potential. This behavior is consistent with a well-known property of OmpF porin, according to which this channel protein undergoes inactivation on both sides of the zero transmembrane potential.

This micromembrane was also used to investigate the spontaneous formation of microdomains, when the distal lipid monolayer is made up of a lipid mixture. Microdomains are in the gel state when they consist primarily of glycolipids and sphingolipids, in a "liquid-ordered" state (so-called "lipid rafts") when they also contain cholesterol, and in a "liquid-disordered" state when they consist primarily

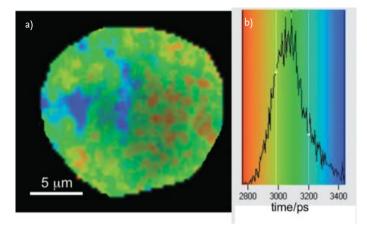


Figure 4.9 (a) Two-photon (excitation wavelength = 760 nm) fluorescence lifetime $21 \times 21 \, \mu m$ image and (b) fluorescence lifetime distribution histogram of a distal monolayer of dioleoylphosphatidylcholine–palmitoylsphingomyelin–cholesterol (47:47:6) mixture, labeled with 1 mol% laurdan, at

room temperature. The mixture is the distal monolayer of the micromembrane. The color code in the fluorescence lifetime imaging microscopy image is that indicated in the histogram. Reprinted from [30] with kind permission from the Royal Chemical Society.

of unsaturated phosphatidylcholines and a small percentage of cholesterol [91]. Rafts are receiving increasing attention since they are considered to regulate the membrane function in eukaryotic cells. Figure 4.9 shows a two-photon fluorescence lifetime image of a distal monolayer consisting of a palmitoylsphingomyelin–dioleoylphosphatidylcholine–cholesterol (47:47:6) mixture, with the addition of 1 mol% of the fluorophore laurdan [30]. The bluish microdomains with an irregular percolative-like shape and lifetimes of 3450 \pm 50 ps are ascribed to the gel phase. The small roundish orange microdomains immersed in the surrounding green matrix have lifetimes of 2830 \pm 50 ps and are attributed to the liquid-ordered phase. Finally, the matrix, with intermediate fluorescence lifetimes, is ascribed to the coexistence of the liquid-disordered phase and of liquid-ordered microdomains of size below the resolution of the microscope.

4.5.2.3 Thiolipid-Spacer-Based tBLMs

A mixture of a short hydrophilic spacer anchored to gold via a sulfhydryl or disulfide group (e.g., mercaptoethanol or dithiodiglycolic acid) and of a thiolipid with its terminal hydrophobic group (e.g., a cholesteryl [8, 9] or a phytanyl group [17]) capable of pinning a lipid bilayer gives rise to a practically aqueous region sandwiched between the short spacer and the lipid bilayer (Figure 4.10); this may favor the accommodation of the extramembrane domains of relatively bulky membrane proteins. Deposition of a lipid bilayer onto such a mixture of a short spacer and of a thiolipid is usually carried out by vesicle fusion or by rapid solution exchange. The resulting lipid bilayer has a capacitance of about $0.5-0.6\,\mu\text{F}\,\text{cm}^{-2}$

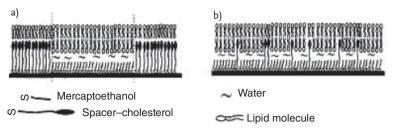


Figure 4.10 (a) Schematic of a micropatterned film consisting of distinct areas covered by a mercaptoethanol monolayer with a lipid bilayer on top, and by a monolayer of a hydrophilic spacer covalently bound to a

cholesteryl group with a lipid monolayer on top. (b) Schematic of a mixed monolayer of mercaptoethanol and spacer—cholesterol molecules with a lipid bilayer on top.

and a resistance higher than $5\,M\Omega\,cm^2$. These values compare favorably with those of conventional BLMs. Ion-channel activity of gramicidin and melittin incorporated in such a gold-supported tBLM and ion-selectivity of gramicidin were verified by EIS [17]. However, here too, the lateral mobility of the lipid bilayer is hindered by the hydrophobic group of the thiolipid molecules, which provides a fraction of the proximal leaflet of the lipid bilayer.

The ability of the volume enclosed between the gold electrode and the lipid bilayer moiety of thiolipid–spacer-based tBLMs to accommodate water molecules and inorganic ions was estimated from the level of conductance induced by the incorporation of a given amount of the ionophore valinomycin. This ionic reservoir was simulated by a capacitance C_s and the lipid bilayer by an $R_m C_m$ mesh. By using a number of thiolipids and spacers of variable length, Cornell and coworkers [25, 92] found that the conductance, $1/R_m$, of the lipid bilayer increases with an increase in the length of the thiolipid and with a decrease in the length of the spacer.

Relatively bulky membrane proteins with extramembrane domains can be accommodated in the bilayer if the areas covered by the short thiolated spacer are sufficiently large. Differences in chain length [93] and chemistry [94] between thiolipids and spacers may hopefully favor the formation of phase-demixed domains on the nanoscale, facilitating protein incorporation. This seems to be the case with mixed monolayers of cholesteryl-based thiolipids and short thioalcohol spacers on gold [4, 8, 95]. A micropatterned layer, consisting of regions of thiolipid molecules with a lipid monolayer on top alternated with regions of mercaptoundecanoic acid with a lipid bilayer on top, was used to incorporate rhodopsin [96, 97], an integral protein that contributes to the closing of the Na⁺ channels of the plasma membrane of the outer segment of vertebrate rod cells.

4.5.3 Polymer-Cushioned Bilayer Lipid Membranes

The spaciousness of the ionic reservoir of tBLMs may not be sufficient to accommodate bulky extramembrane domains of membrane proteins. The problem is

particularly serious with cell-adhesion receptors, whose functional extramembrane domains can extend to several tens of nanometers. This problem can be circumvented by separating the lipid bilayer from the solid substrate using soft polymeric materials of typically less than 100 nm thickness, which rest on the substrate and support the bilayer. These stratified films are often referred to as polymer-cushioned or polymer-supported bilayer lipid membranes (for a review, see [98]). This approach reduces the nonspecific binding of proteins to the solid support and the frictional coupling between proteins and the support, preventing the risk of protein denaturation due to direct contact between protein subunits and the bare support surface. In some cases, the cushion may assist self-healing of local defects in lipid bilayers deposited on macroscopically large supports.

To form a thermodynamically stable polymer-lipid composite film on a solid support in aqueous solution and to avoid the formation of polymer blisters, several requirements should be fulfilled. The interaction between the lipid bilayer and the substrate surface must be repulsive. In fact, if the net force acting per unit area (disjoining pressure) is negative, continuous thinning of the interlayer results in film collapse, that is, dewetting, giving rise to regions of tight local contact between the lipid bilayer and the substrate surface (so-called pinning centers). Polymercushioned membranes are fairly unstable if the attractive interfacial forces between the polymer and the lipid bilayer are relatively weak. In this case, the bilayer can easily detach from the polymer cushion. On the other hand, attractive forces that are too strong may decrease the lateral mobility of the bilayer. A compromise should therefore be found between a sufficient stability of the polymer-lipid interface and lateral mobility. A possible strategy consists of enhancing the stability of polymer-cushioned membranes via attractive electrostatic interactions [99]. A different approach to stabilization of polymer-cushioned membranes was adopted by Naumann et al. [100] and by Wagner and Tamm [101] by tethering the polymer both to the substrate and to the membrane.

Usually, polymer cushions are anchored to supports such as glass, silica, and mica using polymers derivatized with alkyl silanes [12] or triethoxysilane for covalent linkage to silanols at the surface of silicate substrates [102]. Polymercushioned lipid bilayers on conducting supports have been investigated only rarely. Spinke et al. [103] and Erdelen et al. [104] described a polymer-supported lipid bilayer anchored to gold. These authors used a methacrylic terpolymer consisting of a hydrophilic main chain that acts as a spacer, a disulfide unit that anchors the polymer to the gold surface, and a hydrophobic lipid-like part that forms a first lipid monolayer upon self-assembly. A second lipid monolayer was formed on top of the first by fusion with DMPC vesicles. Polymer-cushioned lipid bilayers have frequently been investigated on the ITO semiconductor. In the case of a lipid bilayer deposited on a regenerated (and thus hydrophilic) cellulose cushion, the swollen polymer film behaves like an aqueous electrolyte. Selective ion transport via ion channels and carrier proteins incorporated in the membrane was quantitatively evaluated by determining the electric resistance of the membrane [105].

4.5.4

S-Layer Stabilized Bilayer Lipid Membranes

One of the common surface structures of archea and bacteria are monomolecular crystalline arrays of protein subunits, called S-layers [106–109]. They constitute the outermost component of the cell envelope of these procaryotic organisms. S-layer subunits can be aligned in lattices with oblique, square, or hexagonal symmetry. Since S-layers are monomolecular assemblies of identical protein subunits, they exhibit pores of identical size and morphology. A group of nonclassical cell wall polymers, called "secondary cell wall polymers" (SCWPs), are attached noncovalently, presumably by a lectin-type interaction, to the S-layer proteins.

Since S-layer subunits of most bacteria interact with each other through non-covalent forces, they can be set free with high concentrations of agents that break hydrogen bonds, such as guanidine hydrochloride or urea. Once the S-layer lattice of a bacterial cell is completely disintegrated and the disintegrating agent is removed by dialysis, the S-layer subunits have the unique ability to reassemble spontaneously in suspension, at the liquid—air interface, on solid surfaces, on spread lipid monolayers, and on liposomes. Recrystallization starts at several distant nucleation points on the surface and proceeds until neighboring crystalline areas meet. In this way, a closed mosaic of differently oriented monocrystalline domains is formed. Recrystallization of isolated S-layer proteins on differently charged solid supports reveals an electroneutral outer S-layer surface and a net negative or net positive inner S-layer surface.

The natural tendency of S-layers to interact with membranes has been exploited to insert them as an intermediate layer between a lipid bilayer and a substrate, giving rise to the so-called ssBLMs. Thus, in the case of bacterial S-layer proteins, it has been demonstrated that protein domains or functional groups of the S-layer lattice interact via electrostatic forces with some head groups of lipid molecules. In addition, the affinity of S-layer proteins for the corresponding SCWPs, which are recognized as specific binding sites, can be used to fabricate complex architectures (Figure 4.11). Thus, a monolayer of SCWPs, suitably thiolated at one end, can be anchored to a gold support, and an S-layer can be spontaneously recrystallized on top of it; finally, a lipid bilayer can be self-assembled on top of the S-layer. Optionally, the lipid bilayer so formed can be further stabilized by recrystallizing an additional S-layer on top of it. Alternatively, a loose monolayer of lipidated SCWP molecules can be bound to an S-layer previously recrystallized on a substrate; a lipid bilayer self-assembled on top of the SCWP monolayer is then firmly anchored by the lipid moieties of the SCWPs, which penetrate the inner lipid leaflet. Optionally, the lipidated SCWP molecules present on the outer lipid leaflet can be bound to a further S-layer. All these architectures have a stabilizing effect on the associated lipid bilayer, leading to an improvement in its lifetime and robustness. The assembly of S-layer structures from solution to a solid substrate, such as a gold-coated glass slide, can be followed using SPR or a QCM-D [106]. The self-assembling process is completed after approximately 45 min. The mass

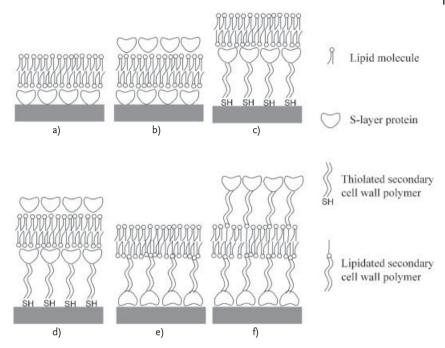


Figure 4.11 Schematic of S-layer stabilized solid supported lipid membranes. (a) S-layer directly recrystallized on gold, with a lipid bilayer on top. (b) Same as (a), with an additional S-layer recrystallized on top of the lipid bilayer. (c) Thiolated SCWPs directly bound to gold and interacting with an S-layer, with a lipid bilayer on top. (d) Same as (c),

with an additional S-layer recrystallized on top of the lipid bilayer. (e) S-layer directly recrystallized on gold and interacting with lipidated SCWPs, which anchor a lipid bilayer. (f) Same as (e), with lipidated SCWPs inserted into the distal lipid monolayer and interacting with an additional S-layer.

increase followed using a QCM-D corresponds to a thickness of about 8–9 nm, in agreement with the value estimated by SPR.

A well-characterized S-layer protein, SbqA from *Bacillus sphaericus* CCM 2177, was used as an ultrathin crystalline water-containing hydrophilic layer between a gold electrode and a lipid bilayer [107]. The SbqA protein recrystallizes in monomolecular square lattices with the neutral outer surface exposed to the aqueous phase and the negatively charged inner surface attached to the gold electrode. Membrane resistances of up to $80\,\mathrm{M}\Omega\,\mathrm{cm}^2$ were observed for DPhyPC bilayers on SbpA. In addition, lipid bilayers supported by SbpA exhibited a long-term robustness of up to two days. Upon incorporating the voltage-gated ion channel alamethicin at open circuit, the membrane resistance dropped from about $80\,\mathrm{M}\Omega\,\mathrm{cm}^2$ to about $950\,\Omega\,\mathrm{cm}^2$, whereas the capacitance did not change.

4.5.5

Protein-Tethered Bilayer Lipid Membranes

In all the biomimetic membranes previously described and allowing the incorporation of proteins, the protein orientation in the membrane is purely casual. At most, if one of the two extremembrane domains of the protein is much bulkier than the other, incorporation in a tBLM occurs preferentially with the bulkier domain turned toward the aqueous phase, in view of the limited spaciousness of the hydrophilic moiety of the tBLM. Moreover, the packing density of the reconstituted proteins in the lipid bilayer is not well controlled. The need for a well-defined protein orientation with respect to the electrode surface is particularly felt with redox membrane proteins, in which the electrons involved in a chain of redox couples are conveyed across the membrane in a well-defined direction.

To overcome this problem, Naumann and coworkers have developed a novel methodology based on tethering proteins, rather than lipids, to electrode surfaces; the lipids are then allowed to self-assemble around the tethered proteins [110]. To this end, a recombinant membrane protein is engineered to bear a stretch of six consecutive histidine residues. A gold surface is then functionalized by attaching a molecule terminated with a nitrilotriacetic (NTA) moiety at one end and with a sulfhydryl group for anchoring to gold at the other end. Complexation of Ni²⁺ ions to both the NTA functionality and the histidines of the stretch causes the protein to be anchored to the gold surface from its solution in detergent, as shown in Figure 4.12. To retain full functional integrity, the membrane protein is incorporated into a lipid bilayer. For this purpose, the protein layer tethered to gold is mixed with detergent-destabilized lipid vesicles of DMPC. By removing the detergent with microporous biobeads, the tethered proteins are surrounded by lipid molecules that form a lipid bilayer around them, as verified by SPR and EIS; a water layer remains interposed between the lipid bilayer and the NTA moiety, acting as an ionic reservoir. At low surface densities of the redox protein, the bilayer does not effectively form, and protein aggregates are observed; on the other hand, at very high surface densities, very little lipid is able to intrude between the closely packed protein molecules [111]. In both cases, redox activity is low. Redox activity is preserved in the biomimetic membrane only at moderate surface coverages, in which a continuous lipid bilayer is present and the protein molecules are not forced to aggregate.

This approach has been adopted to investigate the function of COX from the proteobacterium *Rhodobacter sphaeroides* [110], the last enzyme in the respiratory electron transport chain of bacteria, located in the bacterial inner membrane. It receives one electron from each of four ferrocytochrome c molecules, located on the periplasmic side of the membrane, and transfers them to one oxygen molecule, converting it into two water molecules. In the process, it binds four protons from the cytoplasm to make water, and in addition translocates four protons from the cytoplasm to the periplasm, to establish a proton electrochemical potential difference across the membrane. In this ptBLM, the orientation of the protein with respect to the membrane normal depends on the location of the histidine stretch

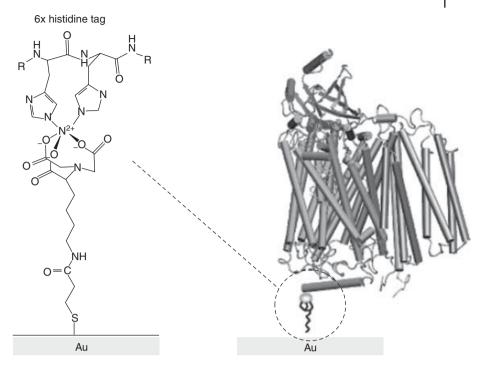


Figure 4.12 Adsorption of COX on Ni–NTA-modified gold surface via the His tag at the C-terminus of subunit I. Two nitrogen atoms of the imidazole rings from two of the histidines of the His stretch coordinate the

Ni²⁺ ion. The coordinating histidine residues are not necessarily adjacent in the primary sequence, as drawn in the figure. Reprinted from [110] with kind permission from the American Chemical Society.

(His tag) within the protein. Two opposite orientations of the protein were investigated, either with the cytochrome c binding side pointing away from the electrode surface or directed toward the electrode, simply by engineering the His tag on the C-terminus of subunit SU I or SU II, respectively. The functional activity of COX was verified by cyclic voltammetry with both protein orientations. In this connection, it should be noted that electron transfer in COX occurs sequentially through the four redox centers Cu_A, heme a, heme a₃, and Cu_B, in the direction from the binding site of cytochrome c, located on the outer side of the bacterial membrane, to its inner side. With the cytochrome c binding site pointing away from the electrode surface, the primary electron acceptor, Cu_A, is far from the electrode surface. Hence, in the absence of cytochrome c, the cyclic voltammogram exhibits only a capacitive current. This indicates that COX is not electrically coupled to the electrode, and direct electron transfer does not take place. When the COX is oriented with the cytochrome c binding site pointing toward the electrode surface, the primary electron acceptor, Cu_A, is also oriented toward the electrode. In this case, the cyclic voltammogram in the absence of oxygen shows a single reduction peak at about $-274\,\mathrm{mV}$ vs. NHE, due to the electroreduction of the enzyme, and a corresponding oxidation peak at about $-209\,\mathrm{mV}$ vs. NHE [112, 113]. The peak currents increase linearly with the scan rate, denoting a surface-confined process. In the presence of oxygen, electrons transferred from the electrode to the redox centers of COX are irreversibly transferred to oxygen, leading to a notable increase of the reduction peak, which now lies at $-202\,\mathrm{mV}$ vs. NHE, and to a continuous electron transfer. The absence of direct electron transfer and of proton electroreduction when COX is oriented with the cytochrome c binding site turned toward the solution confirms the orientation dependence both of direct electron transfer and of transmembrane proton transport.

4.6 Conclusions

The use of electrochemical techniques such as EIS, charge transient recordings, and cyclic voltammetry for the investigation of biological systems is becoming increasingly popular, just as is the application of the concepts of electrochemical kinetics and of the structure of electrified interfaces to the interpretation of the electrochemical response.

Much work is presently being done to realize biomembrane models consisting of a lipid bilayer anchored to a solid electrode through a hydrophilic spacer and satisfying those requirements of ruggedness, fluidity, and high electrical resistance that are necessary for the incorporation of integral proteins in a functionally active state. The capacitive currents resulting from the activation of ion pumps, transporters, ion channels, and channel-forming peptides incorporated in these biomembrane models can be analyzed over a broad potential range by electrochemical techniques, which are far less expensive than other techniques presently adopted.

The realization of these biomembrane models allows fundamental studies of the function of integral proteins. Biomimetic membranes are ideally suited to elucidate many problems in molecular membrane biology. This will open the way to the elucidation of structure-function relationships in ligand-receptor and protein-protein interactions. Moreover, the development of biomimetic systems that incorporate therapeutically or diagnostically important natural proteins will open the door to the realization of sensors targeting biological analytes. Many practical applications are foreseen for these sensors, such as the detection of drug candidates modulating the function of ion channels and pumps or targeting membrane receptors. In this respect, there is a strong need to develop novel, rapid, and highly sensitive methods for drug screening, capable of selecting and analyzing a huge number of compounds. At present, screening of pharmacologically active compounds follows traditional procedures that apply time-consuming ligandbinding studies and receptor-function tests separately. Thus, for instance, the function of ion channels and transporters is traditionally characterized in detail by patch-clamp studies, which investigate the proteins in their natural environment, the cellular membrane. These assays are tedious to perform and difficult to automate at high throughput, making the investigation of many samples difficult. The lack of knowledge about the different functions of these channels is due to a lack of specific inhibitors, which are unavailable due to the lack of efficient measuring systems. Present ligand-binding experiments identify only ligands to already known binding sites on the protein(s) of interest and neglect other potentially more interesting sites. Moreover, they cannot easily differentiate between agonists and antagonists. Thus, the direct, predominantly electrochemical determination of the function of ion channels and pumps in biomembrane models reconstituted from purified components addresses a strongly felt need for the development of new drug candidates or diagnostic test systems.

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5

Enzymatic Fuel Cells

Paul Kavanagh and Dónal Leech

5.1 Introduction

The rapid depletion of fossil fuels and the environmental concerns associated with their combustion have created a strong demand for new and improved technologies capable of harnessing energy from sustainable sources. The automobile industry's development of the hydrogen fuel cell as a "green" alternative to the traditional, polluting, combustion engine is one such initiative. Although hydrogen fuel cells are environmentally friendly (only producing water as a byproduct), major obstacles, such as cost and hydrogen production/storage issues, must be overcome before this technology could seriously compete in the marketplace.

An interesting spin-off from conventional fuel cell technology is the development of biofuel cells (BFCs) which utilize biological catalysts, in place of metal catalysts, to catalyze the reactions at the anode and cathode. Due to the versatile nature of the biocatalysts, BFCs are not limited to hydrogen as a fuel and can derive power from a wide range of organic substrates. Two types of BFCs are currently being explored for technological application: microbial fuel cells (MFCs) and enzymatic fuel cells (EFCs). MFCs use whole microorganisms, such as bacteria and algae, to produce and oxidize fuels (e.g., sugars and hydrogen) through fermentation or metabolism. This area is extensively reviewed elsewhere [1-5] and will not be covered in this chapter. EFCs use isolated enzymes, derived from microorganisms, to catalytically oxidize a specific fuel at the anode and reduce the oxidant, usually oxygen, at the cathode (shown schematically in Figure 5.1).

In 1964, Yahiro et al. reported the first EFC using glucose as a fuel and oxygen as an oxidant [6]. Shortly thereafter EFCs were investigated with specific applications in mind, such as using an EFC to power an artificial heart or to provide power to biomedical devices such as cardiac pacemakers [7]. It soon became clear that EFCs could not meet the energy and operational lifetime demands required for such devices and, for the next 25 years, this research area was largely abandoned. A re-emergence occurred in the late 1990s, partly driven by progress in (i) biosensor design and enzyme electrochemistry, particularly in terms of achieving high, and increasingly stable, current densities at modified electrodes and

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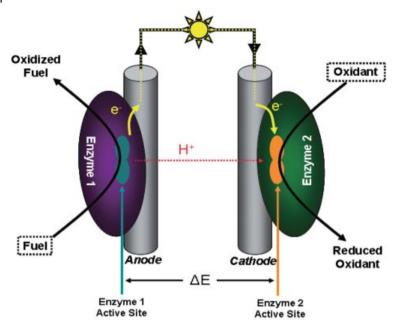


Figure 5.1 Schematic of the processes in a membraneless EFC.

(ii) microelectronics, where ever smaller and lower-energy-consuming devices are being manufactured. Such advances reignited interest in integrating EFCs with specific low-energy-demanding microelectronic biomedical devices [7].

As with the hydrogen fuel cell, EFCs must also compete with state-of-the-art batteries. Lithium-iodine batteries are currently used to power implantable devices, such as cardiac pacemakers, and can last for 5-6 years. Therefore, to compete in this market, a fully implantable EFC should have an operational lifetime of greater than 5 years. This seems quite a challenge considering that, to date, an implanted EFC generated intermittent power for 10 days [8]. One niche application which shows potential centers on the development of an integrated miniature EFC-sensor system for diabetes management. It is envisaged that these "semiimplantable" miniature EFCs will provide power for the lifetime of glucose sensors (typically less than a week) and be discarded after their first and only use [7], thereby eliminating the need for long-term stability. Other in vivo applications have been investigated, such as generating power from carbohydrates contained in plants [9]. Regardless of the intended application, the power densities, operational lifetime, and biocompatibility of such devices must be further investigated and improved upon before commercial applications can be realized. Indeed, whether or not EFCs will ever meet the operational and cost requirements to compete in the marketplace remains open to debate. Nonetheless fundamental EFC research will aid in providing new insights into redox enzyme structure-activity and stability in solution and at surfaces, surface and immobilization chemistry, and enzyme electrode electrochemistry and contribute to our fundamental understanding of catalysis, permitting future applications using biomimetics to be exploited.

This chapter is intended to provide a selective overview of the most recent developments in EFC research. Several excellent and comprehensive reviews focused on aspects of EFC research have appeared over the past few years [7, 10–16]. We focus here on developments that may lead to deployment of an implantable glucose-oxidizing, oxygen-reducing EFC, as an example of the extensive research on EFC bioelectrochemistry. Developments, focused on using other fuels, such as hydrogen, alcohols, or other sugars, or oxidants will not be considered in detail.

5.1.1

Enzymatic Fuel Cell Design

An EFC consists of two electrodes, anode and cathode, connected by an external load (shown schematically in Figure 5.1). In place of traditional nonselective metal catalysts, such as platinum, biological catalysts (enzymes) are used for fuel oxidation at the anode and oxidant reduction at the cathode. Judicious choice of enzymes allows such reactions to occur under relatively mild conditions (neutral pH, ambient temperature) compared to conventional fuel cells. In addition, the specificity of the enzyme reactions at the anode and cathode can eliminate the need for other components required for conventional fuel cells, such as a case and membrane. Due to the exclusion of such components, enzymatic fuel cells have the capacity to be miniaturized, and consequently micrometer-dimension membraneless EFCs have been developed [7]. In the simplest form, the difference between the formal redox potential (E^0) of the active site of the enzymes utilized for the anode and cathode determines the maximum voltage (ΔE) of the EFC. Ideally enzymes should possess the following qualities.

- A high specificity for fuel/oxidant.
- Retain good catalytic properties and high stability under cell operational conditions.
- Enzymes should not compete with each other for substrates (fuel and oxidant).
- Enzyme reactions should not generate any harmful side-products.

The chosen electrode material should be conductive and inert within the potential range of the cell. Materials composed of allotropes of carbon and, to a lesser extent, gold are most commonly used. The cell should be designed to minimize overpotentials due to kinetics, ohmic resistance, and mass transfer of fuel in order to maximize cell voltage (ΔE) and current (i) generation. In addition, all cell components should be mechanically stable within their operating environment.

5.1.2

Enzyme Electron Transfer

Redox enzymes typically consist of an apoenzyme (the protein component of an enzyme) and at least one coenzyme (a small nonproteinaceous electroactive

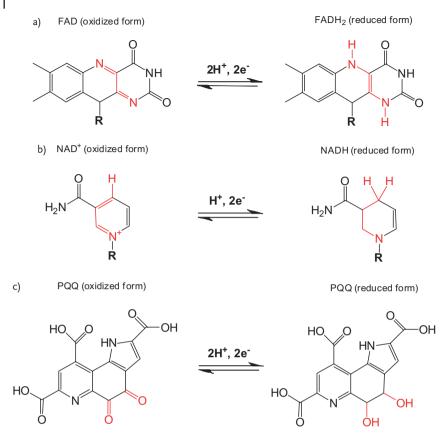


Figure 5.2 Enzyme cofactor structures and redox processes for: (a) FAD/FADH $_2$, (b) NAD $^+$ / NADH, and (c) PQQ. R represents adenosine diphosphate.

molecule). The presence of the coenzyme is essential for facilitation of electron transfer between the enzyme and the substrate, which in EFCs can be the electrode surface. The coenzyme can be either tightly bound within the enzyme structure (coenzyme) or released from the enzyme's active site during the reaction (cofactor). Common coenzymes/cofactors include flavin adenine dinucleotide (FAD), nicotinamide adenine dinucleotide (NAD), and pyrroloquinoline quinone (PQQ) (Figure 5.2).

If the active site of the enzyme is located sufficiently close to the electrode surface electrons can be transferred directly from the enzyme to the electrode as depicted in Figure 5.3a. In the case of an anodic reaction, the electrode replaces the natural co-substrate (such as oxygen) as an electron acceptor. This process is known as direct electron transfer (DET), often categorized as "third-generation" enzyme electrodes in the biosensor literature, and is the most elegant and simplest method of bioelectrocatalysis between an enzyme active site and an electrode.

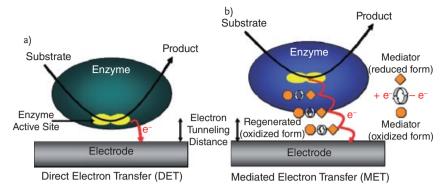


Figure 5.3 Schematics of (a) direct electron transfer (DET) and (b) mediated electron transfer (MET) between an enzyme active site and an electrode to oxidize a substrate.

Although DET has been reported for a wide range of enzymes [13, 17–19], several challenges need to be overcome to achieve significant rates of DET, leading to appreciable current densities, between active sites and solid electrode surfaces. For example, the rate of electron transfer is related exponentially to the distance of closest approach between an electron donor and acceptor [20], resulting in negligible rates for distances beyond 2 nm. This means that DET can only take place when an electrode is placed within this distance to an enzyme active site. In addition, for those enzymes that have active sites sufficiently exposed to permit DET to take place, correct orientation of each enzyme at the electrode surface is required to maintain the active site at the distance of closest approach. Finally, even when an electrode can approach sufficiently close to an active site to achieve DET, usually taking the place of a redox substrate/co-substrate, this may not necessarily lead to generation of a bioelectrocatalytic current for substrate electrolysis, as the electrode can block access to the active site of the co-substrate/substrate.

As an alternative to DET, small, artificial substrate/co-substrate electroactive molecules (mediators) can be used to shuttle electrons between the enzyme and the electrode (Figure 5.3b). This involves a process in which the enzyme takes part in the first redox reaction with the substrate and is re-oxidized or reduced by the mediator which in turn is regenerated, through a combination of physical diffusion and self-exchange, at the electrode surface. The mediator circulates continuously between the enzyme and the electrode, cycled between its oxidized and reduced forms, producing current. This process is known as mediated electron transfer (MET).

In MET, the thermodynamic redox potentials of the enzyme and the mediator should be accurately matched. The tuning of these potentials is of critical importance to EFC design as this will have a major bearing on cell voltage and catalytic current. When compared to the redox potential of the enzyme, the mediator should have a redox potential that is more positive for oxidative biocatalysis (at anode) and more negative for reductive biocatalysis (at cathode). For efficient electron transfer,

it has been proposed that mediators and enzyme active sites should have a redox potential difference of approximately 50 mV, to provide a driving force for electron transfer [21]. More detailed studies suggest that, at least for biocatalytic oxygen reduction by a laccase enzyme within a redox hydrogel, an optimum compromise between driving force and current is obtained at 170 mV difference in redox potentials between mediator and enzyme active site [22]. Other considerations in design and selection of mediators include the following.

- The mediator dimensions should be sufficiently small to penetrate the protein structure and access the enzyme active site.
- The mediator should exhibit fast electron exchange rates with both the enzyme
 and the electrode to ensure that the mediator is not limited by electrode kinetics
 and to minimize competition with the enzyme natural substrate, if present.
- The mediator should be chemically stable in both the oxidized and reduced forms, and exhibit rapid diffusion and/or self-exchange, to provide stable and efficient electron transport through a three-dimensional layer.
- The mediator should present a redox potential sufficiently removed from all other electroactive species to avoid interferences and unwanted catalysis.
- The mediator and the mediated reaction should be insensitive to changes in pH or ionic strength.
- The mediator should be easily immobilized, to enable fabrication of devices.

Mediators can be either present in solution or immobilized at electrode surfaces. If contained in solution, separation of anodic and cathodic chambers is necessary to maintain a potential difference (ΔE) between the electrodes and to prevent crossover reactions. For eliminating the need for case and membrane all catalytic components (i.e., enzyme and mediator) should be immobilized at the electrode surfaces. The mediator and enzyme can be immobilized by adsorption onto electrodes, entrapment behind or within membranes and gels, linkage functionalization to the electrode surface, or integration into a polymer layer. Immobilization of mediator and enzyme is essential for cell miniaturization, and consequently EFC prototypes intended for *in vivo* applications tend to follow this approach. Therefore the following sections of this chapter will concentrate predominately on DET and MET for anodes and cathodes in immobilized systems for current/power generation, with some reference to solution-phase catalysis.

Comparison of current densities, and the potential for onset and for maximum current, is hampered when attempts are made, as will be undertaken in the following sections, to report on results from diverse sources, as there is as yet poor adherence to normalization of such data to standard or accepted conditions. The variability in experimental approaches includes differences, often unspecified, in specific surface areas of the electrodes used, the control of mass transport of fuel/oxygen to the surface, the coverage and activity of the enzyme on the surface, the stability of the system, the operational (and optimal) conditions (pH, ionic strength,

buffer and chloride ion concentration, and temperature), and the method used to estimate current density (amperometry at fixed potential, cyclic voltammetry at various scan rates or in a BFC circuit) or power density (fixed load, potentiostatic or galvanostatic control). Where possible all potentials here are reported vs. Ag/AgCl (3 M NaCl) reference electrode. Recent focus on the need to adopt a standardized platform for testing of BFC electrodes, mediators, and configurations, to allow exchange and comparison of results, and to help understand contributions to improve bioelectrocatalysis at electrodes is to be welcomed [22–25].

5.2 Bioanodes for Glucose Oxidation

As in conventional fuel cells, the fuel is oxidized at the anode to release electrons and protons. The selection of the biocatalyst (redox enzyme) will determine the fuel consumed. Redox enzymes, oxidoreductases, are classified according to the nature of reactions that they catalyze. Of these, oxidases, which use oxygen as natural electron acceptor, and dehydrogenases, which transfer hydrogen to either NAD/NADP or a flavin, are most widely used. Comparison of current densities, and the potential for maximum current, is hampered by variability in the conditions used in each research report. Nonetheless, an attempt is made in Table 5.1 to compare the operational performance of bioanodes using a range of redox enzymes and electron transfer strategies (DET and MET) and under a diverse set of conditions. As recent EFC development has been predominately driven by potential in vivo applications, most research has focused on using glucose as a fuel due to its relatively high concentration in blood (about 6-8 mM). Glucose oxidase (GOx) from Aspergillus niger is the most widely used enzyme for glucose oxidation in EFCs owing to its remarkable stability, substrate selectivity, high electron turnover rate, and commercial availability. This enzyme catalyzes the oxidation of β-Dglucose to gluconolactone while reducing oxygen (as the natural electron acceptor) to hydrogen peroxide:

Glucose + GOx-FAD
$$\rightarrow$$
 Gluconolactone + GOx-FADH₂ (5.1)

$$GOx-FADH_2 + O_2 \rightarrow GOx-FAD + H_2O_2$$
 (5.2)

Aspergillus niger GOx is a dimer, composed of two identical $80\,\mathrm{kDa}$ subunits, each containing a tightly bound FAD cofactor (Figure 5.2a) which has a redox potential of $-0.32\,\mathrm{V}$ (vs. Ag/AgCl) at pH = 7 [26]. The tightly bound FAD center is deeply buried (about 1.5 nm) within the protein structure, thus inhibiting DET at smooth electrode surfaces. DET to GOx has, however, been reported to be observed when using carbon nanotubes (CNTs) to molecularly "wire" the FAD active site [27–31]. However, the characteristic redox peaks of FAD are largely masked by background currents. In addition, the presence of CNTs within the protein structure may block substrate access to the enzyme active site and thus deactivate the enzyme. The most persuasive report of DET to a fully active GOx has been presented by Mano

 Table 5.1
 A comparison of the performance of bioanodes.

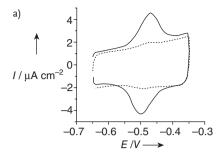
Sugar oxidizing enzyme	Surface	Mediator	Cell conditions	Potential (V) ^{a)}	Current density (mAcm ⁻²) (approx.) ^{b)}	Ref.
Aspergillus niger GOx	Crosslinking to glassy carbon electrode via ultraviolet irradiation	Ferrocene–benzophenone- modified polydimethylacrylamide polymer	pH = 5.2, 37 °C, N ₂ saturated atmosphere, 1000 rpm rotation 15 mM glucose 125 mM glucose	0.45	0.7	[33]
Aspergillus niger GOx	Epoxide crosslinking to glassy carbon electrode	Ferrocene-modified linear polyethyleneimine polymer Dimethylferrocene-modified linear polyethyleneimine polymer	pH = 7.4, 37°C, 100 mM glucose, stirred solution	0.24	2.0	[34]
Aspergillus niger GOx	Epoxide crosslinking to carbon fiber electrode	$\begin{aligned} & \text{PVI-[Os(4,4'\text{-}dimethyl-2,2' -} \\ & \text{bipyridine}_2\text{CI]}^{1/2+} \\ & \text{coacrylamide} \end{aligned}$	pH = 5, 23 °C, 15 mM glucose	0.2	0.2	[35]
Aspergillus niger GOx	Epoxide crosslinking to carbon fiber electrode	Polyvinypyridine-[Os(4,4' - dimethoxy-2,2' - bipyridine);Cl] ^{+/2+}	pH = 7.4, 37.5 °C, 15 mM glucose	0	0.6	[36]
Aspergillus niger GOx	Epoxide crosslinking to carbon fiber electrode	$\mathrm{PVI-[OS(4,4'\text{-}diamino-2,2'} \ \cdot \ \mathrm{bipyridine)_2Cl]^{+/2+}}$	pH = 7.4, 37.5 °C, 0.14M NaCl, 15mM glucose	-0.05	0.2	[37]
Aspergillus niger GOx	Epoxide crosslinking to carbon fiber electrode	Polyvinylpyridine– $[Os(N,N'-dimethyl-2,2' \cdot biimidazole),Cl]^{2r/3+}$	pH = 7.2, 37.5 °C, argon saturated atmosphere 15 mM glucose	-0.1	1.3	[38]
Penicillium pinophilum GOx	Epoxide crosslinking to carbon fiber electrode	Polyvinylpyridine– $[Os(N,N'-dimethyl-2,2'$ - biimidazole),Cl] ^{2+/3+}	pH = 5, 37 °C, 32 mM glucose, 500 rpm rotation	-0.1	0.8	[39]

							5.2 Bioan	odes for Gli	ıcose Oxidation	237	
[32]	[40]		[41]		[42]			[43]	[44]	[45]	(Continued)
0.2	0.2	0.7		0.8	1	0.1	0.3	1.2	0.02	4	
-0.2	0.2	9.0	0		0.1			0.2	0	0.4	
pH = 7.4 , 45 mM glucose, 500 rpm rotation	pH = 7 10 mM glucose	100 mM glucose 40 mM glucose	pH = 7, 37 °C, 1 mM NAD, 1000 rpm rotation	20mM glucose	pH = 6, 20 °C. 10 mM NAD	15 mM glucose	90mM glucose	pH = 5, 25 °C, 50 mM glucose, 500 rpm rotation	pH = 7, 25 °C, 20 mM glucose	pH = 5, 25 °C 200mM fructose, stirred solution	
No mediator	Vitamin K ₃ -modified poly(L-lysine)		Vitamin K ₃ -modified poly(L-lysine)		Poly(methylene blue)			$PVI-[Os(4,4'\text{-}dimethyl\text{-}2,2' \ \cdot \ bipyridine)_2Cl]^{7/2+}$	1-Methoxyphenazine methosulfate	No mediator	
Adsorption to glassy carbon electrode	Adsorption onto a glassy carbon electrode	Adsorption onto a carbon fiber electrode	Adsorption onto a Ketjen black modified	glassy carbon electrode	Glutaraldehyde crosslinking to	poly(methylene	composite, deposited on glassy carbon electrode	Epoxide crosslinking to glassy carbon electrode	Glutaraldehyde crosslinking to carbon paste electrode	Adsorption onto a Ketjen black-modified carbon paper electrode	
Aspergillus niger GOx (deglycosylated)	Bacillus sp. GDH (NAD	diaphorase	GDH (NAD dependent),	diaphorase	GDH (NAD dependent)	•		Acinetobacter calcoaceticus GDH (PQQ dependent)	Acinetobacter calcoaceticus GDH (PQQ dependent) mutant	Gluconobacter sp. FDH	

Table 5.1 (Continued)

Sugar oxidizing enzyme	Surface	Mediator	Cell conditions	Potential (V) ^{a)}	Current density (mA cm ⁻²) (approx.) ^{b)}	Ref.
Gluconobacter sp. FDH	Alkanethiol-modified gold nanoparticles deposited on carbon fiber paper	No mediator	pH = 5, 25 °C 200 mM fructose	0.4	14	[46]
Gluconobacter industrius FDH	Adsorption to cellulose–MWCNT-modified glassy carbon electrode	No mediator	pH = 5 , 15mM fructose	0.3	0.8	[47]
Agaricus meleagris PDH	Epoxide crosslinking to graphite rod electrodes	Polyvinylpyridine– $[Os(N,N'-dimethyl-2,2' \cdot biimidazole)3Cl]^{2+/3+}$	pH = 8.5, 25 °C, 3 mM glucose, flow rate 1 ml/ min	-0.1	0.02	[48]
		PVI-[Os(4,4'-dimethoxy- $2,2'$ -bipyridine) $_2$ CI] $^{+/2+}$		-0.1	0.15	
		$PVI-[Os(4,4'\text{-}dimethyl\text{-}2,2'\text{ -} bipyridine})_2CJ]^{+/2+}$		0.1	0.15	
		$PVI-[Os(2,2'\ \cdot\ bipyridine)_2CI]^{+/2+}$		0.15	0.16	
Phanerochaete	Epoxide crosslinking to	No mediator	pH = 3.5, 25°C, $100 mM$	0.0	0.06	[49]
sordida CDH	SWCNTs deposited on pyrolytic graphite electrode	Polyvinylpyridine– $[Os(N,N'-dimethyl-2,2' - biimidazole),Cl]^{2+/3+}$	lactose	0.0	0.6	

Potential at which a steady-state current for oxidation is observed, corrected to the Ag/AgCl (3M NaCl) reference. Current densities are for the projected two-dimensional geometric underlying electrode area. a) b)



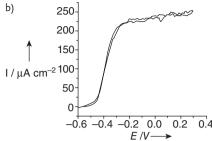


Figure 5.4 (a) Cyclic voltammograms of GOx (dotted curve) and dGOx (solid cirve) adsorbed on glassy carbon electrodes, 20 mM phosphate buffer, pH = 7.4, 37 °C, scan rate of of 5 mV s⁻¹, 500 rpm, argon atmosphere). 20 mV s⁻¹, argon atmosphere. (b) Direct electrooxidation of glucose (45 mM) on a

monolayer of deglycosylated Aspergillus niger GOx adsorbed on a glassy carbon electrode $(20 \,\mathrm{mM})$ phosphate buffer, pH = 7.4, scan rate From [32] with permission from Wiley-VCH.

and coworkers [32] who report a redox couple in cyclic voltammograms at -0.49 V vs. Ag/AgCl, attributed to the FAD/FADH2 redox process within a deglycosolated GOx (dGOx) adsorbed on a glassy carbon electrode (Figure 5.4a), although this redox potential is more negative than that observed by others for GOx-FAD/ FADH₂. In the presence of 45 mM glucose a current density of 235 µA cm⁻² at -0.2 V vs. Ag/AgCl is observed indicating that catalytic activity of the enzyme is retained (Figure 5.4b).

Despite the simplicity of DET, current densities are usually limited as only monolayer coverage is possible and enzymes may be electrically insulated due to incorrect orientation on the electrode surface. The simplest way to overcome these limitations is to immobilize the enzyme within a polymeric mediator matrix on the electrode surface. This enables multilayer enzyme coverages and, if the mediator can access the enzyme active site, the orientation of enzyme becomes irrelevant.

Much of the early research on MET to GOx stems from the development of enzyme electrodes for "second-generation" MET glucose sensing [50]. This involved replacing oxygen, used in "first-generation" glucose biosensors, with a synthetic electron acceptor to improve sensitivity and accuracy for determination of glucose concentrations in blood. Cass et al. [51] first demonstrated the use of ferrocene derivatives as mediators co-immobilized with GOx at a pyrolytic graphite electrode. This led to numerous studies examining ferrocene-GOx electron transfer reactions [52-56] including a comprehensive study by Forrow et al. [56] investigating the influence of ferrocene derivative structure on GOx mediation. Although adequate for single-use glucose biosensing, ferrocene derivatives are not particularly well suited as anodic mediators in EFCs due to their relatively high redox potentials (about 0.35 V vs. Ag/AgCl), and the instability of their oxidized form. The high redox potentials contribute to large thermodynamic losses in an EFC as shown by Liu et al. [57] using ferrocene monocarboxylic acid as an anodic mediator of GOx. Bunte *et al.* [33] recently described an EFC anode based on films of a ferrocene–benzophenone-modified polymer crosslinked with GOx via ultraviolet irradiation on glassy carbon. Although a high current density was achieved (about $1.2\,\mathrm{mA\,cm^{-2}}$ at $0.45\,\mathrm{V}$ vs. Ag/AgCl), the potential remains relatively high for practical use in an assembled EFC. A more appropriate bioanode was developed by Meredith *et al.* [34] using GOx crosslinked with dimethylferrocene-modified polyethyleneimine polymer on a glassy carbon electrode, to provide about $2\,\mathrm{mA\,cm^{-2}}$ at $0.24\,\mathrm{V}$ vs. Ag/AgCl at pH = 7.4. The electron-donating alkyl groups not only shifted the redox potential of the ferrocene but also resulted in better stability compared to unsubstituted ferrocene–polymers, reflected by superior power retention in assembled EFCs.

To date, osmium redox polymers are the most successful mediators for GOx in terms of current density and stability. As with ferrocenes, much of the early research concerning osmium redox polymers as mediators for GOx centered on their use in enzyme electrodes for glucose sensing [58]. Heller and coworkers [59, 60] first reported the use of the redox polymer [Os(2,2'-bipyridine)₂(polyvinylimid azole)₁₀Cl]^{+/2+}, depicted in Figure 5.5a, as an efficient mediator for glucose oxidation by GOx when co-immobilized with GOx at a glassy carbon electrode. This class of water-soluble redox polymer has the advantage of forming, when crosslinked with enzymes, hydrogel films on electrodes, permitting ingress and egress of substrate, solvent, and counter-ions. Shifting the Os(II/III) redox potential from +0.25 V vs. Ag/AgCl to lower, more suitable, values can be achieved through introduction of electron-donating functional groups on the bipyridyl ligands coordinated to the osmium redox center. De Lumley-Woodyear *et al.* [61] applied this approach to lower the redox potential by replacing the 2,2'-bipyridine

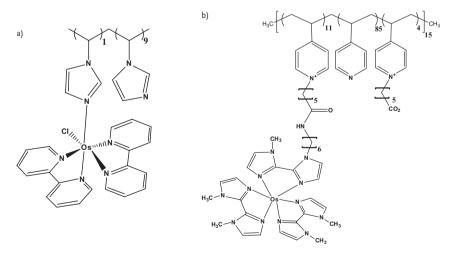


Figure 5.5 Structure proposed for the osmium redox polymers (a) $[Os(2,2'-bipyridine)_2(PVI)_{10}CI]^+$ and (b) polyvinylpyridine– $[Os(N,N'-dimethyl-2,2'-biimidazole)_3CI]^{2+}$.

ligands of osmium with 4,4'-dimethyl-2,2'-bipyridine, resulting in biocatalytic oxidation of glucose at about +0.1 V vs. Ag/AgCl [35, 61]. Further refinement using 4,4'-dimethoxy-2,2'-bipyridine [36] and 4,4'-diamino-2,2'-bipyridine [37] as ligands yielded redox polymers with potentials of -0.1 V and -0.15 V vs. Ag/AgCl, respectively. Hydrogels formed using the latter polymer crosslinked with GOx on carbon fiber electrodes yielded current densities of about 200 µA cm⁻² at about -0.1 V vs. Ag/AgCl in solutions containing 15 mM glucose [37]. Electron transport within these redox hydrogels is controlled by the process of self-exchange between the oxidized and reduced osmium centers, which requires segmental chain motion of the polymer backbone to bring the centers close enough to enable exchange to occur, and can limit the current generated for glucose oxidation. This process may be characterized as diffusional, and an apparent charge (electron) transport diffusion coefficient can be estimated using voltammetry. The apparent electron transport diffusion coefficient, Dapp, for a polyvinylimidazole (PVI)-[Os(4,4'diamino-2,2'-bipyridine)₂(PVI)Cl]^{+/2+} hydrogel was estimated as 1.2×10^{-9} cm² s⁻¹ [62]. Subsequently, an improved $D_{\rm app}$ of $5.8 \times 10^{-6} \, {\rm cm^2 \, s^{-1}}$ for electron transport through an osmium-based hydrogel was achieved by Mao et al. [62] using the novel redox polymer polyvinylpyridine–[Os(N,N'-dimethyl-2,2'-biimidazole)₃Cl]^{2+/3+} (Figure 5.5b). The increase in D_{app} was attributed to the 13-atom-long flexible spacer arm situated between the polyvinylpyridine polymer backbone and an alkyl functional group of an $[Os(N, N'-dialkylated-2, 2'-biimidazole)_3]^{2+/3+}$ redox center, which increases the rate of electron transferring self-exchange collisions between the osmium centers. A current density of 1.5 mA cm⁻² at -0.1 V vs. Ag/AgCl was obtained at a carbon fiber electrode coated with crosslinked films of GOx and this polymer in solutions containing 15 mM glucose (pH = 7.4) in phosphate buffered saline at 37°C [38]. With a view to improving current generation at lower glucose concentrations, Mano [39] replaced A. niger GOx ($K_m = 33 \,\mathrm{mM}$ for glucose) [63] with Penicillum pinophilum GOx ($K_m = 6.2 \,\mathrm{mM}$ for glucose) [64] as biocatalyst in this hydrogel. Although improved current densities were achieved at pH = 5 using hydrogels containing P. pinophilum GOx at low glucose concentrations, operation of an EFC under physiological conditions would be hindered by the instability of *P. pinophilum* GOx at temperatures above 40° C in the range of pH = 7–8.

Lowering the K_m value of a fuel-oxidizing enzyme is one of the many challenges faced when utilizing natural wild-type enzymes for EFC applications. Other issues, depending on the intended application, include enzyme immobilization and orientation, temperature and pH stability, salt inhibition, and substrate specificity. Such limitations may be overcome through protein engineering [65–68]. For example, recent reports by Schwaneberg and coworkers [66–68] describe a medium-throughput screening system for improving the properties of *A. niger* GOx through directed protein evolution. Using this method several mutants were obtained which displayed improved pH and thermal resistance, and higher activity for β -D-glucose oxidation compared to wild-type *A. niger* GOx. Despite these improvements, the natural reaction of GOx with oxygen (Eq. 5.2) represents a significant drawback for inclusion of this enzyme at the anode as oxygen competes with the electrode, or artificial electron acceptor, for electrons and in so doing can produce

harmful hydrogen peroxide. In addition, oxygen consumption at the anode will reduce the concentration of oxidant available to the cathode in a membraneless EFC.

This has led to an increased focus on the use of enzymes which oxidize fuels yet do not produce hydrogen peroxide, such as the dehydrogenases. The majority of dehydrogenases contain a loosely bound NAD+ cofactor (Figure 5.2b) which acts as a two-electron and one-proton carrier. The NAD⁺ cofactor ($E^0 = -0.52 \,\mathrm{V}$ vs. Ag/ AgCl at pH = 7) [69] itself is not an effective mediator due to the large overpotential required for direct oxidation at solid electrodes and lack of reversibility of the NAD+/NADH redox process. Several reports describe appreciable current densities for oxidation of glucose catalyzed by NAD+dependent glucose dehydrogenase (GDH) at low overpotentials (>1 mA cm⁻² at about 0.1 V vs. Ag/AgCl) using diaphorase, which catalyzes NADH oxidation, coupled to reduction of vitamin K₃ as mediator, all adsorbed at high-surface-area carbon electrodes [40-42, 70]. Yan et al. [42] employed polymethylene blue as mediator for NAD+-dependent GDH, adsorbed at single-walled carbon nanotube (SWCNT) electrodes, generating currents of $0.33 \,\mathrm{mA\,cm^{-2}}$ at $0.1 \,\mathrm{V}$ vs. Ag/AgCl in buffered solutions (pH = 6) containing 60 mM glucose. Tsujimura et al. [43] first reported the use of a POO-dependent GDH at the anode of an EFC operating at physiological conditions. The PQQ cofactor (Figure 5.2c), which has a redox potential of -0.13 V vs. Ag/AgCl at pH = 7 [71], is tightly bound to the enzyme and is thus more practical than its NAD+dependent counterpart. However, the wild-type PQQ-dependent GDH used is unstable, severely impeding its performance at physiological conditions [70]. Sode and coworkers produced several PQQ-dependent GDH mutants which show increased thermal stability [44] and improved DET properties [44, 71, 72] for use in EFCs operating under physiological conditions. In addition, Durand et al. [73] recently reported a soluble PQQ-GDH mutant, obtained by site-directed mutagenesis, which displayed twofold higher activity towards glucose oxidation compared to wild-type GDH.

In addition to glucose, other saccharides, such as fructose, have been studied as potential fuels. Fructose dehydrogenase (FDH), for example, has been used in a number of studies for oxidation via DET of fructose at carbon black-modified [45, 46] and CNT-modified [47] electrodes. Kamitaka et al. [45] observed current densities of about 4 mA cm⁻² at carbon paper electrodes modified with FDH in solutions containing 200 mM fructose at pH = 5. Gorton and coworkers have published extensively on the use of cellobiose dehydrogenases (CDHs) in biosensors and as putative bioanodes in EFCs [17, 49, 74-79]. CDH, which contains a flavin and heme domain, connected by a flexible linker region, is capable of both DET and MET, as depicted in Figure 5.6. DET occurs through intramolecular electron transfer from the substrate-oxidizing flavin to the heme domain followed by DET from the heme to the electrode surface [49]. Alternatively, electron transfer from the flavin domain can be facilitated through mediation by an appropriate mediator such as an osmium redox complex [49]. SWCNTs on graphite electrodes modified with CDH sourced from Phanerochaete sordida produced DET and MET current densities of 0.06 and 0.6 mA cm⁻², respectively, in solutions containing 100 mM

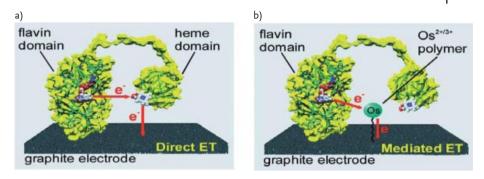


Figure 5.6 Schematic showing (a) DET and (b) MET for CDH. From [49] with permission from the American Chemical Society.

lactose [49]. Recent research has focused on production of CDHs that can efficiently oxidize glucose, instead of lactose, under physiological conditions, for application as glucose-oxidizing bioanodes in EFC configurations [17, 79]. Bioanode coulombic efficiencies can be greatly increased through combination of rationally selected enzymes to provide for more extensive (deeper) oxidation of the biofuel [80]. This was first demonstrated in an EFC anode by Palmore *et al.* [81] using a mixture of solution-phase dehydrogenases to catalyze the successive oxidation of methanol to carbon dioxide. Subsequently, Minteer and coworkers employed multiple-enzyme cascades for full or partial oxidation of ethanol [82, 83], pyruvate [84], and glycerol [85] in EFC assemblies. Recently, modified electrodes combining CDH and pyranose dehydrogenase (PDH) have been shown to be capable of extracting up to six electrons from one molecule of glucose [86].

5.3 Biocathodes

Several studies focused on understanding the factors that contribute to provide power in BFCs utilize hybrid systems, combining anodes that exploit the advantages of a biological catalyst with oxygen reduction reaction (ORR) catalysts used in traditional fuel cell research, such as platinum, platinum alloys, and other metal alloys [84, 85]. Such hybrid biological fuel cells, however, still require an ion-exchange membrane to separate the anolyte from the catholyte, and to efficiently transfer charge, usually protons, to maintain electro-neutrality and sustain the proton-coupled reduction of oxygen to peroxide or water. This is necessary as the ORR catalysts are not selective towards oxygen reduction alone, and may also oxidize the fuel, and/or be passivated by products of the fuel oxidation pathway. In addition the ORR at platinum at relatively low temperature and neutral pH has been shown to occur less efficiently than at electrodes modified with selected biological catalysts [87]. Stability, inhibition, and modulation of the biocatalytic

processes remain, however, major problems for technological advances in the adoption of prototype biocatalytic cathodes on an industrial scale.

The theoretical thermodynamic reduction potential for oxygen to water is +1.23 V vs. NHE at pH = 0, or +0.82 V vs. NHE at pH = 7. The direct reduction of oxygen at metal-free carbon electrodes in neutral electrolytes is, however, hampered by slow electrode kinetics [88]. Catalysts, such as platinum, are therefore used to increase electrode kinetics in traditional fuel cell cathodes. However, the use of expensive, and nonselective, platinum catalysts is less compatible with operation of miniaturized membraneless fuel cells for portable systems or for application to in vivo power generation. An additional disadvantage of oxygen reduction at both metal-free carbon and platinum electrodes is that reduction occurs, at neutral pH, via a two-electron process to produce peroxide, a toxic reactive oxygen species. Recent research on the ORR has focused on replacement of platinum catalysts by less expensive metal alloys and co-catalysts, with some degree of success, beyond the scope of this chapter. The biological catalysts most widely studied for application to ORR in BFCs are a class of enzymes termed multiple copper oxidases (MCOs) that include laccases (EC 1.10.3.2), ascorbate oxidases (EC 1.10.3.3), bilirubin oxidases (EC 1.3.3.5), and others, that can reduce oxygen directly to water without apparent release of a peroxide intermediate. The first report of a MCO in a fuel cell for reduction of oxygen [89] used a solution-phase laccase isolated from Pyricularia oryzae and 2,2'-azinobis(3-ethylbenzothiazoline-6-sulfonate) (ABTS) as a diffusional mediator.

The MCOs generally contain at least four copper atoms in their complex active site, classified according to the electron paramagnetic resonance signals [19, 90-92]. One of these copper atoms, termed type 1 (T1), responsible for the blue color of copper proteins and enzymes, contains copper coordinated to at least two histidines and is the substrate oxidizing site of the MCOs. This substrate oxidation is coupled through intramolecular electron transfer via a conserved cysteinehistidine bridge to a trinuclear cluster of copper, consisting of one type 2 (T2) and two type 3 (T3) coordinated copper, approximately 1.3 nm distal to the T1 site, that reduces oxygen directly to water. While the exact detailed mechanism of oxygen reduction has yet to be clarified, recent computational, spectroscopic, and electrochemical studies on wild-type and mutant MCOs [92] indicate that oxygen binds to the trinuclear copper cluster of the reduced MCO, to form a peroxide intermediate, followed by intramolecular electron transfer and O-O bond cleavage to yield a fully oxidized, native intermediate, which reverts to the reduced form by accepting four electrons consecutively from the substrate with release of two molecules of water, as depicted in Figure 5.7.

Laccases are the most widely studied of the MCOs for application as biological ORR catalysts. Laccases, classed as polyphenol oxidases, catalyze the oxidation of diphenols, amines, and some inorganic ions as substrates, coupled to the reduction of oxygen to water. The substrate binding pocket at the T1 copper is rather open, resulting in the broad substrate range for the laccases, and this substrate range can be extended by the use of mediators [93], resulting in technological application of laccases in a range of applications, from biosensors [94–97], biob-

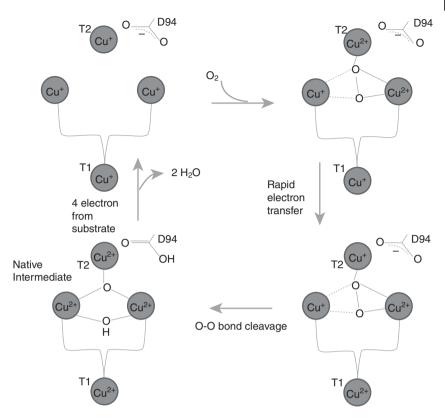


Figure 5.7 Proposed mechanism for reduction of oxygen to water at MCO active sites. Adapted from [92] with permission from the American Chemical Society.

leaching, and biodegradation [98-101], to EFCs. Laccase was first isolated by Yoshida [102] in 1883 from tree lacquer of Rhus vernicifera, with laccases subsequently isolated from plant, fungal, or, more recently, bacterial and insect sources [103]. Interest in the use of laccases as ORR catalysts stems from the reported low overpotential for ORR of a range of fungal laccases compared to that of platinum. The low overpotential for ORR has been ascribed to the relatively high reduction potential of their T1 copper sites, the capacity of the MCOs to utilize multi-atomic reaction sites, the influence of residues close to the active site on the reaction mechanism, and the structural flexibility of both the catalyst protein shell and the Cu–Cu interatomic distances of the active site, in comparison to simpler inorganic metal clusters [92, 103, 104]. The fungal laccases, in general, have higher T1 site redox potential, about 0.58 V vs. Ag/AgCl, than the plant laccases (e.g., 0.23 V vs. Ag/AgCl for the R. vernicifera laccase), with ligand coordination geometry around the copper atom, and in particular the presence of weakly axially coordinated residues, contributing to the differences. The MCOs have also been shown to transfer electrons directly with electrode materials, allowing estimation of the T1 redox potential using voltammetry, and possible correlation with structure and activity.

DET to a laccase was first reported by Yaropolov's group for enzymes adsorbed onto carbon electrodes [105]. Subsequent studies have investigated DET to the copper sites of MCOs leading to their classification, as suggested by Shleev *et al.* [19, 106], into separate groups, based upon the redox potential of the T1 site. The plant, bacterial, and some fungal laccases have a low T1 potential of about +0.23 V vs. Ag/AgCl; bilirubin oxidases (BODs) and some fungal laccases possess T1 sites of middle potential between +0.27 and +0.51 V vs. Ag/AgCl; and fungal laccases, especially isolated from *Trametes* sources, possess high T1 potentials of about +0.58 V vs. Ag/AgCl. Initial studies of MCOs as ORR catalysts, not surprisingly therefore, focused on the use of these *Trametes* laccases.

Unfortunately, the high-redox-potential fungal laccases are inhibited by hydroxyl ions, with maximal activity centered around pH = 4-5, and, to a lesser extent, by chloride ions [107]. This limits their use in neutral pH or saline conditions, for example as biocathodes in a putative implantable EFC. This low activity of most fungal laccases under pseudo-physiological conditions has led to increased focus on MCOs that are more active under those conditions. These include BODs [108], certain fungal laccases, such as that isolated from Melanocarpus albomyces [109– 111], bacterial MCOs such as the copper efflux oxidase (CueO) isolated from the periplasmic space of Escherichia coli [112-116], CotA laccase isolated from Bacillus subtilis [117, 118], and small laccase from Streptomyces coelicolor (SLAC) [119, 120]. The first report on a BOD-based biocatalytic oxygen cathode focused on homogeneous ABTS-mediated reduction of oxygen at carbon felt electrodes using a BOD from Myrothecium verrucaria in phosphate buffer (pH = 7.0) [109]. However, the potential of the T1 site of these MCOs is of low to middle classification only, with activity at neutral pH for the ORR apparently being compromised by a decrease in the onset of the MCO-catalyzed ORR potential [108, 121].

DET to MCOs, for bioelectrocatalytic reduction of oxygen, is of immense interest to biocathode research and development, as this usually enables ORR onset at the T1 copper reduction potential, which can be close to the thermodynamic potential for oxygen reduction for the high-redox-potential laccases. For the reasons outlined above, DET to MCOs is surface specific. DET for the ORR is reported to occur through the four-electron reduction of oxygen to water, with the T1 copper site as the proposed linkage between electrode and MCO, for MCOs physically adsorbed at spectroscopic graphite and carbon aerogels [114, 122-124]. However, DET for the ORR is thought to occur through a different mechanism for MCOs on bare gold [123, 125, 126]. The earliest studies, focused on examining the possibility for DET to MCOs, were conducted by adsorption of fungal and plant laccases on carbon black [127] or spectroscopic graphite electrodes [128] of high surface areas, providing proof of DET through bioelectrocatalytic oxygen reduction at low overpotentials at MCO-modified surfaces compared to bare carbon electrodes. Comparison of current densities, and the potential for maximum ORR current, is hampered, however, by variability in the conditions used in each research report. Nonetheless, an attempt at a limited compilation of the more recent configurations of MCO-based biocathodes using DET for the ORR can be made, as summarized Table 5.2.

 Table 5.2
 A comparison of the performance of biocathodes for the ORR, based on DET to MCOs.

МСО	Surface	Conditions for ORR	Potential (V) ^{a)}	Current density (mA cm ⁻²) ^{b)}	Ref.
Rhus vemicifera laccase	EDC coupling to a mercaptopropionic acid-treated gold electrode	pH = 7.0 , 21 °C, aeration of cell	0.1	0.080	[129]
Myrothecium verrucaria BOD	Adsorption to spectroscopic graphite electrode	pH = 4.0	0.3	0.03	[128]
Myrothecium verrucaria BOD	Adsorption onto a polylysine-modified plastic formed carbon electrode (high edge plane density)	pH = 7.0 , 25°C, aeration of cell, 1400 rpm rotation	0.25	0.85	[121]
Trametes versicolor laccase Myrothecium verrucaria BOD	Glutaraldehyde crosslinking to MWCNT, solubilized using cellulose derivatives, on glassy carbon electrode	pH = 6.0 , aeration of cell pH = 7.0 , aeration of cell	0.45	0.06	[130]
Myrothecium verrucaria BOD	EDC coupling to MWCNT, deposited on glassy carbon electrode	pH = 7.0, aeration of cell	0.33	0.04	[131]
Trametes sp. laccases	Adsorption to carbon aerogel-modified carbon paper electrode	pH = 5.0 , 25 °C, aeration of cell, 3000 rpm rotation	0.4	6.5	[124]
Myrothecium verrucaria BOD		pH = 7.0 , 25 °C, aeration of cell, 2000 rpm rotation	0.3	3.5	
Rhus vernicifera laccase Trametes sp. laccase	Adsorption to highly oriented pyrolytic graphite	pH = 5.0 , 25 °C, aeration of cell	0.1	0.052	[45]
Myrothecium verrucaria BOD Trametes sp. laccase	Adsorption to carbon aerogel-modified carbon paper		0.4	0.205	
Trametes hirsuta laccase	EDC coupling to aminophenylmodified low density graphite electrode	pH = 5.1, 27°C	0.2	0.5	[122] –
				9)	(managaga

Table 5.2 (Continued)

МСО	Surface	Conditions for ORR	Potential (V) ^{a)}	Current density (mA cm ⁻²) ^{b)}	Ref.
Myrothecium verrucaria BOD	EDC coupling to organic carboxylates (PQQ best) adsorbed on MWCNT,	pH = 7.0, aeration of cell unstirred	0.25	0.5	[125]
	deposited on gold electrode	1300 rpm stirring		1.6	
E. coli CueO	Adsorption onto a Ketjen black- modified glassy carbon electrode	pH = 5.0, 25 °C, aeration of cell, $10000 \mathrm{rpm}$ rotation	0	12	[114]
		pH = 7.0 , 25 °C, aeration of cell, 1000 rpm rotation	0	4	
Cerrena unicolor laccase	Adsoption onto CNT-loaded glassy carbon electrode with Nafion binder	pH = 5.2 , 22 °C, aeration of cell	0.2	0.058	[132]
E. coli CueO	Adsorption onto a Ketjen black- polytetrafluoroethylene binder- modified carbon paper electrode	pH = 5.0, 25 °C, air diffusion configuration	-0.2	20	[116]
Cerrena unicolor laccase	Entrapment within carbon nanoparticle-loaded sol–gel on indium tin oxide electrode	pH = 4.8, 22 °C, aeration of cell	0.2	0.08	[133]
Myrothecium verrucaria BOD	Entrapment within carbon nanoparticle loaded sol-gel	22°C, aeration of cell nH = 5	0.2	0.23	[134]
		p = Hq		0.255	
		pH = 7		0.13	
Trametes hirsuta laccase	Adsorption to a gold nanoparticle dispersion on glassy carbon electrode	pH = 4.0	0.5	0.005	[135]

Potential at which a steady-state current for the ORR is observed, corrected to the Ag/AgCl (3M NaCl) reference. Current densities are for the projected two-dimensional geometric underlying electrode area. a) b)

Addition of conducting particles and tubes possessing nanometric dimensions has been postulated to improve DET for ORR by orienting the MCOs in a matrix of conductive electrode material that can approach the T1 copper site. For example, increased current density for ORR is observed for Trametes sp. laccase adsorbed onto three-dimensional carbon aerogel-modified electrodes, compared to that observed when the enzyme is adsorbed onto highly oriented pyrolytic graphite [45]. Addition of CNTs [124, 125, 131, 134], and carbon [114, 124, 133] or gold nanoparticles [135] to an electrode matrix, either drop-coated or by incorporation into a sol-gel [135], has also yielded substantial currents for the ORR when MCOs are either adsorbed or coupled to the nanostructures. Other approaches for achieving increased current density focus on retaining the MCOs in a hydrogel close to the active electrode surface, for example by using a polylysine matrix, resulting in a reaction-layer model where the enzyme, BOD in this case, can diffuse to and from the electrode [121]. An elegant approach for achieving improved DET current density for the ORR using a laccase, isolated from Pycnoporus cinnabarinus, proposed retention to a pyrolytic graphite electrode surface via a surface layer of anthracene, introduced via in situ diazotization of anthracene amine followed by electroreduction of the anthracene-diazonium salt. It was proposed that the anthracene orients and retains the laccase through interaction with a hydrophobic pocket close to the T1 active site, permitting DET, yielding a current density that is twice that for laccase adsorbed to an unmodified electrode. More importantly, this study reported that laccase on the anthracene-modified electrode maintained activity for ORR at levels exceeding 50% of the initial activity for over eight weeks, when the electrodes were stored hydrated at 4°C between experiments (Figure 5.8) [136].

More recently, isolation of a MCO from the periplasmic space of E. coli, proposed to have a role in copper regulation [112], has led to its use as an ORR biocatalyst [113-116]. This CueO, while requiring a fifth copper to yield oxidase activity in the presence of mediators such as ABTS or phenylenediamine, has been shown to be a highly efficient catalyst for the ORR through DET at nanostructured carbon surfaces [114]. These surfaces are prepared by addition of a slurry of Ketien black particles and binder to glassy carbon electrodes, and yield current densities exceeding $12 \,\mathrm{mA\,cm^{-2}}$ at pH = 5.0 for the ORR. This, however, comes at a thermodynamic cost, as steady-state ORR current is observed at a potential of 0V vs. Ag/ AgCl, approximately 0.4V more negative than the potential for steady-state ORR at electrodes modified with fungal laccases.

An approach focused on fabrication of nanostructured three-dimensional electrodes and introduction of surface modifications for tethering/retention in an optimal orientation of the MCOs to permit DET to the T1 site from the electrode shows great promise for the production of biocathode prototypes for application to EFCs. A systematic study of such electrodes modified with each of the MCOs available, reporting on their activity for ORR, using DET, under defined conditions of pH, mass transport, and temperature is not yet available, and would be a valuable contribution to advance the technological application of EFCs. A welcome recent focus is normalization of ORR, based on DET to Trametes versicolor adsorbed on porous carbon-based electrode materials, to electrode volume and to electrode

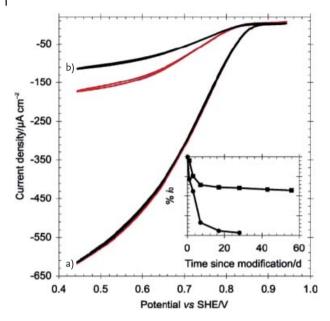


Figure 5.8 Cyclic voltammogram (5 mV s⁻¹, 2500 rpm) recorded for laccase adsorbed to (a) anthracene-modified and (b) unmodified pyrolytic graphite electrodes immediately following laccase deposition (black) and upon exchange of electrolyte (red). The inset shows

long-term change in current density recorded at 0.44 V vs. SHE for modified (squares) and unmodified (circles) electrode, relative to the first wave with no enzyme in solution.

Adapted from [136] with permission from the Royal Society of Chemistry.

Brunauer–Emmet–Teller (BET) estimated surface areas [23]. For example, from Figure 5.9, graphite felt electrodes show poor volume-normalized ORR current density compared to carbon nanofibers and multiwalled carbon nanotube (MWCNT)-based electrodes. However, the results also reveal that CNTs and porous carbon tubes exhibit dramatically lower ORR current densities when normalized to BET surface area, while graphite felt electrodes perform better, perhaps indicative of agglomeration of the carbon tubes, preventing enzyme adsorption over the entire area. Further research on methods to permit dispersion of nanotubes, while retaining electrical conductivity and adsorption of enzymes oriented for DET, is warranted.

Recent research on biocathode design has focused on gas-diffusion layers at the cathode to afford air-breathing biocathode configurations. Such configurations can overcome the limitation on current imposed by the availability of oxygen to the cathode due to its low solubility and diffusion coefficient in the aqueous electrolytes used heretofore. For example, CueO was adsorbed onto porous carbon paper, modified with carbon nanoparticles (Ketjen black) using hydrophobic binders, to yield biocathodes that could operate in air, permitting oxygen diffusion from air to the particles with minimal loss of electrolyte, as illustrated schematically in

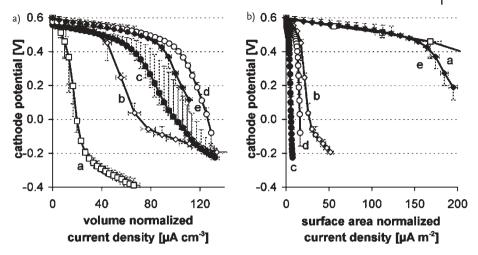


Figure 5.9 Cathode polarization plots for electrodes with laccase adsorbed to graphite felt (curves a), porous carbon tubes (curves b), SWCNTs (curves c), MWCNTs (curves d), and carbon nanofibers (curves e).

(a) Comparison of cathode performance with

a current density normalized to volume; (b) comparison of cathode performance with a current density normalized to BET surface area. From reference [23] with permission from Elsevier.

Figure 5.10 [116]. Such biocathodes yielded current densities in the mAcm⁻² range. However, the use of CueO as a biocatalyst meant that steady-state currents for ORR were only observed at potentials more negative than $-0.2\,\mathrm{V}$ vs. Ag/AgCl. It was also suggested that the current was limited by supply of protons, as the current was dependent on buffer concentration in the electrolyte. Use of a 1.0 M citrate electrolyte (pH = 5.0), resulted in a steady-state current density of $20\,\mathrm{mA\,cm^{-2}}$, the highest yet reported for the ORR at a biocathode.

Advances in nanostructured conducting materials for DET have resulted in impressive current densities for the ORR, and application of these three-dimensional materials to DET from MCOs other than CueO may provide biocathodes with the characteristics suitable for an implantable EFC. While a DET approach using MCOs can provide for ORR at potentials approaching the thermodynamic reduction potential for oxygen, the current density achievable in this approach still relies upon intimate contact, and correct orientation, of the MCO to a conducting surface. Use of a mediator, capable of close interaction with the T1 site of the MCOs, and with a redox potential tailored to permit rapid electron transfer to the T1 site, can eliminate the requirement for direct contact in the correct orientation between MCO and electrode, and offer the possibility of a three-dimensional biocatalytic reaction layer on electrodes for higher ORR current densities.

Initial reports on mediated ORR using MCOs focused on assessing the capability for ORR using solution-phase mediators with solution-phase MCOs [89].

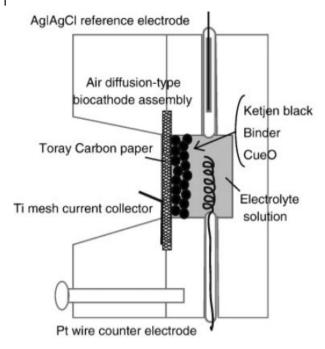


Figure 5.10 Schematic representation of the air diffusion biocathode half-cell based on CueO adsorbed on carbon particle-modified carbon paper electrode. Adapted from [116] with permission from Elsevier.

Immobilization of both mediator and MCO on relatively inert (to ORR) carbonbased electrodes can provide biocathodes for miniaturized membraneless BFCs. Trudeau et al. [96] were first to report on oxygen reduction by films of immobilized mediator and laccase, formed by crosslinking a Trametes versicolor laccase to the osmium-based redox polymer in Figure 5.5a, physisorbed on carbon electrodes. Although the laccase biocatalytic electrode for ORR was developed for detection of modulators of enzyme activity, steady-state current densities of greater than 125 µA cm⁻² were achieved at potentials of about +0.15 V vs. Ag/AgCl in oxygensparged acetate buffered solutions (pH = 4.5) [137]. Further improvements in the performance of redox polymer hydrogel-based biocathodes for the ORR were achieved by tailoring, through synthesis, the structure of the osmium ligands to yield redox polymers matching closely the redox potential of the T1 site of the MCOs. For example, substitution of the chloride ligand of a [Os(4,4'-dimethyl-2,2'-bipyridine)(2,2':6',2"-terpyridine)Cl]⁺ complex with imidazole units of PVI yields a redox polymer that may be co-immobilized with laccase from Coriolus hirsutus on carbon cloth fiber. The resulting biocathode can operate at mA cm⁻² current densities at $+0.42 \,\mathrm{V}$ vs. Ag/AgCl for the ORR in buffer of pH = 5, 37°C, when rotated at 4000 rpm [138]. Co-immobilization of the redox polymer described above with a laccase from *Pleurotus ostreatus*, which has been reported to retain a high level of activity at pH = 7, yielded biocathodes capable of ORR in 0.1 M NaCl solution (pH = 7) at 37 °C [139]. Insertion of a more flexible tether between the polymer backbone and the redox complex, in an approach adopted for glucoseoxidizing anodes, is reported to yield improved performance, in terms of lower overpotentials and higher current densities, for mediated ORR in redox polymer hydrogels. This approach was implemented by coupling of a [Os(4,4'-dimethyl-2,2'-bipyridine)₂(4-aminomethyl-4'-methyl-2,2'-bipyridine)]⁺ complex to a quaternized polyvinylpyridine polymer. The resulting redox polymer showed a twofold improvement in current density, and a decrease in overpotential, for the ORR at pH = 5, when co-immobilized with a fungal laccase at carbon fiber electrodes, compared to films prepared using a redox polymer without the tether [95]. Other studies have focused on matching the redox polymer potential to that reported for the T1 site of BODs, to overcome the reported chloride and pH sensitivity of the fungal laccases. A redox polymer prepared by substitution of one of the chloride ligands of a Os(4,4'-dichloro-2,2'-bipyridine)2Cl2 complex with imidazole units of a copolymer of PVI and polyacrylamide can provide ORR current densities of 0.7 mA cm⁻² at +0.3 V vs. Ag/AgCl in unstirred phosphate buffered saline at 37 °C when co-immobilized with M. verrucaria BOD on carbon cloth electrodes [140]. Use of a Trachyderma tsunodae BOD, claimed to have a T1 redox potential of +0.44 V vs. Ag/AgCl, yields current densities of 6.25 mA cm⁻² for the ORR at +0.3 V vs. Ag/AgCl with electrodes rotated at 4000 rpm in oxygenated phosphate buffered saline at 37°C [141]. Further purification of the enzyme yielded films providing current densities of 9.5 mA cm⁻² under these conditions [142]. More recent reports have focused on ORR using mediation to laccases that are active in neutral electrolytes. For example, SLAC produced higher current densities for the ORR at pH = 7 compared to that obtained with a Trametes versicolor laccase, when coimmobilized with the redox polymer in Figure 5.5a on carbon electrodes [120]. The use of a redox polymer with lower redox potential than those used with BOD or high-potential fungal laccases is required, as the reported redox potential of SLAC is about +0.3 V vs. Ag/AgCl. Nonetheless, current densities of mAcm⁻² are achieved for the ORR at 0.1 V vs. Ag/AgCl in electrolyte at pH = 7 for films of SLAC and the redox polymer adsorbed to high-surface-area carbon composite electrodes with mass transport aided by rotation of the electrode. The recently reported crystal structure for this enzyme, indicating a trimeric form containing 12 copper atoms [143], points to its unique characteristics over other MCOs, and may provide a route for tailoring of its structure to yield variants of increased stability and redox potential at neutral pH. An alternative laccase active in neutral electrolyte, isolated from Melanocarpus albomyces, can catalyze the ORR under pseudo-physiological conditions when co-immobilized with the redox polymer in Figure 5.5a on smooth glassy carbon electrodes, displaying current density of 0.38 mA cm⁻², but again only at lower redox potentials than that observed for the more acidophilic fungal laccases [110].

A series of recent reports have highlighted the improvement in current density that can be achieved by use of electrodeposition for formation of redox hydrogels, moving from relatively smooth surfaces to structured roughened surfaces [144],

and by systematic variation in the structure of the redox complex and polymer backbone using osmium-based redox polymers as mediators [22, 144–148]. For example, electrodeposition, via electrochemically induced ligand exchange of labile chloride ligands of an osmium complex for imidazole units in a polymer, to provide crosslinked Trametes versicolor-containing redox hydrogels on carbon surfaces yielded higher specific activity for the ORR compared to chemically crosslinked hydrogels, postulated to be a result of thinner, more homogeneous film formation [144]. Using porous carbon fiber paper electrodes, with mass transport controlled using electrode rotation, ORR current densities of greater than 10 mA cm⁻² for both electrodeposited and chemically crosslinked *Trametes versicolor*containing redox hydrogels, at 0.2 V vs. Ag/AgCl, could be obtained. Alternatively, incorporation of both functional groups for coordination to redox complexes and functional groups to provide for electrodeposition via protonation or de-protonation induced by local pH alteration at electrode surfaces by electrolysis of water provides a facile means for control of redox hydrogel formation [146, 147]. Initial reports using this procedure focused on an Os(2,2'-bipyridine)2Cl2 complex, ligand-exchanged to an imdazolyl group on an electrodeposition polymer that also incorporated carboxylic acid groups, permitting co-deposition with BOD, by protonation in acidic electrolyte. ORR activity was observed at 0V vs. Ag/AgCl, with local ORR over the biofilm visualized using scanning electrochemical microscopy [146]. Further refinement of the redox polymer illustrated in Figure 5.11 provided for a biofilm containing Trametes hirsuta laccase on glassy carbon electrodes that vielded ORR current densities of 0.3 mA cm⁻² at 0.4 V vs. Ag/AgCl in oxygensaturated buffer (pH = 4.0) [147].

A systematic approach using *Trametes versicolor* laccase co-immobilized with osmium-based redox polymers has highlighted the importance of redox complex loading [144, 145], and of the redox potential of the redox polymer [22], on ORR current density. Increasing ORR current density is observed with increased difference in redox potential between the T1 site of laccase and redox polymer, but maximum power from an EFC is predicted to result using a redox polymer of redox potential 0.17 V more negative than the T1 site of laccase. It would be inter-

Figure 5.11 Scheme for the synthesis of a pyridinylimidazolyl ligand, its copolymerization with acrylic acid (AA) and butyl acrylate (BA), and subsequent ligand substitution reaction with an osmium complex to yield a redox polymer. From [147] with permission from Elsevier.

esting to extend this study to other MCOs, and to electrodeposited redox hydrogels. Indeed, in a preliminary study we reported a comparison of ORR currents observed for three redox polymers co-immobilized with BOD and Trametes hirsuta and Melanocarpus albomyces laccases [148], with highest ORR current density observed for redox hydrogels of BOD and the redox polymer in Figure 5.5a.

5.4 Assembled Biofuel Cells

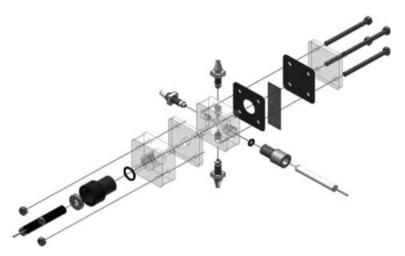
We review in this section progress focused on development of devices capable of power generation through oxidation of glucose and reduction of oxygen. Studies aimed at device implementation as an implanted system to harvest energy from the body's own fuel under physiological conditions should focus on oxygen and glucose concentrations of about 10⁻⁴ M and about 6 mM, respectively, in solutions containing about 150 mM chloride, buffered to a pH of 7.4, thermostatically controlled at 37°C. The velocity of blood in blood vessels is of the order of 1 to 10 cm s⁻¹. Unfortunately, to date, few studies have been undertaken in a configuration which permits exploration of EFC response under these conditions, usually because of limitations in condition choice due to biocatalyst sensitivity to one or other of the variables. Other body fluids or configurations may be considered, for instance EFCs using saliva, tears, or sweat or using high-energy-density fuels such as methanol or carbohydrates in devices for portable power.

Prototype EFC devices have been reported for both implanted power and portable power generation that point to technological application of devices, and provide a foundation to target improvements in device configuration and performance. These prototypes, to date, are based on demonstration of the concept using EFCs assembled from two chambers, separated by a membrane, containing diffusible biocatalysts and mediators adsorbed onto high-surface-area carbon [8, 40]. For example, mediated biocatalysis of glucose oxidation and oxygen reduction using vitamin K₃, diaphorase, NADH, and GDH in the analyte and ferricyanide and BOD in the catholyte separated by cellophane membranes can provide power at pH = 7.0, in a demonstration unit for powering portable devices [40]. An airbreathing configuration adopted to enhance oxygen supply to the cathode resulted in a single EFC, operated as a "biobattery" with glucose introduced as a bolus at 0.4 M level, delivering maximum power density of 1.45 mW cm⁻² at 0.3 V. A parallel stack structure achieved a power output of 50 mW from a 40 cm³ volume that was expanded to provide over 100 mW from 80 cm³ volume, when two such systems were connected in series, providing sufficient power to operate a toy car or portable music player over at least two hours. Another important demonstration of the potential application of EFCs focused on evaluating the performance of EFCs implanted in the retroperitoneal space of a rat [8]. The EFC was based on mediated biocatalysis of glucose oxidation and oxygen reduction using ubiquinone, catalase, and GOx confined behind a dialysis membrane at a high-surface-area carbon anode and quinone and a polyphenol oxidase (a tyrosinase) at a similar cathode,

both electrodes then wrapped in an expanded polytetrafluoroethylene membrane. Maximum specific power density of $24.4 \mu W\,\text{cm}^{-3}$ was obtained at $0.13\,\text{V}$, with stable production of $2\,\mu\text{W}$ of power observed over several hours, for the implanted system. Although little detail is provided on the performance of anode or cathode individually the platform proposed is amenable to simple interchanging of biocatalysts and mediator to help evaluate relative performances of a range of implanted EFCs.

An innovative alternative approach to the use of membranes in constructing prototype EFC assemblies for powering portable devices focuses on the use of either concentric porous tubes of graphite [149, 150] or of delivery of fuel and oxidant via laminar flow within a microchannel, to help prevent interference of fuel or oxidant in the cathodic or anodic reaction [41, 151–155]. The concentric glucose–oxygen EFC utilizes the outer surface of a porous carbon tube as cathode, and the inner surface of a second, larger diameter, porous tube encasing the first tube as anode, with delivery of oxygenated buffer through the inner tube bore while maintaining the rest of the system in deoxygenated 10 mM glucose buffer (pH = 7.4) [149]. Stability of the system was enhanced by grafting BOD and GOx at cathode and anode precoated with an electropolymerized layer of polyamino-propylpyrrole, and by trapping mediators, ABTS and 8-hydroxyquinoline-5-sulfonic acid, at cathode and anode using a diepoxide reagent, to yield a system providing maximum power of $20\,\mu\text{W}\,\text{cm}^{-2}$ at $0.20\,\text{V}$ in $10\,\text{mM}$ glucose solutions over a 45-day period when used intermittently [150].

Microfluidic EFCs are designed to operate using laminar flow delivery of almost parallel streams of fuel and oxidant within the microchannel, preventing crossover and mixing [151-155]. For example, delivery in a polydimethylsiloxane-fabricated Y-shaped microchannel of separate streams of glucose analyte (pH = 7) and oxygen in catholyte (pH = 3) for oxidation by $GOx-Fe(CN)_6$ [3], and reduction by laccase-ABTS at gold anodes and cathodes, respectively, produced a maximum power density of 110 μW cm⁻² at 0.3 V at room temperature [154]. Further optimization of electrode and channel size, and spacings, led to smaller ohmic drop and depletion zone within the system providing power density up to 0.55 mW cm⁻² at 0.3 V [155]. Of course, operation of microfluidic pumps and the requirement to add mediator and enzyme to the channel streams mitigate against adoption of microfluidic EFCs for energy harvesting to power portable electronic devices in this configuration. Progress depends upon immobilization and stabilization of enzymes, and mediators if required, onto channel electrodes, as demonstrated within microchannels fueled with 5 mM glucose and 1 mM NAD+, oxidized at high-surface-area carbon anodes modified by co-adsorption of vitamin K₃-modified polylysine, diaphorase, and NAD-dependent GDH. This anode was coupled to oxygen reduction at a platinum cathode, to provide 32 µW cm⁻² at 0.29 V [41]. The same anode has been coupled to a cathode, constructed by adsorption of BOD onto high-surface-area carbon for the ORR via DET, to examine the effect of electrode configuration on cell performance. Results indicate that placement of the cathode upstream from the anode can protect the anode from oxygen and lead to an increase in maximum cell current [153]. Prior to the appearance of these demon-



modular stack half-cell showing the central reaction chamber with reference electrode inlet (bottom right) and solution filling ports, the working electrode plate with inlet for the

Figure 5.12 Three-dimensional drawing of the glassy carbon electrode (bottom left), and the counter electrode plate separated by gasket seal, all held leak-free together using bolts (top right). From [25] with permission from Wiley.

stration devices, a range of reports on the assembly, and the performance, of BFCs have appeared. These reports, with some notable exceptions, mostly focus on demonstration of the capability to generate power under a defined condition (pH, temperature, biocatalyst, surface, etc.), usually with little consideration of the stability of such power or of the consequences for powering devices. Comparison of power output from such assemblies under the range of conditions utilized thus becomes, as is evident from previous comparison of bioanode and biocathode performance, problematic. A welcome contribution to methodologies for comparison of bioelectrocatalytic fuel cell electrodes is the use of a standardized fuel cell set-up [25]. The set-up is actually a half-cell, anode, with a platinum mesh as counter electrode, assembled, as presented in Figure 5.12, to permit flow presentation of fuel to a glassy carbon anode. The reproducibility of the glassy carbon anode surface area was verified using voltammetry of a solution-phase ferricyanide redox probe. Reproducibility of voltammetry for deposited polymethylene green and for the voltammetric response of the deposited film to NADH oxidation is presented, with a view for use as a bioanode for NADH-dependent dehydrogenase reactions. Results from the use of such a standardized methodology should prove invaluable for comparison of surfaces, mediators, and enzymes in bioanodes and BFCs, and are eagerly anticipated!

A series of reports, focused on systematic approaches to optimizing choice of biocatalyst and either redox polymer-based mediator and/or carbon-based support, to provide for a glucose-oxygen EFC are noteworthy for the wide-ranging contribution to technological advances. This series of studies, based on initial reports of redox polymer-mediated biosensors for glucose [58–61] and oxygen [96, 137], focused on systematic selection and optimization of biocatalyst, mediator, and surface for anode and cathode, as detailed in the previous sections, and their combination to provide prototype EFCs. For example, initial demonstration of power density of $137 \mu \text{W cm}^{-2}$ at 0.37 V in 37 °C aerated buffer at pH = 5 and chloride-free medium containing 15 mM glucose was provided by a cell constructed using carbon fiber electrodes, modified by crosslinking GOx to [Os(4,4'dimethyl-2,2'-bipyridine)₂(PVI)Cl]⁺ and laccase (Coriolus hirsitus) to [Os(4,4'dimethyl-2,2'-bipyridine)₂(2,2',6',2"-terpyridine)₂²⁺ tethered to a PVI polymer [35]. Systematic improvements in anodic redox polymer, by inclusion of a flexible tether to improve charge transport diffusion, and a more appropriate redox potential, provided power density of 268 µW cm⁻² at 0.78 V [156]. When the same refinement procedure was followed for the cathode [87], a power density of 350 µW cm⁻² at $0.88\,\mathrm{V}$ was obtained, in buffer pH = 5, a cell voltage only $0.3\,\mathrm{V}$ lower than that of the reversible cell voltage for glucose-oxygen.

Recent reports have highlighted improvement in power densities for assembled EFCs using osmium redox polymer mediators by purification of GOx [157], or replacement of GOx isolated from Aspergillus niger with GOx isolated from Penicillium pinophilum, used in the anode [39]. For example, a biofuel using purified A. niger GOx at the anode provided twice the power density than obtained using the "as-received" GOx, when operated using a BOD cathode in 15 mM glucose in phosphate buffered saline (pH = 7.4) [157]. Use of GOx isolated from P. pinophilum versus that from A. niger at a redox polymer anode coupled to a laccase cathode yielded a threefold increase in power density, to 0.28 mW cm⁻² at 0.88 V, at glucose concentrations of 5 mM, approaching those in serum [39]. Unfortunately, the instability of the P. pinophilum GOx in neutral electrolyte precluded operation under physiological pH conditions. In addition, most fungal laccases lose activity in neutral electrolyte, and undergo inhibition by chloride. In an EFC we have developed, a maximum power density of 16 µW cm⁻² at a cell voltage of 0.25 V can nonetheless be obtained in pseudo-physiological conditions of 10 mM glucose in phosphate buffer (pH = 7.4) at 37 °C, for graphite electrodes with an A. niger GOx crosslinked to [Os(4,4'-diamino-2,2'-bipyridine)2(PVI)Cl]+ as anode and Trametes versicolor laccase crosslinked to [Os(1,10-phenanthroline)₂(PVI)₂]²⁺ as cathode. A maximum power density of $40\,\mu\mathrm{W\,cm^{-2}}$ was observed for this cell at pH = 5.5 where laccase is more active [158]. As discussed previously, replacement of acidophilic fungal laccases with biocatalysts that are active towards the ORR in neutral electrolyte can provide for improved ORR current densities, but not necessarily power output. An initial GOx-BOD fuel cell operating in pseudophysiological conditions, with appropriate selection of the redox polymer structure to reflect the decreased T1 copper redox potential in BOD compared to fungal laccase, yielded a power density of 50 μW cm⁻² at 0.5 V. In an illustration of the trade-off between cell voltage and maximum power, use of a redox polymer for the anode with a more positive redox potential increased maximum power density to 244 µW cm⁻² but at the expense of a lower cell voltage of 0.36 V [36]. Again, inclusion of a flexible tether to improve charge transport diffusion for the anodic redox polymer improved anodic current density, producing a power density of 430 µW cm⁻² at a cell voltage of 0.52 V [159]. One week of continuous operation of this cell at 0.52 V, resulted in an approximately 6% loss in power density per day. The operation of this miniaturized carbon fiber-based EFC in a grape, chosen for its high glucose content (>30 mM), was demonstrated, with power density of 47 μW cm⁻² observed when the cell is placed in the oxygen-deficient grape center, whereas 240 uW cm⁻² is obtained when the cell is placed near the grape skin. with a cell voltage of 0.52 V for both cases [9]. Further improvement in output for this EFC was achieved by increasing the redox site density of the anode redox polymer, resulting in a glucose flux-limited current density increase of 20% and an overpotential decrease of 50 mV [160]. This optimized EFC operated at +0.60 V with a $480 \,\mu\text{W}\,\text{cm}^{-2}$ power density in phosphate buffer of pH = 7.2 containing 0.1 M NaCl, 15 mM glucose at 37.5 °C, losing about 8% of its power each day of operation.

Increased power output of these redox polymer-mediated EFCs can be obtained by use of three-dimensional conducting materials as supports for the biocatalytic films, provided mass transport does not become rate limiting. This has been demonstrated by Gao et al. [161] through a comparison of carbon fiber-based electrodes with carbon microwire electrodes, formed via particle coagulation spinning of CNTs. Maximum power density of 0.74 mW cm⁻² at 0.57 V was obtained for the miniaturized membraneless EFC in an unstirred phosphate buffer saline solution (pH = 7.2) under air, containing 15 mM glucose at 37 °C, more than ten times that obtained for the carbon fiber-based system.

5.5 Conclusions and Future Outlook

Bioelectrocatalytic studies have proven that enzymes as biocatalysts in EFC electrodes can perform better than conventional catalysts, providing advantages in terms of rates of reaction and specificity. This advantage can be transformed into provision of power with low overpotentials, and with ease of miniaturization, because removal of membranes and casings can be undertaken due to enzyme specificity, providing for niche opportunities for EFCs in the production of power using implanted or portable, miniaturized, devices. Challenges, nonetheless, remain in order to advance the knowledge gained over the past decade of research into technological products. Issues that require further attention relate to stabilization of power output of assembled cells and to increasing power output. Power output may be improved through use of optimized biocatalyst (and mediator if appropriate) on three-dimensional surfaces and deeper oxidation of fuels using multiple enzymes and enzyme cascades. An additional consideration is the cell voltage required to provide power to microelectronic devices. Cell voltages of greater than 0.5 V can be delivered under certain conditions, but most studies to date only report voltages of 0.2–0.3 V, requiring therefore stacking of cells to power electronic devices.

A systematic comparison of EFC benchmark figures of merit on stability, as was the case for reporting of power density benchmarks, is hampered by the lack of a defined measure of stability in each study. Many studies simply report on storage stability using either intermittent or single-shot measurement of current or power, with ill-defined storage conditions provided. Future studies should therefore focus on reporting operational stability of assembled cells under a defined load, in defined conditions, as is the case in traditional fuel cell research. Stability of power output is affected by enzyme deactivation and inhibition and by desorption/ leaching of enzyme, and mediator if present, from the electrode surfaces. Recent reports highlight, for example, the effect of other endogenous serum components on EFC operation [17, 162-164]. For example, a glucose-oxygen EFC, based on DET for CDH (Corynascus thermophilus) oxidation of glucose and BOD reduction of oxygen, both enzymes simply adsorbed onto spectroscopic graphite rod electrodes, results in an operational half-life of more than six hours in buffer but only less than two hours in serum [17]. Others have reported that oxidation of uric acid in the presence of oxygen destabilizes BOD, leading to a decrease in performance of a BOD biocathode to only a few hours in urate-containing serum, compared to over a week in buffer [164]. Blocking access of urate to the enzyme can be achieved by, for example, coating the biocathode with a cubic-phase lyotropic liquid crystal that has channels modified using an anionic lipid, to increase mechanical strength and reduce urate permeation [165]. Desorption of enzyme and/or mediator from the electrode surface can also lead to a decrease in EFC performance, and attempts to prevent this range from entrapment behind or within membranes and crosslinking of films, to modification of electrode surface chemistry to provide for anchoring sites for chemical tethering of layers to the surface. For example, we have recently reported on improved stability of layers of redox complexes [166, 167], GOx [168], and MCOs in redox hydrogels [148] by chemical crosslinking of the layer to surface functional groups on carbon electrodes. The functional groups are introduced using in situ diazotization of an aryl amine, followed by electroreduction of the aryldiazonium to carbon-carbon couple to the electrode. Such an approach, combined with crosslinking and overcoating with additional membranes, could provide for EFCs with increased stability in serum.

EFCs operating on fuels other than glucose show promise as devices for powering portable electronics, and recent reports highlight advances in prototype design of cells using either fructose [45–47] or ethanol [169] as fuel in the anolyte, and either FDH or alcohol dehydrogenase as biocatalyst entrapped in three-dimensional conducting electrodes to provide for DET to the anode, in a membraneless EFC configuration. Maximum power densities of 0.85 mW cm⁻² [45] and 0.126 mW cm⁻² [47] at a cell voltage of about 0.4 V are observed in the fructose studies using either carbon black or MWCNTs as the three-dimensional electrode, and a fungal laccase or BOD as biocathode element, respectively, again making it difficult to compare performance.

The use of enzyme cascades, in an effort to extract more than two electrons from the fuel, as is the case for GOx oxidation of glucose, is a promising direction for improving power output for portable devices [80–86, 170]. Since the initial report, demonstrating complete oxidation of methanol to carbon dioxide based on NAD+/NADH regeneration in the analyte by diaphorase and benzylviologen coupled to three dehydrogenases [81], advances have focused on efforts to mimic metabolic pathways by combining multiple enzymes in the analyte. For example, a mimic of the citric acid cycle, based on immobilization of dehydrogenases at, and electrocatalytic oxidation of NADH on, poly(methylene green)modified carbon electrodes, provides for an ethanol-air (platinum-based) fuel cell that provides 8.7-fold more power compared to using a single alcohol dehydrogenase system in a Nafion-separated system [170]. Mediatorless, DET, glycerol oxidation using PQQ-dependent dehydrogenases and an oxalate oxidase coimmobilized on CNT-coated carbon paper can also be used to provide a glycerolair fuel cell with a maximum power density of 1.32 mW cm⁻² in 100 mM glycerol solutions [85].

In conclusion, despite over a decade of increased attention, in the most recent phase of interest in EFC research, performance of operating prototype devices remains limited in terms of power output and stability. At a fundamental level, insight into enzyme electron transfer and power production can be gained by systematic studies on integration of surfaces, catalysts, substrates, co-substrates, and artificial substrates (mediators). Unfortunately, wide-scale comparison between devices (anodes, cathodes, assemblies), in terms of operating conditions, performance benchmarks, and stability benchmarks, is hampered by a lack of conformity to standardized procedures for testing, and reporting of data. In addition, there is scope for greater understanding of the role that each component plays in an integrated system to provide for an EFC, by comparison of performance to models. Relatively few examples exist to date of detailed studies comparing response to models, to enable identification of limiting factors in reaction kinetics, mass transport, and system design. Niche application areas for assembled EFCs, such as self-powered miniaturized disposable sensors [171], replaceable power for implantable sensors [7, 21, 172], or boosting power to electronic devices in remote locations using energy-dense fuels [45-47, 170], will emerge. Ultimately, the understanding gained from biocatalyst-based research may also lead to development of better biomimetics, mimicking not only the catalytic active site of a biocatalyst, but also the binding site and the structural flexibility provided by the protein assembly surrounding the active site, to provide for improved "smallmolecule" catalysts in more traditional fuel cell systems, providing high power densities at low cost.

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6

Raman Spectroscopy of Biomolecules at Electrode Surfaces

Philip Bartlett and Sumeet Mahajan

6.1 Introduction

Surface-enhanced Raman spectroscopy (SERS) is a remarkable phenomenon. The effect was first reported by Fleischmann et al. over 35 years ago [1] in an experiment in which they recorded the Raman spectrum of pyridine at a roughened silver electrode surface. From the start the effect provoked controversy because of its large magnitude and unexpected nature. Raman scattering is inherently inefficient with only a small fraction of the photons, typically around 1 in 10¹⁰, being Raman scattered, the rest, the vast majority, being elastically, or Rayleigh, scattered. Thus to observe clear spectra for a monolayer of pyridine molecules adsorbed at the silver solution interface was unprecedented. In their original experiments Fleischman et al. used an electrochemically roughened silver surface but even so the increased surface area alone could not account for the many orders of magnitude increase in the Raman signal. At the time Fleischman et al. were aware of the unusually high intensity of the Raman signals and careful to exclude the possibility of multilayer formation [2]. This first publication, with its surprising results, sparked the interest of other groups and it was followed a few years later by publications by Jeanmarie and Van Duyne [3] and by Albrecht et al. [4] confirming the effect and demonstrating that the anomalous intensity of the spectra (a factor or 10⁵ to 10⁶) was the result of some kind of hidden resonance or a new type of Raman enhancement [5]. The term "surface enhanced Raman spectroscopy" was coined by Van Duyne and coworkers soon after to describe the effect [5]. This started a great interest in the effect and a search for detailed understanding of it and its application, which continues to this day.

One of the problems of SERS is that it is relatively easy to achieve but difficult to control. This has led, and continues to lead, to publications that claim to have new surfaces that show SERS but which in reality offer no advantage over existing roughened surfaces, or publications that claim insight into the mechanism of the phenomenon but which are in fact readily explained within the basis of current understanding of SERS [6]. As a consequence a vast and somewhat confusing literature has built up around the subject over the past 30 years and this is a

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problem for those who wish to apply the technique to address specific chemical or biochemical questions or to use it for analytical purposes.

One of the attractions of SERS is that it is ideally suited to *in situ* electrochemical measurements because of the very high surface selectivity of the technique and the fact that water is a very weak Raman scatterer. As a result SERS is able to provide vibrational spectra for molecules at electrode surfaces under realistic operating conditions. In this chapter we focus on the applications of SERS in bioelectrochemistry, focusing on studies carried out at electrode surfaces under potential control. There are a number of excellent reviews in the literature of various aspects of SERS covering both the general field [6–8], as well as more specific topics including the biochemical applications of SERS [9, 10], analytical prospects [11–13], surfaces for SERS [14–17], and electrochemical SERS [18–20]; and the reader is directed to these for specific details. In this chapter we begin with a brief overview of SERS and SERS surfaces before reviewing the particular applications in bioelectrochemistry.

6.2 Raman Spectroscopy

Raman scattering is an inelastic process in which the change in wavelength of the scattered photon is associated with the excitation (Stokes) or relaxation (anti-Stokes) of vibrational modes of the scattering molecule (Figure 6.1). Raman spectroscopy, like infrared spectroscopy, thus gives information about molecular structure because the different functional groups within molecules have different characteristic vibrational frequencies—a kind of "molecular fingerprint" for the

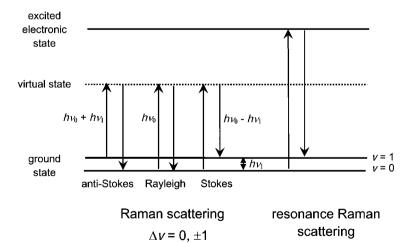


Figure 6.1 Spectroscopic transitions for Rayleigh scattering, Stokes and anti-Stokes Raman scattering, and resonance Raman scattering.

molecule. A Raman spectrum is thus a plot of scattered intensity as a function of the shift in energy, hv_1 , with respect to the excitation energy, hv_0 . As in infrared spectroscopy, only certain vibrational modes are active as determined by the selection rules. The virtual state, shown in Figure 6.1, can be thought of as a short-lived distortion of the electron density of the molecule brought about by the oscillating electric field of the exciting light.

The intensity for a transition from an initial state i to a final state f is given by [21, 22]

$$I_{f,i} = \frac{8\pi e^4}{9c^4} \omega_0 \omega_{sc}^3 \sum_{\rho,\sigma} \left| (\alpha_{\rho,\sigma})_{f,i} \right|^2 I(\omega_0)$$
(6.1)

where $I_{f,i}$ is the number of scattered photons, $I(\omega_0)$ is the incident number of photons at the incident angular frequency ω_0 , ω_{sc} is the angular frequency of the scattered light, e is the charge on the electron, c is the velocity of light, ρ is the polarization of the laser electric field at the molecule, σ is the direction of the Raman-scattered electric field at the molecule, and α is the transition polarizability tensor. From Eq. (6.1) we can see that the Raman scattering is first order in the intensity of the incident laser light. The primary selection rule for Raman spectroscopy states that only vibrations that change the polarizability of the molecule are Raman active; this is different from the primary selection rule for infrared spectroscopy which requires a change in the dipole for the vibrational mode to be active. Thus Raman spectra and infrared spectra are similar but not identical. In addition the vibrational selection rule requires $\Delta v = \pm 1$ so the Stokes and anti-Stokes shifts give information about the ground-state vibrational frequencies for the molecule (see Figure 6.1). Extensive compilations of infrared and Raman frequencies can be found in the literature. The intensity of Raman bands depends on the polarizability tensor, α (see Eq. (6.1)). Thus the intensities of Raman bands can be very different from the corresponding intensities in the infrared spectrum. Strong Raman bands are expected for molecules with extended π systems, for molecules with large electron-rich atoms (e.g., S–S bonds, C–Cl bonds, C–I bonds), and for molecules with multiple bonds (e.g., C=C, C≡N) as these are all easily polarized. By the same argument the Raman spectra of weakly polarizable bonds (C-H, C-O, C-C, etc.) will generally be weak. Thus water is a very weak Raman scatterer-a significant benefit for studies in aqueous solution.

Even so Raman scattering is a very weak process with typically only 1 in 10¹⁰ scattered photons being Raman scattered and the vast majority being elastically (or Rayleigh) scattered [23]. The cross sections for Raman scattering are very small, typically 6 to 8 orders of magnitude smaller than fluorescence cross sections [23] and, in the absence of resonance effects, decrease with the fourth power of the excitation wavelength. This wavelength dependence reflects the fact that light of shorter wavelength is scattered more efficiently. Another advantage of using shorter wavelengths for Raman measurements is that it is easier to detect light at shorter wavelength. However, for biological samples these advantages are offset by the fact that fluorescence becomes more likely and the risk of sample

degradation at high laser intensity is increased. The commonly used wavelengths, reflecting the commonly available lasers, are 1064 and 785 nm in the near-infrared, and 632.8, 532, 514.5, and 488 nm in the visible.

When the laser photon energy is close to (pre-resonance) or matches (resonance) the energy difference between the ground state of a molecule and an excited vibronic state the intensity of Raman scattering is significantly increased. This is referred to as resonant Raman scattering. In this case the vibrations that show the resonant enhancement are those involving atoms or bands associated with the chromophore while Raman bands for parts of the molecule not associated with the chromophore are not enhanced. In biological systems this can be used to advantage to study specific parts of large redox proteins, for example the heme in cytochrome c [24]. For a comprehensive discussion of Raman spectroscopy, including the practical aspects, the reader is directed to the book by McCreery [23].

6.3 SERS and Surface-Enhanced Resonant Raman Spectroscopy

The key to the SERS effect, and to its utility as a tool to study processes *in situ* at electrode surfaces, is the large magnitude and short range of the surface enhancement. In the early days of the technique there was much discussion and disagreement over the origin and nature of this surface enhancement. It is now generally agreed that the enhancement is made up of two contributions: a chemical enhancement and an electromagnetic enhancement. Of the two the electromagnetic enhancement is the larger, contributing a factor of greater than 10⁴ as compared to a contribution of typically no more than 10² for the chemical enhancement.

The chemical enhancement model [22] postulates a chemical bonding interaction between a molecule and a metal surface such that new electronic levels are produced by overlap of the molecular orbitals of the adsorbed molecule and the orbitals of the metal. Then, provided these new electronic levels are close to resonance with the exciting laser light, there will be an enhancement in the strength of the Raman scattering for the adsorbed molecule. Clearly this mechanism is very short range as it requires significant orbital overlap between the molecule and the metal surface (i.e., the molecules must be chemisorbed) and it will be highly specific to the precise structure of the molecule and choice of metal since it requires the appropriate matching of orbital energies and symmetries [25].

In contrast electromagnetic enhancement [6] relies on the intensification of the local electromagnetic field at the metal surface and the interaction of this localized electromagnetic field with the molecules close to, but not necessarily directly in contact with or chemisorbed at, the metal surface. The electromagnetic enhancement is of longer range than the chemical enhancement decaying over a distance of the order of 50 to 100 nm.

For electromagnetic enhancement the exciting laser light must first couple to the metal surface to produce a surface plasmon. A surface plasmon is an interaction between free surface charges and the electromagnetic field [26–28] (Figure

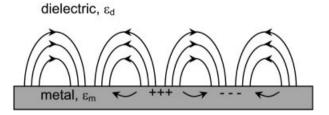


Figure 6.2 Combined electromagnetic wave and surface charge character of a surface plasmon.

6.2). Surface plasmons (also called surface plasmon polaritons) are characterized by a frequency-dependent surface wave vector

$$\mathbf{k}_{\rm sp} = \frac{\omega}{c} \left(\frac{\varepsilon_{\rm d} \varepsilon_{\rm m}}{\varepsilon_{\rm d} + \varepsilon_{\rm m}} \right)^{1/2} \tag{6.2}$$

where ω is the frequency of the electromagnetic field, c the velocity of light, and $\epsilon_{\!\scriptscriptstyle d}$ and $\epsilon_{\!\scriptscriptstyle m}$ the frequency-dependent permittivities of dielectric and metal, respectively. For surface plasmons to be possible ε_{d} and ε_{m} must have opposite signs. A consequence of the interaction between the electromagnetic field and the free surface charges is that the field perpendicular to the surface decays exponentially. Once the light is coupled to a flat surface as a surface plasmon it will propagate along the surface but will be slowly attenuated due to resistive losses in the metal. The greater the conductivity of the metal at the plasmon oscillation frequency the further the light will propagate. This depends on the complex permittivity of the metal. On a periodically structured or nanostructured surface, scattering of the surface plasmons can lead to the formation of standing waves or localized surface plasmons.

These surface plasmons can interact with the molecules close to, or at, the metal surface leading to Raman scattering which generates new surface plasmons shifted in energy by amounts corresponding to vibrational quanta for the molecules. The new surface plasmons then radiate light from the surface and can be detected by a spectrometer (Figure 6.3). The high magnitude of the electromagnetic contribution to the surface enhancement arises because there is a contribution from both the light coupling in to the surface and a contribution from the light coupling out from the surface [19, 29]:

$$I_{\text{SERS}} \propto \left| \frac{E_{\text{local}}(\omega_0)}{E_{\text{inc}}(\omega_0)} \right|^2 \left| \frac{E_{\text{local}}(\omega_{\text{sc}})}{E_{\text{inc}}(\omega_{\text{sc}})} \right|^2 \sum_{\rho, \sigma} \left| (\alpha_{\rho, \sigma})_{f, i} \right|^2 I(\omega_0)$$
(6.3)

where ω_0 is the incident and $\omega_{\rm sc}$ the emitted frequency, and $E_{\rm inc}$ is the incident and E_{local} the local field strength. The first two terms describe the electromagnetic enhancement; the summation term contains the chemical enhancement contribution. As for the Raman effect itself, the SERS intensity is linear in the laser light intensity, $I(\omega_0)$.

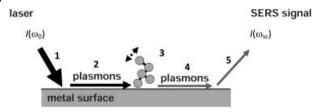


Figure 6.3 The SERS process. 1. The impinging laser light, ω_0 , excites plasmons at the metal surface. 2. These plasmons convey optical energy into the molecule. 3. The molecule undergoes Raman scattering taking

up a vibrational quantum of energy ($\omega_0-\omega_{\rm sc}$): Stokes scattering. 4. A plasmon at longer wavelength is produced. 5. This plasmon decays away into an emitted photon, $\omega_{\rm sc}$, which is detected in the spectrometer.

In both cases, coupling in and coupling out, the enhancement goes as the square of the local electromagnetic field and this is the origin of the very large enhancements that are possible in SERS—a 100-fold enhancement in the in-coupling and out-coupling electric fields at the metal surface will produce an electromagnetic contribution to the enhancement of 10⁸. Often one sees in the literature the statement that the enhancement goes as the fourth power of the local electric field but this is actually a simplification [30–32], since, as shown by Eq. (6.3), the in- and out-coupling fields are at different frequencies. This has been clearly demonstrated experimentally in angle-resolved SERS measurements at gratings and structured surfaces where in-coupling and out-coupling are clearly separated [33–36].

In the literature there is often some confusion over the precise definition of the enhancement factor as measured experimentally and indeed there are several complications, and frequently assumptions, inherent in the estimation of enhancement factors. This is discussed in depth in a very useful article by Le Ru *et al.* [37].

The best-known metals for SERS are the coinage metals, Cu, Ag, and Au, but SERS has been observed on many other metals. The relative size of the enhancements for different metals depends on the wavelength-dependent optical properties of the metal [38]. In the visible region enhancements are strong on Cu, Au, and particularly Ag because of the optical properties of those metals. For other metals the effect is much less. Nevertheless there have been considerable efforts to extend the application of SERS to other metals, such as Pt, Pd, Rh, Ru, Fe, Co, Ni, etc., because of their importance in catalysis and electrochemistry [19, 39].

In addition to the Raman selection rules described above there are surface selection rules that apply for SERS because the process occurs close to metal surfaces [40–42]. The SERS surface selection rule predicts that the vibrational bands that have contributions from the Raman polarizability tensor component α_{zz} , where z is the surface normal, will be most intense with weaker contributions from vibrational bands which have contributions from α_{xz} and α_{yz} . This is essentially because the electric field of the exciting light is enhanced in the direction of the surface normal (Figure 6.2). The surface selection rule for Raman spectroscopy is more complex than that for infrared spectroscopy. Modes with the bond axis parallel to

the surface often contain a substantial polarizability component in the direction of the surface normal and can, therefore, be SERS active. The surface selection rule means that the SERS spectrum is sensitive to the orientation of the molecule at the metal surface. For example, for an aromatic ring standing perpendicular to a metal surface the in-plane vibrational modes are expected to be much more intense in the SERS spectrum than the out-of-plane vibrational modes, whereas if the molecule is adsorbed flat on the surface this will change [43]. Pemberton *et al.* [44] have described a method to use SERS to determine the orientation of molecules with low symmetry using the relative intensities of vibrational modes for functional groups (such as –CH₃) which have multiple vibrational modes of known spatial relation. This approach has been used by Szafranski *et al.* [45, 46] to determine the tilt angle for aromatic thiols adsorbed at an Au electrode surface and to study the change in tilt angle with electrode potential.

The SERS enhancement falls quite sharply with distance from the surface; however, there is no generally agreed description for the distance dependence. This is probably because of a number of factors. First, it is quite difficult experimentally to establish the distance dependence because of the difficulty in unequivocally placing a probe molecule in the same orientation, at well-defined distances from a surface. Various approaches have been used including a polymer spacer layer [47], Langmuir-Blodgett layers [48] and multilayers [49, 50], self-assembled thiol monolayers [51, 52], and single-stranded DNA tethers [53, 54]. When using intervening layers there is the problem that this complicates the interpretation of the data since the intervening layer necessarily alters the plasmon field distribution and energy. Also, if a molecule is simply tethered to a surface by tethers of different length it is difficult to be certain that the molecule does not approach closer to the surface or change orientation [55]. Second, there are clearly differences in the distance dependence of the contributions from the chemical enhancement (which will be very short range) and the electromagnetic enhancement (expected to be longer range). Third, one can expect the distance dependence of the electromagnetic enhancement to depend on the surface geometry and to be different within a cavity or on a nanoparticle surface. There is, however, general consensus that the enhancement falls sharply with distance. Nevertheless, because the enhancement is so strong it is still possible to record spectra for molecules some tens of nanometers from a surface [54].

A further resonant contribution to the enhancement is possible when the molecule has an electronic transition in resonance, or close to resonance, with the exciting laser so that one obtains surface-enhanced resonant Raman scattering or surface-enhanced resonant Raman spectroscopy (SERRS). In these circumstances the resonant enhancement typically contributes an additional factor of between 100 and 1000 to the intensity of the signal and this makes SERRS a particularly attractive approach for analytical applications because of its extremely high sensitivity.

SERS and SERRS are powerful techniques for *in situ* electrochemical studies for several reasons. First, water is a weak Raman scatterer and is transparent in the visible region of the spectrum so it is relatively easy to carry out *in situ*

measurements without interference from the solvent. Second, the SE(R)RS effect is highly surface selective so the spectra are obtained only from those molecules very close (typically 50 nm or less) to the electrode surface. Third, the technique gives molecular information about the species at the electrode surface including information about orientation and changes in molecular structure or bond strength.

In electrochemical SERS there can be a significant dependence, up to 100 cm⁻¹ V⁻¹ or more, of the band frequencies on the electrode potential for adsorbates and molecules close to the electrode surface. This is referred to as the Stark effect or Stark tuning and can have contributions from the effect of the potential on metal-adsorbate bonding as well as from the purely electrostatic effect of the field at the electrode surface on the vibrational force constant [56]. For example, the Stark shifts seen for CO on Pt [56-58], CO on Ni [59], and NO on Pt [58] all have significant contributions from the effect of the electrode potential on the metal-adsorbate bonding as well as from purely electrostatic effects. When the functional group is remote from the metal surface, so that there are no direct bonding interactions, the Stark shift is entirely due to changes in the electric field in the double layer caused by changes in the electrode potential. In this case the magnitude of the Stark shift depends on the magnitude of the electric field and the dipole moment of the functional group so that groups with a large dipole such as CN are the most sensitive. This effect has been used very elegantly by Oklejas et al. to measure the interfacial electric fields in diffuse double layers using SERS [60, 61]. They used mixed monolayers of alkanethiol and a longer alkanethiol with a terminal CN moiety on an Ag electrode (Figure 6.4) to position the CN probe at different positions within the double layer. In this case, for measurements in 10 mM NaClO₄, they observed a Stark tuning rate of about 10 cm⁻¹ V⁻¹.

Excellent reviews of electrochemical SERS, including details of instrumentation and applications, can be found in the work of Tian and the group in Xiamen [18–20]. One of the key impediments to the wider application of electrochemical SERS has been the problem of generating suitable, stable, reproducible electrode surfaces that show strong SERS signals. This is discussed below.

6.4 Comparison of SE(R)RS and Fluorescence for Biological Studies

Fluorescence-based methods are the most commonly used for biological labeling and studies of biological interactions. They require the covalent attachment of a fluorescent label to the biological molecule of interest that can then used in a number of different measurement formats including fluorescence resonance energy transfer. However these fluorescence-based techniques are not always compatible with *in situ* electrochemical measurements because of the quenching of fluorescence by the electrode surface. In contrast SERRS is an attractive alternative approach that is gathering support for biological applications [15, 16, 20, 62–64] and which is compatible with *in situ* electrochemical measurements.

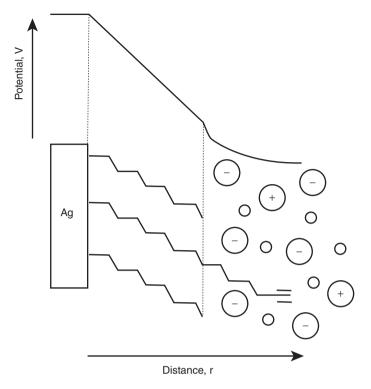


Figure 6.4 Schematic of the Ag–SAM–aqueous interface region showing the nitrile group extending into the double layer. Reprinted with permission from Oklejas, V.,

Sjostrom, C., and Harris, J.M. (2002) *Journal of the American Chemical Society*, **124**, 2408. Copyright 2002 American Chemical Society.

Both fluorescence and SERRS require labeling of the biological component, both are highly sensitive techniques, and both have been shown to be capable of single-molecule detection under favorable conditions [65–68]. Labels used for biological studies must fulfill several requirements [69]: they should not perturb the specificity of the biological recognition system under study; they should be robust and reproducible, stable on storage, in contact with the sample, and during measurement; they should give a linear response over the appropriate concentration range with high sensitivity; and the signal should be readily discernable from the background. These criteria are met by a wide range of Raman labels. Thus commonly available fluorescent labels can be used in SERRS and there are also a large number of Raman dyes [70–72] and nonfluorescent labels available [73]. This range of labels can be further extended by simple molecular modifications or the synthesis of new dyes with distinctive spectra [71, 74].

Compared to fluorescence, SERRS has some significant advantages. Unlike fluorescence-based methods, SERRS measurements are unaffected by quenching by oxygen or other species, are less sensitive to photobleaching, and are inherently more sensitive. For example, Sabatte *et al.* [75] showed that the limit of detection for SERRS in an immunoassay application was around 1000 times lower than for the corresponding fluorescence assay once account is taken of dilution effects. A similar improvement of three orders of magnitude in the limit of detection for SERRS over fluorescence was reported by Faulds *et al.* for a DNA assay [76].

For multiplex measurements, when compared to fluorescence, SERRS also offers significant advantages. In multiplex measurements fluorescence has the disadvantage that the electronic spectra produced are broad (typically 50 to 100 nm full width at half maximum) and therefore overlap so that the technique is limited to the simultaneous measurement of around four dye labels [69, 77]. In contrast, SERRS uses the vibrational Raman spectrum of the label as a spectroscopic molecular fingerprint. As a result the information content of the spectra is much higher and, because the vibrational bands are much narrower (about 1 nm full width at half maximum), spectral overlap is much less of a problem. Thus using SERRS it is possible to readily identify the components of a mixture without extensive separation procedures [78] and it has been estimated in the literature that simultaneous measurement with up to 30 SE(R)RS labels should be possible [79].

SE(R)RS is well suited as a technique for the detection and discrimination of DNA due to its high sensitivity and advantages over fluorescence [63]. The first applications of SE(R)RS to DNA detection by Vo-Dinh's group used roughed silver surfaces to achieve surface enhancement [80–82] but this approach suffers from problems of reproducibility. Using silver colloids Faulds *et al.* have demonstrated the quantitative detection of dye-labeled oligonucleotides by SERRS [83].

There are several examples of multiplexed SERRS measurements as applied to DNA assays [69, 70, 73, 77, 84–89] reported in the literature and a few examples of immunoassays. Thus Woo *et al.* [90] described a triplex immunoassay for the detection of bronchioalveolar stem cells, Cui *et al.* [91] used two dye labels to carry out a duplex sandwich immunoassay, Jun *et al.* [92] reported a triplex immunoassay using SERS, and Lutz *et al.* [93] described a quadruplex platelet activity state plate binding assay.

Finally, the instrumentation required for SE(R)RS measurements is not necessarily large or complex. Indeed, commercial, handheld spectrometers are available. In essence the technique requires a laser for excitation of the surface, a filter (notch or edge) to exclude the Raleigh-scattered radiation, a monochromator, and a detector to collect the Raman-scattered light [19, 20, 23].

6.5 Surfaces for SERS

Early studies of SERS were plagued by irreproducibility. The studies relied on electrochemically roughened coinage metals (Ag, Au, or Cu) with ill-defined and unstable surfaces. As a result the intensity of the spectra varied by orders of magnitude from place to place on the surface, from one surface to another, and over

time. Thus although SERS was in principle, since it gives molecular structurespecific information with exceptionally high surface selectivity, ideally suited for in situ electrochemical studies it was dogged by problems. There was no way to be certain the spectra came from molecules in typical surface environments at the surface, as opposed to reflecting the properties of a small subset of molecules in an atypical environment; the method was restricted to studies of Ag, Au, and Cu electrodes and it was impossible to make quantitative measurements because of the irreproducibility and instability of the substrates.

This led to a lot of attention being focused on ways of roughening electrode surfaces, including potential- or current-controlled oxidation-reduction cycles, chemical etching, electrodeposition and template synthesis, and the development of optimized recipes for roughening different surfaces [11, 18-20]. It also led to many attempts to characterize these roughened surfaces down to the nanoscale, and a particular focus on generating "hotspots"-those regions on the surface where the random nanostructure was, by chance, such that there was a high surface enhancement. In turn this led to the interest in using colloids and in particular aggregates of colloids in an attempt to generate hotspots for SERS. While these colloidal aggregates are useful for solution studies it remains a challenge to control the aggregation process and use them in quantitative measurements [94]. This focus on the role of hotspots continues to the present where it is still common to read in papers that hotspots are essential for SERS even though this is clearly not the case. For in situ electrochemical studies relying on hotspots to give reliable SERS is not a productive approach; rather, the key to developing substrates for electrochemical SERS must be to make controlled surfaces where the nanostructure is stable and robust.

A lot of the published work on electrochemical SERS uses roughened Ag or Au electrodes because these give very large SERS enhancements. However, Ag and Au are not the only electrode materials of interest and consequently significant efforts have been made, particularly by Tian's group, to extend SERS to other transition metal surfaces [18, 19, 39]. The problem here is that, in the visible region of the spectrum, the enhancements that can be achieved on metals such as Pt or Pd are orders of magnitude less than for Ag or Au because of the optical properties of the metals. One way to overcome this is by roughening the metals or by synthesizing shaped nanoparticles of the metals that can be assembled on an electrode surface [19, 20, 95, 96]. An alternative approach, pioneered by Weaver's group [97-99], is the so-called borrowed SERS technique [18]. In borrowed SERS a thin (one to five atoms thick) overlayer of the metal of interest is formed on top of a SERS-active surface prepared from Ag or more usually Au. The technique can also be used with core-shell nanoparticles which have a metal shell over a SERS-active core [100]. The metal overlayer must be very thin otherwise it damps the plasmons in the underlying SERS-active surface and prevents the electromagnetic enhancement of the SERS. This obviously causes a problem because it is difficult to form perfectly pinhole-free overlayers of a metal just a few atoms thick. This has been addressed by Weaver's group using a "pinhole free" deposition technique [101-103]. It is also a potential problem because one cannot always be sure that the

electrochemical properties of a very thin overlayer have not been altered from those of the bulk metal.

6.6 Plasmonic Surfaces

One of the key developments in SE(R)RS over the last 10 or so years has been the development of structured surfaces designed to achieve strong SERS enhancements by controlling the plasmonics of the surface and thus the electromagnetic enhancement. This is a very active area of research which overlaps with work in the wider fields of nanophotonics and optical metamaterials [104, 105]. A number of excellent reviews of the area have been published in the last few years [11, 15–17, 26–28, 106].

The plasmon resonances of metallic nanostructures can be modeled using classical electromagnetism and analytical solutions are available for simple highly symmetrical cases such as spherical particles. For more complex nanostructures it is necessary to use numerical methods such as the discrete dipole approximation [107], finite difference time domain methods [108, 109], or boundary element techniques [110]. However, these calculations are complex and computer intensive so that it is not possible to accurately model extended nanostructures. An important approach to understanding the plasmonics of these more complex nanostructures is the concept of plasmon hybridization [111-113]. The idea of plasmon hybridization exploits the fact that there are similarities between the way plasmons behave and the behavior of electrons in atomic orbitals so that, in the same way that atomic orbitals can combine to form new molecular orbits if the energies and symmetries are correct, plasmons in complex nanostructures can be envisaged as being made up from combinations of interacting plasmons. This is illustrated in Figure 6.5 for the example of a nanoshell particle. In the plasmon hybridization model the plasmon resonance of the nanoshell is built up by combining the plasmon resonances for a nanosphere and for a nanocavity. For these two simple, symmetric structures Mie theory gives the plasmon frequencies [111] for the sphere as

$$\omega_{S,l} = \omega_B \sqrt{\frac{l}{2l+1}} \tag{6.4}$$

and for the cavity as

$$\omega_{\mathrm{C},l} = \omega_{\mathrm{B}} \sqrt{\frac{l+1}{2l+1}} \tag{6.5}$$

where l is the angular quantum number and ω_{B} is the bulk plasmon frequency.

Provided the plasmon resonances of the component structures are of the same symmetry and similar energy they can be combined together to form hybrid plasmon resonances in the same way that molecular orbital theory is used to combine together atomic orbitals to form bonding and antibonding molecular orbitals. As shown in Figure 6.5 the l=1 modes for the cavity and sphere mix to

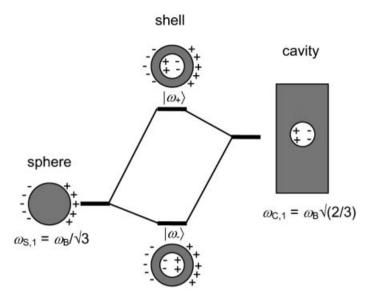


Figure 6.5 Energy level diagram for plasmon hybridization in a metal nanoshell by combination of the plasmons for a sphere and a cavity. The two nanoshell plasmons are symmetrically and antisymmetrically coupled and have energies given by Eq. (6.6).

produce two new modes with the higher energy ω_{+} mode corresponding to the antisymmetric coupling (this would be the antibonding orbital in molecular orbital theory) and the lower energy ω mode corresponding to the symmetric coupling. The plasmon frequencies for these hybridized plasmons are given by [112]

$$\omega_{l\pm}^2 = \frac{\omega_{\rm B}^2}{2} \left[1 \pm \frac{1}{2l+1} \sqrt{1 + 4l(l+1) \left(\frac{a}{b}\right)^{2l+1}} \right]$$
 (6.6)

where *a* is the inner and *b* the outer radius of the nanoshell.

This plasmon hybridization approach [114] has been used very effectively to describe the plasmonics of a number of core-shell [112, 115, 116] and other structures [114, 117, 118] including effects on extended nanostructured surfaces [119-121].

For electrochemical SE(R)RS measurements it is necessary to have a continuous conducting surface to use as the electrode and this rules out the use of many nanoparticle and core-shell-type structures used for SE(R)RS studies. In the next section we concentrate on electrode structures for SE(R)RS.

6.7 SERS Surfaces for Electrochemistry

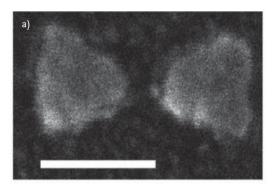
Over the last few years there has been an increasing move away from roughened electrode surfaces for electrochemical SE(R)RS with all their inherent limitations towards structured surfaces which show large, reproducible, repeatable surface enhancements [19, 20].

Knoll *et al.* [36] carried out a systematic study of SERS for Langmuir–Blodgett films of cadmium arachidate with different numbers of monolayers on Ag gratings. They clearly demonstrated, by a careful study of the angle dependence of the spectra, that the enhancement was caused by coupling with surface plasmons on the grating [122] and that it depended on both coupling of light in and coupling of the Raman scattered light out from the grating (see Figure 6.3). Subsequently Baltog *et al.* [34, 123] demonstrated a 10⁴ enhancement of SERS of copper phthalocyanine on an Ag grating due to the resonant excitation of delocalized surface plasmons. Again they showed that the Raman-scattered light emerges at particular angles corresponding to resonance of the outgoing light with the grating and different from the angle for resonance of the incoming laser light. Essentially similar conclusions were reached by Kahl and Voges [35]. These initial studies on gratings pointed the way toward developing structured surfaces for SERS which can give surface enhancements without high surface roughness or randomly distributed hotspots.

There are several ways to structure surfaces on a scale close to the wavelength of light in order to generate surfaces with strong SERS enhancements. These can be divided into top-down methods, such as electron beam lithography, focused ion beam milling, and soft lithography, or bottom-up methods that rely on self-assembly to build up the structures. Electron beam lithography is an attractive approach because it allows a high degree of control over the structures that can be made, but it is an expensive approach requiring access to specialized equipment and is unsuited to large-scale fabrication.

Structures made by top-down methods are very useful in exploring the SERS effect and the role of surface plasmons [124]. One structure which has attracted attention because of its interesting plasmonic properties [118] is the nanohole array. Brolo et al. [125] used focused ion beam milling to fabricate square arrays of sub-wavelength holes on an Au film with periodicities of 560, 590, and 620 nm. They found enhancements in the SERRS for oxazine at the surface when irradiated at 633 nm. Anema et al. [126] carried out similar experiments for oxazine on square arrays of 133 nm diameter nanoholes in Cu. Their arrays had periodicities between 360 and 740 nm and were made in 144 nm thick Cu films on glass. The highest SERRS intensity, with an estimated enhancement factor of 10⁴, was observed for the array with a periodicity of 578 nm when using 633 nm irradiation. These results were found to be consistent with calculations of the electric field intensity as a function of the array periodicity. Reilly et al. investigated SERS on arrays of 200 nm diameter holes in a 50nm Ag film as a function of lattice spacing [127]. They concluded, on the basis of quantitative measurements for a nonresonant system, that the enhancement originated from two multiplicative contributions: a contribution from field localization near the aperture edges and a contribution from the surface roughness between the apertures. They concluded that the aperture contribution was a factor of around 600 and that from surface roughness around 10⁵, giving a total enhancement of 6×10^7 in the optimal case.

Im *et al.* [128] have used optical lithography combined with atomic layer deposition to form vertically oriented metal—dielectric—metal nanogap structures designed to achieve significant field enhancements with strong localization within the gap. The method was used to fabricate a number of different nanogap structures and it was shown that the local enhancement increased as the size of the nanogap decreased from 20 to 5 nm. Dinish *et al.* [129] used deep-ultraviolet lithography to fabricate nanogaps between circular Au–Ag structures and obtained SERS intensities slightly greater than those for metal film over nanosphere (MFON) structures (see below) for 2-napthalenethiol. Jackel *et al.* [130] used electron beam lithography to fabricate bowtie structures (two Au nanotriangles oriented tip to tip with a nominal gap of 35 nm) on an indium tin oxide-coated fused silica substrate and used the structure to study electrochemical SERS of *p*-mercaptoaniline at individual bowtie structures. They found substantial changes in the shapes of the bowties caused by two electrochemical cycles between –0.4 and +0.4 V vs. Ag/AgCl in 0.1 M NaClO₄ (Figure 6.6).



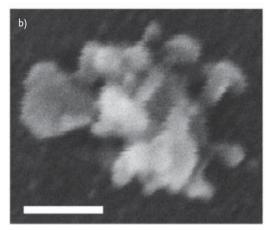


Figure 6.6 Scanning electron microscopy images of Au nano-bowtie structures (a) before and (b) after cycling in sodium perchlorate solution. The scale bars are

100 nm. Adapted from Jackel, F., Kinkhabwala, A.A., and Moerner, W.E. (2007) *Chemical Physics Letters*, **446**, 339, with permission from Elsevier.

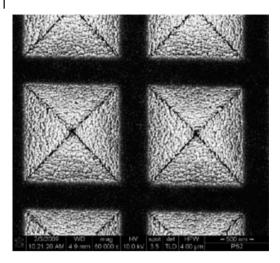


Figure 6.7 Scanning electron microscopy image of part of a Klarite[®] substrate showing some of the Au-coated inverted pyramidal pits. Reprinted from Ignat, T., Munoz, R., Irina, K. *et al.* (2009) *Superlattices and Microstructures*, **46**, 451, with permission from Elsevier.

Lithography and anisotropic etching of (100)Si has also been used as a way to produce structures for SERS. In this case the etching produces pyramidal pits in the surface with a fixed apex pit angle of 70.5°. These are then coated with Au by evaporation to produce a nanostructured surface (Figure 6.7). In this case the optical properties of the surface depend on the geometry of the pits (their spacing and the size of the opening or aperture). These structures are commercially available under the trade name Klarite® and have been used for SERS measurements by several authors including studies with molecularly imprinted polymers [131], supramolecular host molecules [132], squarines [74], and dye-labeled DNA [133, 134]. Meyer *et al.* [135] have compared the SERS of rhodamine 6G on Klarite® and several other surfaces. This type of structure has been fabricated by Ohta and Yagi [136] and used for electrochemical SERS measurements with pyridine in 0.1 M KCl between –0.06 V and +0.84 V vs. RHE.

Templated self-assembly approaches have the significant advantage over top-down approaches of being more generally accessible, since they do not require clean-room access and access to specialized expensive equipment, and are amenable to scale-up. In particular, methods based on nanosphere lithography and the assembly of colloidal templates have been very successfully applied to fabricate SE(R)RS substrates. There are a number of variations on the approach, producing several different types of structure (Figure 6.8), but all begin with the assembly of a closed packed array of uniform spherical particles to act as the template or mask. This approach originates with the work of Deckman and Dunsmuir on "natural lithography" [137].

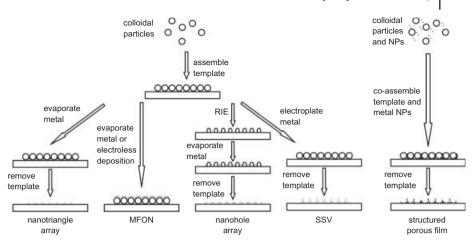


Figure 6.8 Different examples of colloidal templating, or "natural lithography," used to make SERS-active surfaces starting from uniform spherical colloidal particles, usually of polystyrene or silica. (NPs, nanoparticles; RIE, reactive ion etching.)

The most commonly used template particles are either polystyrene or silica spheres. These can be synthesized or obtained commercially and are available in a range of sizes (from about 100 nm upwards) with a narrow coefficient of variation (<2%) in diameter. The size variation is important because the more uniform the size of the particles the more regular the packing. The particles can be assembled into templates in a variety of ways, including evaporation of the solvent [138], convective self-assembly [139–141], Langmuir–Blodgett deposition [142, 143], or by withdrawing the substrate slowly from a suspension of the colloidal particles [144]. The optimum methods for the assembly of the silica and polystyrene particles differ because of the differences in their densities and hence sedimentation rates. Once assembled on the surface the particles act as a template to define the deposition.

These templates have been used in various different ways to make SE(R)RS substrates. Van Duyne and coworkers [106, 145] used evaporation of metal (typically Ag or Au) through the template. The metal evaporation occurs by line of sight so only the areas of the substrate exposed between the template spheres become coated by metal. Thus when the template is removed the substrate is left covered in an array of metal triangles oriented tip to tip (Figure 6.9) like an ordered array of the bowtie structures [130] discussed above. By varying the size of the template spheres, the amount of metal deposited and by using templates of one or of two monolayers of spheres and by varying the angle of metal deposition, they were able to make a range of structures with different sizes and geometries [146–148]. These surfaces have been used extensively by Van Duyne's group for localized surface plasmon studies [146–154] and as SE(R)RS substrates [145, 155, 156]. The plasmonics and SERS on these surfaces are dependent on the geometry [157, 158]

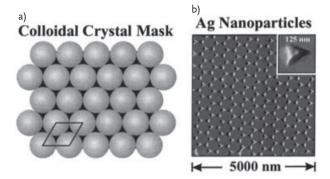


Figure 6.9 (a) Schematic of a single-layer colloidal template. The dark areas show the exposed parts of the underlying surface. (b) Representative ambient contact mode atomic force microscopy image of the array of nanotriangles formed by thermal evaporation of 48 nm of Ag through a single-layer colloidal

template (sphere diameter 542 nm) followed by removal of the template by sonication in methylene chloride for 3 min. Reprinted with permission from Haynes, C.L. and Van Duyne, R.P. (2001) *Journal of Physical Chemistry B*, **105**, 5599. Copyright 2001 American Chemical Society

and dominated by the small gap regions between the triangular structures. Haynes and Van Duyne have measured enhancement factors in excess of 10^8 for SERS of benzenethiol on these Ag structures and in excess of 7×10^9 for SERRS [155], although there was a significant amount of scatter in the measurements. As far as we are aware these triangular structures have not been used for electrochemical SERS, although such structures have been made on an indium tin oxide surface [159].

If a thicker metal film is evaporated so that it totally covers the template spheres a two-dimensional grating structure is formed [145]. This is referred to by Van Duyne and coworkers as a MFON structure [160] (Figure 6.10). Evaporated Ag film over nanosphere (AgFON) or Au film over nanosphere (AuFON) surfaces give large SERS enhancements [156, 161, 162], although data from Im et al. [128] indicate that the SERS comes from randomly distributed hotspots across the surface. Bantz and Haynes [163] have described an alternative approach to the preparation of AgFON surfaces that uses electroless plating of Ag to form the metal film rather than evaporation. The resulting surfaces are less topographically homogeneous but still exhibit strong surface enhancement. Van Duyne's group have used MFON surfaces to make biosensors [164-166] and have shown that for AgFON structures made with silica templates the thermal stability is good [167]. These MFON structures can be used for electrochemical SERS as there is now a continuous conducting surface. Van Duyne's group have used AuFON and AgFON surfaces for electrochemical SE(R)RS and surface-enhanced hyper-Raman scattering studies [156, 168]. For example, Dick et al. have studied the distance and orientation dependence of electron transfer to cytochrome c at thiol-coated AgFON electrodes [169]. A significant advantage of this type of structured surface over roughened

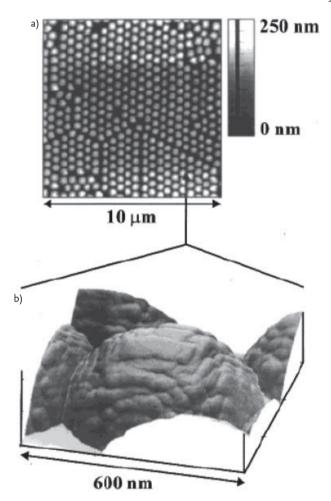


Figure 6.10 Ambient contact mode atomic force microscopy images of an AgFON structure. The structure was made by depositing 200 nm of Ag over 542 nm diameter polystyrene spheres. (a) The array of spheres and (b) image of one Ag-coated

sphere showing the substructure roughness. Reprinted with permission from Dick, L.A., McFarland, A.D., Haynes, C.L., and Van Duyne, R.P. (2002) *Journal of Physical Chemistry B*, **106**, 853. Copyright 2002 American Chemical Society.

electrode surfaces is the significantly greater stability, so that the electrode can be cycled to extremely negative potentials (-1.3 V vs. Ag/AgCl) without loss of SERS activity [160]. AgFON surfaces have also been used for borrowed SERS by coating with a thin layer of Pt [170].

Comparison of the enhancement factors between triangular nanoparticle arrays formed by nanosphere lithography and the MFON structures [8] shows that although the enhancement factor is about 10 times larger for the triangular

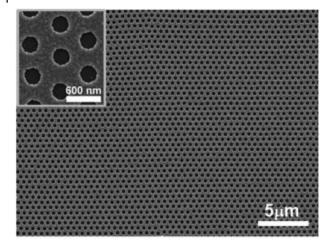


Figure 6.11 Scanning electron micrographs of tape and solvent cleaning. Reprinted with a single-crystalline hexagonal nanohole array formed from a template of 600 nm polystyrene spheres by size reduction using reactive ion etching followed by deposition of 50 nm of Ag deposition and removal of the template with

permission from Lee, S.H., Bantz, K.C., Lindquist, N.C. et al. (2009) Langmuir, 25, 13685. Copyright 2009 American Chemical Society.

nanoparticle array, the SERS signals are larger for the MFON structures because there is a larger active area covered by molecules.

In a variation of the nanosphere lithography approach Lee et al. [138] used reactive ion etching to reduce the size of polystyrene spheres before evaporation of Ag onto the structure. Removal of the spheres then produces a nanohole array (Figure 6.11) similar to that fabricated by focused ion beam milling. For benzenethiol Lee et al. obtained enhancement factors of up to 8×10^5 for these nanohole structures.

Self-assembled templates of colloidal particles have also been used to prepare metallic structures where the voids between the template spheres are filled with metal. Kubo et al. [144] used a dipping method to produce metal-coated colloidal crystal films by first depositing monodisperse 300 nm silica spheres onto a glass substrate in the presence of 10 nm Ag nanoparticles. This produced, after calcining, a structure in which the Ag nanoparticles were immobilized on the silica spheres. SERS signals of p-toluenethiol of this structure were 40 times stronger than for a flat Ag film and 3 times larger than for electrochemically roughened Ag with good reproducibility from place to place on the surface.

Velev and coworkers [140, 171] used a similar approach of co-assembling colloidal particles with Au nanoparticles. In this case they removed the colloidal template to produce a porous Au inverse opal structure (Figure 6.12). They found evidence for SERS activity on these structures but concluded that the enhancements were due to the nanoscale surface roughness stemming from the discrete aggregated Au particles [140].

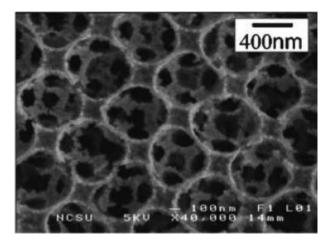


Figure 6.12 Scanning electron microscopy image of a mesoporous Au film made by co-assembly of colloidal template particles and Au nanoparticles following removal of the

template. From Kuncicky, D.M., Prevo, B.G., and Velev, O.D. (2006) *Journal of Materials Chemistry*, **16**, 1207. Reproduced by permission of the Royal Society of Chemistry.

Lu *et al.* [172] used a template assembled from uniform silica spheres to produce both an ordered macroporous Au–Ag nanostructure and an ordered hollow Au–Ag nanostructured film by electroless deposition. Both films showed SERS activity but were rather rough on the nanoscale and the authors attributed the surface enhancement to the presence of interconnected nanostructured aggregates and nanoscale roughness.

The problem with the nanoparticle infiltration and electroless deposition approaches to fill the colloidal templates with metal as described above is that it is very difficult, if not impossible, to accurately and systematically control the structure and the deposited metal is rough on the nanoscale and often porous so that the surface area is poorly defined. These problems can be overcome by using electrodeposition through the template. Electrodeposition is a volume-filling method and the thickness of the electrodeposited metal film is readily controlled [173]. The resulting films have very low nanoscale roughness and their structure can be readily varied. This approach has been developed at the University of Southampton over the last 10 years. The surfaces are prepared by electrodeposition of a metal, typically Au, through a close-packed monolayer of uniform polystyrene spheres assembled on a flat conducting surface [141, 173]. After deposition of the metal the polystyrene spheres are removed by dissolution in an organic solvent to leave a film with a regular array of sphere segment voids (SSVs) (Figure 6.13). The diameter of the SSVs is controlled by the choice of the polystyrene spheres (typically around 400-900 nm in diameter) and the thickness of the metal film is controlled by the charge passed during deposition (typically 30 to 80% of the sphere diameter) [174].

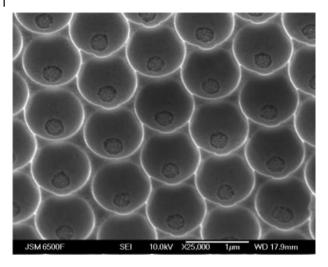


Figure 6.13 Scanning electron microscopy image of an Au SSV structure. Note the smooth electroplated metal walls and top surface. The rough circular areas at the bottom of each cavity are the evaporated Au substrate.

These SSV surfaces show strong reproducible SE(R)RS [175–177] with enhancement factors of 2.7×10^7 for benzenethiol on an Au SSV surface with typically about 10% variation from place to place across the surface [178]. By careful analysis of the reflection spectra of these nanostructured metal surfaces as a function of sphere diameter, film thickness, type of metal, light polarization, azimuthal angle, and angle of incidence it is possible to build up a detailed picture of the plasmonics on these structured surfaces and to use this knowledge to design surfaces for particular applications [33, 110, 120, 121, 174, 178–182]. These studies reveal that there are both Bragg-type propagating and Mie-type localized plasmons on the SSV surfaces and that these plasmons couple together [121].

This detailed understanding of the localized electromagnetic field on SSV metal surfaces due to the plasmonic resonances makes these surfaces ideal platforms for SE(R)RS [175, 178, 183]. Thus we have been able to design SSV surfaces suitable for SERS measurement in the ultraviolet [184], in the visible [175, 178, 183], and in the near-infrared [176] regions of the spectrum by varying the sphere diameter and film thickness. We have also been able to produce films of Pt and Pd, metals which do not show strong enhancements because of their fundamental optical properties, with surface enhancements equal to the highest reported values [185] for these metals. Our understanding of the detailed plasmonic properties of the SSV surfaces has enabled us to clearly demonstrate, by making angle-resolved measurements, the importance both of coupling light from the laser into the surface and of coupling the Raman-scattered light out to the detector [33]. We have demonstrated the relationship between the SERS intensity and the plasmonics of the surfaces [186], we have shown that the contribution of resonant enhancement

on the SSV surfaces is around three orders of magnitude [177], and we have examined in detail the origins of the broad SERS background and how it arises on the SSV surfaces [187].

SSV surfaces are ideally suited for electrochemical SE(R)RS studies because of their low surface roughness, high surface enhancement, and good stability. They have been used for electrochemical SE(R)RS studies of pyridine [175], flavin [183, 188], adenine [184], β -thioglucose [189], and an osmium redox hydrogel [188], and for discrimination of DNA mutations [190, 191]. Recent work by Jose *et al.* has also demonstrated enhanced fluorescence on SSV surfaces [192].

6.8 Tip-Enhanced Raman Spectroscopy

It is appropriate to conclude this part of the chapter, before going on to review the literature on SE(R)RS of biomolecules at electrode surfaces, by briefly describing tip-enhanced Raman spectroscopy (TERS) since this rapidly developing technique offers the potential for studies at molecular resolution. In TERS a metal nanoparticle or metalized tip (usually Ag or Au) with an apex diameter of about 25 nm is illuminated by a laser as it is scanned across the surface (Figure 6.14). The tip is used to locally amplify and confine the electromagnetic field, in effect creating a local hotspot which can be scanned across the surface. The first examples of this approach were reported in 2000 [193–195]. Since then the approach has been

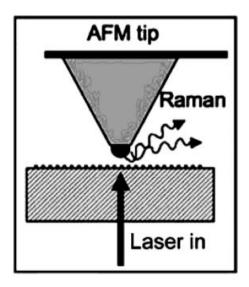


Figure 6.14 Schematic of a modified atomic force microscopy (AFM) tip used for TERS. Reprinted from Yeo, B.-S., Stadler, J., Schmid, T. *et al.* (2009) *Chemical Physics Letters*, **472**, 1. Copyright (2009), with permission from Elsevier.

applied to study a number of biomolecules at surfaces including oxidized glutathione on Au [196], histidine on Ag [197], cytochrome c [198], adenine [199, 200], and other DNA nucleotides and bases [201, 202], and RNA on mica [203]. For further details of TERS the reader is referred to several recent reviews [204–207].

The idea of TERS is to have a localized probe that, ideally, stimulates SERS but without perturbing the chemical system under study. This concept has been taken forward by Tian's group in recent work on silica- or alumina-coated Au nanoparticles in so-called shell-isolated nanoparticle-enhanced Raman spectroscopy, or SHINERS [208]. The idea is that the thin (a few nanometers) silica or alumina coating renders the particles inert so that they can be scattered over the surface of interest to generate SERS signals without perturbing the surface chemistry, rather like a large random array of TERS tips.

6.9 SE(R)RS of Biomolecules

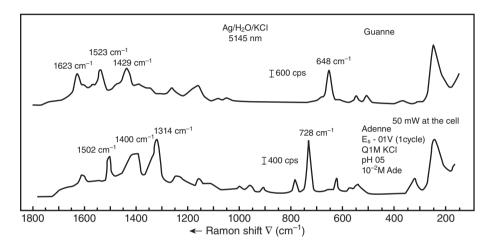
As mentioned earlier in this chapter, water is a very weak Raman scatterer and therefore does not obscure SERS signals from biomolecules in aqueous solution. In addition SERS is very well suited as an *in situ* technique for monitoring changes in surface chemistry, including the adsorption and orientation of molecules. Thus SERS is an ideal complement to bioelectrochemical studies and provides surface-selective, molecule-specific information. Frequently it is possible to increase SERS intensities by manipulating adsorption of biomolecules through changes in the electrode potentials and this represents a further, significant advantage.

In the remainder of this chapter we review the applications of SE(R)RS in studies of biomolecules at electrode surfaces. We review SE(R)RS as applied in bioelectrochemistry, concentrating attention on studies where the interfacial potential is under direct electrochemical control. In general, and for the sake of brevity, we exclude any detailed discussion of SE(R)RS studies of biomolecules at colloidal nanoparticles where the interfacial potential is undefined; for accounts of these studies the reader is directed to some recent reviews [209–214].

6.9.1 DNA Bases, Nucleotides, and Their Derivatives

Nucleic acids and their constituents were among the first biomolecules to be studied on electrodes with SERS. Among the first reports on nucleic acid components studied on roughened Ag electrodes are those on adenine and its nucleotide derivatives by Koglin and coworkers [215, 216]. Their work showed that surface Raman spectra were stronger than their solution-based counterparts and they reported a detailed spectroelectrochemical study [216] where the intensity of the peaks was found to be strongly dependent on the applied potential and on the preparation conditions for the substrate. A strong peak at around 740 cm⁻¹ was visible for all adenine derivatives including adenosine, adenosine monophosphate

(AMP), and adenosine triphosphate (ATP) in both the studies, and a band at 248 cm⁻¹ was assigned to the Ag-phosphate vibration. In general the intensities of all peaks were found to decrease with increasing negative potentials for all of the nucleotide derivatives. Spectra of other nucleotide bases and of adenine, recorded on Ag electrode surfaces, were reported by Koglin *et al.* [217]. The SERS spectra of adenine, guanine, cytosine, and thymine adsorbed on a polycrystalline Ag electrode surface showed prominent SERS bands at 728, 648, 798, and 782 cm⁻¹, respectively, corresponding to their ring-breathing modes (Figure 6.15). The



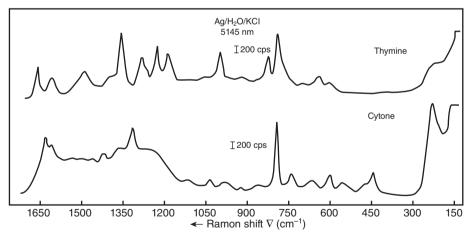


Figure 6.15 SERS spectra of protonated guanine, adenine, thymine, and cytosine (0.3 M HCl; 1×10^{-2} M guanine/adenine; pH = 0.5; electrode potential vs. sat. Ag/AgCl electrode, -0.1 V; electrode area, 0.2 cm²; laser power at the cell, 100 mW). The Ag electrode

was activated by one triangular voltage sweep between –0.1 and +0.2 V at a sweep rate of 50 mV s⁻¹. Reprinted from Koglin, E., Séquaris, J.M., and Valenta, P. (1982) *Journal of Molecular Structure*, **79**, 185. Copyright (1982), with permission from Elsevier.

spectral band positions are sensitive to pH and shift quite a lot in acidic media for the purine bases (adenine and guanine) compared to their position at pH = 7. Another study was carried out on DNA bases by Otto *et al.* in which they found typical differences between the normal Raman and SERS spectra [218]. They also saw changes in the SERS spectra with changes in potential but they did not study this in detail. The differences in shape and intensity of the carbonyl stretching vibrations are related to the different modes of adsorption with respect to the surface. The SERS spectra of DNA bases on surfaces are dependent on the adsorption process which, in turn, can be controlled by potential but is also clearly dependent on co-adsorption of ions [219] and the relative strength of the adsorbate—surface interaction as indicated by the heat of desorption as studied on Au surfaces [220].

SERS spectra are sensitive to proximity to the surface and the orientation of molecules, surface preparation, ions in solution, pH, and electrochemical potential, among other factors; these can lead to differences in the spectra reported for the same compound, giving rise to apparent contradictions in the literature. A case in point is the SERS spectra of adenine reported on Ag electrodes with both flat [216] as well as perpendicular [221] orientations being postulated. Proper assignment of each of the bands is extremely important in predicting orientations and this can be achieved by numerical simulations and comparison to experimental spectra [222]. Using electromagnetic theory-based surface selection rules, Giese and McNaughton conclude that on Ag colloids adenine adopts a more perpendicular orientation while on electrodes it has a more tilted orientation. That this is the case for Au surfaces has been reported by Kundu et al. for nanoshell surfaces where they have studied adenine, AMP, and polyadenine [223]. They have also carried out density functional theory (DFT) calculations which support the experimental conclusion that the adenine adsorbs "end-on" with the exocyclic NH2 oriented almost normal to the surface.

Similarly the Raman and surface-enhanced Raman spectra of guanine and uracil have been recorded under a range of experimental conditions, including surface potential, and their spectra predicted by DFT calculations at the (B3LYP/6- $31^{++}G(d,p)$) level [224, 225]. Sardo *et al.* have recently studied a chemotherapeutic agent, 5-fluorouracil, on Ag nanostructures including electrodes [226]. They also carried out DFT calculations to deduce the positions of the main bands in the Raman spectra and studied the potential dependence on an Ag electrode to probe the charge transfer mechanism. Based on their calculations, and comparison with experimental spectra, they were able predict that the N_1 deprotonated anion is responsible for the spectral features. These studies highlight the importance of theoretical work in understanding the orientation of species at electrode surfaces and the possible adsorption behavior of neutral and charged molecules.

Not only can changes in orientation be studied and rationalized using SERS measurements but also the relative proportion of adsorbates can be determined. Watanabe and coworkers have studied potential-dependent changes in the spectra of adenosine, cytidine, and their mixtures among other derivatives of these nucleosides on Ag surfaces [221]. Cytidine displays weaker intensities than adenosine

and they show very different potential dependence. Although it is clear that the relative proportions can be worked out by appropriate calibration, careful control of the conditions (such as the electrode potential) used when recording SERS spectra is necessary for quantification of the respective amounts.

Apart from adenine, much SERS work has been carried on thymine under electrochemical control. Cunha et al. studied the potential-dependent behavior of thymine at Ag electrodes [227]. Their detailed study demonstrates that determination of the potential of zero charge (PZC) is important for understanding the adsorption behavior of biomolecules at electrodes. Furthermore, changes in the spectra with potential can be rationalized on the basis of changes in adsorption behavior depending on whether one is positive or negative of the PZC. Drawing on the assignments from Aroca and Bujalski [228] the possible changes in orientation of thymine at potentials positive and negative of the PZC were deduced [227]. However, the most striking evidence seems to be the change of the low-frequency band at 216 cm⁻¹ (Ag-O) to 226 cm⁻¹ (Ag-N) on going from -0.3 to +0.6 V vs. Ag/ AgCl/KCl(sat.), suggesting that the thymine adsorbs through O₈ in the first case and a ring nitrogen in the second (Figure 6.16a). Based on other potentialdependent changes, a model for adsorption of thymine on Ag was presented, as shown in Figure 6.16b. Another SERS study on thymine was carried out on a roughened Cu electrode by Shang et al. [229]; in this case potential-dependent signals were observed. The surface attachment (chelation) models were verified with DFT calculations leading to the conclusion that thymine molecules are likely to be adsorbed through N₁ and O₇ (through the O=C-N moiety). Besides changes in intensities and the relative heights of different peaks as a function of potential, SERS peaks also undergo shifts in position with electrode potential (Stark shift; see Section 6.3). The extent of this Stark shift is often related to the charge transfer

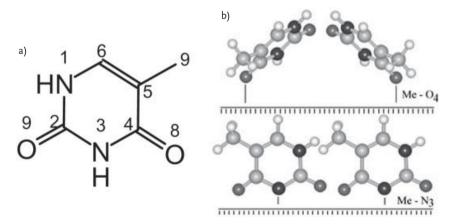


Figure 6.16 (a) Molecular structure of thymine. (b) Adsorption model of thymine on an Ag electrode. With kind permission from Springer Science+Business Media: Cunha, F., Garcia, J.R., Nart, F.C. *et al.* (2003) *Journal of Solid State Electrochemistry*, **7**, 576, figure 5.

mechanism and can be indicative of chemical interaction with the surface. In one such study using ultraviolet excitation, Hao and Fang found that the C=O bending and ring-breathing modes of thymine at 618 and 755 cm⁻¹, respectively, underwent a blue shift (to higher frequencies) of about 3 to 19 cm⁻¹ on going from -0.1 to -1.3 V vs. Ag/AgCl on an Au electrode [230]. Recently Cui *et al.*, in their study of co-adsorption with perchlorate, found that thymine showed the weakest ability to co-adsorb on Au electrodes among DNA bases although the protonated forms of the bases were found to be more easily adsorbed at negative potentials [219]. These studies clearly demonstrate the importance of considering the chemical nature of the molecule under study and show that SERS is a sensitive probe of the nature of the interaction at the electrode interface.

6.9.2 DNA and Nucleic Acids

Most of the early work on SERS of nucleic acids themselves was carried out with colloidal nanoparticles [231–233]. However, DNA has also been extensively studied by SERS in tandem with electrochemistry. Early reports consisted of qualitative observation of the SERS spectra of nucleic acids and their possible conformational changes [215, 234, 235]. However, a proper study of DNA from calf thymus and the effect of adsorption on electrodes was reported by Brabec and Niki [236]; in this case they primarily observed adenine peaks in the spectra. Barhoumi et al. also found, in their SERS study of Au nanoshell substrates, that thermally uncoiled single-stranded DNA (ssDNA) sequences gave spectra dominated by the adenine and that thermal pretreatment resulted in better reproducibility as compared to the untreated oligonucleotides [237]. Furthermore, Brabec and Niki were able to prove that double stranded DNA (dsDNA) could be adsorbed on electrode surfaces without denaturation [236]. Interestingly they observed a change in the relative intensity of the adenine band (as shown in Figure 6.17) on denaturation of dsDNA on the surface suggesting that the helix-coil transition could be followed using SERS. It is pertinent to point out that Barhoumi et al. did not observe an increase in the adenine signal on thermal denaturation of dsDNA for the model sequences on their Au surfaces [237]. Nevertheless, spectra showing that the conformational rigidity of a DNA duplex is maintained on an Ag surface upon adsorption were provided by Koglin and Sequaris [238]. That the conformation of even a triplex oligonucleotide is preserved on adsorption on Ag electrode surfaces was shown by Fang et al. [239]; by carrying out potential-dependent SERS measurements they concluded that positive potentials were more conducive to the adsorption of the negatively charged triplex. At negative potentials (around and below the PZC of -0.8 V vs. Ag/AgCl) there was a decrease in the intensity of the peaks due to possible desorption.

Monolayers of nucleic acids are an important element in many putative biosensors. Some studies, mainly carried out over the last decade, have reported nucleic acid monolayers on surfaces. For example, monolayers of ssDNA and dsDNA were studied by recording both *ex situ* and *in situ* SERS spectra on Au electrodes [240].

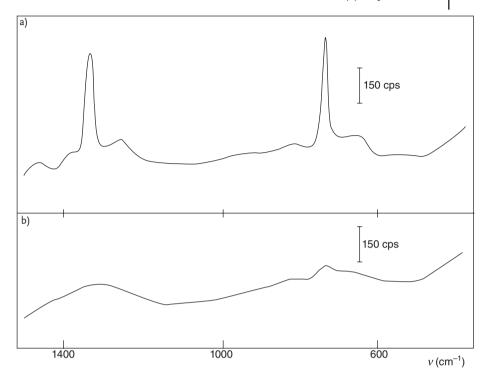


Figure 6.17 Surface Raman spectra of (a) thermally denatured DNA and (b) native DNA adsorbed at an Ag electrode polarized to -0.1 V in 0.1 M KCl with 1 mM phosphate buffer (pH = 7.0). DNA concentration was

0.2 mg/mL. Waiting time at -0.1 V was approx. 40 min. Reprinted from Brabec, V. and Niki, K. (1985) *Biophysical Chemistry*, **23**, 63, with permission from Elsevier.

In this work Dong *et al.* suggest that the duplexes undergo a vertical to horizontal transition in orientation on going from negative to positive potential. Furthermore, it was demonstrated by electrochemical scanning tunneling microscopy and electrochemical SERS on Au electrodes that dsDNA formed highly ordered and compact monolayers stable over a wide range of potentials [241]. The dsDNA was essentially arranged as rods parallel to each other and a change in orientation occurred only at positive potentials around +0.8 V vs. SCE. In contrast ssDNA adopted a coiled-like conformation.

Work on surfaces has proceeded more in the direction of developing sensing applications by using hybridization to surface-bound oligonucleotides which act as "probes" for solution "targets" with complementary sequences. Vo-Dinh and colleagues were the first to demonstrate this concept of a "SERS gene probe" using cresyl violet-labeled sequences [242]. Instead of signals from the nucleic acid, the stronger signature peaks of the label are detected. This strategy has been further developed for applications [79] including that for HIV [82] and cancer (BRCA-1 gene implicated in breast cancer) [81, 243] detection. This concept of hybridization

sensors is very attractive for genomic analysis and incorporation onto DNA chips. Recent work has integrated on-chip DNA analysis by SERS with microfluidics wherein the platform consists of a structured surface on which probe oligonucleotides are immobilized [244]. In this regard an interesting study was reported by Huh *et al.* who used electrokinetically controlled microwells on a microfluidic device to increase the concentration of DNA for SERS detection of single nucleotide polymorphisms [245]. Although the SERS signals generated are nanoparticle based, the detection strategy is unique employing ligation at the mismatch site (called the ligation detection reaction) to bring the nanoparticle close to a dye label for detection.

Another way to develop DNA-based sensors is to utilize the reverse of hybridization, that is, the duplex denaturation process. The ability to controllably denature dsDNA on electrode surfaces and to monitoring this transition by SERS have been utilized by Mahajan et al. to distinguish between mutations [191, 246]. Although labeled oligonucleotides were used, the method relies on the concept of monitoring the helix-coil transition-denaturation by SERS as alluded to by Brabec and Niki [236]. The denaturation process can itself be induced thermally or electrochemically resulting in "melting" profiles which are dependent on the sequences and the number of mismatches, and thus allow distinction between healthy and mutated gene sequences [246]. The electrochemical denaturation option is particularly attractive for integration with gene chips for increased addressability, miniaturization, and rapid analysis. This has been applied for discriminating between the wild-type and triplet deletion (Δ F508 mutation) sequences in solutions containing the unpurified polymerase chain reaction products as shown in Figure 6.18. The electrochemically induced melting method, called "SERS-Emelting," has been further extended to analyze short tandem repeats, which are commonly used to determine genetic profiles in forensic applications [190].

Recent work has focused on improving SERS labels for hybridization detection and the process of hybridization. Use of peptide nucleic acids as uncharged probes for DNA detection [247] offers some advantages in studying interactions with neutral species. New anthraquinone derivatives as electrochemical labels (redox indicators) for in situ SERS monitoring have been reported by Kowalczyk et al. [248]. An interesting approach to label-free detection for DNA hybridization has been reported by Halas and coworkers where they use SERS on nanoshell surfaces [249]. By replacing all the occurrences of adenine in the DNA probe sequence with 2-aminopurine, an artificial adenine substitution that preserves the hybridization characteristics, they were able to use the strong SERS band of adenine itself at 736 cm⁻¹ as a marker for hybridization. Although it is clear that great progress has been made in nucleic acid detection and in developing sensing applications, gene detection, and in particular quantification, remains an issue. There is some indication that this issue could be addressed, for example, by internal referencing using what is referred to as the "gold-plasmon" band in Au nanoparticle films [250]; however, this is still an active area of research for the entire SERS community.

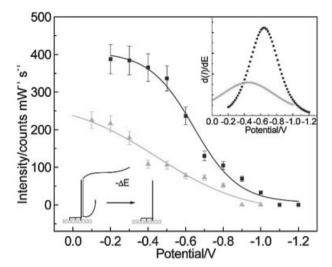


Figure 6.18 Resonant SERS-melting profiles of the polymerase chain reaction products for wild type (squares) and Δ F508 mutation (triangles) using the 1347 cm⁻¹ SERRS band for Cy5. A schematic of the dehybridization process of the polymerase chain reaction products is shown. The first derivatives of the melting profiles are shown in the inset. The mutation has a melting potential of -0.44 V,

while that of the wild type is -0.64V vs. SCE. Spectra were acquired with a single static scan of 2s exposure at 1.5 mW laser power in 10 mM Tris buffer (pH = 7). Reprinted with permission from Mahajan, S., Richardson, J., Brown, T., and Bartlett, P.N. (2008) *Journal of the American Chemical Society*, **130**, 15589. Copyright 2008 American Chemical Society.

6.9.3 Amino Acids and Peptides

For historical reasons most of the work on amino acids is on electrochemically roughened electrodes. Nevertheless these studies provide important information on the adsorption and orientation of amino acids at interfaces. Phenylalanine (Phe), tyrosine (Tyr), and tryptophan (Trp) are by far the most studied amino acids using SERS. Phe, Tyr, and Trp was first studied by Nabiev *et al.* on Ag electrodes in 1981 [251]. In that work they showed detection at a sensitive level and clear differences between the spectra. While there are other studies of these amino acids that discuss the orientation and peak assignments on Ag colloids these quite often contradict each other and in some cases completely different spectra are reported for the same compound. In view of this probably the most comprehensive study of amino acids is that by Stewart and Fredericks in which 19 amino acids were studied on roughened Ag electrodes, their spectra analyzed, and molecular orientations proposed [252]. Owing to the large shift of the bands at 721 and 620 cm⁻¹, attributed, respectively, to the deformation and wagging modes of the carboxylate

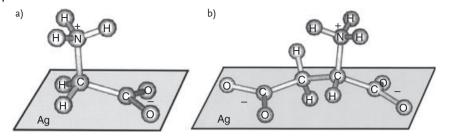


Figure 6.19 Proposed orientation of (a) glycine and (b) aspartic acid on a roughened Ag surface. Reprinted from Stewart, S. and Fredericks, P.M. (1999) *Spectrochimica Acta A*, **55**, 1641. Copyright 1999, with permission from Elsevier.

group, in the SERS spectrum as compared to the normal Raman spectrum the authors concluded that glycine adsorbs through the carboxylate group (Figure 6.19) which exists in its ionized form with the amino group pointing away from the surface. Suh and Moskovits have also studied the SERS of glycine, although with Ag colloids [41]. There is little similarity between their the spectra and those obtained by Stewarts and Fredericks [252]. The mode of adsorption is also different in the case of colloids, with the authors concluding that both the amino and carboxylic groups are attached to the surface since strong peaks for the C–N and carboxyl (–COO⁻) vibrations at 1032 and 1384cm⁻¹ were observed. It is pertinent to point out that the same vibration is assigned to 1413cm⁻¹ in the study of Fredericks and Stewart [252]. This highlights the fact that SERS spectra depend strongly on the adsorption mode and hence on the nature of the surface, pH, and surface preparation, so that comparison between spectra and between experiments needs to be done carefully, not only when comparing results between nanostructured surfaces and colloids but also between different colloid preparations.

Nevertheless, we will try to briefly summarize the key features of SERS for amino acids as observed on Ag electrode surfaces primarily based on the study by Fredericks and Stewart [252] but referring to other authors where relevant. The strongest bands are obtained from amino acids with aromatic side chains such as Phe, Tyr, and Trp, each of which is adsorbed through the carboxyl group with the aromatic ring perpendicular to the surface. The carboxy bands shift in SERS on adsorption with respect to solution Raman, and the absence of C=O peaks above 1650 cm⁻¹ clearly suggests that it is in its ionized form. The acidic amino acids, aspartic and glutamic acids, are structurally quite similar yet their spectra are distinguishable. Carboxylic acid peaks are more intense than for the others because they each have two carboxyl groups and both attach to the Ag surface (Figure 6.19). Similarly although glutamine and asparginine are quite similar, their spectra allow differentiation. Broadly the amide C=O bond is seen along with two types of NH₂ vibration which are shifted compared to the solution Raman spectra. The amide deprotonated form appears at 1100 and 1116 cm⁻¹ and shifts more with respect to normal Raman, indicating chemical interaction of the amide bond, through the nitrogen lone pair, with the surface. The amino acids with aliphatic side chains, namely glycine, alanine, leucine, valine, isoleucine, and proline, can be distinguished based on their aliphatic C–C vibrations in the region around 850 to 950 cm⁻¹. As expected, with increasing length of the aliphatic side chain the amine terminus moves farther away from the surface and therefore the relative peak intensity for it decreases, except for proline. The alcohol-containing amino acids show no significant difference as compared to their aliphatic counterparts and show no evidence for interaction of the hydroxyl group with the Ag surface.

With sulfur-containing side chains, as in cysteine and methionine, the Ag–S interaction dominates. Cysteine adsorption on Ag has been studied separately and its potential dependence observed by Watanabe and Maeda [253]. While the adsorption of cysteine could be followed by observing the disappearance of the S–H band at 2576 cm⁻¹, the shift (650 to 670 cm⁻¹) and broadening of the C–S stretching peak indicated the formation of a strong Ag–S bond. Furthermore, although the adsorbed cysteine could not be oxidized to disulfide, cystine (the dimeric form of cysteine in which the two cysteines are linked by a disulfide bond) adsorption could be reversibly cycled between the disulfide and monothiolate forms by varying the potential between +0.3 and -0.3 V vs. Ag/AgCl (Figure 6.20).

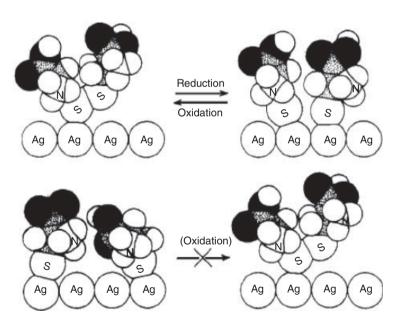


Figure 6.20 Surface adsorption layer model to account for experimental observations. The dotted, black, and white spheres represent carbon, oxygen, and hydrogen atoms, respectively. The atoms are scaled referring to the Ag–Ag distance (2.88Å) for the Ag(111)

plane together with known van der Waals radii and interatomic distances. Reprinted with permission from Watanabe, T. and Maeda, H. (1989) *Journal of Physical Chemistry*, **93**, 3258. Copyright 1989 American Chemical Society. Adsorption and oxidation of glycine, serine, and threonine were studied on roughened Pt and Au electrode surfaces using SERS [254]. In alkaline solutions (0.1 M NaOH) dissociative adsorption was found to take place with a peak appearing at 2040 cm⁻¹ on Pt which is stable over a wide range of potentials. It was assigned to a CN vibration. On Au this peak appears only at positive potentials, resulting from oxidation of the amino acids to give cyanide which is then adsorbed on the surface. The dissociation of glycine is easier and starts in the hydrogen potential region while for serine and threonine dissociation occurs along with the oxidation of CHOH groups at positive potentials.

Among the basic amino acids, SERS of histidine has been studied on a roughened Cu electrode [255] in water and D₂O solutions. Depending on the pH of the solution histidine exists in five different ionic forms Eq. (6.7), which can be distinguished by the shift in Raman/SERS peak positions as studied by Martusevicius *et al.* [255]:

$$H_4His^{2+} \xleftarrow{pK_a=1.8} H_3His^{+} \xleftarrow{pK_a=6.0} H_2His^{0} \xleftarrow{pK_a=9.1} HHis^{-} \xleftarrow{pK_a=14.0} His^{2-} \tag{6.7}$$

Correspondingly, different adsorption states of histidine were observed depending on pH and electrode potential [255]. In acidic solutions at pH = 1.2 the imidazole ring of the adsorbed histidine remains protonated and is not involved in chemical coordination with the surface. The anion in the solution, SO₄²⁻ or Cl⁻, also seems to play a role. At pH = 3.1 three different adsorption states of histidine are observed depending on the potential. It adsorbs with the protonated imidazole ring oriented perpendicular to the surface at potentials more positive than -0.2 V vs. NHE. At neutral pH histidine is adsorbed through the deprotonated nitrogen atom of the imidazole ring and the carboxyl group at potentials positive of -0.2 V vs. NHE. At more negative potentials the interaction is only through the amino group or the imidazole ring. In alkaline solutions at pH = 11.9 histidine is adsorbed on the Cu surface through the neutral imidazole ring. However, Stewart and Fredericks find that for histidine at neutral pH on an Ag surface the interaction is primarily through the carboxyl group [252]. The existence of chemically distinct states on Ag substrates has also been reported in the TERS study by Deckert-Gaudig and Deckert, although the zwitterionic state (His⁰) was the dominant one [197]. It is postulated to be adsorbed with its neutral imidazole ring and with the carboxyl moiety bound to the Ag surface with an upright geometry predicted for the ring.

Similar results have been obtained for dipeptides and tripeptides studied on Ag surfaces, with the adsorption primarily through the carboxyl terminus and the SERS spectra dominated by the residue nearest the surface [256]. With an increase in chain length the strength of the amide bands increased while that of carboxyl peaks decreased. Although good-quality and distinguishable spectra could be obtained, it is evident that SERS is not effective for determining the secondary structure of peptides and proteins as the spectra were mostly dominated by the amino acid side chains. In cysteine-containing dipeptides the spectral signature of the aromatic amino acids dominated the spectra as studied on Au nanoshell

SERS substrates [257]. The dipeptide data were used as the basis set for predicting and understanding the spectral characteristics of a cell-penetrating peptide, penetratin, which, although much longer (19 amino acids), still overwhelmingly displayed only aromatic signatures.

6.9.4

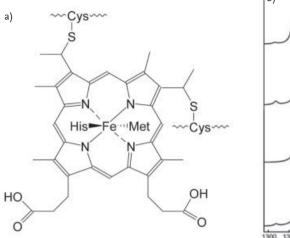
Proteins and Enzymes

Studying proteins and enzymes with SE(R)RS is not simple especially if information regarding the native state is required. This is because their secondary and tertiary structure can be easily distorted near an electrode surface. Thus significant care, in terms of the correct functionalization of the surface, is required if the native structure is to be preserved. However, the variation in the structure of a protein at different potentials at the electrode, that is, with changes in the structure of the double layer, is itself a question of great interest. For the purposes of our review we will distinguish between proteins in terms of their electroactivity. Thus we first discuss redox-active proteins and enzymes, which have been widely studied in bioelectrochemistry, and after that move on to nonelectroactive proteins and enzymes.

6.9.4.1 Redox Proteins

Cytochromes, and in particular cytochrome c (Cyt c), are among the best studied redox proteins by electrochemical SE(R)RS. They have metallocenters which shuttle electrons by reversible oxidation-reduction and thus carry electrons in photosynthetic and respiratory processes. In aerobic organisms Cyt c transfers electrons to the membrane-bound enzyme complex cytochrome c oxidase (CcO or COX) resulting in reduction of oxygen to water. In mitochondria, Cyt c transfers electrons between two inner membrane-bound proteins, cytochrome c reductase (CcR) and CcO by interaction at surface binding sites on these two redox partners [258]. Cyt c immobilized on an electrode surface ("bare" or functionalized) provides a model system wherein changes with electrode potential can be studied by SERS in an environment free from spectroscopic interference from other entities in the CcO complex. Nonresonant SERS studies on Cyt c such as that by Niaura et al. [259] are relatively rare. The redox transition, monitored by SERS at 633 nm on roughened Ag electrodes by following the band at 742 cm⁻¹, was found to be -0.11 V vs. SHE largely in agreement with the formal potential of -0.17 V vs. SHE. Application of potentials more negative than -0.7 V vs. SHE led to irreversible changes in the spectra and the appearance of amide III vibrational modes (corresponding to the protein backbone) suggesting large conformational changes and possibly unfolding. They also observed bands due to the aromatic residues of the protein in their spectra.

Cyt c has a heme-based redox center (Figure 6.21a) which plays the key role in the electron transfer reaction. The heme moiety (based on the porphyrin structure) has electronic absorptions around 400 nm (called the "Soret" band) and a weaker band around 530 nm (called the "Q" band). Thus experiments can be carried out



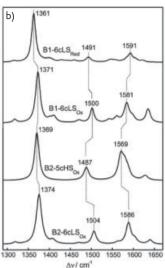


Figure 6.21 (a) Structure of heme center in cytochromes. (b) Resonant Raman component spectra of native (B1) and non-native (B2) forms Cyt c. From Murgida, D.H. and

Hildebrandt, P. (2008) *Chemical Society Reviews*, **37**, 937. Reproduced by permission of the Royal Society of Chemistry.

under different excitation conditions resulting in selective, resonant enhancement of different signature vibration (symmetric or nontotally symmetric) modes of the heme moiety [260]. The peak positions also give information on the spin and oxidation states of the heme unit [261]. The marker band region from 1200 to 1700 cm⁻¹ has typical peak positions which can be used to characterize the metallocenter-protein complex. A complete assignment of the resonance Raman peaks in Cyt c under various conditions has been published [262]. In particular ν_4 , corresponding to C-N stretching vibration, is extremely sensitive to the redox state because the Fe center interacts strongly with the porphyrin core [24]. Furthermore, Cyt c has been shown to exist in its native hexa-coordinated low-spin or 6cLS (termed B1) state, where the central Fe-porphyrin is stabilized by His 18 and Met 80 axial ligands, as well as in a non-native state (termed B2), which lacks the Met 80 ligand, by in situ SERRS study on Ag electrodes [263-265]. Owing to bands in the marker band region which are sensitive to the spin state of the heme (see Figure 6.21b) complete characterization the state of Cyt c as well as its evolution as a function of redox potential is possible by SERRS.

SERRS studies on electrodes can not only probe at a very sensitive level but can also selectively probe the heme group giving electronic and conformation information, without interference from rest of the protein. In one of the first of such studies, Cotton *et al.* used roughened Ag electrodes to record the potential-dependent SERRS of Cyt c and myoglobin [266]. They were able to observe the

shift in the marker band peaks at two different potentials, -0.2 V and -0.6 V vs. SCE, corresponding to a ferric low-spin and ferrous low-spin state, respectively. It was subsequently shown by Hilderbrandt and Stockburger that direct adsorption onto Ag induces some change in the spin state of the Cyt c as monitored by SERRS, indicating structural modification as compared to the native state [267]. With SERRS monitoring of the heme moiety of Cyt c at the electrode surface it was also shown that at low temperature the low-spin state was dominant while at room temperature a mixture of high- and low-spin states occurred [267]. Although the spin transitions remained reversible with temperature, indicating no denaturation at the electrode surface, electronic interaction did exist between the surface and the heme, possibly through the charged ligands, leading to a different crystal field splitting energy compared to the native state. By comparing the relative intensities the authors also concluded that the orientation of the heme-porphyrin ought to be perpendicular to the surface rather than parallel as found for free porphyrin molecules. The non-native conformational states of Cyt c have also been spectroscopically characterized by SERRS [265]. The non-native state B2 can itself consists of different sub-states depending on whether the vacant site is occupied by a water molecule or by a histidine ligand (His-33 or His-26) contributed by another peptide segment [265, 268]. Thus studying Cyt c by surface-enhanced Raman techniques on electrode surfaces is very attractive as it can give information regarding conformation and redox potential [269, 270]. It has been proposed that not only are redox transitions related to conformational changes in the protein but also that they may be even controlled by them [271, 272]. In this respect it can be said that the redox-controlled (and reversible) formation of the B2 state at the electrode might provide insight into deciphering the mechanism of electron transfer mediated by changes in conformation. For example, based on the observed spin state a model for the biochemical mechanism of interprotein electron transfer by Cyt c between the CcR and CcO was initially proposed [267] and subsequently refined [24]. A schematic of the electron transfer reaction between Cyt c and CcO is shown in Figure 6.22.

Electrochemical studies in combination with SE(R)RS can afford dynamic information regarding conformational change and coupled electron transfer processes. In this context the technique of time-resolved SERRS as developed by Hildebrandt and coworkers to obtain insight into the electron transfer kinetics of heme proteins [273–275] should be mentioned. The technique consists of a rapid potential step initiating the process coupled with synchronous monitoring of the SE(R)RS spectra after a specified delay time (δ , the probe interval). To improve signal-tonoise ratio the process has to be repeated several times using the same potential step; the transient behavior is built up by repeating the process for different probe times. Analysis of such kinetic data affords the heterogeneous electron transfer constant $(k_{\rm FT})$. However, the need to repeat the potential steps limits the application of the technique to systems that are stable and chemically completely reversible. Subsequent improvements in the technique, including a better cell design with a rotating disc electrode to reduce photodegradation [276], and in the measurement technique by using two-color time-resolved SERS, have been

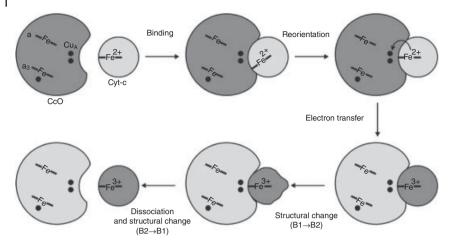


Figure 6.22 Schematic of the electron transfer reaction between Cyt c and CcO. From Murgida, D.H. and Hildebrandt, P. (2008) *Chemical Society Reviews*, **37**, 937. Reproduced by permission of the Royal Society of Chemistry.

implemented so that real-time kinetic information is generated [277] providing new insights into protein electron transfer dynamics. This technique of time-resolved SERRS has been successfully applied to study Cyt c as well as to a variety other heme proteins immobilized at electrodes [278–282].

For proteins, it is clear that the surface functionalization has a critical affect on the SERRS intensities observed due to the difference in orientation that it can induce and due to the affect on the electrochemistry. Self-assembled monolayers (SAMs) of thiols can provide a twofold advantage by (i) preventing denaturation and (ii) defining the orientation by either electrostatic or covalent binding. This particular aspect has been utilized elegantly by Dick et al. [169] to use SERRS to study heterogeneous electron transfer. By varying the chain length of carboxylterminated alkanethiols on AgFON substrates (see Section 6.7) they showed that the electrochemistry, as followed by SERRS, was most reversible for mercaptohexanoic acid and became completely irreversible for mercaptoundecanoic acid. In the latter case the SERRS signals were 10 times weaker. A similar SERRS study by Murgida and coworkers has shown that, by controlling the chain length of carboxyl-terminated thiols on the surface, the formation of a non-native state of Cyt c can be controlled. Apparently the non-native redox state can be varied from 76% to 0% on going from C2 to C11 SAMs [264] clearly suggesting the role of the electric field in determining the conformation of the redox center of the protein. Furthermore, close to the surface (for C2 and C3 SAMs) the interfacial redox process is found to be coupled to proton transfer as studied by monitoring the kinetic isotope effect with SERRS [263]. Obviously this proton coupling is also distance dependent and decreases with increasing distance from the surface. The study clearly demonstrates that the redox processes of proteins at electrode surfaces, and presumably other biomolecules, may not be entirely controlled by the electron transfer step, warranting careful interpretation of the data. Indeed, improved electrochemical performance, as demonstrated by cyclic voltammetry and SERS [283], can be achieved by the correct functionalization of the electrode such as by using 1,4-dithiane. Nevertheless the utility of these thiol-modified surfaces in understanding the functional aspects of protein complexes was established in the study by Dick et al. where their SERS-based titration (binding) curves gave association constants similar to that for the cytochrome c peroxidase-Cyt c complex [169]. Again stressing the role of functionalization in studying electron transfer rates, Grosserueschkamp et al. [276] used their time-resolved SERRS technique to study the electron transfer rates for Cyt c at mercaptoethanol-modified Ag electrodes. They obtained oxidative and reductive electron transfer rates of 46 and $84 \, \mathrm{s}^{-1}$, respectively, at least an order of magnitude lower than that found on Au electrodes. Recently Sezer et al. showed that the SERRS signals for Cyt c can be improved by using covalent attachment to Ag-(C₁₁-NH₂)-Au. In this case they obtained signals 15 times larger than those for Ag alone when using their multilayer structures [284]; however, the electron transfer deviated from Nernstian behavior upon covalent attachment while for electrostatically bound Cyt c fast long-distance electron transfer rates (49 s⁻¹) were observed.

One of the key advantages of SE(R)RS is the orientation information that it can provide as a function of the electrode potential. This can be crucial in understanding the functioning of redox proteins. As discussed above, Murgida and coworkers, among others, have studied the interaction of Cyt c on bare or SAM-coated electrodes. Studies of the conformational changes on interaction with other biomolecules coated on electrodes are an interesting prospect. A SERRS study by Jiang et al., in conjunction with circular dichroism measurements, indicated that interaction with DNA does not affect the secondary structure of Cyt c, although the heme microenvironment was disturbed at the electrode-electrolyte interface [285, 286]. Cyt c adsorbed at a DNA-modified metal electrode showed voltage-dependent orientation resulting in non-Nernstian behavior at high electrode potentials.

Among other classes of redox proteins [258] those possessing Fe₂-S₂ clusters and blue Cu metallocenters have been explored by SERS on electrodes. Putidaredoxin (Pdx) is a 12kD protein with an Fe2-S2 redox center that performs two stepwise one-electron transfers from Pdx reductase to the terminal cytochrome mono-oxygenase in Pseudomonas putida PpG786. Reipa et al. studied the reductionoxidation of Pdx on bare roughened Ag electrodes and obtained spectroscopic evidence for quasi-reversible charge transfer between -0.6 and -0.3 V vs. Ag/AgCl [287]. Azurin, another nonheme protein [258], has also been studied on an Au electrode by a combination of electrochemistry and SERRS. In that work it was demonstrated that the Cu site of the immobilized protein remained intact upon adsorption and underwent reversible reduction-oxidation [288].

6.9.4.2 Other Proteins

We now turn our attention to nonredox proteins. SERS spectra of IgG adsorbed on roughened Ag electrodes were obtained by Grabbe and Buck [289]. The spectra

contained bands attributed to the amino acid residues that were in close contact with the surface but interestingly did not include any amide backbone vibrations or solution bands. Usually these amide bonds, the bonds making up the protein backbone, produce the strongest bands in the Raman spectra of proteins, often indicating denaturation. Nevertheless, the vibrational frequencies depend on the secondary structure and change in position and intensity depending on whether they are part of a β -pleated sheet or α -helix. Again only amino acid side residues were observed in SERS indicative of the mode of adsorption on the Ag electrode. Adsorption was greatest at the PZC and at lower ionic strengths. The authors used the variation in band width and the relative intensity of the peaks to describe the molecular rearrangements.

Spectra of lysozyme on an Ag electrode have been recorded. The authors did not see peaks corresponding to the amide I, amide III, and S-S bands [238, 290] although Hu et al. were able to observe these on colloids [291]. Thus although chemisorption seems to be the key to what may be observed from the spectra, the orientation of the adsorbed protein and its side chains can be inferred from analysis of the spectra. Similarly the SERS spectrum of bovine insulin has been observed with high enhancement factors by Chumanov et al. [290]. A further, detailed, electrochemical SERS study for bovine insulin was carried out by Reipa et al. on an Ag electrode over the potential range of -0.2 to -1.2 V vs. Ag/AgCl [292]. Conformational changes with potential can be seen and the rupture of the disulfide bonds takes place sequentially with increasing negative potentials leading to a decrease in the α-helix conformation. Recently Drachev et al. have reported a protein sensor based on nanostructured adaptive Ag films using SERS detection [293]. The sensor can differentiate at the sub-monolayer level between two insulin isomers, human insulin and its analog insulin lispro, that differ by only two neighboring amino acids.

6.9.4.3 **Enzymes**

SE(R)RS spectra of enzymes are essentially similar to those of proteins except in those cases where they possess a distinct site that is chromophoric. In the latter case heme-based enzymes have again been predominantly studied by SERRS. Enzymes can have multiple redox centers (or multiple chromophores) or are often intimately associated with a cofactor; careful interpretation of the spectra is required to take account of these effects. If the cofactor is chromophoric then it can turn out to be an advantage as SERRS monitoring can be used. For example, Holt and Cotton used the SERRS from associated flavin adenine dinucleotide (FAD) to study the catalytic activity of glucose oxidase (GOx) studied by SERRS and combined electrochemistry [294]. Although successful as a strategy, they observed unusual potential shifts which could be attributed to the formation of a flavin-Ag⁺ complex. The results caution against the use of Ag surfaces in solution SERS, as H₂O₂, an enzymatic reaction product of GOx, oxidized the Ag ultimately leading to flavin–Ag⁺ complex formation. In another study the same authors found that free flavin was an interferent when recording spectra from commercially available GOx [295]. Potential-dependent SERS spectra of alkaline phosphatase, horseradish peroxidase, and lactoperoxidase were obtained by Razumas and coworkers on Ag electrodes [296, 297]. They also found that alkaline phosphatase and lactoperoxidase form a complex with Ag ions. Nevertheless, they were able to interpret the conformation and orientation changes as a result of changes in potential on the basis of the changes in the amino acid signatures. In another study Grabbe and Buck observed the SERS spectrum of an anti-human immunoglobulin G alkaline phosphatase conjugate (anti-IgG) adsorbed on an Ag electrode as a function of potential and the ionic strength of the buffer [298]. SERS spectra were recorded at -0.5 and -0.8 V vs. SCE. Upon reduction at -0.5 V most of the peaks lost intensity. In particular, the peaks corresponding to tryptophan and cysteine residues including those corresponding to C-S (675 cm⁻¹), S-S (511 cm⁻¹), and Ag-S (389 cm⁻¹) were affected.

Microperoxidase is the heme-containing peptide portion of Cyt c that retains peroxidase activity. Several microperoxidases are available with different numbers of amino acid residues. The conformation of microperoxidase-11 (the microperoxidase with 11 amino acids in the peptide) adsorbed on roughened Ag electrodes was studied using Fourier transform SERS and shown to be adsorbed via the α helical polypeptide chain [299]. As expected the characteristic amide I and III bands for the protein backbone were the strongest. Similarly microperoxidase-8 was studied by combined SERRS and electrochemistry where it was shown that the heme existed in the penta-coordinated state and could bind cyanide as the sixth ligand [300].

Iron tetraphenylporphyrin was immobilized on an electrode coated with poly(γethyl-I-glutamate) functionalized with imidazole pendant arms [301]. Striking similarities with the model Cyt c3 system, which is a tetraheme protein, were observed. To mimic and understand biological function proper immobilization is necessary without disrupting the active site. CcO suspended in a lipid membrane was attached through a histidine-tag linker to an electrode and studied by SERRS [302]. It was demonstrated that the structures of the active sites remained intact in the enzyme and that electron transfer from the electrode could take place. Recently, biomimetic immobilization of CcO on a metal electrode has been carried out and studied by a combination of fast-scan cyclic voltammetry and SERRS [303]. This allowed the kinetic analysis of all four redox centers in the enzyme providing insights in to how the proton translocation might be coupled to electron transfer in such complex systems. In another study where CcO, embedded in a phospholipid bilayer, was tethered to an Ag electrode via a histidine tag it was found that the intramolecular electron transfer and proton translocation were perturbed due to the effect of the interfacial electric field although the heme active site structures were preserved [304].

The importance of the functional layer on an electrode to the interaction of an enzyme with the surface is brought out by Kudelski's studies with Cu-containing tyrosinase (a phenol oxidase) [305] and laccase [306]. Using ω -functionalized thiols he showed that electron transfer was not prevented between the electrode and the enzyme. That the local environment of the enzyme is important has also been demonstrated in a recent study of human sulfite oxidase using SERRS and cyclic

voltammetry [307]. The native heme structure of the cytochrome b5 domain and the functionality of the enzyme were preserved on the electrode surface. It was further found that the heterogeneous electron transfer rate was directly affected by the ionic strength of the buffer.

6.9.5 Membranes, Lipids, and Fatty Acids

Studies of membranes, lipids (including phospholipids), and other such amphiphilic biomolecules on electrodes are rather limited due to the redox inactivity of these materials. Nevertheless, a few reports exist of studies of bilayer membranes and lipid amphiphiles on SERS surfaces and are mentioned here as exemplar studies due to the unique insight they offer.

SERS of adsorbed dipalmitoylphosphatidylcholine (DPPC) lipids was studied on an Ag electrode [308]. The potential-dependent behavior showed that the ratio of trans and gauche structures of the acyl chain, corresponding to bands at 1105 and 1142 cm⁻¹, respectively, changed around -1.05 V vs. SCE. However, in a later paper Guo et al. suggest that the adsorption and the potential-dependent order-disorder phase transition involves the phosphocholine head group [309]. Hybrid bilayer membranes wherein the layer in contact with the surface is formed by a thiolipid or a long-chain alkanethiol are robust and useful biomimetic model interfaces [310, 311]. Such tethered bilayer membranes on Au display good blocking properties, are well oriented, and have "fluid-like" properties [311]. Leverette and Dluhy characterized hybrid bilayer films composed of a long-chain alkanethiol, 1-dodecanethiol, and the phospholipid DPPC on Ag island films [312]. The DPPC layer was deposited by the Langmuir-Blodgett technique over the dodecanethiolfunctionalized SERS surface. While spectral features from the dodecanethiol were strong and readily identified, the changes observed in the spectrum after addition of DPPC were quite subtle. The ratio of the antisymmetric to symmetric methylene stretches (also called the order parameter) reflects both intermolecular lateral chain interactions as well as intramolecular (i.e., conformational) chain order, with a decrease indicative of intramolecular disordering of the monolayer [313]. By employing deuterated DPPC the C-H stretches were shifted to around 2100 cm⁻¹ from 2900 cm⁻¹ allowing insight into the order of the transferred Langmuir-Blodgett layer along with the immobilizing layer of dodecanethiol [312]. The order parameter, based on C-D peaks and the position of the symmetric stretch of deuterated methylene, indicated the as-transferred gel-like state of the film.

As noted above lipid bilayer films supported on electrodes represent an interesting bioelectrochemical interface where the potentials are of the same order as those of physiological systems, and can be readily modulated. Nevertheless, this area remains less studied except in the case of enzyme complexes such CcO where the hydrophobic environment of the lipids is necessary to maintain the integrity of the enzyme when immobilizing it on electrodes [304, 314]. Studies of the interaction of drugs with lipid membranes are important from many points of view, particularly considering the fact that nearly 50% of drug molecules have mem-

brane proteins as their targets. Among the first such reports, the interaction of various anthracyclines was studied with asymmetrical planar supported bilayers on Ag surfaces [315]. Different anthracyclines showed different interaction behavior depending on the hydrophobic-hydrophilic balance in the lipid bilayers. Halas and coworkers have recently examined the intercalation of the drug ibuprofen with lipid bilayers on nanoshells [316]. Using SERS measurements at two different pH values and various concentrations the authors are able to conclude that the drug molecule is intercalated into the lipid layer using primarily the hydrophobic effect and that the membrane is disrupted at higher concentrations. Studies such as this would enormously benefit if carried out on continuous conducting SERS surfaces and lead to more insight besides being able to corroborate conclusions such as membrane disruption. Recently Millone et al. used dimyristoylphosphatidylcholinesupported bilayers on dithiothreitol-treated roughened Au electrodes for an in situ electrochemical SERS study of the incorporation of methylene blue and FAD into the lipid layer [317]. Spectroscopically the lipids do not show any significant signal in the 1200-1700 cm⁻¹ region while the characteristic bands for methylene blue and FAD are clearly visible. Based on their results the authors conclude that methylene blue is able to penetrate the bilayer easily reaching the Au interface due to its lipophilic positively charged nature, while FAD, due to its negative charge, remains in the outer part of the bilayer and therefore cannot be electrochemically detected.

6.9.6 Metabolites and Other Small Molecules

Neurotransmitters

Anodic voltammetry has conventionally been applied for detection of neurotransmitters and is most commonly the final step at the end of a chromatographic elution even for ex vivo analysis. In general these electrophysiological techniques are limited to easily oxidizable molecules and are not free of interferences such as ascorbate. Hence, SERS presents a unique tool for their interference-free detection at a very sensitive level. Furthermore, although colloidal nanoparticles might provide a better alternative for in cellulo work [318], SERS at electrodes provides unique insight into adsorption behavior and can be ideal for in vitro detection methodologies serving as a means to enhance signals and increase discrimination between different neurotransmitters. Among the earliest work in this area is that by Morris and coworkers who reported the detection of various catecholamines on Ag electrodes [319]. Distinguishable spectra with good signal-to-noise ratio were obtained for dopamine, norepinephrine, 3-methoxytyramine, and epinine in phosphate buffer of pH = 7.2. The intensities were maximized when the electrodes were polarized around -0.9 V vs. SCE or below. The PZC of Ag under these conditions is -0.9 V, which clearly indicates that these neurotransmitters do not adsorb well on positively charged surfaces. Furthermore under these conditions no interference from ascorbate, glutathione, or acetylcholine was observed. The lowest detection limit for dopamine was found to be 0.3 µM, which is much lower than

the physiological limit of 1 to $100\,\mu\text{M}$, with seemingly modest detectors and moderate laser powers. A polymer-coated electrode probe has been developed to show the application of neurochemical measurements with SERS in protein-rich matrices without fouling [320].

Another important neurotransmitter is histamine. It is also involved in several biochemical functions such as regulating physiological function in the gut. Its detection by conventional electrochemical means is difficult as it is not readily reducible and its oxidation coincides with that of water. SERS of histamine adsorbed on Ag electrodes was observed by Morris and coworkers who found potential-dependent intensities [321]. Due to the imidazole moiety some of the peaks are pH dependent due to the ionizable nitrogens. Histamine is adsorbed in its neutral form at all pH values at negative potentials. At potentials more positive than $-0.5\,\mathrm{V}$ vs. Ag/AgCl, the cation is adsorbed in acidic media. The maximum intensities were obtained at $-0.8\,\mathrm{V}$ vs. Ag/AgCl. In addition to orientation changes brought about by the change in electrochemical potential the adsorption is also clearly affected. Nevertheless, the analytical utility of SERS detection was aptly demonstrated by showing the ability to differentiate between histamine and its primary metabolite, methyl histamine.

Much more work has been undertaken with colloidal nanoparticles over the past decades for developing various protocols for detecting neurochemicals and their metabolites. Recently it has even been shown that this SERS strategy can be used to image dopamine and norepinephrine release in cells [318] demonstrating the potential of the technique. However, issues with fast detection for live cell work and selectivity still remain.

6.9.6.2 Nicotinamide Adenine Dinucleotide

Nicotinamide adenine dinucleotide (NAD+) is an important co-enzyme which plays a significant role in many biological processes [258]. The molecular structure of NAD⁺ is shown in Figure 6.23. Studies on NAD⁺ with SERS-active Ag and Au electrodes have been carried out primarily to understand its absorption behavior and orientation. Involvement of adenine in adsorption was clear from initial experiments on electrodes and similarly the vertical orientation of the adenine moiety could be inferred [322]. Some differences were found in the SERS spectra of NAD+ between Ag and Au electrodes. While adsorption on Ag was primarily through the exocyclic amino group and N₇ of the adenine, on Au the orientation changed from N₇ to N₁ at -0.25 V vs. SCE. It was further found by Xiao and Marwell that Fourier transfer SERS in the near-infrared region gave better signals on an Au electrode and significant modulation in intensity of various peaks with potential was observed [323]. However, the structural orientation and assignment of peaks for the nicotinamide moiety could only be determined after careful experimentation. The potential dependence of various peaks has also helped in assignment of the peaks and disentanglement of the nicotinamide bands from those of adenine. Only after careful experiments were the same authors able to correctly assign the ring vibration of nicotinamide at $1575\,\mathrm{cm}^{-1}$ which appeared at $-0.4\,\mathrm{V}$ vs. SCE [324]. Furthermore, some variations in peak positions and observed

Figure 6.23 The structure of NAD+.

spectra are apparent between different reports. For example, a $1340\,\mathrm{cm^{-1}}$ band assigned to the N_7 – C_5 bond in the adenine moiety in NAD⁺ is a strong feature in the spectra reported by Xiao *et al.* [324] while it is not observed in the study by Yang *et al.* [325]. This not only highlights the fact that reproducible SERS surfaces are essential for bioanalysis but also that careful consideration needs to be given to electrochemical potential, pH, and surface characteristics (adsorbed ions). Additionally, it is found that the spectra from NAD⁺ changes with adsorption time as a full monolayer is formed [325]. The band at $735\,\mathrm{cm^{-1}}$, assigned to in-plane NH₂ bending and a ring-breathing mode of adenine, appears only after 8 h, also suggesting that the adenine moiety might be changing its orientation from flat to vertical. Reorientation takes place from a flat adsorption mode for both the adenine and the nicotinamide moieties such that the final organized monolayer has the adenine vertically oriented while the ring and the carboxamide lie flat on the chemically roughened Ag surface.

6.9.6.3 Flavin Adenine Dinucleotide

Flavins are an important cofactor involved in many biological electron transfer processes [258]. They commonly occur as FAD, flavin mononucleotide, and riboflavin. For example, FAD is the cofactor in GOx. Cotton and Holt could even detect it at a very sensitive level of 10^{-10} M on a roughened Ag electrode by SERRS although in their study on GOx it was an interference [295]. Flavins can undergo oxidation–reduction in either a single two-electron transfer step or two

one-electron transfer steps. The preference towards either of these mechanisms depends on the interaction with metal ions, pH, and the polarity of the matrix; SERS measurement made under electrochemical control can help to determine whether an intermediate state is involved. An unstable semiquinone state was detected on reduction of flavin in acidic media signifying that two one-electron transfers are involved in the overall two-electron reduction via this radical intermediate [326]. This was confirmed in the study by Abdelsalam $et\ al.$ where they covalently attached the flavin moiety (isoalloxazine ring) to an Ag SSV surface [183]. The $in\ situ$ electrochemical SERS spectra recorded at different potentials are shown in Figure 6.24. Additionally by quantitative analysis of their SERS data, based on a modified Nernst equation, they were able to estimate a redox potential, E', of $-0.354\,\mathrm{V}$ vs. SCE and an interaction parameter, E', of E', of E', of E', of indicates repulsive interactions between the immobilized molecules.

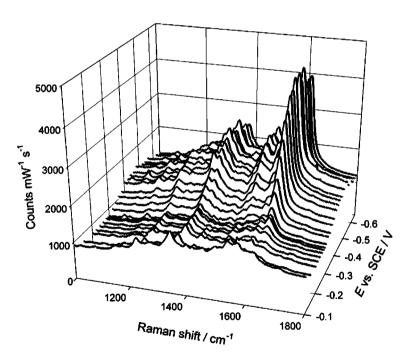


Figure 6.24 In situ electrochemical SERS spectra for flavin immobilized on an SSV Ag film produced using template spheres of 900 nm diameter and 540 nm thickness with a He–Ne laser (633 nm, single 10 s accumulation, 3 mW laser power, recorded in Tris buffer, pH = 7). The sample was purged

with nitrogen for 20 min at the beginning of the experiment and then for 1 min between each measurement. Reprinted with permission from Abdelsalam, M., Bartlett, P.N., Russell, A.E. et al. (2008) Langmuir, 24, 7018. Copyright 2008 American Chemical Society.

6.9.6.4 Bilirubin

Bilirubin is a lipid-soluble metabolite of hemoglobin that is transported through the bloodstream as an albumin complex to the liver where it is esterified for excretion in urine. It plays an important role in the pathology of many diseases. Its Raman spectrum has been difficult to observe because of its sensitivity to photoisomerization and decomposition. However, under conditions of SERS on Ag electrodes this is suppressed allowing potential-dependent spectra to be observed in its free form as well as complexed with cyclodextrins and albumin [327]. The band intensities are higher at more positive potentials implicating adsorption of anionic species on the electrode.

6.9.6.5 Glucose

Glucose is an important metabolite. However, its small Raman cross section makes it a difficult species to study by SERS. Approaches with colloidal nanoparticles, utilizing chemistry on the surface and solution to generate species with higher Raman cross sections such as using horseradish peroxidase and GOx coimmobilized on nanoparticles with o-phenylenediamine in solution to generate azoaniline on reaction with glucose, have been reported [328]. Detection of azoaniline, chosen because of its relatively high SERS cross section, enabled a dynamic range of 0.5 to 32 mM which is larger than the physiological range of 3.5 to 6.1 mM. Further work on glucose sensing has been carried out by Yonzon et al. using AgFON surfaces [166] using pre-concentration of the glucose molecules in alkanethiol monolayers to increase the sensitivity. Several improvements in sensor performance, such as increased temporal stability, could be achieved by switching to AuFON surfaces [165]. Recently Lipkowski and coworkers studied electrochemical SERS of β-thioglucose on Ag SSV surfaces (see Section 6.7) and compared them to electrochemically roughened electrodes [189]; the results for the SSV surfaces were found to be much more reproducible. Ab initio calculations were used to predict and carry out the assignments of the observed spectra. The results of this study showed that, under the experimental conditions, the β-thioglucose film anomerized into a mixture consisting of both α - and β -thioglucose anomers.

6.10 Conclusion

In this chapter we have provided an overview of the SE(R)RS technique and reviewed the literature in the context of bioelectrochemical studies. It has been impossible, due to the enormous amount of SERS literature, to be exhaustive and all-inclusive. Rather, we have tried to give an up-to-date account focusing on developments in the field of SERS of biomolecules and including progress made towards understanding the basic phenomena. The bulk of the historical literature covers roughened electrode surfaces but there is now an increasing move towards reproducible SERS surfaces. We believe that SERS is at an exciting stage where

the ability to fabricate reproducible, stable SERS-active surfaces, and the growing understanding of the SERS phenomenon on these surfaces, will make possible much more quantitative measurements. These studies will not only help in advancing the understanding of interfacial behavior of biomolecules, which itself is important for biosensor and biofuel development, but will also, we believe, provide insights into the biological processes themselves. We hope that we have been successful in bringing out the advantages and novel insight that can be obtained by combining SE(R)RS with bioelectrochemistry.

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7

Membrane Electroporation in High Electric Fields

Rumiana Dimova

7.1 Introduction

The autonomy of a cell, the basic building unit of most living creatures, is ensured by a bounding membrane. The scaffold of this membrane is made of a double lipid layer, which is basically impermeable to all substances in the cellular environment except for water. The mechanical and rheological properties of the bilayer define the response of the membrane to external perturbations. The membrane "intolerance" towards letting solute molecules easily cross it creates the main obstacle in biomedical applications where drugs or genes have to be introduced into cells. One approach finding broad use nowadays in overcoming the barrier functions of membranes relies on the temporary bilayer perforation when exposed to strong electric fields. This phenomenon, called electroporation, is the main focus of this chapter.

The main topics to be covered build upon our knowledge of the mechanical and rheological properties of membranes and their response to perturbations. In the remaining parts of this introductory section, some of these properties are briefly described to set the basis for a discussion of the behavior and response of membranes to external forces. The following sections consider in detail the morphological changes and poration electric fields can induce in vesicles made of membranes in different phases, and the effects of media environment and various molecular inclusions in the lipid bilayer, that is, the specific membrane composition. Finally, some application aspects of the work are discussed.

7.1.1

Giant Vesicles as Model Membrane Systems

The field of membrane structure and characterization is attracting the attention of a growing number of researchers. The basic research in this area builds upon studies performed on the simplest and minimal systems mimicking cell membranes, namely model membranes. Examples of such model membranes are

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lipid monolayers at the air—water interface, solid-supported bilayers, black lipid membranes, vesicles, and bilayer stacks. Among them, vesicles or liposomes are membrane "bubbles" formed by bending and closing up of a lipid bilayer. They are the most natural systems, because, in terms of shape and structure, they are closest to membranes of cells and cell organelles.

Various experimental techniques have been developed for preparing liposomes of different sizes (from nanometers to tens of micrometers) [1]. The largest ones, several tens of micrometers in size, are called "giant vesicles" [2] and are an extraordinarily convenient system for studying membrane behavior [3, 4]. They are well visible under an optical microscope using various enhancing techniques like phase contrast, differential interference contrast, or confocal and conventional fluorescence microscopy, the latter two being particularly useful in distinguishing domains on membranes (Figure 7.1). Thus, giant vesicles allow for direct manipulation and observation of membrane interactions and responses to external perturbations. On the contrary, working with conventional vesicles (a few hundreds of nanometers) usually involves the application of indirect methods and techniques for observation. In addition, their small sizes often raise questions about effects due to high membrane curvature when molecular interactions are considered. In contrast, giant vesicles, which have sizes in the micrometer range (i.e., comparable to the sizes of cells), and therefore have nearly zero membrane curvature, reflect the properties and behavior of cell plasma membranes.

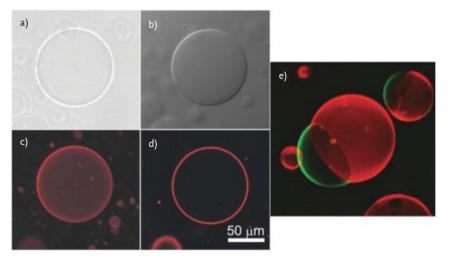


Figure 7.1 Snapshots of the same giant vesicles observed under different microscopy modes: (a) phase contrast; (b) differential interference contrast; (c) projection averaged confocal microscopy; (d) equatorial section confocal microscopy. Adapted from [3] by

permission of IOP Publishing Ltd.
(e) Confocal three-dimensional projection image of vesicles with immiscible fluid domains visualized with fluorescent dyes, which preferentially partition in one or the other lipid phase.

The two most popular techniques for the formation of giant vesicles (other available methods are summarized in [4]) are spontaneous swelling or gentle hydration, introduced by Reeves and Dowben [5] and electroswelling, introduced by Angelova and Dimitrov [6]; for a brief description of these two preparation protocols, see [3]. These two methods were further developed and improved by several groups (e.g., [4, 7-10]). Interestingly enough, the underlying mechanism of the electroformation protocol, which is based on exposing lipid layers to alternating electric fields, is still poorly understood even though widely used. Both protocols yield giant vesicles with sizes in the range of a few tens of micrometers.

Giant unilamellar vesicles (GUVs) are increasingly employed for quantitative characterization of the physicochemical properties of lipid membranes with various compositions, but also to study membrane-related processes like cell adhesion, phase separation and domain formation, protein sorting in lipid rafts, endoand exocytosis, uptake of various molecules, and protein mobility, to mention just a few of the studied fields. Examples of using GUVs as simple model systems for unraveling certain physicochemical properties of biological membranes include lipid domain formation [11-13], mechanical and rheological properties of the entire vesicle [14] or of the membrane [3, 15, 16], lipid dynamics, membrane growth [17–20], membrane adhesion [21–25], wetting phenomena [26, 27], budding and fission [28-32], and membrane fusion [33-37]. Giant vesicles are also a very practical tool to study the response of membranes to external perturbations like hydrodynamic flows [38], locally applied forces [39], micromanipulation [40], and electric fields [41, 42]. This chapter focuses on the effects of strong electric pulses on model membranes as exhibited by the behavior of GUVs exposed to such pulses. The vesicle response will be interpreted in view of general concepts of membrane biophysics.

7.1.2 Mechanical and Rheological Properties of Lipid Bilayers

The physical properties of lipid bilayers are those that define their response to external perturbations. Knowing the mechanical and rheological characteristics of lipid membranes will prepare us to tackle problems related to stress induced in bilayers by electric fields and the phenomena that it triggers, for example, dynamics of vesicle and cell deformation, bilayer instability, electroporation, and electrofusion.

As a simple depiction of a lipid bilayer, one can consider it as a film or a slab, which may be curved, compressed or dilated, and sheared. At physiological temperatures most natural lipid membranes are fluid. Therefore, within this slab, the lipid molecules are free to move. Below the lipid phase transition temperature, single-component membranes crystallize. In this so called "gel" phase, the relative motion of lipids and membrane inclusions is principally hindered. The fluidity of the membrane and resistance to shear in the plane of the film are characterized by the shear viscosity, η_s (or the diffusion coefficient of the lipids). Typical values

of η_S lie in the range 1×10^{-9} – $5\times 10^{-9}\,\mathrm{N\,s\,m^{-1}}$ [16] for fluid membranes, but for gel-phase membranes divergence is observed [43]. One may equivalently define a viscosity η_D related to the dilation and compression of the membrane. The value of η_D is of the order of $3.5\times 10^{-7}\,\mathrm{N\,s\,m^{-1}}$ [44].

Phospholipid membranes in the fluid phase are very soft: the energy required for their bending is comparable to the thermal energy. The bilayer bending rigidity, κ , which characterizes how easy it is to curve the lipid bilayer, is typically of the order of $0.9 \times 10^{-19} \text{J}$ [45–47], which is equivalent to $20k_{\text{B}}T$, where k_{B} is the Boltzmann constant and T is the absolute temperature. Thus, fluid membranes fluctuate due to thermal noise. These fluctuations can be directly observed on tensionless giant vesicles under the microscope. They are the basis of the so-called fluctuation spectroscopy method used to measure membrane bending rigidity [48–55]. For gel-phase membranes, the bending rigidity increases significantly, and a few degrees below the main phase transition temperature it reaches values of the order of $(15–20) \times 10^{-19} \text{J}$ (about $350k_{\text{B}}T$) [43, 56, 57].

Weak tensions applied to a fluid membrane smooth out bilayer undulations. At high tensions the membrane can be stretched leading to a change in the area per lipid molecule. The stretching elasticity modulus, K_a , characterizing this response is of the order of that of a rubber sheet with the same thickness (about 4 nm). Typical values of K_a for fluid membranes lie in the range 200–300 mN m⁻¹ [47] and for gel-phase membranes can reach values of 850 mN m⁻¹ [58]. Upon stretching, a lipid bilayer can sustain tensions up to several mN m⁻¹. At a certain critical tension, also known as the lysis tension, σ_{lys} , the membrane ruptures. For fluid membranes, σ_{lys} is of the order of 5–10 mN m⁻¹ [59, 60]. Note that the membrane tensile strength depends on the tension loading rate [61]. Membranes in the gel phase can sustain higher tensions and rupture at higher values of σ_{lys} [62].

After rupture or poration, the rearrangement of the lipids to close the bilayer sheet is energetically favorable because in this way the hydrophobic tails of the lipid molecules are shielded from exposure to water. The energy penalty of closing a hole in the membrane is described by the edge tension, γ , which is of the order of several piconewtons [63]. The edge tension plays a strong role in processes of pore stability and resealing as in electroporation, which is discussed in more detail in Section 7.5.1.

7.2 Electrodeformation and Electroporation of Membranes in the Fluid Phase

Vesicles exposed to electric fields deform. The response of GUVs to electric fields has been the subject of extensive investigation. When exposed to AC electric fields, as a stationary state they attain ellipsoidal shapes-prolate or oblate-depending on the field frequency and media conductivity [42, 64]. Initiated by the seminal work of Winterhalter and Helfrich [65], this effect has been considered theoretically [66–74] and experimentally [42, 64, 67, 75–77], whereby interesting dynamics and flows in the membrane and in the surrounding medium were observed

[78, 79]. This chapter mainly discusses the response of giant vesicles to short DC rectangular pulses with duration in the range $100\,\mu\text{s}$ – $5\,\text{ms}$. Parallels to the vesicle behavior in AC fields will be also drawn. In this section, only vesicles in the fluid phase will be considered. In Section 7.3, we will discuss the response of vesicles in the gel phase and compare it to that of fluid vesicles.

While vesicle deformation in AC fields concerns stationary shapes, DC pulses induce short-lived shape deformations. In different studies, the pulse duration has been typically varied from several microseconds to milliseconds, while studies on cells have investigated a much wider range of pulse durations—from tens of nanoseconds to milliseconds and even seconds [80], as discussed in other chapters of this book. Various pulse profiles, unipolar or bipolar, as well as trains of pulses have been also employed (e.g., [81, 82]). Because the application of both AC fields and DC pulses creates a transmembrane potential, vesicle deformations of similar nature are to be expected in both cases. However, the working field strength for DC pulses is usually higher by several orders of magnitude. Thus, the degree of deformation can be different.

Vesicle deformation induced by DC pulses has been studied theoretically [83–85]. The majority of experimental studies were preformed on small vesicles of hundreds of nanometers in size [86–88], but their size did not allow for direct observation of the deformation dynamics. Employing fast digital imaging, recently we succeeded in revealing vesicle deformation [89], whereby the vesicle response was recorded with high temporal resolution of up to 30 000 frames per second (fps), that is, acquiring an image every 33 μs . The GUVs were observed to deform into prolate ellipsoids during the pulse and subsequently relax back to their initial spherical shape.

The degree of deformation of an ellipsoidal vesicle can be characterized by the aspect ratio of the two principal radii, a and b (see the inset of Figure 7.2). For a/b = 1 the vesicle is a sphere. In the absence of poration, that is, relatively weak or short pulses, the relaxation can be described by a single exponential with a characteristic decay time, τ_1 . Figure 7.2 gives one example of the response of a giant vesicle, which is initially spherical. The maximum deformation of this vesicle under the applied pulse conditions corresponds to about 10% change in the vesicle aspect ratio. The degree of vesicle deformation depends on the initial tension of the vesicle as well as on the excess area [90], the latter being defined as an excess compared to the area of a spherical vesicle with identical volume.

The typical decay time for the relaxation of nonporated vesicles, τ_1 , is of the order of 100 μ s. It is set by the relaxation of the membrane tension achieved at the end of the pulse. The membrane tension, σ_{el} , acquired during the pulse, also referred to as "electric tension," arises from the transmembrane potential, Ψ_m , built across the membrane during the pulse. Lipid membranes are impermeable to ions and, in the presence of an electric field, charges accumulate on both sides of the bilayer, which gives rise to this transmembrane potential [91]:

$$\Psi_{\rm m}(t) = 1.5R|\cos\theta|E\left[1 - \exp\left(\frac{-t}{t_{\rm c}}\right)\right]$$
(7.1)

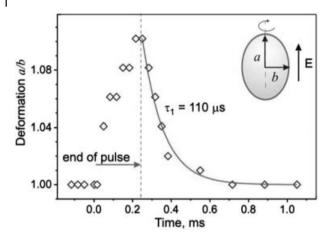


Figure 7.2 Time dependence of the degree of deformation, a/b, of a vesicle exposed to a square-wave DC pulse with field strength $E=100\,\mathrm{kV}\,\mathrm{m}^{-1}$ and pulse duration $t_\mathrm{p}=250\,\mathrm{\mu s}$. The solid curve is an exponential fit with a decay time τ_1 as indicated. The inset

schematically illustrates the shape of the deformed vesicle and the principal radii. The dashed line indicates the end of the pulse. Adapted from [89] by permission of the Biophysical Society.

where R is the radius of a spherical vesicle, θ is the tilt angle between the electric field and the surface normal, t is time, and t_c is the charging time as defined by

$$t_{\rm c} = RC_{\rm m} \left(\frac{1}{\lambda_{\rm in}} + \frac{1}{\lambda_{\rm ex}} \right) \tag{7.2}$$

Here, λ_{in} and λ_{ex} are the conductivities of the solutions inside and outside the vesicle, respectively. Equations (7.1) and (7.2) are valid only for a nonconductive membrane.

The effective electrical tension, σ_{el} , induced by the transmembrane potential, Ψ_{mr} , is defined by the Maxwell stress tensor [59, 89, 92]

$$\sigma_{\rm el} = \varepsilon_{\rm m} \left(\frac{h}{2h_{\rm e}^2}\right) \Psi_{\rm m}^2 \tag{7.3}$$

Here h is the total bilayer thickness (\approx 4 nm), $h_{\rm e}$ is the dielectric thickness (\approx 2.8 nm for lecithin bilayers [93, 94]), and $\varepsilon_{\rm m}$ is the membrane permittivity (\approx 2 $\varepsilon_{\rm 0}$, where $\varepsilon_{\rm 0}$ is the vacuum permittivity). For vesicles with some initial tension $\sigma_{\rm 0}$, the total tension reached during the pulse is simply

$$\sigma = \sigma_0 + \sigma_{\rm el} \tag{7.4}$$

As mentioned above, the decay time for the relaxation of nonporated vesicles, τ_1 , is defined by the total membrane tension at the end of the pulse. The tension relaxation requires relative displacement of the lipid molecules in the bilayer,

whose intermolecular distance is increased during the pulse, that is, the membrane is stretched. This relative displacement is characterized by the dilational viscosity η_D as introduced in Section 7.1.2. Thus, τ_1 relates mainly to the relaxation of membrane stretching: $\tau_1 \sim \eta_D/\sigma$. For membrane tensions of the order of $5\,\mathrm{mN\,m^{-1}}$ (which should be around the maximum tension before the membrane ruptures, $\sigma \approx \sigma_{lys}$) and for typical values of η_D , one obtains $\tau_1 \sim 100\,\mu\text{s}$, which corresponds to the experimentally measured value (see Figure 7.2).

Above some electroporation threshold, the transmembrane potential Ψ_m cannot be further increased, and can even decrease due to transport of ions across the membrane [91, 95]. The phenomenon of membrane electroporation can also be understood in terms of tension. If the total membrane tension exceeds the lysis tension σ_{lys} , the vesicle ruptures. This corresponds to building up a certain critical transmembrane potential, $\Psi_m = \Psi_c$. According to Eqs. (7.3) and (7.4), this poration potential Ψ_c depends on the initial membrane tension σ_0 as previously reported [59, 89, 90, 96, 97]. The critical transmembrane potential for cell membranes is about 1 V (e.g., [98, 99]).

Vesicle poration induced by DC pulses has been studied extensively, initially mainly on small vesicles [88, 100, 101]. Experiments on giant vesicles are of special relevance because their size allows for direct observation using optical microscopy [37, 89, 102–107]. The pores can reach various sizes depending on the location on the vesicle or cell surface (e.g., [99, 108] and work cited therein). For the case of plane parallel electrodes, the poration occurs predominantly in the area at the poles of the vesicle facing the electrodes. This is because the transmembrane potential attains its maximal value at the two poles as expressed by the angular dependence in Eq. (7.1). With optical microscopy, only pores that are of diameter larger than about 0.5 μ m can be resolved. We refer to them as macropores. The lifetime of macropores, τ_{pore} , observed in vesicles in the fluid state, varies with pore radius, r_{pore} [104], and depends on the membrane edge tension, γ , and the membrane dilatational viscosity: $\tau_{\text{pore}} \approx 2r_{\text{pore}}\eta_{\text{D}}/\gamma$. For phosphatidylcholine vesicles with low tension, the lifetime τ_{pore} is typically shorter than 30 ms [89]. The effect of membrane edge tension on pore resealing is discussed in detail in Section 7.5.1.

Figure 7.3a shows the time dependence of the deformation of vesicles in which macropores were observed. The maximum deformation is much higher than that observed for nonporated vesicles (compare with the aspect ratio a/b in Figure 7.2). The typical relaxation time is $\tau_2 \approx 7 \pm 3$ ms. The relaxation process associated with τ_2 takes place during the time interval when pores are present (shaded region in Figure 7.3a; snapshots of a porated vesicle are shown in Figure 7.3b). Thus, τ_2 is determined by the closing of the pores: $\tau_2 \approx \eta_D r_{pore}/2\gamma$. The edge tension γ is of the order of 10^{-11} N [109, 110]. For a typical pore radius of $1\,\mu$ m one obtains $\tau_2 \approx 10$ ms. When the vesicles have some excess area, the relaxation proceeds in two steps: a fast relaxation characterized by τ_2 and a second, longer, relaxation with decay time τ_3 : 0.5 s $< \tau_3 < 3$ s [89]. This relaxation time is related to the presence of some excess area available for shape changes corresponding well to the experimentally measured value.

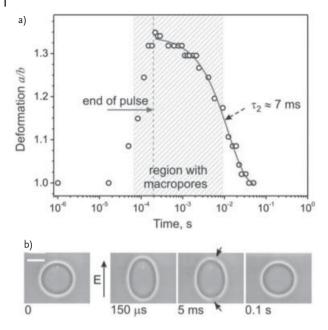


Figure 7.3 Deformation of macroporated phosphatidylcholine vesicles exposed to rectangular DC pulses with amplitude $E = 200 \,\mathrm{kV} \,\mathrm{m}^{-1}$, and pulse duration $t_p = 200 \,\mathrm{\mu s}$. (a) Time dependence of the degree of deformation a/b of a vesicle. Time t = 0 was set as the beginning of the pulse. The dashed line indicates the end of the pulse. The relaxation of the vesicle is described by a single exponential fit (solid curve) with a decay time τ_2 as indicated. The shaded area

indicates the time interval when macropores were optically detected. (b) Snapshots from another vesicle before, during, and after poration. The time after the beginning of the pulse is indicated below each micrograph. The scale bar corresponds to $15\,\mu m$. The arrows in the third image point to the porated zones, which are visualized as interruptions in the bright halo around the vesicle. Adapted from [89] by permission of the Biophysical Society.

7.3 Response of Gel-Phase Membranes

As discussed in Section 7.1.2, the mechanical and rheological properties of membranes in the gel phase differ significantly from those of fluid membranes. These differences introduce new features in the response of gel-phase membranes to electric fields. We compared the response to DC pulses of a vesicle in the fluid phase with that of a vesicle in the gel phase. The applied DC pulses were weak enough not to induce formation of macropores in the membranes. Figure 7.4a shows the deformation of the two vesicle types in response to DC pulses with duration of $300\,\mu s$. To achieve similar maximal degree of deformation in vesicles with comparable radii, stronger pulses have to be applied to the gel-phase vesicle as compared to the fluid one. Pulses with field strength of about $100\,kV\,m^{-1}$ do not produce optically detectable deformations in gel-phase vesicles, while strong

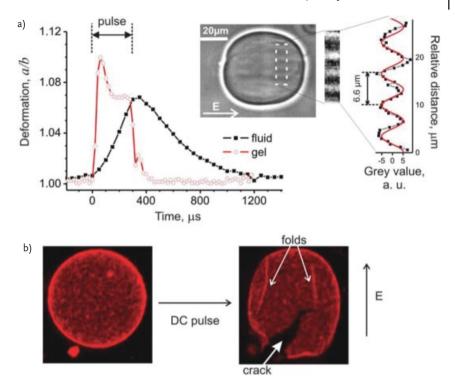


Figure 7.4 Deformation and electroporation of gel-phase vesicles. (a) Deformation response of a gel-phase vesicle with a radius of 22 µm, and a fluid-phase vesicle with a radius of 20 µm. The applied rectangular DC pulses were of duration of 300 µs as indicated. The field strength of the pulses was 500 and 80 kV m⁻¹ for the gel and the fluid vesicle, respectively, which is below the corresponding poration thresholds for the two types of membranes. The gel-phase vesicle exhibits an intrapulse relaxation arising from wrinkling of the membrane. The vesicle wrinkling is visible in the inset snapshot recorded at time $t = 200 \,\mu s$. A magnified and enhanced section of the image indicated with a dashed rectangle is given to the right of the image, showing the membrane wrinkling parallel to the electric field direction. The gray

value intensity from such a section is plotted and fitted with a sinusoidal function. The corresponding wavelength of the wrinkles is about 6.6 µm as indicated. (b) Electroporation of a gel-phase vesicle with radius 25 μm observed with confocal microscopy. Before the pulse, the vesicle has a spherical shape. After applying a pulse with field strength of 600 kV m⁻¹ and duration of 300 µs, the vesicle cracks open and folds as indicated by the white arrows. The field direction is indicated with a vertical arrow. The second image was recorded a few seconds after the end of the pulse. A relatively large crack is visible in the vesicle as shown in the three-dimensional projection of the vesicle top part. Adapted from [111] by permission of the Royal Society of Chemistry.

pulses of about $500\,\mathrm{kV}\,\mathrm{m}^{-1}$ applied to fluid-phase vesicles cause poration. The fluid vesicle gradually deforms and reaches maximal deformation (maximal aspect ratio a/b) at the end of the pulse as in Figure 7.1. The gel-phase vesicle responds significantly faster, and exhibits a relaxation with a decay time of about $50\,\mu\mathrm{s}$ already during the pulse. This unusual intrapulse relaxation was found to be due to wrinkling of the membrane [111] as shown with the inset in Figure 7.4a. Typical wavelengths of the wrinkles, Λ , lie in the range $5-8\,\mu\mathrm{m}$ and were found to obey laws for wrinkling of elastic sheets [112–114]: $\Lambda = 2(\pi^2 L^2 \kappa/\sigma)^{1/4}$, where L is the characteristic length of the system, which in our case is the vesicle size, $L \approx 2R$. The bending stiffness, κ , of membranes in the gel phase is orders of magnitude greater than that of fluid membranes (see Section 7.1.2), which is one of the reasons why wrinkling of fluid vesicles is not observed. Furthermore, fluid membranes have zero shear modulus and deform smoothly rather than exhibiting wrinkles.

The behavior of gel-phase vesicles exposed to stronger pulses above the poration threshold is also significantly different from that of porated fluid membranes. While pores in GUVs made of lipids in the fluid phase reseal within a few tens of milliseconds (Section 7.2), gel-phase giant vesicles exhibit long-living pores [42, 111]. The pores may resemble cracks on solid shells (Figure 7.4b), which remain open for minutes. Similar arrest was reported for mechanically induced pores in membranes below the main phase transition temperature [58]. Having in mind that the pore lifetime is defined by the membrane viscosity ($\tau_{\text{pore}} \approx 2r_{\text{pore}}\eta_{\text{D}}/\gamma$, see Section 7.2), it is easy to understand the long lifetime of pores in gel-phase membranes. The membrane viscosity diverges when the membrane crosses the main phase transition [43]. Thus, the resealing process is strongly suppressed. The irregular shape of the pores in gel-phase vesicles may be further indicative of the relatively low edge tension in such membranes.

Above we discussed the behavior of macroporated giant vesicles. Electroporated gel-phase vesicles with sizes around 100 nm were reported to reseal within milliseconds [100, 115]. For pores created in such vesicles, the pore radius, $r_{\rm pore}$, should be in the nanometer range, yielding pore lifetimes, $\tau_{\rm pore}$, that are three orders of magnitude shorter than those of the micrometer-size pores in giant vesicles. Furthermore, the fields applied in references [100, 115] were of the order of 30 kV cm⁻¹. Such pulses induce Joule heating resulting in a temperature increase of up to 15 K [115], which can bring the membranes close to and even above the pretransition temperature of the investigated lipid and lead to a decrease in the membrane viscosity. Finally, recent coarse-grain simulations on gel-phase vesicles with similar sizes suggest that below the main phase transition temperature a fraction of the lipids in such highly curved membranes remain in the fluid phase [116], which may further facilitate pore closure. To summarize, the response of small and giant vesicles in the gel phase to DC pulses above the poration threshold differs significantly.

One additional feature that differentiates the response of gel-phase from fluidphase membranes is the critical poration threshold. As demonstrated in Figure 7.4a, pulses that porate fluid vesicles lead only to deformation of gel-phase GUVs. The critical transmembrane potential for the latter was found to be in the range 8–10 V [111]. This value is significantly higher than the critical potential of 1 V reported for fluid membranes [59, 98, 99]. Thus, membranes in the gel phase can sustain higher tensile stresses (see Eqs. (7.3) and (7.4)). This is also confirmed by micropipette aspiration experiments showing that fluid-phase dimyristoylphosphatidylcholine membranes undergo lysis at tensions around 2–3 mN m⁻¹, but when in the gel phase, the membranes rupture at tensions above 15 mN m⁻¹ [62]. It is important to note that the rupture process depends on the loading rate [61, 117, 118]. At high loading rates a membrane can sustain much higher tensions before it ruptures. Similar behavior was demonstrated by simulation studies, where fluid dipalmitoylphosphatidylcholine (DPPC) bilayers were shown to spontaneously rupture at tensions exceeding 90 mN m⁻¹ [119].

7.4 Effects of Membrane Inclusions and Media on the Response and Stability of Fluid Vesicles in Electric Fields

In the previous two sections we discussed the electrodeformation and electroporation of vesicles made of single-component membranes in water. In this section, we consider the effect of salt present in the solutions. The membrane response discussed above was based on data accumulated for vesicles made of phosphatidylcholines (PCs), the most abundant fraction of lipids in mammalian cells. PC membranes are neutral and predominantly located in the outer leaflet of the plasma membrane. The inner leaflet, as well as the bilayer of bacterial membranes, is rich in charged lipids. This raises the question as to whether the presence of such charged lipids would influence the vesicle behavior in electric fields. Cholesterol is also present at a large fraction in mammalian cell membranes. It is extensively involved in the dynamics and stability of raft-like domains in membranes [120]. In this section, apart from considering the response of vesicles in salt solutions, we describe aspects of the vesicle behavior of fluid-phase vesicles when two types of membrane inclusions are introduced, namely cholesterol and charged lipids.

7.4.1

Vesicles in Salt Solutions

In the presence of salt in a vesicle exterior, unusual shape changes are observed during an applied DC pulse [107]. The vesicles adopt spherocylindrical shapes (Figure 7.5) with lifetimes of the order of 1 ms. These deformations occur only in the presence of salt outside the vesicles, irrespective of their inner content (note that, in the absence of salt in the external solution, the vesicles deform only into prolates; see Figure 7.5a). When the solution conductivities inside and outside are identical, $\lambda_{\rm in} \approx \lambda_{\rm ex}$, vesicles with square cross section are observed (Figure 7.5d). For the case $\lambda_{\rm in} < \lambda_{\rm ex}$, the vesicles adopt disc-like shapes (Figure 7.5c), while in the

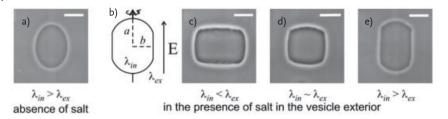


Figure 7.5 Deformation of vesicles in the absence and in the presence of salt at different conductivity conditions when subjected to DC pulses $(200\,kV\,m^{-1},\,200\,\mu s)$. (a) In the absence of salt in the external solution, prolate deformation is observed. (b) Schematic of a cross section of a vesicle, which has adopted spherocylindrical

deformation (a cylinder with spherical caps) when salt is present in the vesicle exterior. The field direction is indicated with an arrow. The presence of salt flattens the vesicle walls into (c) disc-like, (d) "square"-like, and (e) tube-like shapes. The scale bars correspond to 15 µm. Adapted from [107] by permission of the Biophysicsl Society.

opposite case, $\lambda_{in} > \lambda_{ex}$, they deform into long cylinders with rounded caps (Figure 7.5e).

Analogy can be drawn between the cylindrical shapes observed in the presence of salt at various conductivity conditions and the elliptical morphologies of vesicles exposed to AC fields. The beginning of Section 2 briefly discussed the latter shapes. In AC fields, vesicles attain different deformations: prolate (with the long principal radius oriented in the field direction) and oblate (with the long principal radius oriented perpendicular to the field direction). For a fixed field frequency in the range 10^4 – 10^6 Hz, the type of deformation depends on the solution conductivities [42, 64]. In particular, vesicles with lower internal conductivity ($\lambda_{\rm in} < \lambda_{\rm ex}$) adopt oblate shapes (a/b < 1), analogously to the disc-like deformation in Figure 7.5c, whereas vesicles with higher internal conductivity ($\lambda_{\rm in} > \lambda_{\rm ex}$) adopt prolate shapes (a/b > 1), analogously to the tube-like deformation shown in Figure 7.5e. However, in AC fields, no flattening of the vesicles is observed.

The cylindrical deformations shown in Figure 7.5 are nonequilibrium shapes and have a very short lifetime, which is why they have not been observed previously when standard video acquisition speed was accessible only. The flattening of the vesicle walls starts during the applied pulse and is observed throughout a period of about 1 ms. So far, cylindrical deformations have not been observed in studies on cells [121, 122]. However, the temporal resolution in those experiments (3.3 ms per image) was not high enough to detect such short-lived deformations.

The formation of the spherocylindrical shapes is not well understood. They are observed not only on lipid vesicles but also on polymersomes (vesicles made of diblock copolymers [123, 124]) [107]. Therefore, lipid-specific effects, for example partial head group charge and membrane thickness, as a possible cause for the observed cylindrical deformations are to be excluded. One possible explanation could be that ions flatten the equatorial zone of the deformed vesicle. During the pulse there is an inhomogeneity in the membrane tension due to the fact that the

electric field is strongest at the poles of the vesicle, and almost zero close to the equator. The kinetic energy of the accelerated ions "hitting" the equatorial (tensionless) region of the vesicle is higher than the energy needed to bend the membrane, thus presumably leading to the observed deformation. To a certain extent, the effect of the ions could be considered as that of tiny particles subject to electrophoresis. Indeed, cylindrical deformations were observed also in quasi-salt-free solutions but in the presence of small negatively charged nanoparticles (40 nm in radius) in the vesicle exterior [42]. Thus, particle-driven flows may be yet another possible factor inducing membrane instability and giving rise to higher order modes of vesicle shape [125]. Another effect that could be considered is related to a local change in the spontaneous curvature of the bilayer due to the ion or particle asymmetry across the membrane [126]. During the pulse, local and transient accumulation of particles or ions in the membrane vicinity can occur inducing a change in the bilayer spontaneous curvature and driving the cylindrical deformations. Furthermore, another influencing factor might be an electrohydrodynamic instability caused by electric fields interacting with flat membranes, which was predicted to increase membrane roughness [127]. Finally, the flexoelectric properties of lipid membranes, first postulated by Petrov and coworkers [128–130], may also be involved as recently proposed [73, 131]. The interplay between surface ion concentration gradients combined with the overall ionic strength and bilayer material properties and tension could be expected to produce the observed cylindrical deformations.

7.4.2 Vesicles with Cholesterol-Doped Membranes

In order to understand the complex behavior of cellular membranes and their response to external perturbations like electric fields, one has to elucidate the basic mechanical properties of the lipid bilayer. The significant expansion in recent years of the field of membrane raft-like domain formation [11, 132, 133] imposes the compelling need for understanding the effect of lipid bilayer composition on membrane properties. Cholesterol, a ubiquitous species in eukaryotic membranes, is an important component in raft-like domains in cells and in vesicles, which motivates studies aimed at understanding its influence on the mechanical properties and stability of membranes.

A widely accepted view in the past regarded cholesterol as a stiffening agent in membranes. The conventional belief was that cholesterol orders the acyl chains in fluid membranes and increases the bending stiffness. This concept was supported by observations of lipids such as stearoyloleoylphosphatidylcholine (SOPC) [134, 135], dimyristoylphosphatidylcholine [53, 136], and palmitoyloleoylphosphatidylcholine (not only in mixtures with cholesterol but also with other sterols) [137]. However, as demonstrated recently [55, 138–140], the bending rigidity of membranes made of dioleoylphosphatidylcholine (DOPC) and cholesterol does not show any significant correlation with the cholesterol content. This suggests that the effect of cholesterol is not universal, but rather specific to the lipid

architecture with respect to unsaturation and acyl chain length, and probably the lipid interfacial region.

The influence of cholesterol on other properties relevant for the stability and electroporation of membranes has not been studied extensively. The inverted-cone shape of this molecule should prevent it from locating at the rim of pores in membranes. Thus, the presence of cholesterol would require more energy to rearrange the lipids along the pore walls leading to an effective increase in the edge tension. Observations in this direction have been reported [141]. In [141] the pores were induced by strong illumination of giant vesicles inducing membrane stretching. Electroporation of giant vesicles and observation on the dynamics of pore closure also supported the notion of cholesterol increasing the edge tension [110]. The associated implication is that the lifetime of pores in such membranes is shorter (as defined above, the pore lifetime is inversely proportional to the edge tension; see Section 7.2). Note that both above-mentioned studies on the effect of cholesterol on pore stability [110, 141] were performed using DOPC membranes. It remains to be clarified whether the tendency will be preserved when other lipids with different degrees of saturation are considered.

Another peculiarity of DOPC-cholesterol GUVs is that they are destabilized and collapse when exposed to DC pulses at which cholesterol-free vesicles (e.g., made of egg phosphatidylcholine (egg PC) or pure DOPC) preserve their stability and only porate and reseal. The cholesterol-doped GUVs burst and disintegrated in a fashion reminiscent of that of charged membranes [142] (see Figure 7.6 for an example of a collapsing charged vesicle). As a quantitative characteristic of mem-

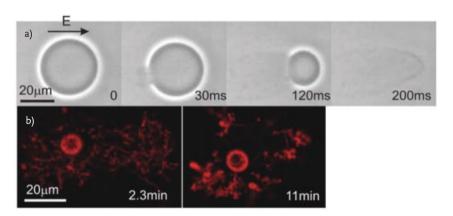


Figure 7.6 Bursting of charged vesicles subjected to electric pulses. The time after the beginning of the pulse is marked on each image. (a) Phase contrast microscopy snapshots from fast-camera observation of a vesicle in salt solution subjected to a pulse with field strength of 120 kV m⁻¹ and duration of 200 μs. The field direction is indicated in

the first snapshot. The vesicle bursts and disintegrates. (b) Confocal microscopy cross sections of a vesicle that has been subjected to an electric pulse and after bursting has rearranged into a network of tubes and smaller vesicles. Adapted from [42] by permission of the Royal Society of Chemistry.

brane stability in electric fields one can consider the critical transmembrane potential, Ψ_c , at which poration occurs. Following [80], at the moment of maximally expanded pore, one can define the critical transmembrane potential as $\Psi_c = 1.5 RE \cos \theta_p$, where R is the initial vesicle radius before poration, E is the field magnitude, and θ_p is the inclination angle defining the location of the pore edge with respect to the vesicle center. This expression can be presented as $\Psi_{\rm c} = 1.5 E (R^2 - r_{\rm pore}^2)^{1/2}$, where $r_{\rm pore}$ is the maximal pore radius. Thus, measuring r_{pore} immediately after applying a pulse with a field strength E provides a rough estimate for the critical transmembrane poration potential Ψ_c . While cholesterolfree membranes made of different lipids porate at similar values of Ψ_c of around 0.9 V, the addition of 17 mol% cholesterol to DOPC bilayers decreases the critical transmembrane potential to around 0.7 V, that is, destabilizes these membranes when exposed to an electric pulse [110]. This finding implies that cholesterol lowers the lysis tension of DOPC membranes. This observation is interesting and unexpected having in mind the somewhat opposite effect of cholesterol observed on SOPC giant vesicles [59, 103], where it was found to increase the critical poration threshold Ψ_c . On the contrary, DPPC vesicles doped with cholesterol appear to porate at transmembrane potentials lower than those of pure DPPC membranes [143]. Finally, for egg PC planar membranes [144] and vesicles [143] no effect of cholesterol on the critical permeabilization potential was observed.

The origin of the cholesterol-induced changes to the critical poration potential could be sought in its effect on altering the membrane conductivity justified in detail in [143]. This behavior may be related to lipid ordering, which may occur to a different extent depending on the type of lipid. In summary, cholesterol, which alters the hydrophobic core of the bilayer, affects the membrane stability in a different fashion depending on the specific molecular architecture of the lipid building the membrane.

7.4.3

Membranes with Charged Lipids

As discussed in Section 7.2, strong electric pulses applied to single-component giant vesicles made of zwitterionic lipids like PCs induce the formation of pores, which reseal within tens of milliseconds. When negatively charged lipids, like phosphatidylglycerol (PG) or phosphatidylserine, are present in a membrane a very different response of the vesicles can be observed, partially influenced by the medium conditions [142].

In buffered solutions containing ethylenediaminetetraacetic acid, PC:PG vesicles with molar ratios 9:1, 4:1, and 1:1 behave in the same way as pure PC vesicles, that is, the pulses induce opening of macropores with a diameter up to about $10\,\mu\text{m}$, which reseal within tens of milliseconds. In nonbuffered solution, the membranes with low fractions of charged lipids (9:1 and 4:1) retain this behavior, but for membrane composition of 1:1 PC:PG, the vesicles collapse and disintegrate after electroporation [142] (see Figure 7.6). Typically, one macropore forms and expands in the first $50-100\,\text{ms}$ at a very high speed of approximately $1\,\text{mm}\,\text{s}^{-1}$.

The entire vesicle content is released seen as darker fluid in Figure 7.6. The bursting is followed by restructuring of the membrane into what seem to be interconnected bilayer fragments in the first seconds, and a tether-like structure in the first minute. Then the membrane stabilizes into interconnected micrometer-sized tubules and small vesicles. Similar behavior is observed for vesicles prepared from lipid extracts from the plasma membrane of red blood cells which also contain a fraction of charged lipids [142]. These observations suggest that vesicle bursting and membrane instability are related to the amount of charged lipid in the bilayer as theoretically predicted in earlier studies [145–147].

As mentioned above, observations of GUVs prepared from lipid extracts from the plasma membrane of red blood cells show a response similar to that of vesicles prepared from synthetic charged lipids. Thus, one would intuitively expect that cells should exhibit similar bursting behavior. However, cell membranes are subjected to internal mechanical constraints imposed by the cytoskeleton, which prevents their disintegration even if their membranes are prone to disruption when exposed to strong DC pulses. Instead, the pores in the cell membrane are stable for a long time [148] and can either lead to cell death by lysis or reseal depending on the media [149, 150]. The latter is the key to efficient electroporation-based protocols for drug or gene transfer in cells. The results discussed in this section suggest that membrane charge might be an important, but not yet well understood, regulating agent in these protocols.

7.5 Application of Vesicle Electroporation

In this section some application aspects of giant vesicle electroporation are considered. In particular, it will be demonstrated that creating macropores in GUVs and observing their closing dynamics can be successfully applied to the evaluation of material properties of membranes. While in Section 7.4.2 we saw that such experiments can be used to characterize membrane stability in terms of the critical poration potential Ψ_c , here we will find out how one can also evaluate the edge tension of porated membranes. In addition, another application based on electroporation, namely vesicle electrofusion, is introduced whereby the use of GUVs as microreactors suitable for the synthesis of nanoparticles is demonstrated.

7.5.1 Measuring Membrane Edge Tension from Vesicle Electroporation

Upon poration, the lipid molecules in a bilayer reorient so that their polar heads can line the pore walls and form a hydrophilic pore [151]. The energetic penalty per unit length for this reorganization is described by the edge tension, which emerges from the physicochemical properties and the amphiphilic nature of lipids. It also gives rise to a force driving the closure of transient pores, playing a crucial role in membrane resealing mechanisms taking place after physical pro-

tocols for drug delivery, such as sonoporation [152] or electroporation [153]. Being able to experimentally measure the edge tension is thus of significant interest for understanding various biological events and physicochemical processes in membranes. However, only a few experimental methods have been developed to directly assess this physical quantity. Among them, an elegant approach was based on rapid freezing of cells with a controlled time delay after electroporation, and examining the pores with electron microscopy [154, 155]. This method, however, provides a static picture of the porated membrane and there is a danger of ice crystal damage.

Only a few previous studies have employed GUVs for estimating edge tension. Observations of open cylindrical giant vesicles exposed to AC fields [109] provided an estimate for the edge tension, but this technique did not allow for good control over the system. In another work, vesicles were porated with an electric pulse, and the pores were kept open by externally adjusting the membrane tension with a micropipette [103]. Even though solid, this approach requires the use of sophisticated equipment like a set-up for vesicle micropipette aspiration. GUVs were also used in [141, 156], where the pore closure dynamics was analyzed in light of a theory developed earlier [44]. However, for the direct visualization of pore closure, the use of viscous glycerol solutions and fluorescent dyes in the membrane was required, both of which potentially influence the edge tension. (The theoretical approach in [44] was also applied to experiments where the membrane was disrupted by laser ablation in [157], but the results of the latter study seem questionable since laser ablation is associated with local evaporation; indeed, the images provided in this work suggest the formation of bubbles in the sample rather than pores in the membrane.)

A much simpler approach introduced recently is based on the electroporation of giant vesicles and observation of the pore closure dynamics with fast digital imaging [110, 158]. The analysis of the pore dynamics was based on the theoretical work of Brochard-Wyart *et al.* [44]. The process of pore closure was observed under phase contrast microscopy with a high-speed digital camera (the acquisition speed was typically above 1000 fps). In this way, the need to use viscous solutions to slow down the system dynamics was avoided and the application of fluorescent dyes to visualize the vesicles as in [104, 141, 156] was not necessary. Note that both glycerol and fluorescent markers may influence the measured value for the edge tension. Vesicle electroporation was induced by applying electric pulses of 5 ms duration and field strength in the range 20–80 kV m⁻¹. The pore dynamics typically consisted of four stages: growing, stabilization at some maximal pore radius, slow decrease in pore size, and fast closure (e.g., data in Figure 7.7). The third stage of slow pore closure is the one that can be used to determine the membrane edge tension applying the dependence derived in [44]:

$$R^2 \ln(r) = -\frac{2\gamma}{3\pi\eta} t + C \tag{7.5}$$

where R and r are the vesicle and pore radii, respectively, γ is the edge tension, η is the viscosity of the aqueous medium, t is time, and C is a constant depending

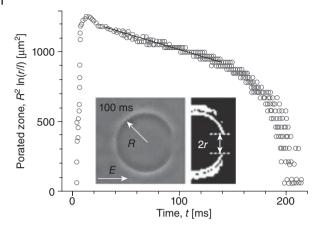


Figure 7.7 Evolution of the porated region in an egg PC vesicle as characterized by $R^2 \ln(r/l)$ as a function of time t (see Eq. (7.5); note that to avoid plotting a dimensional value in the logarithmic term, we have introduced $l = 1 \mu m$). The open circles are experimental data and the solid line is a linear fit, whose slope yields the edge tension γ . The inset shows a raw image (left) of a porated vesicle

with a radius of $17 \mu m$ 100 ms after being exposed to an electric pulse with duration of 5 ms and amplitude of $50 \, kV \, m^{-1}$. The field direction is indicated with an arrow. The right-hand side of the inset is an enhanced and processed image of the vesicle half facing the cathode. The inner white contour corresponds to the location of the membrane. The pore radius is schematically indicated.

on the maximal pore radius reached. Then, one only has to consider the linear part of $R^2 \ln(r)$ as a function of time corresponding to the slow closure stage. Linear fit of this part is characterized by a slope a and the edge tension γ is estimated from the relation $\gamma = -(3/2)\pi\eta a$. Figure 7.7 illustrates the analysis performed on one egg PC vesicle. Measurements on many vesicles yielded an average value of $\gamma = 14.3 \,\mathrm{pN}$ for the edge tension of such membranes.

Using this approach, one can measure the edge tension in membranes of various compositions, thus characterizing the stability of pores in these membranes, and evaluate the effect of various inclusions. For example, it was found that the addition of cholesterol to DOPC membranes increases the edge tension confirming previously reported results [141]. The inverted-cone shape of cholesterol prevents it from locating at the rim of pores. Thus, the presence of cholesterol requires more energy to rearrange the lipids along the pore walls increasing the edge tension. Surprisingly, doping DOPC membranes with another cone-shaped type of lipid like dioleoylphosphatidylethanolamine (DOPE) was found to decrease the edge tension, that is, DOPE has a pore-stabilizing effect [110]. Presumably, the molecular architecture of phosphatidylethanolamine (PE)-lipids leading to their tendency to form an inverted hexagonal phase, which facilitates fusion and vesicle leakage (e.g., [159]), is also responsible for stabilizing pores. A plausible explanation for this behavior is also provided by the propensity of PE to form interlipid hydrogen bonds [160, 161], that is, inter-PE hydrogen bonding in the pore region can effectively stabilize pores.

As demonstrated above, the edge tension is a sensitive parameter, which effectively characterizes the stability of pores in membranes. Compiling a database for the effect of various types of membrane inclusions will be useful for understanding the lifetime of pores in membranes with more complex compositions, which is important for achieving control over medical applications for drug and gene delivery in cells.

7.5.2

Vesicle Electrofusion

The phenomenon of membrane electrofusion is of particular interest, because of its widespread use in cell biology and biotechnology (e.g., [162–164] and the references cited therein). The application of electrofusion to cells can lead to the creation of multinucleated viable cells with new properties (this phenomenon is also known as hybridization) (e.g., [164]). In addition, electroporation and electrofusion are often used to introduce molecules like proteins, foreign genes (plasmids), antibodies, and drugs into cells.

When a DC pulse is applied to a couple of fluid-phase vesicles, which are in contact and oriented in the direction of the field, electrofusion can be observed. Vesicle orientation (and even alignment into pearl chains) can be achieved by application of an AC field to a vesicle suspension. This phenomenon is also observed with cells [164, 165] and is due to dielectric screening of the field. When the suspension is dilute, two vesicles can be brought together via the AC field and aligned. A subsequent application of a DC pulse to such a vesicle couple can lead to fusion. The necessary condition is that poration is induced in the contact area between the two vesicles. The possible steps of the electrofusion of two membranes are schematically illustrated in Figure 7.8a. In Sections 7.5.2.1 and 7.5.2.2, consideration will be given to the fusion of vesicles with different membrane composition or different composition of the enclosed solutions.

7.5.2.1 Fusing Vesicles with Identical or Different Membrane Composition

Membrane fusion is a fast process. The time needed for the formation of a fusion neck can be rather short as demonstrated by electrophysiological methods applied to the fusion of small vesicles with cell membranes [166–169]. The time evolution of the observed membrane capacitance indicates that the formation of the fusion neck is presumably faster than 100 µs. Direct observation of the fusion of giant vesicles recently confirmed this finding and suggested that this time is even shorter [36, 37]. An example of a few snapshots taken from the electrofusion of two GUVs with identical membrane compositions and in the presence of salt is given in Figure 7.8b. The overall deformation of each vesicle as seen in the second snapshot corresponds to the spherocylindrical shapes as observed with individual vesicles in the presence of salt (see Section 7.4.1). From such micrographs, one can measure the fusion neck diameter, denoted by *L* in Figures 7.8a and 7.8b, and follow the dynamics of its expansion as shown in Figure 7.8c. From the data, two stages of the fusion process can be distinguished (note that the data are displayed in a semi-logarithmic plot): an early stage, which is very fast and with average

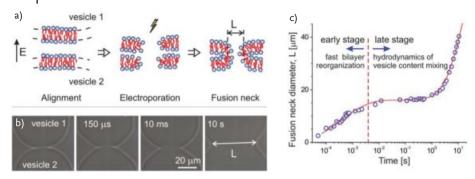


Figure 7.8 Membrane electrofusion.

(a) Schematic of the possible steps of the electrofusion process: two lipid vesicles are brought into contact (only the membranes in the contact zone of the vesicles are sketched), followed by electroporation and formation of a fusion neck of diameter *L*. The pore sizes are not to scale but almost an order of magnitude larger than the bilayer thickness.

(b) Micrographs from the electrofusion of a vesicle couple. Only segments of the vesicles are visible. The external solution contains 1 mM NaCl, which causes flattening of the

vesicle walls in the second snapshot (see Section 7.4.1). The amplitude of the DC pulse was $240\,kV\,m^{-1}$, and its duration was $120\,\mu s$. The time after the beginning of the pulse is indicated on the snapshots. (c) Time evolution of the fusion neck diameter, L, formed between two vesicles with radii of about $15\,\mu m$. The solid curve is a guide to the eye. The vertical dashed line indicates the border between the two stages in the fusion dynamics. Adapted from [41] by permission of the Royal Society of Chemistry.

expansion velocity of about $2 \times 10^4 \mu m \, s^{-1}$, followed by a later slower stage with an expansion rate that is orders of magnitude smaller (about $2 \, \mu m \, s^{-1}$). The early stage is governed by fast relaxation of the membrane tension built during the pulse, whereby the dissipation occurs in the bilayer. Essentially, the driving forces here are the same as those responsible for the relaxation dynamics of nonporated vesicles (as characterized by τ_1 in Section 7.2). Thus, the characteristic time for this early stage of fusion, $\tau_{\rm early}$, can be expressed as $\tau_{\rm early} \sim \eta_{\rm D}/\sigma$, where the membrane tension σ should be close to the tension of rupture $\sigma_{\rm lys}$. Thus for $\tau_{\rm early}$ one obtains a value of $100 \, \mu \rm s$, which is in agreement with the experimental observations for the time needed to complete the early stage of fusion. Linear extrapolation of the data in the early stage predicts that the formation of a fusion neck with a diameter of about $10 \, \rm nm$ should occur within a time period of about $250 \, \rm ns$ [37]. It is quite remarkable that this time scale of the order of $200 \, \rm ns$ was also obtained from computer simulations of a vesicle fusing with a tense membrane segment [169].

In the later stage of fusion, the neck expansion velocity slows down by more than two orders of magnitude. Here the dynamics is mainly governed by the displacement of the volume of fluid around the fusion neck between the fused vesicles. The restoring force is related to the bending elasticity of the lipid bilayer [36, 37].

Fusing two vesicles with membranes of different composition can provide a promising tool for studying raft-like domains in membranes [11, 12, 133, 170, 171].

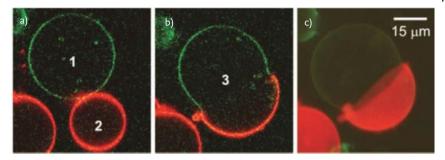


Figure 7.9 Creating a multidomain vesicle by electrofusion of two vesicles with different membrane composition as observed with fluorescence microscopy. (a, b) Images acquired with confocal microscopy scans nearly at the equatorial plane of the fusing vesicles. (a) Vesicle 1 is made of sphingomyelin and cholesterol (7:3) and labeled with one fluorescent dye (green). Vesicle 2 is composed of dioleoylphosphatidylcholine and cholesterol (8:2) and labeled with another

fluorescent dye (red). (b) The two vesicles were subjected to an electric pulse $(220\,\text{kV}\,\text{m}^{-1},\,\text{duration}\,300\,\mu\text{s})$ and fused to form vesicle 3. Because the lipids with this final membrane composition form immiscible fluid phases, the resulting vesicle has two domains. (c) A three-dimensional image projection of vesicle 3 with the two domains formed from vesicles 1 and 2. Adapted from [41] by permission of the Royal Society of Chemistry.

In particular, vesicle electrofusion is a very attractive experimental approach for producing multicomponent vesicles of well-defined composition [36]. One example for the fusion of two vesicles with different membrane composition is shown in Figure 7.9. To distinguish the vesicles according to their composition, two fluorescent markers have been used. In this particular example, one of the vesicles (vesicle 1) is composed of sphingomyelin and cholesterol in 7:3 molar ratio and labeled in green. The other vesicle (vesicle 2) is composed of DOPC and cholesterol in 8:2 molar ratio and labeled in red (with fluorescence microscopy, the two vesicle can be distinguished by their color). Thus, the membrane of the fused vesicle is a three-component one. At room temperature, this lipid mixture separates into two phases, liquid ordered (rich in sphingomyelin and cholesterol) and liquid disordered (rich in DOPC), which is why the final vesicle exhibits immiscible fluid domains. The exact composition of each of these domains is not well known because lipids may redistribute among the domains. However, from the domain area and the area of the initial vesicles before fusion, one can judge whether there is redistribution of cholesterol, and eventually calculate the actual domain composition.

7.5.2.2 Vesicle Electrofusion: Employing Vesicles as Microreactors

Vesicle fusion can be employed to scale down the interaction volume of a chemical reaction and reduce it to a few picoliters or less. In other words, fusion of two vesicles of different content can be used for the realization of a tiny microreactor [172–176]. The principle of fusion-mediated synthesis is simple: the starting reagents are separately loaded into different vesicles, and then the reaction is

triggered by the fusion of these vesicles, which allows the mixing of their contents. The success of this approach is guaranteed by two important factors. First, the lipid membrane is impermeable to the reactants such as ions or macromolecules. Second, fusion can be initiated by a variety of fusogens such as membrane stress [169, 177, 178], ions or synthetic fusogenic molecules [37, 179–181], fusion proteins [182], or electric fields [36, 176]. Among these approaches, electrofusion becomes increasingly important because of its reliable, fast, and easy handling. An immediate benefit of this strategy is that precise temporal control of the synthesis process can be easily achieved.

Synthesis of nanoparticles in such microreactors is of particular interest. Indeed, cells and microorganisms themselves have been reported to have the unique ability to synthesize inorganic nanoparticles such as CdS, ZnS, gold, and silver [183–185]. The tentative interpretation of this observation is related to the mediation ability of specific molecules such as inorganic-binding peptides [186]. Nevertheless, the underlying processes are still not well understood at the molecular scale. Therefore, attempts to perform similar reactions in simplified artificial systems become very important towards detailed exploration of biological synthesis mechanisms and biomimetic fabrication [185].

Although nanoparticle synthesis in vesicle nanoreactors has been well documented [187–190], much effort is still needed to perform such synthesis in GUVs so as to elucidate the biological mechanism of nanoparticle synthesis in cells and construct novel functional "artificial cells" for advanced technological applications. Other processes in giant vesicles occurring in their enclosed compartments or at the membrane surface have also been studied [172, 174, 175, 191–194]. Furthermore, one great and exclusive advantage of GUVs is that giant vesicles and the corresponding products of their interaction can be visualized in real time under a light microscope, providing the potential possibility for on-line monitoring of material growth at micrometer and submicrometer scales.

Success in this direction was recently reported for the synthesis of CdS quantumlike nanoparticles [176]. The protocol applied is schematically illustrated in Figure 7.10a. Two vesicle populations loaded with reactant A (CdCl₂) or B (Na₂S) are mixed in A-, B-free isotonic solution. To be able to distinguish the two vesicle types from each other, the membranes of A-loaded and B-loaded vesicles are labeled by small fractions of red and green fluorescent dyes, respectively, added to the main lipid building the membrane. Similar to pearl-chain formation in cell suspensions, the two types of vesicles align along the direction of an exogenous AC field. This field-induced self-arrangement makes reactive vesicles match well for the CdS synthesis reaction: half of the aligned vesicle couples are A-B couples. After that, a strong DC pulse is applied (typically pulses of 50–200 kV m⁻¹ field strength and 150-300 µs duration suffice) to initiate vesicle fusion and, thus, the reaction between A and B. The product, in this case quantum-dot-like CdS nanoparticles (with sizes between 4 and 8 nm as determined from transmission electron microscopy [176]), is visualized under laser excitation as a fluorescent bright spot in the fusion zone (see the second snapshot in Figure 7.10b). The intensity from this spot increases gradually from 0s (the starting time point of electrofusion), and

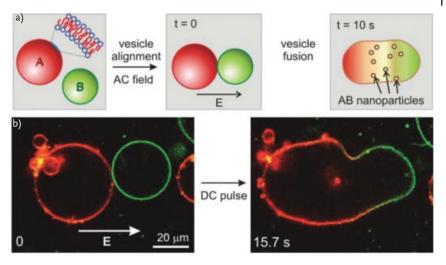


Figure 7.10 Electrofusion of giant vesicles as a method for nanoparticle synthesis.
(a) Schematic of the electrofusion protocol. Two populations of vesicles containing reactant A or B are mixed (in A- and B-free environment) and subjected to an AC field to align them in the direction of the field and bring them close together. A DC pulse initiates the electrofusion of the two vesicles and the reaction between A and B proceeds to the formation of nanoparticles encapsulated

in the fused vesicle. (b) Confocal scans of vesicles loaded with $0.3\,\text{mM}$ Na_2S (red) and $0.3\,\text{mM}$ CdCl_2 (green) undergoing fusion. The direction of the field is indicated in the first snapshot. After fusion (second snapshot), fluorescence from the product is detected in the interior of the fused vesicle. The time after applying the pulse is indicated on the micrographs. Adapted from [176] by permission of Wiley-VCH Verlag.

reaches maximum around 10 s. One can therefore infer that the reaction begins at the electrofusion point and quickly reaches a balance around 10 s. Obviously, this protocol provides us with a visualizing analytical tool to follow the reaction kinetics with high temporal sensitivity. This novel and facile method is especially suitable for the on-line monitoring of ultrafast physicochemical processes such as photosynthesis, enzyme catalysis, and photopolymerization, which usually require complex and abstracted spectroscopy techniques at present. These results also show that even without the mediation of biomacromolecules, nanoparticles can still be synthesized in biological compartments. This outcome provides a new insight in the developing research on biomineralization mechanisms.

7.6 Conclusions and Outlook

The issues addressed in this chapter demonstrate that cell-sized giant vesicles provide a very useful model for resolving various effects of electric fields on lipid membranes because vesicle dynamics can be directly observed with optical

microscopy. It has been shown that the vesicle response to electric fields can be interpreted and understood considering the basic mechanical properties of membranes.

Until recently, the dynamics of vesicle relaxation and poration, which occur on microsecond time scales, has eluded direct observation because the temporal resolution of optical microscopy observations with analog video technology is in the range of milliseconds. The recent use of fast digital imaging has helped us to characterize membrane deformation and poration and also to discover new features of the membrane response arising from the presence of charged lipids or cholesterol in the membrane and nanoparticles in the surrounding media, and to compare the response of gel-phase membranes to fluid-phase ones. Due to this high temporal resolution, new shape deformations, such as spherocylindrical ones, have been detected. The observations of vesicle fusion revealed the presence of two stages of the fusion process. Finally, a novel application of membrane electrofusion was introduced, which allows the construction of vesicles with fluid domains.

In conclusion, the reported observations demonstrate that giant vesicles as biomimetic membrane compartments can be of significant help to advance fundamental knowledge about the complex behavior of cells and membranes in electric fields and can inspire novel practical applications.

Acknowledgments

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8 Electroporation for Medical Use in Drug and Gene Electrotransfer

Julie Gehl

8.1 Introduction

Using electric pulses to manipulate cell membranes started in a community of physicists, actually with contributions from three different continents. The idea that dielectric breakdown of membranes could be used in medicine gained new life with the groundbreaking paper of Neumann *et al.* in 1982 [1], showing that DNA could be transferred to eukaryotic cells in a custom-built electroporation chamber. Eventually other researchers started to work with a short-circuited gel electrophoresis system in order to obtain electrical discharge which could be used.

As is known to molecular biologists, DNA electrotransfer to bacteria is a standard method, and a small pulse generator is everyday equipment in most laboratories. However, whereas the molecular biologist would not hesitate to let billions of bacteria succumb to side effects of high-field electroporation, this view is of course not shared by those working with mammalian cells—and doctors working in the clinic would have safety as the major priority [2].

In order to work with mammalian cells, pulse parameters need to be much better controlled, and this is efficiently done with a square wave electroporator [3]. Here, the pulse amplitude and duration may be individually controlled, allowing a much better adaptability to the target cells or tissues as well as to the molecule that is to be transferred. Another important stage of the development has been the availability of square wave pulse generators designed and approved for clinical use.

Alongside this technological development, important studies have been performed, enabling a much more detailed understanding of how cells and tissues respond to electric fields [4–6], and how the technologies may most optimally be used to transfer molecules or ablate tumor tissue [7–10]. This chapter is devoted to the medical applications of electroporation-based technologies, including how we may understand their scientific basis and exploitation.

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8.2

A List of Definitions

Nomenclature tends to grow with the expansion of the field, and it would be useful to start this chapter with some definitions. These are of course debatable; indeed in the course of history scientific battles have been fought over single words and their use. The following is a list of terms and their meaning as used in this chapter.

Electroporation

Electroporation is use of electric fields to cause cell membrane permeabilization.

Electropermeabilization

This is the same as above. Electropermeabilization is technically more correct, as indeed what we know is that permeabilization structures are created, not necessarily that these have a particular shape, as the word "pore" would imply. However, as the word electroporation has taken precedence and is overwhelmingly used in the literature, this is also the term used in this chapter.

Electrotransfer

This term covers the use of electric pulses to assist transfer of molecules into cells. In principle a membrane would not necessarily need to be measurably porated or permeabilized, possibly only sufficiently destabilized or perturbed to allow electrophoretic transfer across it. Examples of uses are DNA electrotransfer, siRNA electrotransfer, etc.

Reversible Electroporation

This term describes permeabilization that may be reversed to a state where the cell membrane is resealed and the cell may continue its life cycle.

Irreversible Electroporation

This describes a state where a cell will reach a point of no return, a point where depletion of energy, ionic imbalance, and other factors will eventually lead to cell death.

Membrane Perturbation

A membrane may be temporarily destabilized to an extent where effects can occur that would not normally be possible-but the membrane is not measurably permeabilized in terms of molecules passing the membrane.

Pore

A large body of literature describes the occurrence of pores, of various sizes and lifetimes, as the result of electroporation. *In vivo* these pores may not necessarily be neatly rounded structures, but rather a mixture of shapes of permeabilization structures depending on the effect of transmembrane protein domains, cell shape, and surrounding matrix on the cell membrane response to the field, i.e. these permeabilization structures could either be circular in form, or have other forms depending on local factors.

Resealing Time

This is generally represented with τ , and defined as the point in time when twothirds of the membrane permeabilization structures have resealed. Generally this is analyzed by measuring transport of molecules across a membrane.

Electrochemotherapy

Electrochemotherapy is the use of electric pulses to increase uptake of cytotoxic drugs. Drugs used in the clinic include bleomycin and cisplatin.

Voltage-to-Distance Ratio

An applied electric field is very often described in terms of units of V cm⁻¹, indicating the applied voltage relative to the distance between electrodes measured in centimeters. This, in principle, does not describe the electric field in any detail since the actual field will vary depending on a number of factors in the electrode geometry. Direct comparison only works if it is the same set of electrodes being used in the same tissue.

8.3 How We Understand Permeabilization at the Cellular and Tissue Level

Before discussing the medical uses of electroporation, it is useful to look at how we may understand the concept of permeabilization at both the cellular and the tissue level. The simple explanation, which will definitely do for a quick introduction to the field, is shown in Figure 8.1. An electric field is applied that surpasses the capacitance of the membrane; dielectric breakdown of the membrane ensues, most prominently at the side of the cell facing the positive electrode since the cell has an intrinsic negative resting potential. Images of electroporation at the single-cell level may beautifully link theory to empirical findings (Figure 8.1).

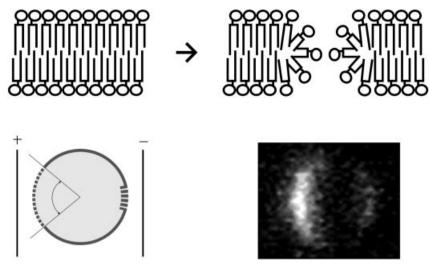


Figure 8.1 Schematic of cell electroporation. A basic understanding is that when the electric field exceeds the capacitance of the cell membrane, permeabilization structures will occur more prominently at the cell pole facing the positive electrode, due to the

inherent negative resting potential of the cell. This is nicely supported by empirical data, the image at lower right showing propidium iodide uptake after pulsation, as described in [37, 38].

However, the situation gets a lot more complex when electroporation is performed in tissues.

- Cells in tissues contain an abundance of proteins with transmembrane domains, meaning that deformation and response to the electric field differ from those of a lipid vesicle, or even a single cell after detachment.
- 2) Cells in tissues are rarely spherical, but rather have all types of shapes, from ellipsoid to square to stellate. This affects deformation in response to the field and the threshold for electroporation may depend on the direction of the field versus the orientation of the cell.
- 3) There are gap junctions, which may mean that one cell may be electrically connected to others, and thereby act as a much larger cell in the electric field [11, 12].
- 4) The influence of neighboring cells and structures [13] has to be taken into account. The field is likely to be inhomogeneous, and depends on the presence of neighboring structures such as high-conductive areas with fluid, or low-conductive areas on bone surfaces. Furthermore, there may be an influence

of permeabilization on neighboring cells, meaning that there is a dynamic development when one long or several shorter pulses are applied.

- 5) In tissues, the extracellular volume is small compared to the intracellular volume–completely the opposite of the situation when performing *in vitro* electroporation where the suspension medium volume by far exceeds the intracellular volume. This means that the threshold for electroporation is lowered [5]. It also means that the cell may tolerate permeabilization better, for example, since loss of ionic homeostasis may be less pronounced [6].
- 6) Furthermore, whereas a suspension medium may be viewed as a liquid, this would be an imprecise description of the extracellular volume. The extracellular volume is somewhere between a gel and a semistructured solid, containing extracellular matrix with complex glycoproteins. Indeed, the cell interior has also been described as a gel or molecular soup [14], so one could view tissue as gel-like or semistructured matter that contains lipid bilayer barriers in rich quantity.

This difference between the *in vitro* situation, which has of course served as a very important research object to understand basic points about electroporation, and tissue response to electric fields is important to highlight because proper attention must be paid to consequences of these differences. Schematic representations of this more complex situation are attempted in Figures 8.2 and 8.3.

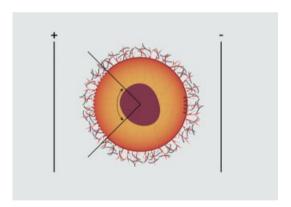


Figure 8.2 Cell electroporation: a more complex view. The cell membrane as shown in Figure 8.1 is of course a simplistic view of the situation. A more detailed (but far from complete) view would be including the cytoskeleton (dotted lines), which has also been shown to be involved in both formation and resealing of permeabilization structures.

Also, transmembrane proteins are schematically depicted, indicating that the bilayer lipid membrane is more than just that, and also that molecules entering into the cell as a result of electrotransfer do so after negotiating both the extracellelular domains of charged proteins and the cell membrane itself.

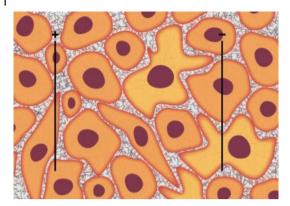


Figure 8.3 Cell electroporation at the tissue level. Yet another layer of complexity is added when the cell is viewed as part of a tissue. The extracellular volume is not a liquid medium where molecules freely diffuse, but rather a dense mesh of structures and

molecules interspersed between cells of varying sizes and shapes, and all linked with transmembrane proteins and intracellular structures. The distribution of molecules and the electric field distribution will necessarily be dynamic and also nonlinearly dynamic.

8.4 Basic Aspects of Electroporation that are of Particular Importance for Medical Use

8.4.1 **Delivery of Drugs**

There are basically two delivery routes for drugs: intravenous delivery and intratumoral delivery [10, 15]. Intravenous delivery is being extensively used for the chemotherapeutic agent bleomycin, with good success. Indeed there are several aspects that make this a favorable solution. First, bleomycin does not readily enter cells. Therefore, an intravenous delivery will have a favorable toxicity profile despite whole-body exposure to the drug. Second, the increase in toxicity is very marked once direct access to the cell cytosol is provided; thus even a limited local dose will exert a sufficient effect. Other drugs to be administered intravenously for delivery during electroporation would need to be able to fulfill this criterion, in order to have both sufficient effects in the treated area as well as a favorable toxicity profile.

Intratumoral delivery (or, in the case of drugs with other purposes, delivery to the local tissue) has the advantage of low whole-body exposure. On the other hand, the distribution of drug in the tissue will be uneven. Thus, washout to the systemic circulation is bound to occur. Washout may be delayed if local analgesic injections contain epinephrine, and also if the injection is timed so that cells are permeabilized immediately around the time of injection, as this will allow the permeabilized cells to take up drug at maximal concentration, and as the cells reseal, the maximal drug concentration could indeed be trapped inside the cells allowing sequestration of the drug intracellularly.

8.4.2 **Delivery of DNA**

DNA is a large and highly charged molecule with very limited diffusion capacity by itself. Injection in suspension may add a hydrostatic pressure that can aid the distribution in a tissue. However, there is also an advantage to this situation, which is that DNA also does not readily diffuse *out* of the injected area easily. Thus it has been shown that pulse delivery even minutes after DNA injection still leads to excellent levels of transgenic expression [16, 17].

Intravenous distribution of DNA has been tried without success. This is because DNA is degraded by DNAases and furthermore the whole-body distribution means that the concentration at the tissue level becomes very low. An intermediate solution is arterial injection, for example in the hepatic artery, and this may lead to an acceptable concentration at the tissue level, with resultant transgenic expression.

Certain tissues may allow distribution of a hydrophilic solution more easily than others, with muscle being an example of a tissue that allows distribution along fibers. In contrast, brain tissue is highly lipophilic in nature, giving a more focal distribution pattern for hydrophilic solutions.

DNA will move in a tissue as a function of electrophoretic forces on the polyanion. This has been measured, and is estimated to have importance only on a very small scale in terms of tissue distribution [18].

It has been shown that DNA must be administered *prior* to the application of an electric field, as DNA is unable to diffuse through cell membranes by itself [16]. DNA electrotransfer (Figure 8.4) works by steps involving membrane adsorption, membrane perturbation (by the electric field), electrophoretic movement of the DNA, and finally cellular processing of the internalized molecule, as nicely reviewed in [19].

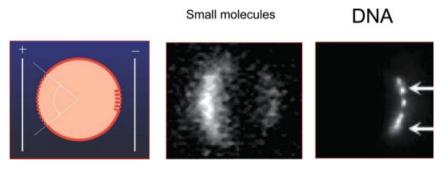


Figure 8.4 Schematic of DNA electrotransfer. Whereas small molecules such as used for chemotherapy diffuse through small permeabilization structures of cell membranes, this is not the case for the highly

charged polyanion DNA. As can be seen on the right, DNA enters the cell moving towards the positive electrode driven by electrophoretic forces, as demonstrated in [40].

8.4.3

Delivery of Other Molecules

A number of other molecules may be delivered by electroporation: oligonucleotides [20], ions [21, 22], dyes [23], and radioactive tracers [24]—just to mention a few. For *in vivo* injection it is important to consider the following.

- 1) Washout of the molecule from tissue. Is it a very rapid process, for example, as seen for ⁵¹Cr-EDTA which has an extremely rapid renal clearance? Or will the molecule remain in the target area much longer, as would be the case for proteins and larger nucleotides?
- 2) Charge of the molecule, which may affect uptake by electroporation at the cellular level, but charge may also lead to sequestration in the extracellular volume. An example of an oligonucleotide class (peptide nucleic acid) and how charge may play an important role are given in Joergensen *et al.* [20].

8.4.4

Delivery of Electric Pulses

Delivery of electric pulses is a science in itself, and a tremendous variety of pulse combinations is possible. Increasing the applied voltage will lead to increased permeabilization to the point of irreversible permeabilization. Increasing the number of pulses will also lead to increased permeabilization, but the correlation is only linear in the range of about 1–20 pulses. Usually the number of pulses chosen is 8–10, which gives a reasonable trade-off.

Pulse length is the third very important issue. Generally, short high-voltage pulses of, for example, $100\,\mu s$ are used to augment uptake of small molecules, as it has been shown that the pulse amplitude defines the extent (and degree) of permeabilization, and so high-voltage pulses will lead to a large area of the membrane being permeabilized and thus open for diffusion of molecules. For DNA, typically long pulses are used, up to $400\,m s$ long and with lower voltage. This allows an electrophoretic effect assisting DNA passage into the cytosol. Pulse characteristics used differ greatly, both as far as what is recommended for individual tissues and what is recommended from individual investigators.

8.4.5

End of the Permeabilized State

Resealing of cells after the end of the permeabilized state happens in part through energy-consuming processes and spontaneous resealing. Cells may succumb due to resealing not occurring before ionic imbalance and adenosine triphosphate loss bringing the life cycle to an end. The threshold for reversible and irreversible permeabilization may be well defined, although in a particular tissue the two phenomena may coexist due to variations in cell size and electrode geometry. Resealing is well described, and occurs in a time scale of the order of minutes.

8.4.6

The Vascular Lock

A particular phenomenon of great importance for the antitumor effect of electrochemotherapy is an antivascular effect. This effect has several elements:

- 1) A direct reflexory constriction at the level of the arterioles, lasting only a few minutes, in direct response to the electric pulses [25].
- 2) A permeabilizing effect on the tumor vessels leading to collapse of intratumoral capillaries [26].
- 3) A high interstitial pressure in the tumor combined with additional edema due to leakage of increased intracellular sodium after permeabilization, leading to further compression of the already collapsed tumor capillaries.
- 4) A direct toxic effect on endothelial cells as well as on tumor cells due to the internalization of bleomycin [27].

All these factors mean that electrochemotherapy is highly useful in the treatment of hemorrhaging tumors [28] (see also Figure 8.6).

8.5

How to Deliver Electric Pulses in Patient Treatment

8.5.1

Pulse Generators and Electrodes

For electrochemotherapy there is at this point only the Cliniporator equipment available (Carpi, IGEA, Italy). This generator allows simultaneous collection of voltage and current traces, and allows storage of treatment data, and is CE marked for clinical use. It works together with two types of needle electrodes as well as plate electrodes for cutaneous applications. It can also be used with endoscopic electrodes (Section 8.8.1), expandable soft tissue electrodes, and pin electrodes for bone metastases (Section 8.8.2). It can be used for gene therapy, as reviewed in [3].

8.5.2

Anesthesia

For smaller tumors, not too numerous and in areas where it may be applied, local anesthesia would be the optimal choice. Generally, an infiltration analgesia around the area treated gives good coverage, and the patient may be discharged immediately after the treatment procedure. It has been shown that the actual applied voltage is important for both the level of muscle contraction and the pain sensation, and therefore needle electrodes with, for example, a 4mm gap may be preferred for the treatment with local anesthesia. Performed according to standard operating procedures, pain sensation should be minimal [15].

For larger tumors, multiple tumors, and tumors in areas where it may be difficult to apply sufficient local anesthesia (e.g., periost involvement in the scalp), general anesthesia is recommended. This may be handled by short-acting anesthetic agents such as propofol combined with short-acting opioids [15]. As pulses should be administered within approximately a 20 min window, the anesthesia can be short and the patient discharged on the same day [9].

8.6 Treatment and Post-treatment Management

Underlying muscle may be exited directly by the electric pulse, leading to a brief contraction. These contractions are limited to the muscle immediately beneath the treatment area and may be alleviated by lifting the lesion during treatment.

When using very high field strengths (e.g., as is used in the irreversible electroporation algorithm), interference with heart rhythm has been observed. This has been alleviated by the development of pulse generators avoiding the refractory phase of the heart cycle. However, for pulses used in electrochemotherapy no adverse events have been reported, despite use of electric pulses directly on the left chest wall.

After electrochemotherapy, necrosis/apoptosis will immediately ensue. If the tumor treated is superficially located, for example in the skin, an ulcerated area will form as a transition between the tumor and scar tissue. In particular in previously irradiated areas, this may take several weeks [9]. Standard wound care may be applied.

8.7 Clinical Results with Electrochemotherapy

8.7.1

Tumors Up to Three Centimeters in Size

The first clinical trial of electrochemotherapy was published in 1993 [29], with treatment of cutaneous metastases from head and neck cancer. This was followed by several studies of patients with malignant melanoma metastases and basal cell carcinoma [30, 31]. In the first study, intravenous or intraarterial bleomycin was used, and in the latter intratumoral injection of 5 U ml-1 was used, for both the tumor and the surrounding area. In the first study, plate electrodes were used, where as in the latter a needle electrode device was used.

In a European Union framework program, four centers worked together on defining standard operating procedures for electrochemotherapy. Three different types of electrodes were used, local drug administration of either cisplatin or bleomycin (here in a 1 U ml⁻¹ concentration) or intravenous bleomycin was used, and either local or general anesthesia. In 41 patients, 171 tumor nodules were treated

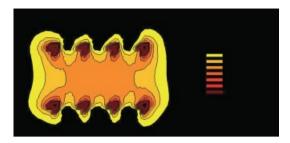


Figure 8.5 Example of electric field distribution. Assuming homogeneous conductivity (which is indeed an unlikely event), the field distribution between two arrays of opposing electrodes is shown. It is obvious that when

speaking of applied voltage at the level of the electrodes, the actual field to which cells are exposed will depend on electrode geometry and conductivity. From [5].

and the result was that 73% of tumors were in complete remission, 11% in partial remission, and other lesions either no change or progression [10]. It was found that the three different administration types gave similar results, except in larger tumors where intravenous bleomycin seemed superior, likely associated with better distribution in the entire tumor volume. Furthermore, it was noted that there was a tissue-sparing effect in normal tissues (Figure 8.5). This is observed for both intravenous bleomycin and local injection of concentration $1 \, \text{U ml}^{-1}$, whereas normal tissue necrosis may ensue with a concentration of $5 \, \text{U ml}^{-1}$. Thus, with the high local concentration there may be a higher tumor response rate but at a cost of more local tissue damage. As electrochemotherapy can be repeated if necessary, it is this author's opinion that a good solution is to not use the higher concentration of bleomycin in order to obtain better preservation of normal tissues, while it will still be possible to retreat the approximately 25% of patients in need of better tumor control.

In 2006, Mir *et al.* [15] described standard operating procedures for electrochemotherapy of tumors up to 3 cm, and these are now frequently referred to as a standard of care. Several studies, most of them based on the 2006 standard operating procedures, have now been published, and confirm both the level of response rates and toxicity profile.

What is clear from the above mentioned studies is that electrochemotherapy works for a number of different tumor histologies. This may initially seem surprising, but is completely logical when taking the scientific background into account. First, the cell membrane is a unifying feature between all types of tumors, and although there may be slight variations in the threshold for permeabilization, if sufficient pulsing parameters are used all cells may be permeabilized. Furthermore, in the case of bleomycin, the increase in cytotoxicity is over 300-fold, which is far beyond what resistance mechanisms in cells may handle. This means (i) that all tumor histologies treated (planocullular carcinoma, adenocarcinoma,







Figure 8.6 Melanoma treatment, showing a patient with ulcerated metastasis from malignant melanoma. (a) Before treatment and (b) after treatment with intravenous bleomycin and electroporation of the tumor area under general anesthesia. As can be seen, there are needle marks outside the

treatment area in the normal tissue, indicating that tumor tissue is more responsive to the treatment (necrosis of lesion) than normal tissue (only needle marks present). (c) Six months follow-up showing complete resolution without recurrence. From [39].

transitocellular carcinoma, hypernephroma, malignant melanoma, basocellular carcinoma) have responded to electrochemotherapy, (ii) that tumors that generally do not show high response rates to chemotherapy (e.g., malignant melanoma; Figure 8.6) respond, and (iii) that tumors previously having progressed on chemotherapy may respond.

8.7.2

Larger Tumors

Whereas the initial studies were focused on assessing the response rates and safety profiles of smaller tumors, it was clear after the first results that also patients suffering from larger tumor recurrences would potentially benefit. One of the first case reports published was the treatment of a large chest wall recurrence of breast cancer, and this subsequently has prompted a phase II clinical investigation of the treatment of these tumors (www.clinicaltrials.gov). Initial results show impressive responses in a heavily pretreated patient population, with a favorable safety profile and patient-reported symptom alleviation.

Particular issues are relevant when treating large tumors. Thus healing time will be delayed as normal tissue needs to grow into the treatment site to replace necrotic tissue. This may be further delayed in previously irradiated tissues, where angiogenesis and wound healing is delayed.

8.8 Use in Internal Organs

After the excellent results on cutaneous tumors, an obvious question has been how to transfer these results to the treatment of internal tumors. At the time of writing, at least three different approaches are being pursued.

8.8.1

Endoscopic Use

An endoscopic applicator with a vacuum device has been developed. This enables endoscopic application and suction of a tumor into the device with subsequent application of drugs/genes and pulses. A study in veterinary clinics in spontaneous tumors in dogs has been carried out successfully (D. Soden, personal communication), and a phase I clinical trial has now opened for patients suffering from colorectal cancer (www.clinicaltrials.gov; NCT01172860).

8.8.2

Bone Metastases

A system with very stiff electrode pins has been developed for the treatment of bone metastases. In this system, a bone metastasis is defined, and pins positioned according to a treatment planning system, in and on all sides of the lesion. After pulse delivery the pins are removed and treatment completed. Extensive preclinical data have been obtained [32] and the first phase I clinical trial has commenced.

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Brain Metastases, Brain Tumors, and Other Tumors in Soft Tissues

A novel electrode has been developed for the use of electrochemotherapy in soft tissues, the primary target being the brain (Figure 8.7). This electrode is expandable, so that an area larger than the insertion point may be treated. Preclinical data have shown results that are in essence comparable with clinical results from skin tumors, with a high level of complete responses [33]. A clinical trial has been opened (www.clinicaltrials.gov).

8.8.4

Liver Metastases

Finally, using the stiff electrode pins, a trial on the treatment of liver metastases has commenced (www.clinicaltrials.gov; NCT01264952) also using a treatment planning system.

8.9

Gene Electrotransfer

Gene transfer is a very exciting application of electroporation. The application is in principle very simple: DNA is injected in the target tissue, pulses are applied, and in this way gene therapy may be performed in a matter of a minute. Gene electrotransfer is nonviral, meaning better patient safety, reduced concerns for staff working safety, as well as markedly reduced cost of the procedure. The fact

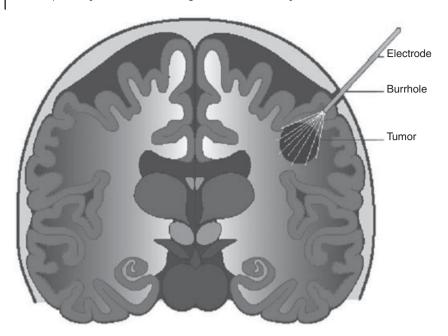


Figure 8.7 Brain electrode. Novel treatment strategies involve enabling the technology to work in internal tumors (see text). An example of this is an electrode developed for use in

soft tissues, exemplified by tumors of the brain. Preclinical results using a prototype electrode are described in [33], and a clinical trial has commenced (www.clinicaltrials.gov).

that transgene expression only occurs in the particular region where injection and pulses are performed has an advantage in that the transfection area may be controlled in extent by the application in itself.

The transgenic expression is transient, as the DNA is not integrated in the host genome. This means that expression will depend on the longevity of the target cells:

- 1) Transfer to muscle cells, which have a very long lifespan, will lead to long-term secretion of the transgene, useful in protein deficiency syndromes.
- Transfer to tumor cells will be short lived, and can be used for expression of proteins with tumoricidal effects, for example, antiangiogenic factors or cytokines.
- 3) Transfer to skin will lead to somewhat longer durations of transgenic expression (a few days to a few weeks), and this may be very useful for vaccinations, possibly also in the treatment of skin diseases.

Gene electrotransfer is a process involving several steps:

- 1) DNA must be approximated to the target cells. As DNA is large and highly charged, diffusion is slow. Once the DNA is in the vicinity of the target cell, it must adsorb to the cell membrane prior to transfer by electric pulses.
- During the pulse application, DNA will interact with the cell membrane, after which DNA will either be internalized or trapped in the membrane (and then possibly later internalized).
- 3) Once in the cytosol, active transport to the nucleus occurs, and the microtubular system as well as the transporter dynein have been shown to play an important role in this process. Actin is also implicated in the early processes, and possibly further mechanisms for intracellular trafficking may coexist.
- 4) In order for the transgenic expression to take place, a number of factors are important. First, the cell must be in good condition, in order to express the transgene, and indeed it has been shown that optimal conditions for expression are at pulsing parameters where the cell is minimally perturbed, in terms of level of membrane permeabilization [6]. Second, choice of promoter will be important for transgenic expression, an important subject which falls outside the scope of this chapter. Third, the transgenic protein may or may not have immunogenic effects, and quenching may affect measurable levels of protein.

8.9.1

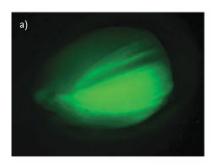
Gene Electrotransfer to Muscle

Muscle is clearly a very interesting target for gene electrotransfer. It is important for gene electrotransfer that the plasmid does not get integrated into the host genome. Muscle cells have a long lifespan, enabling long-term expression of the transgene. Thus, it has been shown that gene electrotransfer to muscle (Figure 8.8) enables highly efficient [16] expression for many months in rodent experiments. In humans it may actually be a life-long expression, yet to be shown. Naturally gene expression to muscle would be ideal for protein deficiency syndromes such as hemophilia or alpha-trypsin-1 deficiency. However, gene electrotransfer may also be used for expressing therapeutic molecules such as anticancer agents. Various inducible systems exist for experimental models, where expression of a transgene may be induced by addition of a certain molecule, for example doxycycline [34].

8.9.2

Gene Electrotransfer to Skin

Gene electrotransfer to skin leads to shorter term expression, from days to a few weeks [7, 35]. The skin consists of a number of different cell types, for example



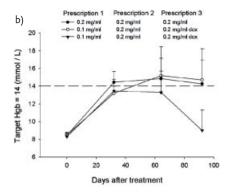


Figure 8.8 Gene electrotransfer to muscle. (a) Through simple injection of naked DNA intramuscularly followed be electric pulses, high-level and long-term expression may be obtained, as seen with transfection using the green fluorescent protein. (b) Transgenic

expression of erythropoietin from a small muscle may in fact lead to therapeutic levels of the transgene, as evidenced by a dramatic increase in hemoglobin (Hgb) levels, erythropoietin stimulating the formation of new erythrocytes. From [34].

keratinocytes, Langerhans cells, and fibroblasts, and transfection levels may depend on depth of DNA injection and pulsing parameters. The skin has a number of different properties, and one very interesting aspect is its immunological properties. DNA vaccinations are likely to play an important role in the future, for several reasons, and in this aspect gene electrotransfer to skin may be very important as it allows good transfection rates of antigenic proteins in an immunologically active organ. Furthermore, the duration of expression of transgenic proteins would fit well with a vaccination strategy. Possible exploitations of gene electrotransfer to skin are represented schematically in Figure 8.9.

8.9.3 Gene Electrotransfer to Tumors

Gene electrotransfer to tumors may lead to high levels of an anticancer molecule directly at the tumor site. It has been shown in a number of preclinical studies how this may be utilized, and also the first clinical study on the use of gene electrotransfer to tumors was published in 2008 [36]. In this study, the plasmid coding for interleukin-12 (IL-12) was transferred to tumors in patients suffering from disseminated malignant melanoma. IL-12 is a cytokine known to have potent antitumor effects but some clinical studies have shown that systemic administration led to high levels of toxicity. By transferring the gene directly into the tumor, high local levels may be obtained while toxicity at the systemic level can be moderate. Indeed, in this study [36], significantly increased levels of IL-12 were shown in transfected tumors while overall toxicity was low. At the same time responses locally at the transfected site were observed, as well as distant responses, likely mediated by an immune response elicited from the treated areas.

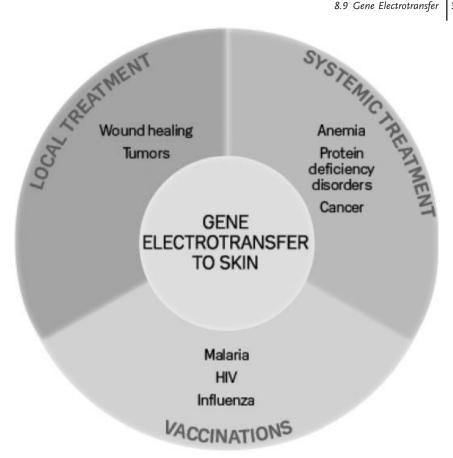


Figure 8.9 Gene electrotransfer to skin. Schematic of how gene electrotransfer to skin may be used in a variety of medical conditions, from vaccination to cancer treatment. Gene electrotransfer to skin gives rise to transgenic expression lasting a couple of weeks [7, 35].

Gene electrotransfer using other molecules may also be envisaged. For example, an ongoing trial is looking at the possibility to obtain responses by transfection of a plasmid coding for an antiangiogenic and antiproliferative molecule (www.clinicaltrials.gov; NCT01045915).

8.9.4 Gene Electrotransfer to Other Tissues

Gene electrotransfer to other tissues, for example kidney, testes and eyes, is extensively reviewed elsewhere and a detailed description falls beyond this chapter.

8.10

Conclusions

Drug and gene electrotransfer has a number of highly interesting perspectives. Clinical feasibility has been shown, and numerous novel applications are in development. Electrochemotherapy is estimated to become an important part of the armamentarium in the treatment of cancer of various histologies, and anatomical locations.

Gene therapy by electrotransfer has potential use in a number of different applications, from vaccinations to treatment of protein deficiencies such as hemophilia, or cancer treatment.

It will be very interesting to follow developments in the coming years, as this platform technology becomes increasingly used in a number of medical applications.

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